

Ultra-High Temperature Ceramic Composites for Leading Edges - 2004

Stanley R. Levine, Elizabeth J. Opila, Jonathan A. Lorincz[†], Mrityunjay Singh*, Raymond C. Robinson*, Donald T. Ellerby[^], and Matthew J. Gasch[#]

NASA Glenn Research Center, Brookpark, OH

[†]Ohio University, Athen, OH

*QSS, NASA Glenn Research Center, Brookpark, OH

[^]NASA Ames Research Center, Moffet Field, CA

[#] Eloret, NASA Ames Research Center, Moffet Field, CA

Ultrahigh temperature ceramics (UHTC) have performed unreliably due to material flaws and attachment design. These deficiencies are brought to the fore by the low fracture toughness and thermal shock resistance of UHTC. If these deficiencies are overcome, we are still faced with poor oxidation resistance as a limitation on UHTC applicability to reusable launch vehicles. We have been addressing the deficiencies of UHTC for the past two years via a small task at GRC that is in the Airframe part of the Next Generation Launch Technology Program. Our focus is on composite constructions and functional grading to address the mechanical issues and on composition modification to address the oxidation issue. The progress on approaches to improving oxidation resistance by alloying and functional grading will be reported.

Keywords: ceramics, composites, carbon fibers, borides, oxidation

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Stanley R. Levine, Elizabeth J. Opilla,
Jonathan A. Lorincz, Raymond C. Robinson,
and Mrityunjay Singh
NASA Glenn Research Center
Cleveland, Ohio 44135
and

Donald T. Ellerby, and Matthew J. Gasch
NASA Ames Research Center
Moffet Field, CA

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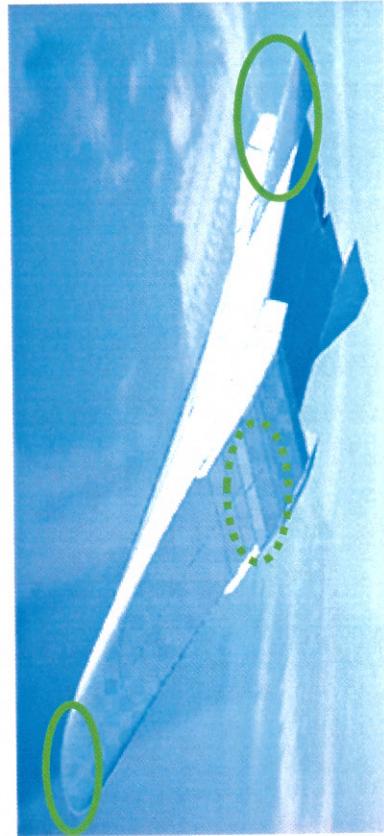


Ultra-High Temperature Ceramic Composites for Leading Edges

OUTLINE

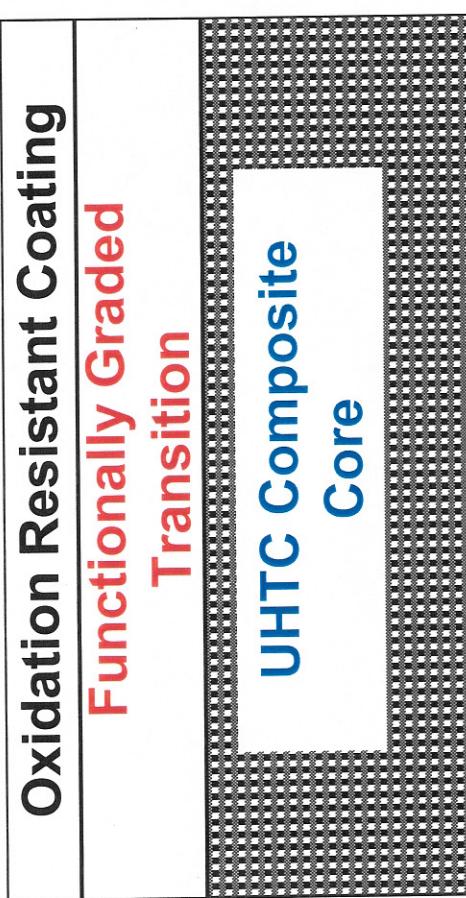
- ♦ Background
- ♦ Objective
- ♦ Ta Modification of ZrB₂ - SiC
 - Furnace oxidation
 - Arc jet test
- ♦ UHTCC
- ♦ Processing
 - Characterization of Starfire UHTCC
- ♦ Concluding Remarks

Ultra High Temperature Ceramic Composites (UHTCC) for Leading Edges



Key Issues

- ◆ Thermal stress resistance
- ◆ Oxidation resistance
- ◆ Temperature capability
- ◆ Architecture optimization



Objective

- ◆ Develop UHTCC for $\geq 3600^{\circ}\text{F}$, reusable, hypersonic vehicle leading edges

Mechanisms for ZrO_2 Protective Scale Enhancement via Ta_2O_5 Additions

- ◆ Ta_2O_5 as a glass modifier
 - Talmey et al
- ◆ Ta^{+5} as a dopant in ZrO_2 lattice
 - Fill O^{-2} vacancies
 - Decrease O^{-2} transport
- ◆ Ta_2O_5 as a major oxide scale constituent
 - Phase V
 - Low melting point

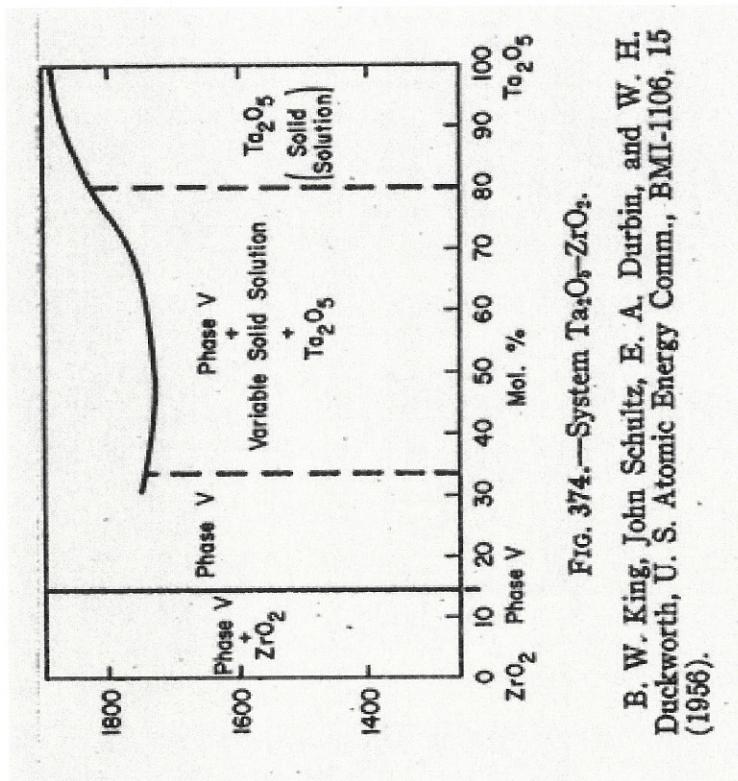


FIG. 374.—System $Ta_2O_5-ZrO_2$.

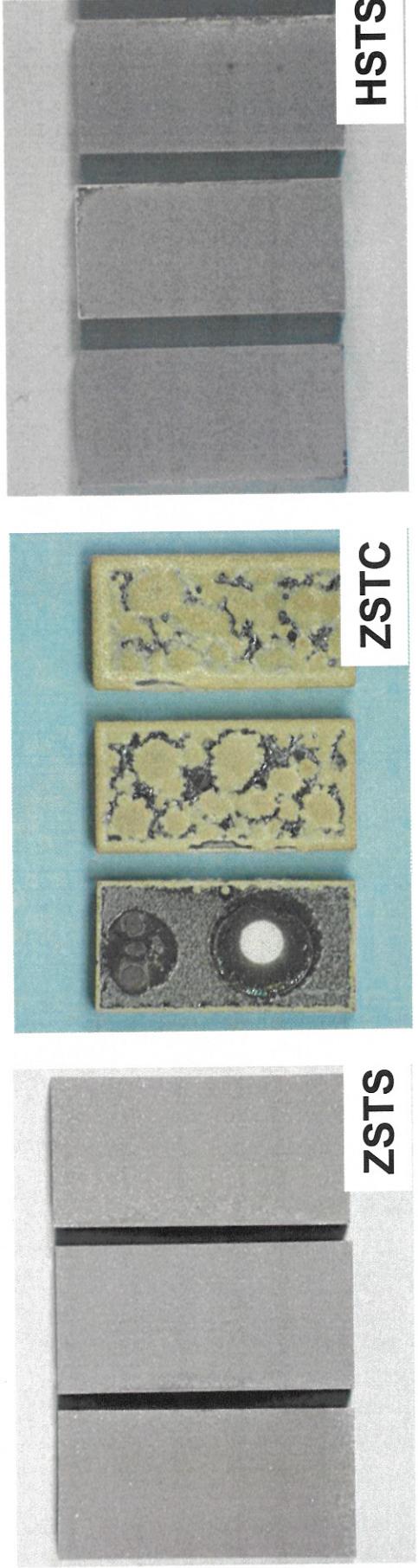
B. W. King, John Schultz, E. A. Durbin, and W. H. Duckworth, U. S. Atomic Energy Comm., BMI-1106, 15 (1956).

Materials Studied

COMPOSITION	DESIGNATION
$\text{ZrB}_2 + 20\text{v/o SiC}$	ZS
$\text{HfB}_2 + 20\text{v/o SiC}$	HS
$\text{ZrB}_2 + 20\text{v/o SiC} + 20\text{v/o TaSi}_2$	ZSTS
$\text{ZrB}_2 + 33\text{v/o SiC}$	ZS+
$\text{ZrB}_2 + 20\text{v/o SiC} + 20\text{v/o TaC}$	ZSTC
$\text{HfB}_2 + 20\text{v/o SiC} + 20\text{v/o TaSi}_2$	HSTS

- ◆ powders ball milled, hot pressed ~ 2000°C, 10 ksi, 2h, vacuum
- ◆ 2.54 x 1.2 x 0.3 cm coupons machined from hot pressed plates
- ◆ coupons cleaned, weighed, surface area determined

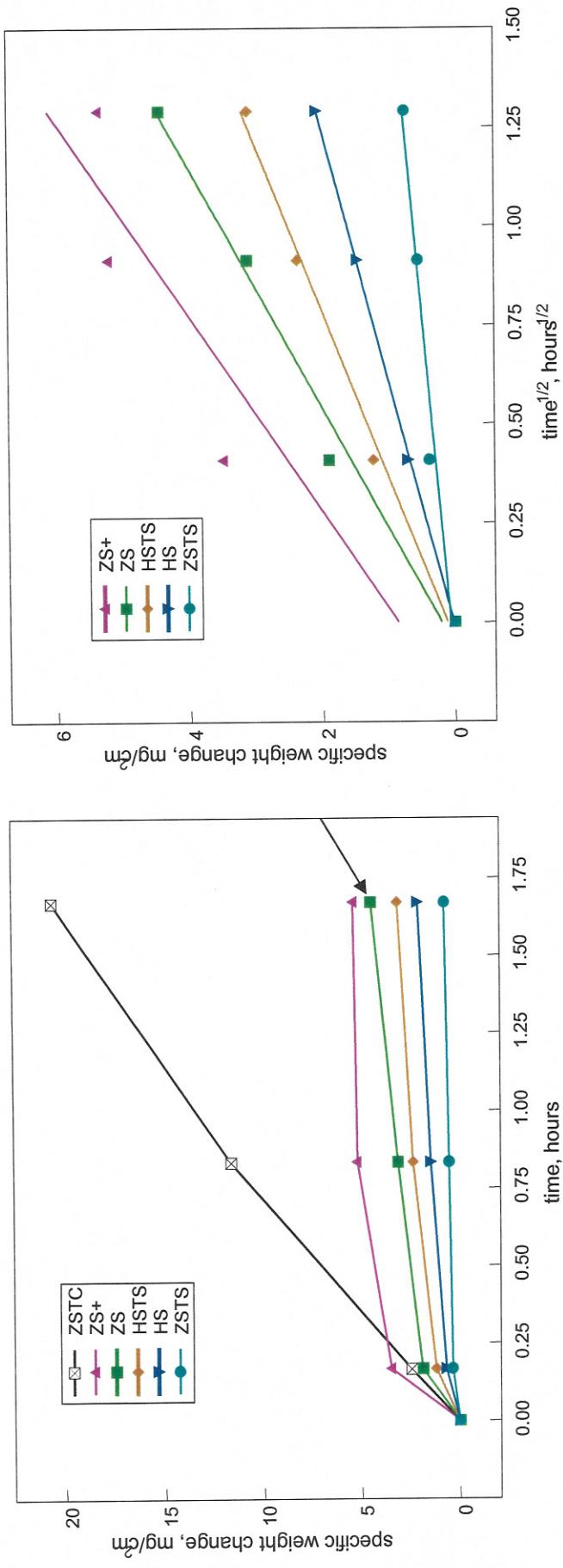
UHTC Coupon Appearance After Oxidation 1627°C , Stagnant Air



Left to right: 1, 5, and 10 ten-minute cycles

Ta-containing compositions show less glass formation

Weight Change of UHTC After Oxidation 1627°C , Stagnant Air



- ◆ Poor results for ZS+ relative to ZSTS show that Ta-addition, not additional Si, is responsible for improved oxidation resistance of ZSTS.
- ◆ TaC additions do not improve oxidation resistance at 1627°C .
- ◆ TaSi_2 additions do not improve HS.

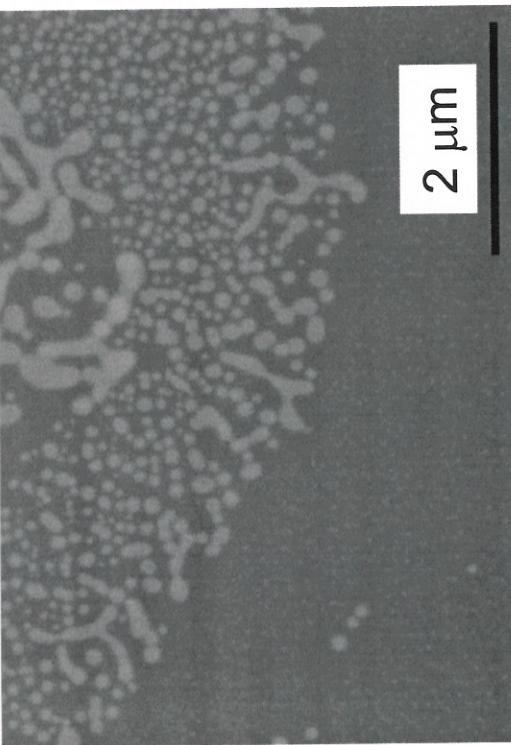
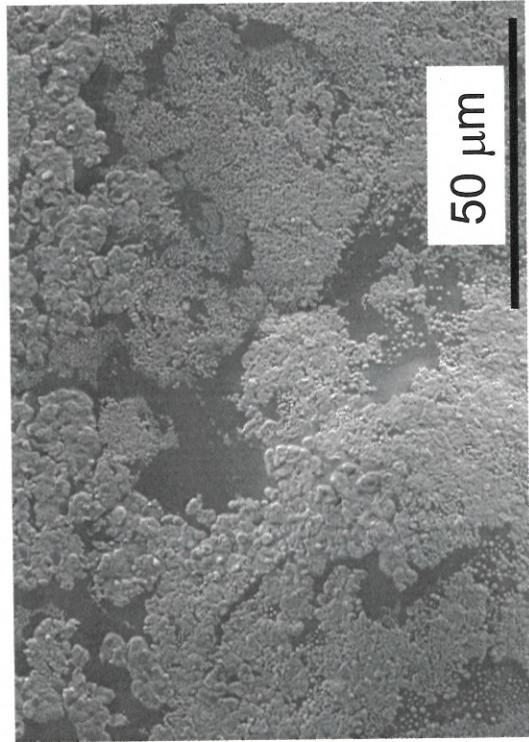
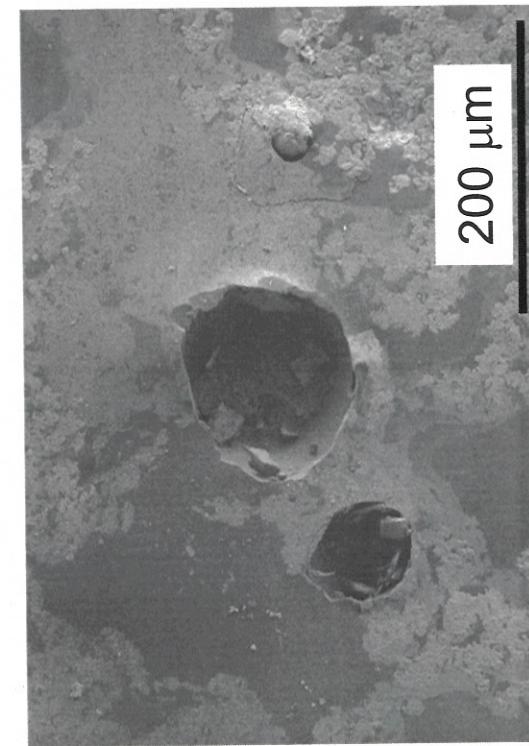
XRD Results for UHTC After Oxidation at 1627°C in Stagnant Air for 100 Min.

composition	phases present
ZS	ZrO₂(m) , ZrO₂(c)
HS	HfO₂(m) , HfO₂(c) , HfSiO₄
ZSTS	ZrO₂(m) , ZrO₂(c)
ZS+	ZrO₂(m) , ZrO₂(c)
ZSTC*	ZrO₂(t) , ZrO₂(m)
HSTS	HfO₂(m) , HfO₂(c) , HfSiO₄

Ta₂O₅·6ZrO₂ (orthorhombic) difficult to distinguish from ZrO₂(c)

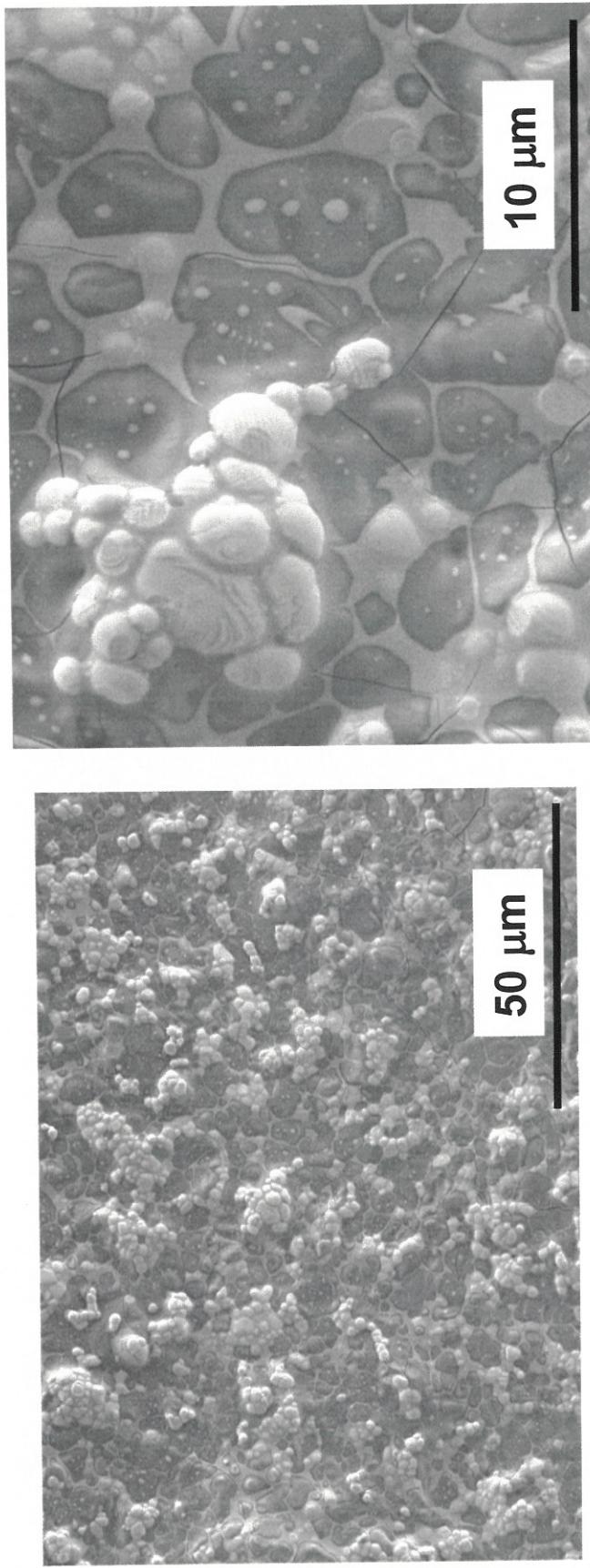
bold = major phase

Surface Microstructure of ZS After Oxidation at 1627°C in Stagnant Air



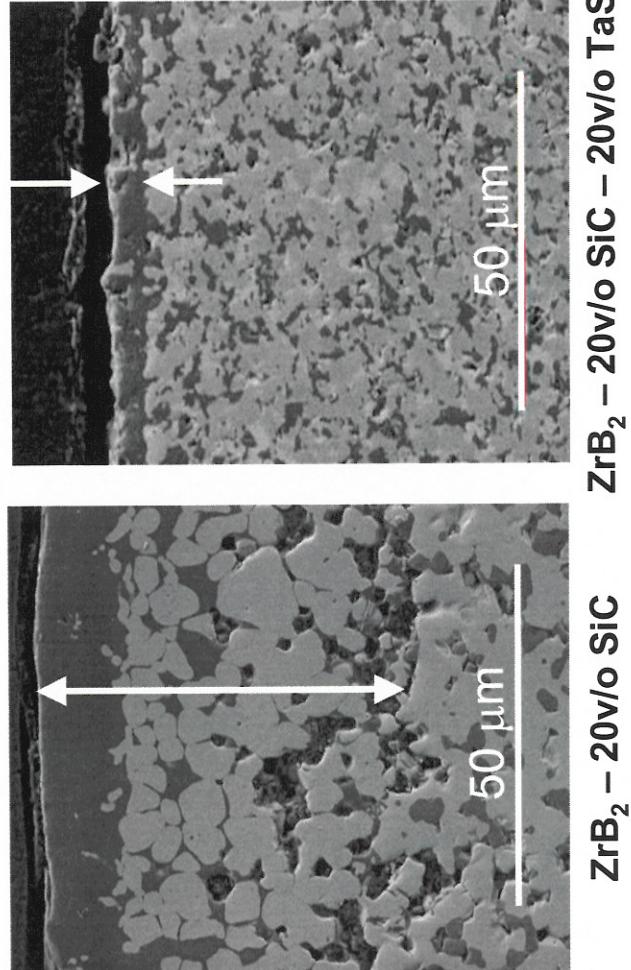
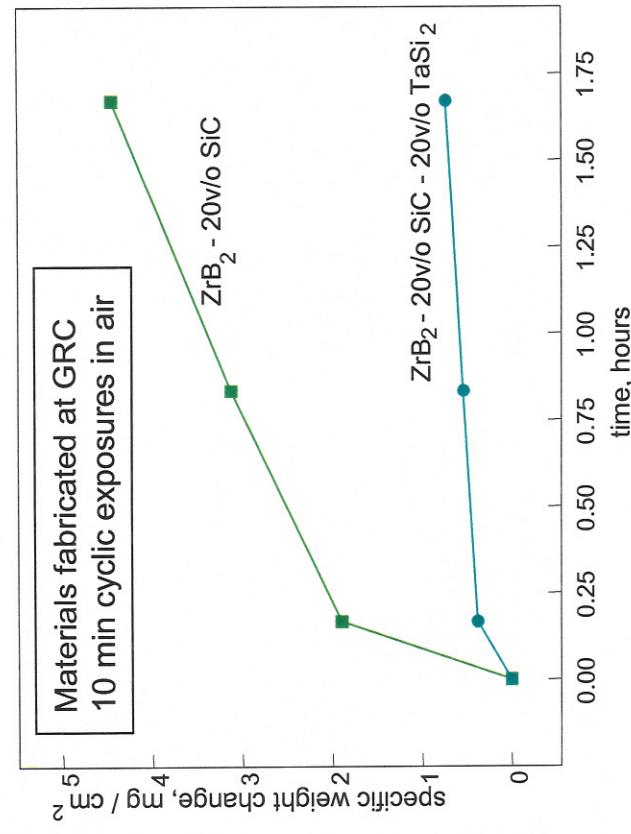
- ◆ Non-uniform dispersion of surface oxide phases.
- ◆ Fine dispersion of second phase in glassy areas.

Surface Microstructure of ZSTS After Oxidation at 1627°C in Stagnant Air

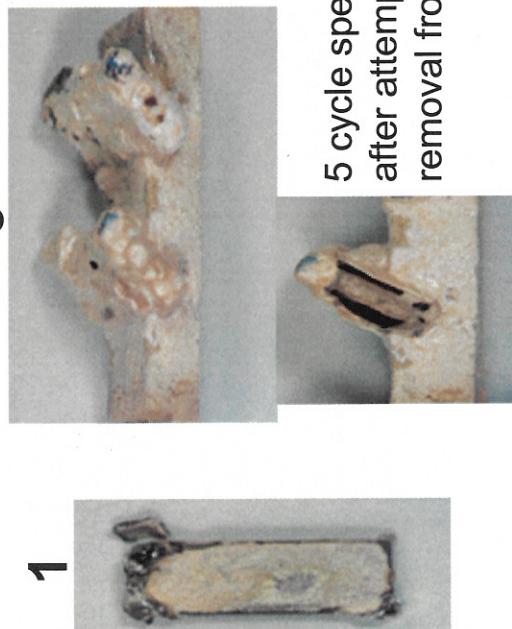
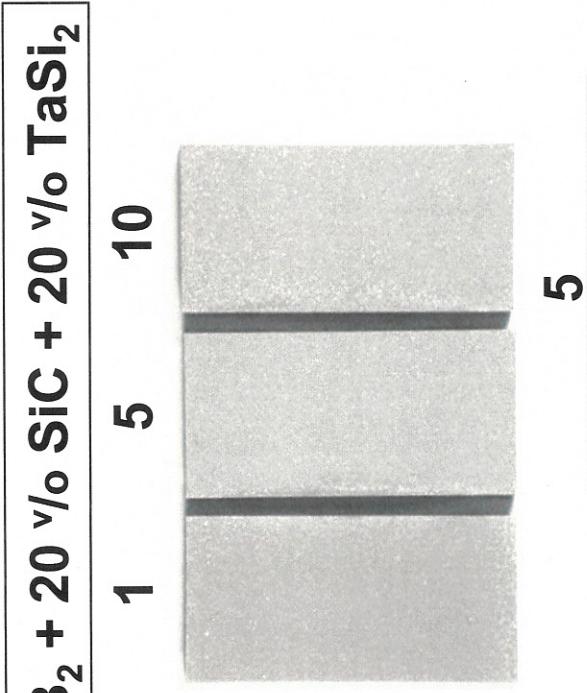
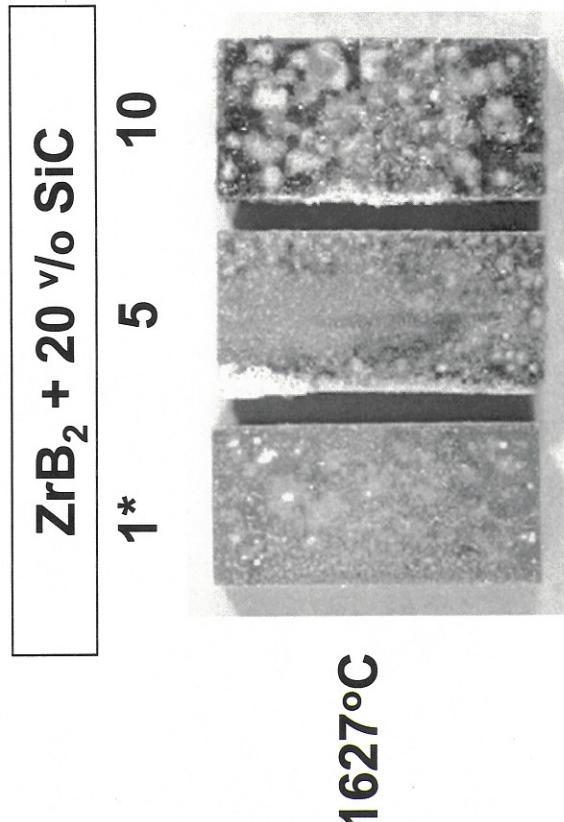


- ♦ Uniform dispersion of surface oxides
- ♦ Evidence of glass immiscibility

Tantalum Additions Improve Oxidation Resistance of ZrB_2 -20v/o SiC at 1627°C

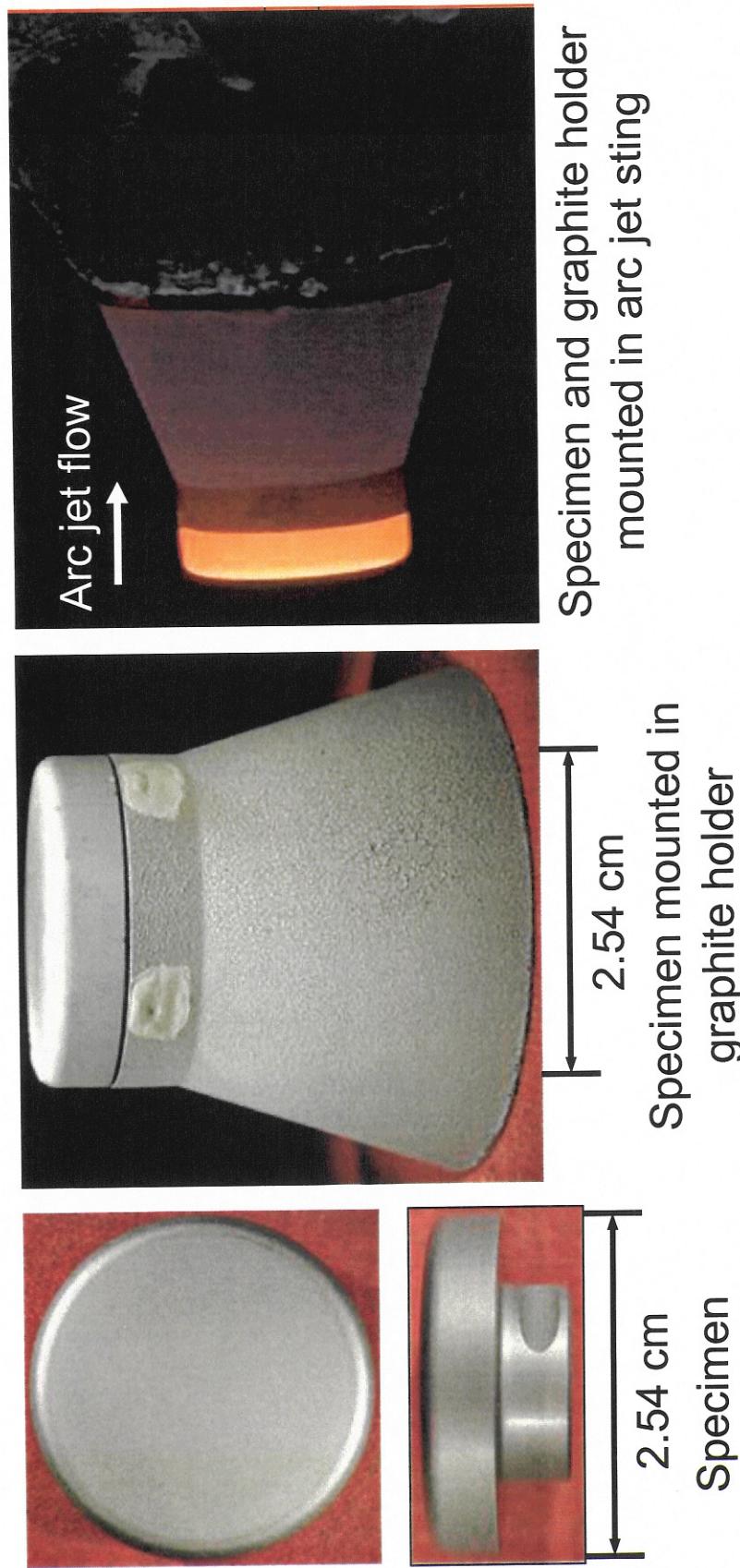


UHTC Furnace Oxidized in Air



* # of cycles (1 cycle = 10 minutes hot & 10 minutes cool)

Arc Jet Test of ZrB₂ - 20 v/o SiC - 20 v/o TaSi₂



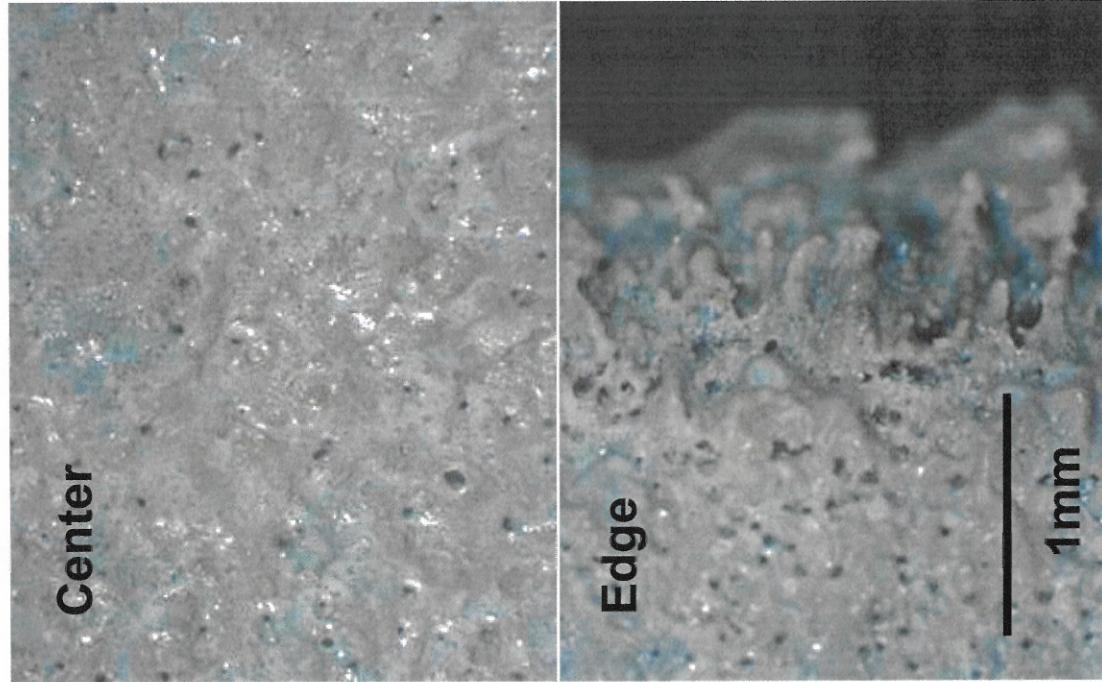
Test Conditions

350 W/cm²
0.07 atm.
600 seconds
% Δwt = -1.4

Temperatures and Heat Flux

Measured Temperature ~1800 °C
Edge Heat Flux Estimate ~ 600 W/cm²
Edge Temp Estimate ~1950 to 2000 °C

Images of GRC Model After Arc Jet Test



XRD Results

- ZrO_2 (m)
- ZrO_2 (c)
- Ta_2O_5

Some Answers

- ◆ Is Ta or Si in TaSi_2 responsible for the improved oxidation resistance of $\text{ZrB}_2 - 20\text{v/o SiC} - 20\text{v/o TaSi}_2$? **Tantalum.**
- ◆ Does the benefit of TaSi_2 additions observed at 1627°C extend to higher temperatures? **No, not at 20v/o addition levels.**
- ◆ Are TaSi_2 additions the best way to add Ta to ZrB_2 -based UHTC?
TaC additions are not effective at 1627°C .
- ◆ Can TaSi_2 additions improve the oxidation resistance of HfB_2 -based UHTC which are already superior to ZrB_2 -based materials? **No, not at 20v/o addition levels.**
- ◆ Can anything be learned about the mechanism by which Ta-additions improve the oxidation resistance of $\text{ZrB}_2\text{-SiC}$ materials at 1627°C ? **Tantalum effects both glass phase and ZrO_2 .**

OBJECTIVES

- ♦ Characterize a UHTC composite plate fabricated by Starfire Systems, Inc.
- CAVEAT: Recognize that little or no development effort went into fabrication of this material. It was a best effort fabrication for NASA LaRC
- ♦ Reveal some of the issues associated with the UHTCC concept

Starfire UHTCC Plate

- ◆ **Constituents**
 - ◆ Zoltek Panex® 30 Carbon Fabric PW06
 - ◆ Starfire Systems' SP-Matrix Polymer
 - ◆ (Allylhydridopolycarbosilane (AHPCS))
 - ◆ HfB₂ Powder
 - ◆ SiC Powder

Processing of Part Number 000928-6-64

Initial Cycle:

- For the initial lay up the **bottom 6 layers of cloth are coated with a SiC/AHPCS slurry and the top 5 layers are coated with a HfB₂ /AHPCS slurry** and assembled in a mold.
- Cured to 400°C and fired to 850°C under inert gas to pyrolyze.

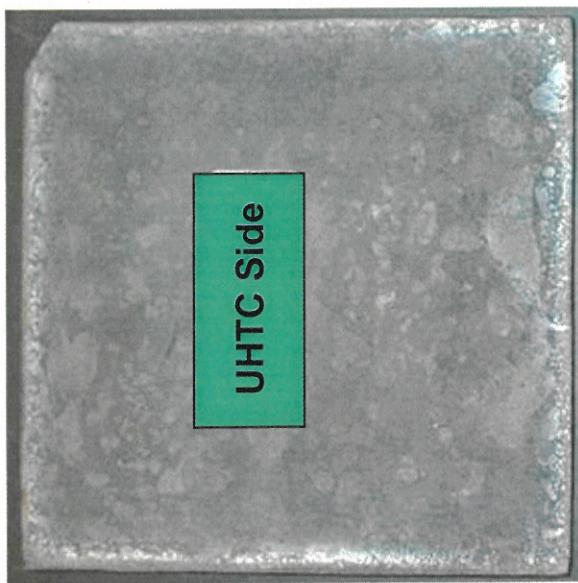
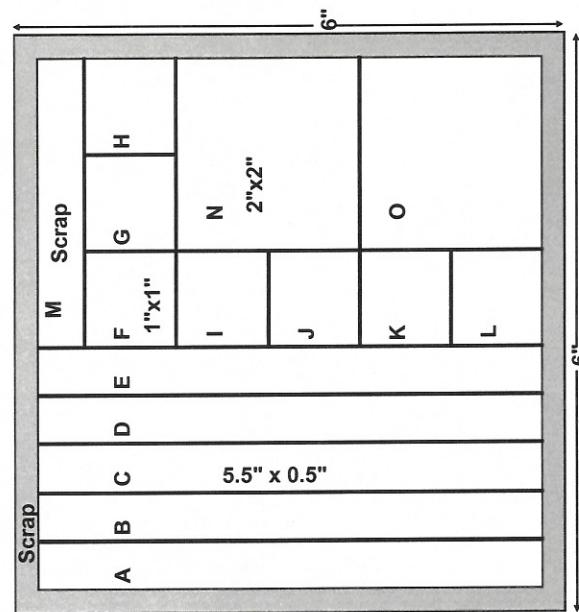
Cycle 2:

- **Coat the HfB₂ side of the plate with more HfB₂ /AHPCS slurry.**
- Fire directly to 850°C under inert gas.

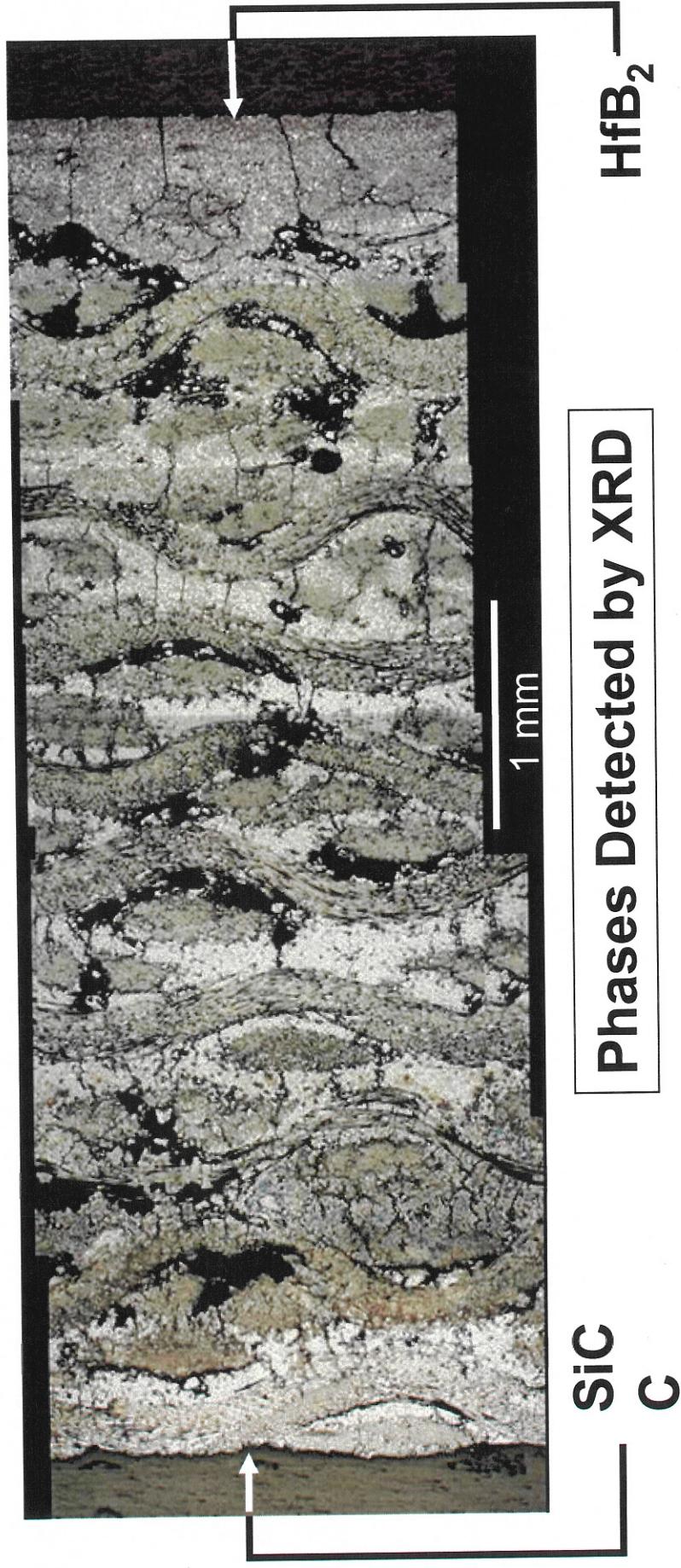
Cycle 3: Repeat cycle 2

Cycle 4 - 10:

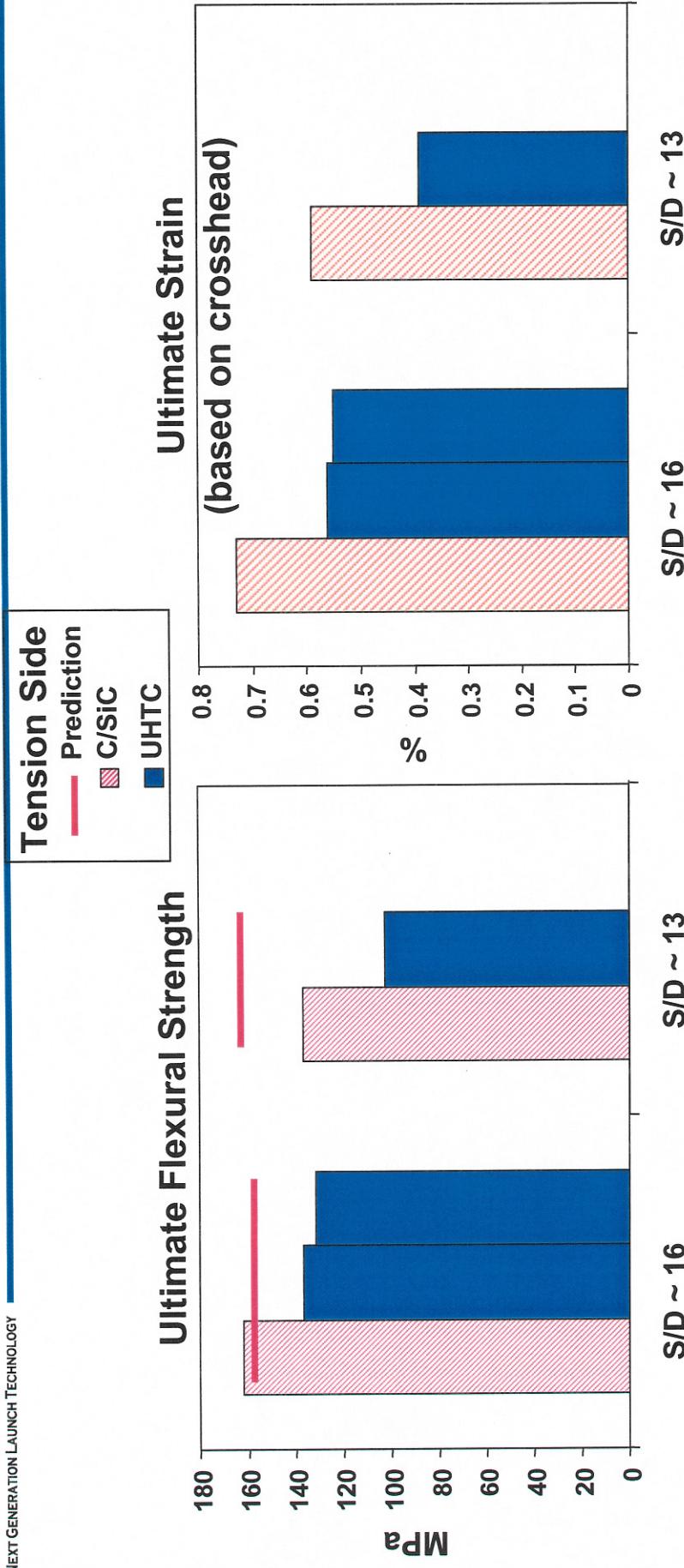
- **Vacuum infiltrate with AHPCS only.**
- Pyrolyze directly to 850°C under inert gas, no clamping necessary.



UHTCC Cross-section



UHTCC Flexural Strength Results

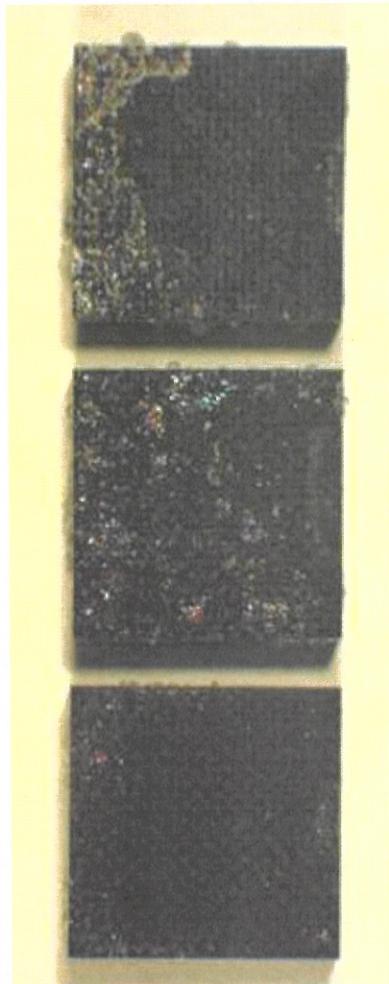


Note: $S/D \geq 16$ necessary for valid 4 point flexural test

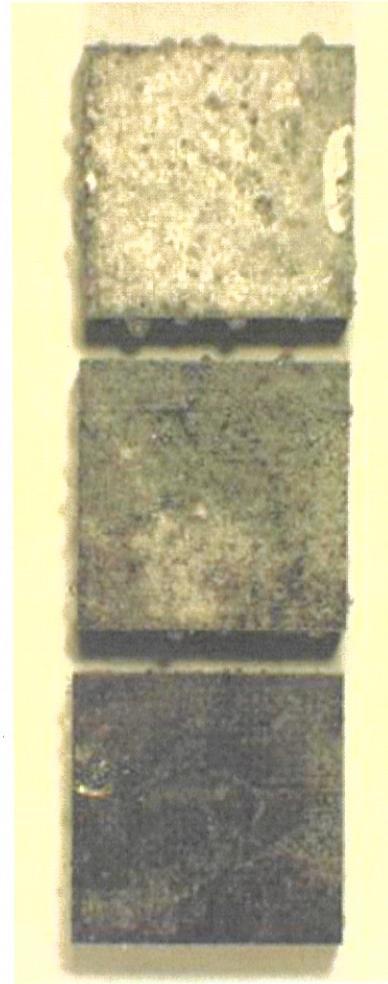
Predictions Based on Beam Theory

- ◆ Calculated load at 0.7% strain
- ◆ Panex 30 minimum property: $E = 193 \text{ GPa}$
- ◆ Rule of mixtures with no matrix contribution

UHTCC After 1627°C Furnace Oxidation



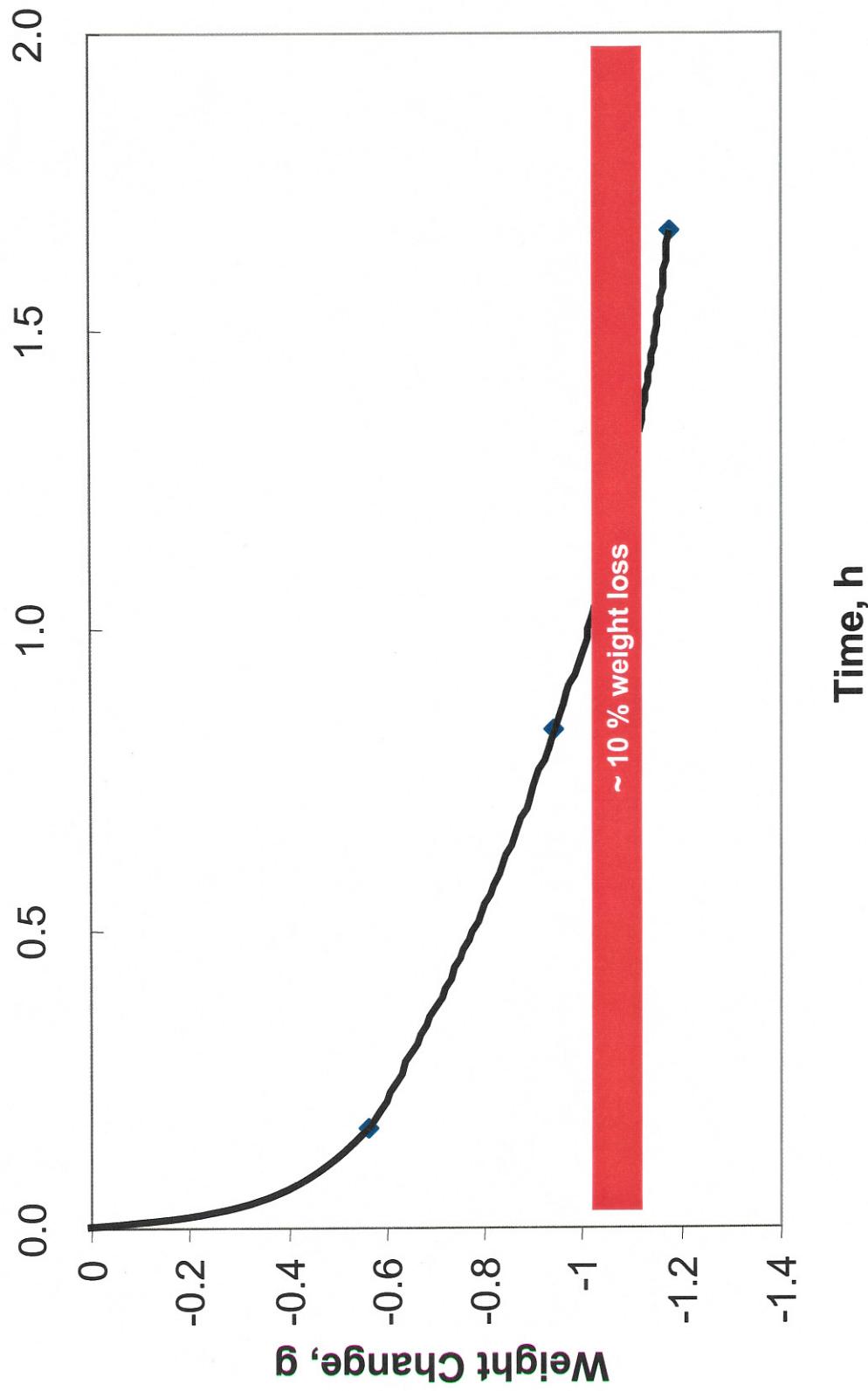
**C/SiC
Side**



**HfB₂
Side**

Cycles	1	5	10
Hours	0.167	0.833	1.667

Furnace Oxidation of UHTCC at 1627°C

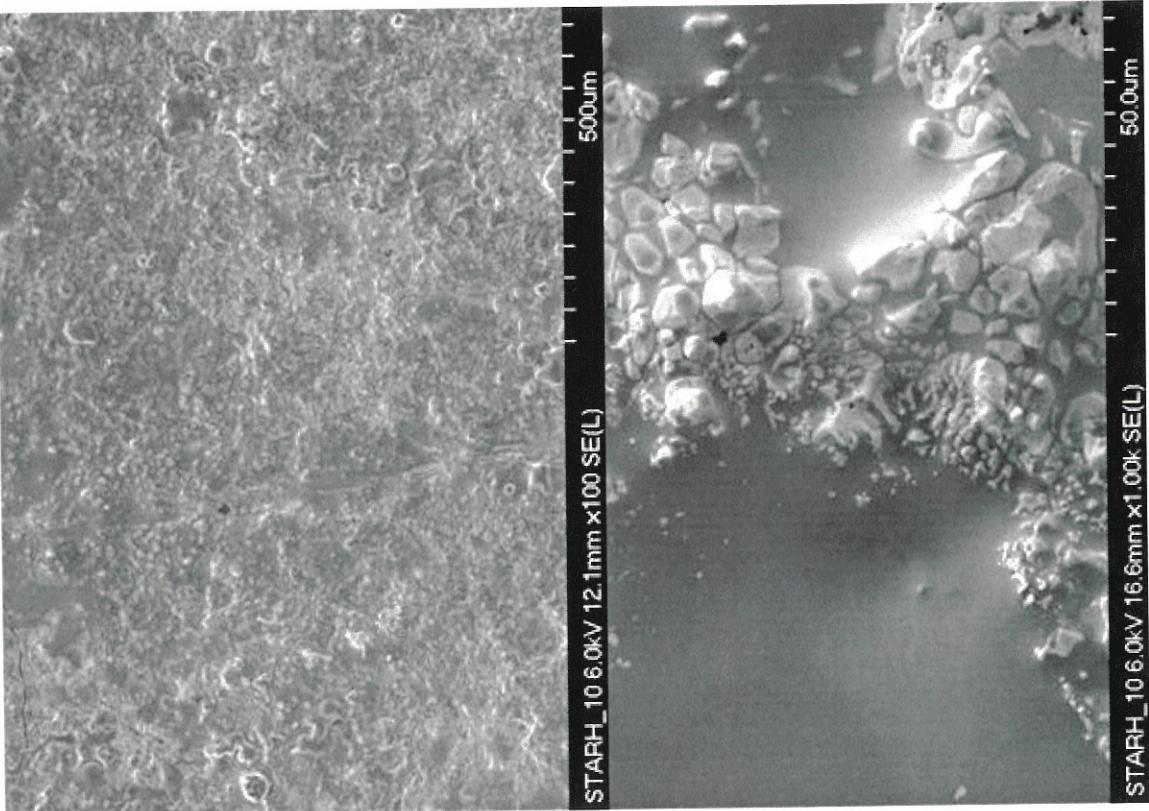


NG/T

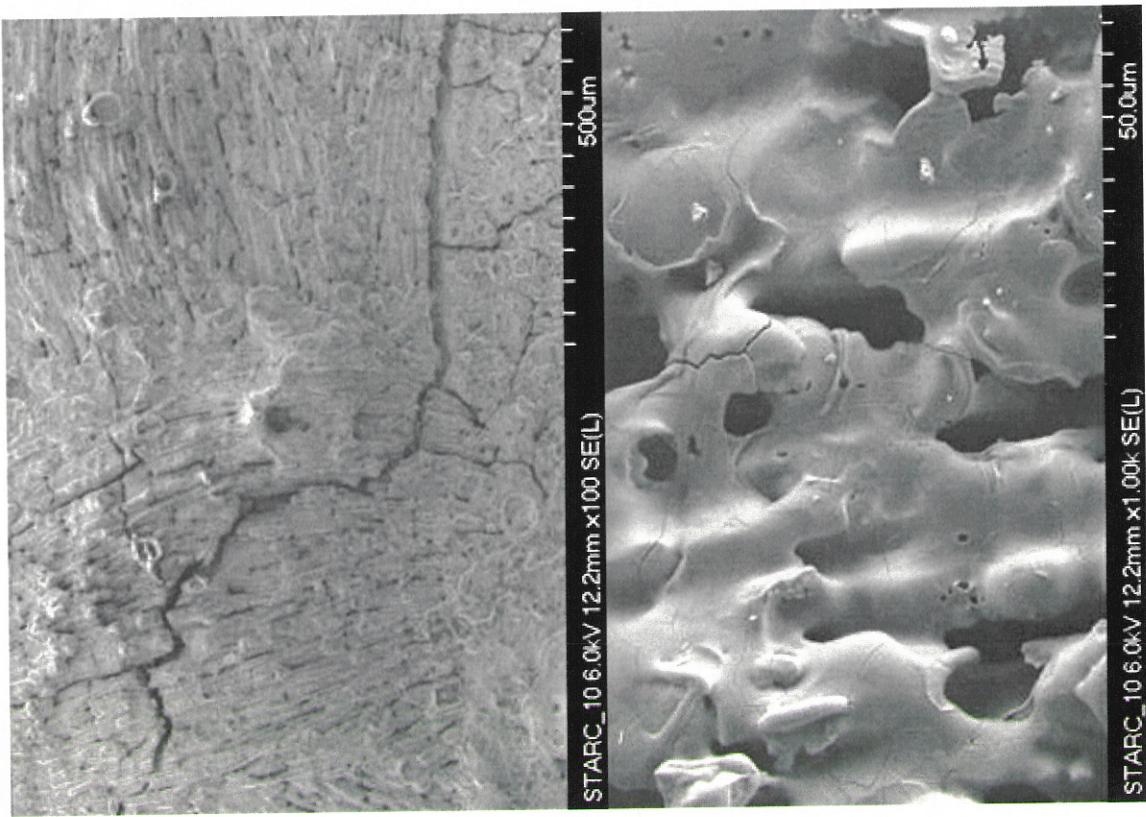
NEXT GENERATION LAUNCH TECHNOLOGY

UHTCC After 10 Ten-minute Cycles in Air at 1627°C

HfB₂ Side



C/SiC Side



STARH_10 6.0kV 12.1mm x100 SE(L)

STARH_10 6.0kV 16.6mm x1.00k SE(L)

STARH_10 6.0kV 16.6mm x1.00k SE(L)

STARH_10 6.0kV 12.2mm x100 SE(L)

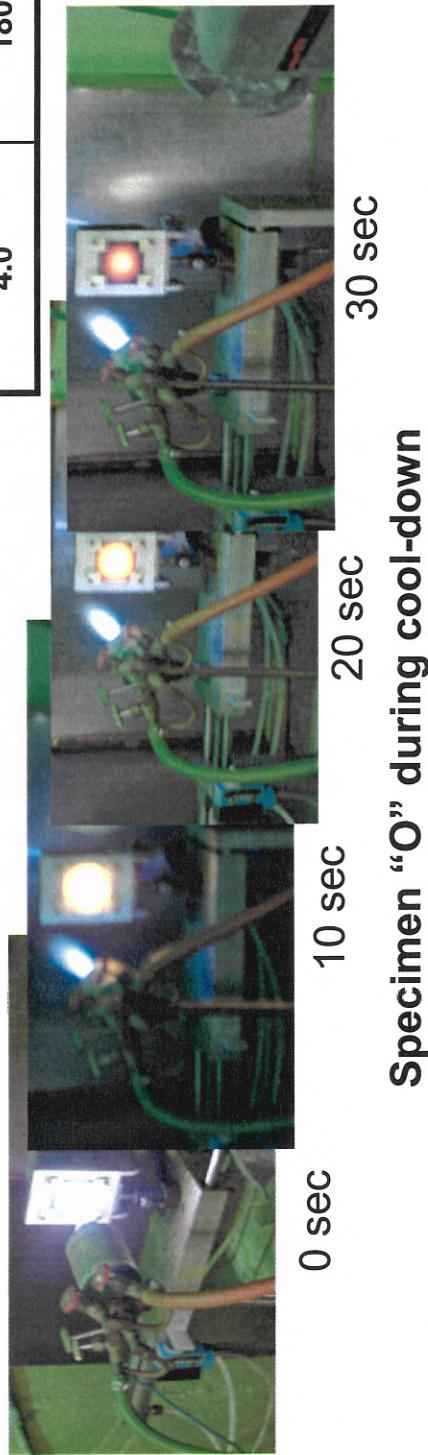
50.0um

50.0um

UHTCC Oxy-Acetylene Torch Test

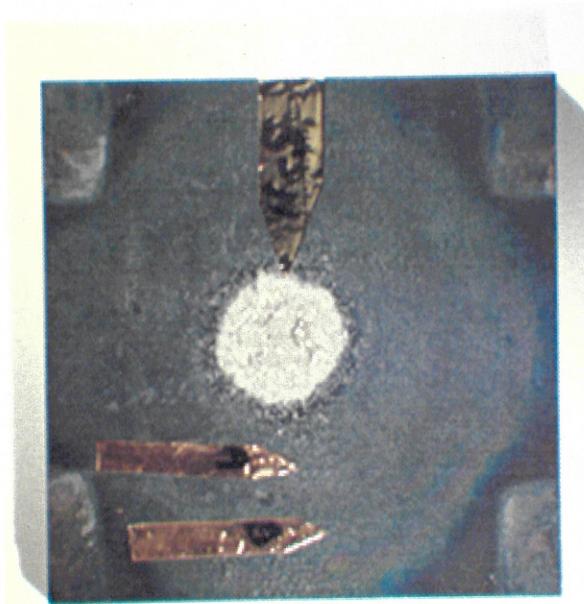
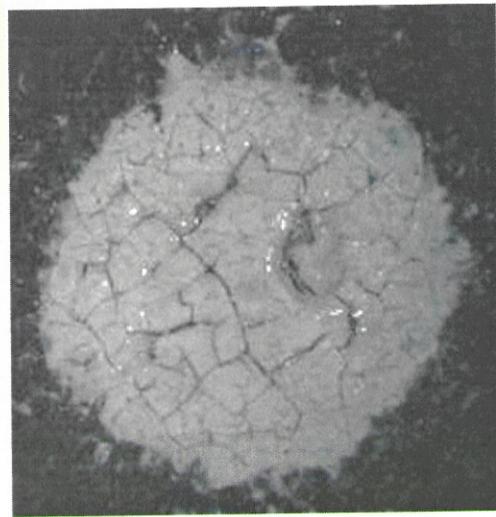
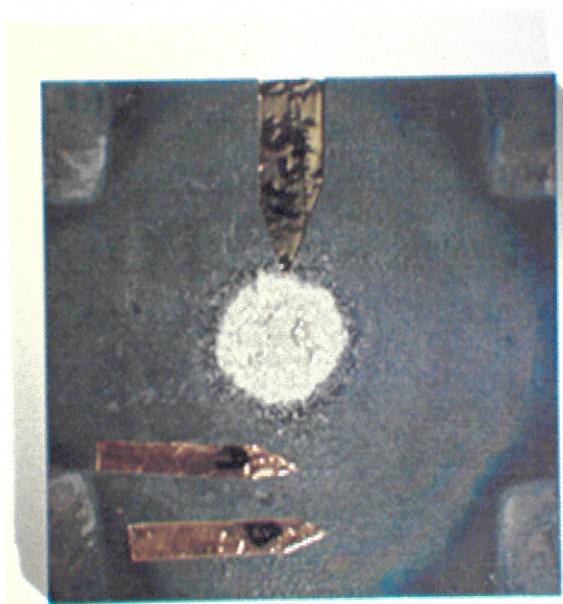
- ◆ Sample O: One 4 min. cycle to 1805°C
 - Temps with Irccon 2 color pyrometer, 980-1760°C range
 - Weight loss = 0.79 g, or 1.9%
- ◆ Sample N: Three ~4 min. cycles
 - Cycle 1 max temp 1815°C
 - Cycle 2 max temp 1915°C
 - Cycle 3 max temp 2015°C
 - Weight loss = 1.95 g, or 4.6%

Sample O	
Time, min.	Temp, °C
0.5	1720
1.0	1750
1.5	1750
2.0	1755
2.5	1765
3.0	1775
3.5	1790
4.0	1805

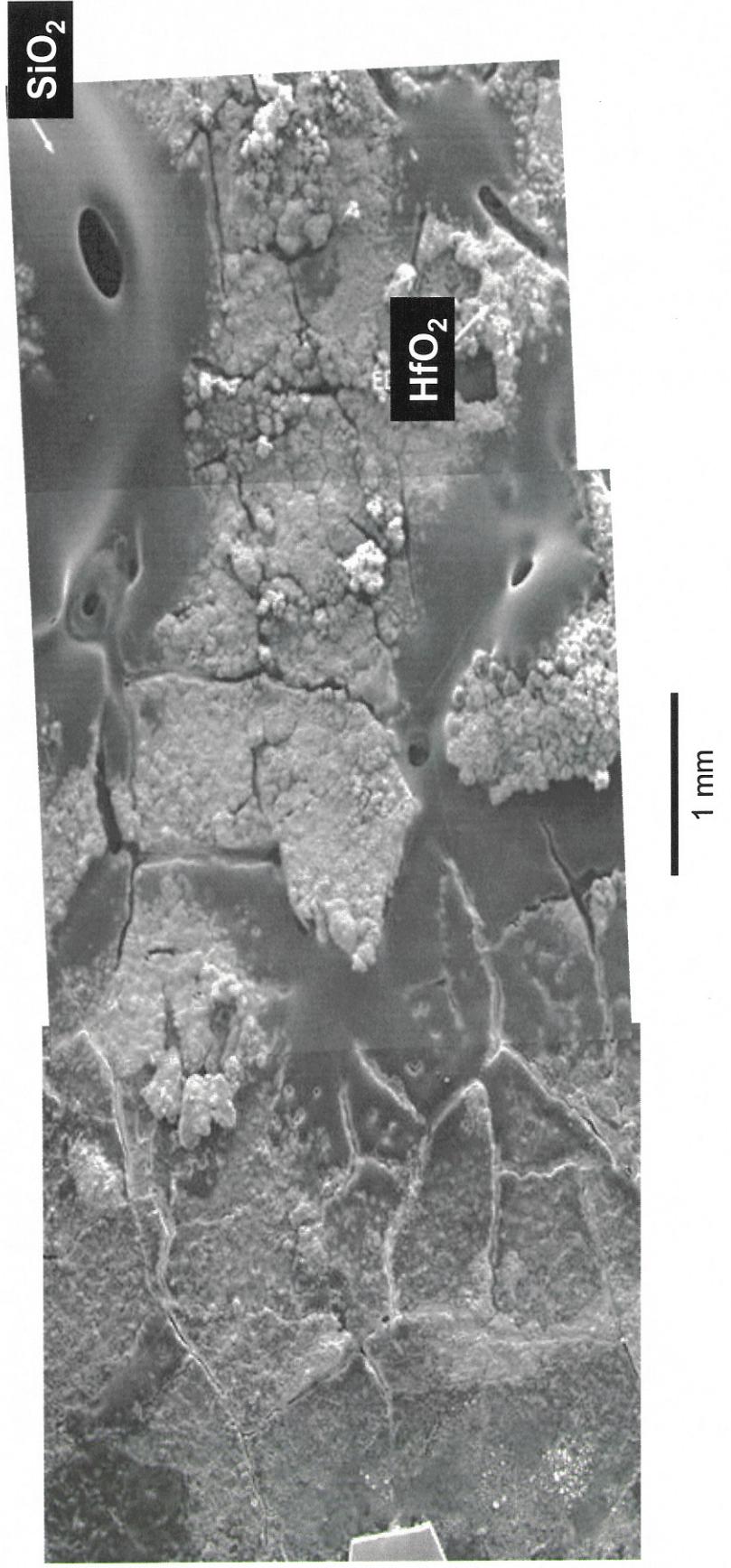


Specimen "O" during cool-down

**UHTC Surface After Three 4-Minute Torch Cycles
to 1815 to 2015°C**

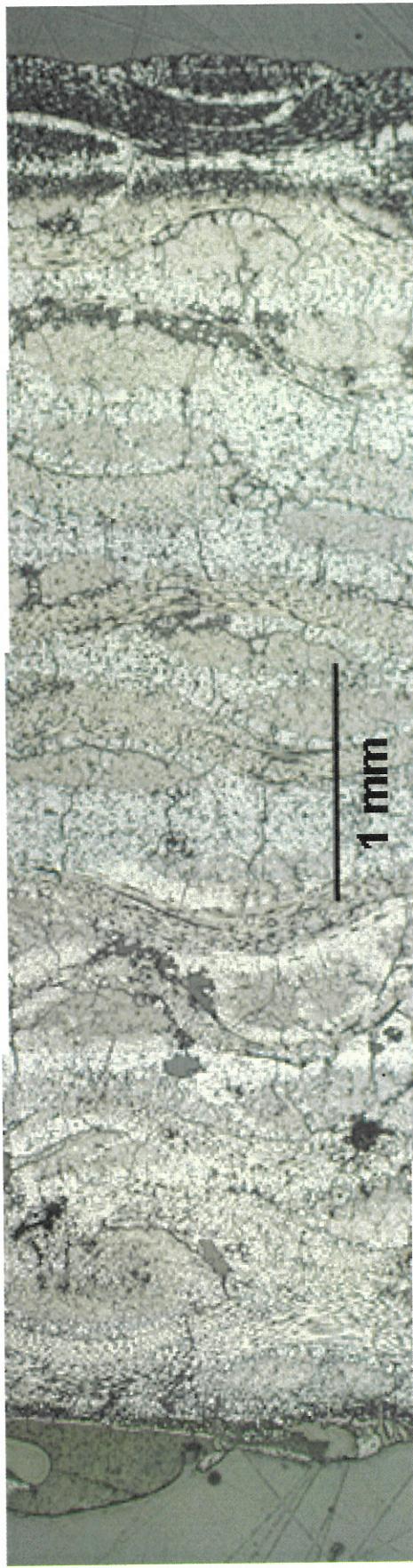


SEM of Center of UHTC Surface After Three Torch Cycles

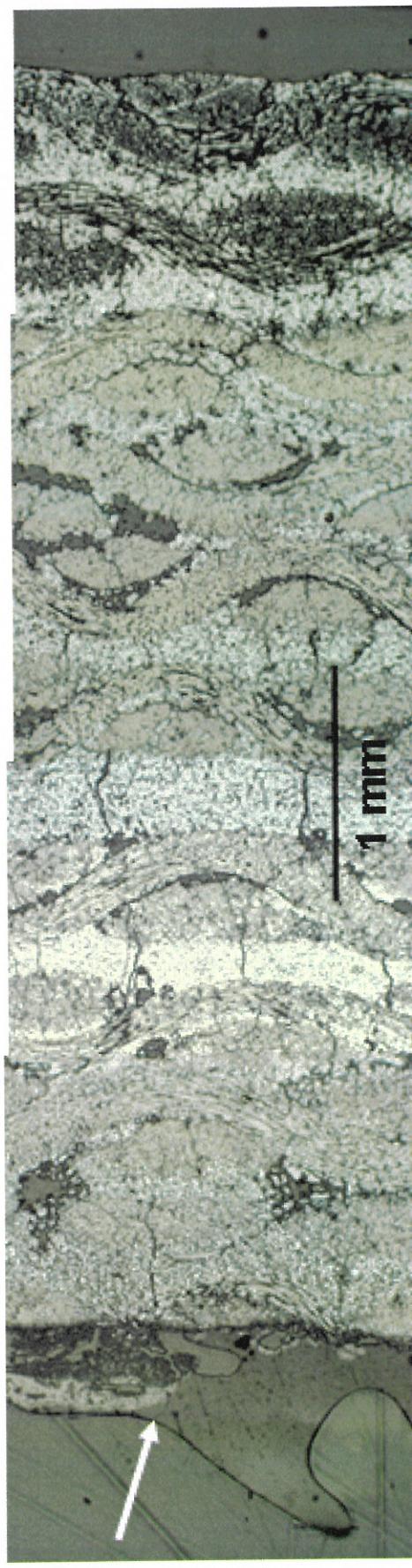


UHTCC Specimen Center Cross Sections After Torch Test

C/SiC
HfB₂

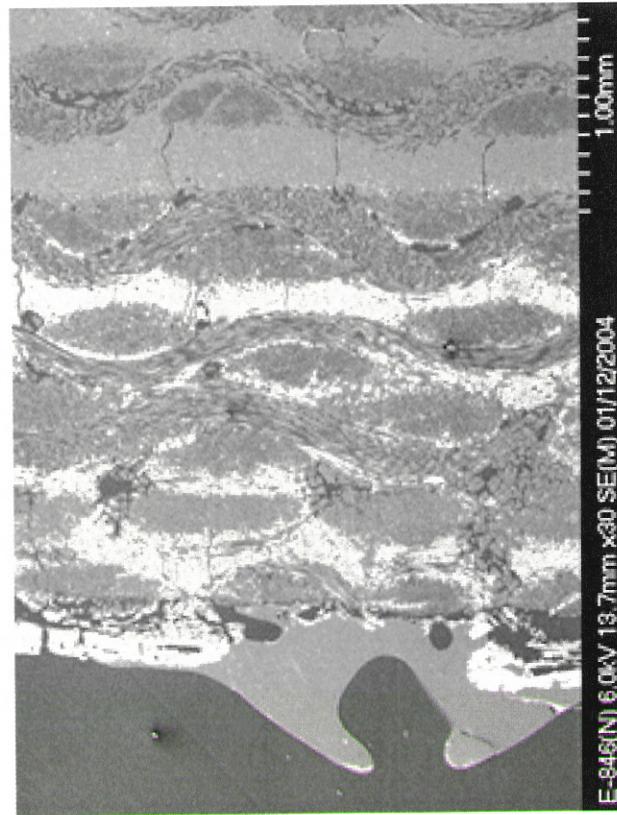
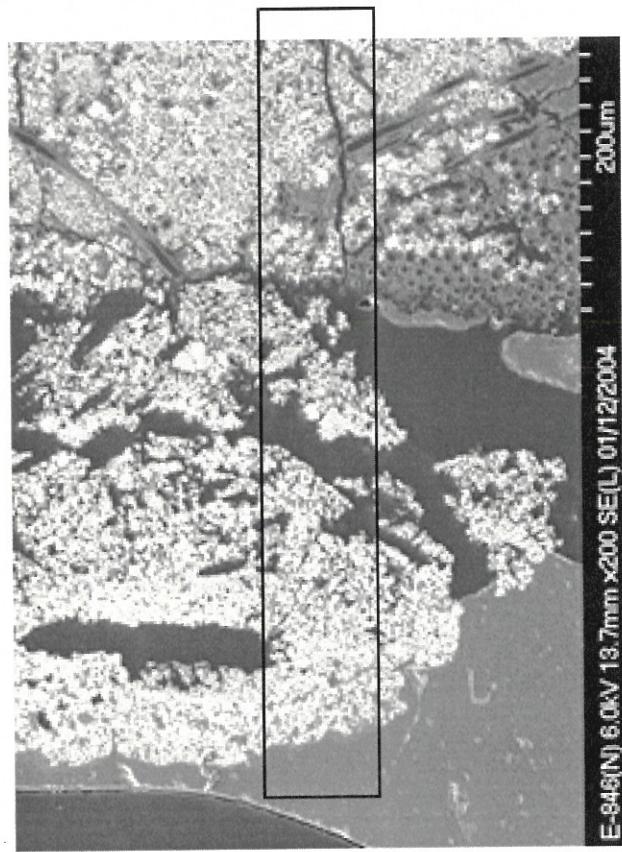


One 4-Minute Cycle to 1805°C

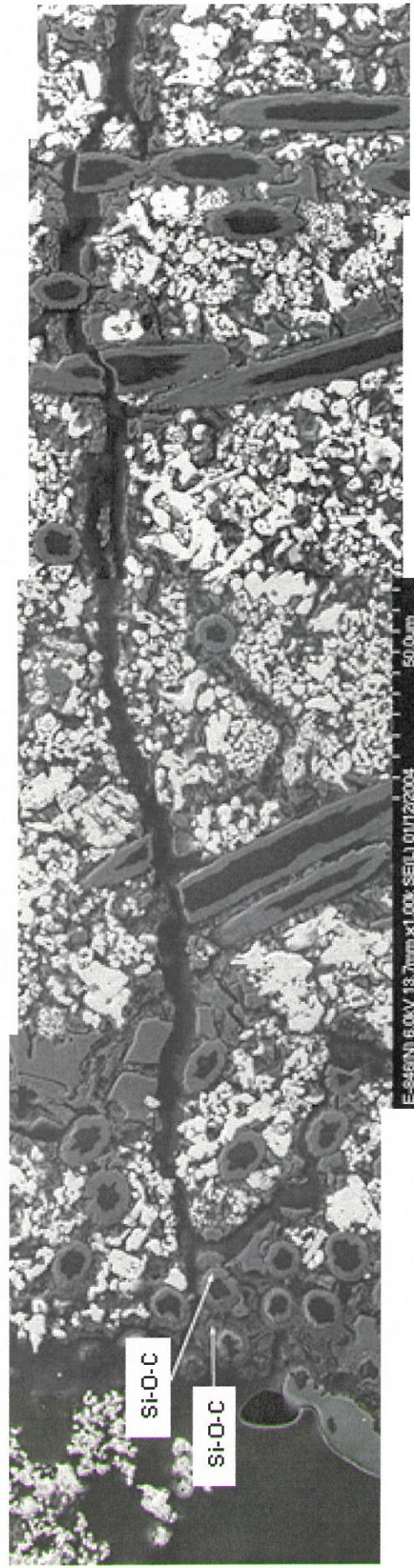
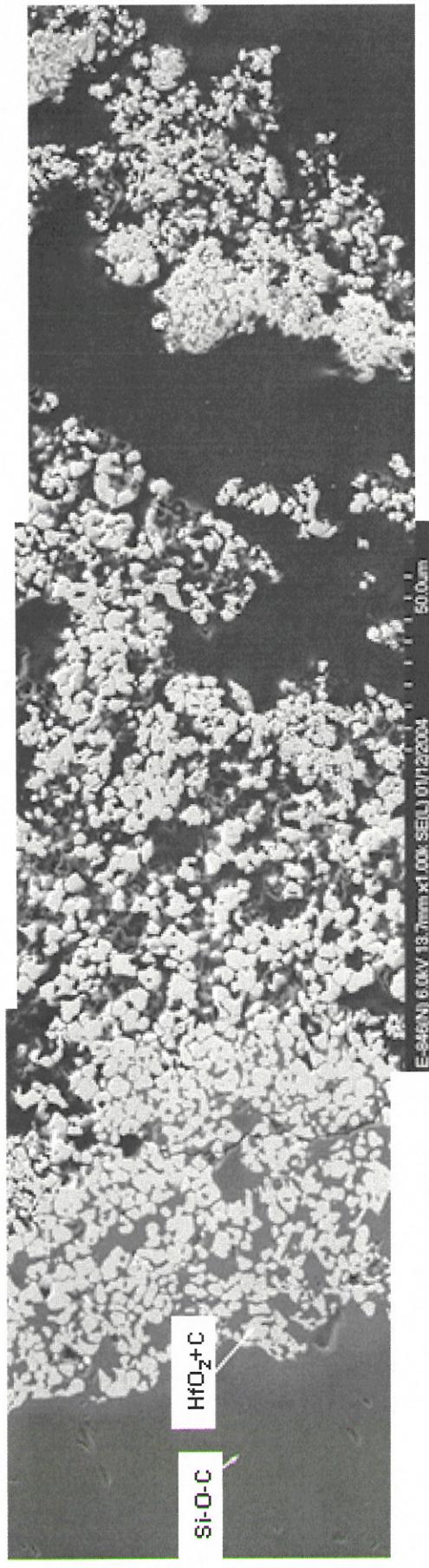


Three 4-Minute Torch Cycles to 1815 to 2015°C

UHTC Side, Center Hot Zone



UHTC Side, Center Hot Zone



Conclusions

- ◆ **TaSi₂ Additions**
 - Good results at 1627°C, but oxide scale too fluid at > ~1800°C
 - Look at lower levels of additive
- ◆ **UHTCC**
 - **Processing**
 - Uniform and through thickness graded microstructure achieved
 - Matrix cracking due to thermal expansion mismatch between C fibers and matrix constituents is a concern
 - **Mechanical Properties**
 - Flexural strength was close to expected values based on rule of mixtures with no matrix contribution
 - Some evidence of composite behavior
 - **Furnace Oxidation**
 - Weight loss and metallography indicated that carbon fiber oxidation occurred rapidly
 - **Torch Test**
 - Based on observed temperature spikes during test, adherence of the HfO₂-rich scale is an area of concern
 - Material withstood ~2000°C (~3600°F), severe heat-up and thermal gradients with no major visible distress, but with significant HfB₂ coating and carbon fiber oxidation revealed by metallography

Future Work

- ◆ Complete oxidation studies focused on lower levels of Ta addition to achieve doping without low melting phase formation
- ◆ Continue UHTCC evaluation
 - Complete metallography on Starfire specimens
 - Evaluate other NASA and industry developed materials
- ◆ Continue UHTCC development
 - Fiber coatings need to be incorporated to address fiber oxidation issues
 - Advanced SiC fibers need to be evaluated to address oxidation and thermal expansion mismatch issues

Acknowledgements

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- ◆ Thanks to Walt Sherwood of Starfire Systems Inc. for providing processing details and for permission to present this study.
- ◆ Thanks to Sarah E. Beckman and Jerome W. Ridge (Eloret Inc.) of NASA Ames Research Center for assistance with arc jet testing.