tacts with the fibers. Because the solder does not wet the nylon, a small void forms between each nylon strand and the solder. The thin solder walls that bound the voids are much more flexible than a solid mass of solder would be; hence, after solidification, the solder can deform to accommodate differential thermal expansion of the various materials present.

This work was done by Robert J. Copeland of TDA Research, Inc., for Johnson Space Center. For further information, contact: TDA Research, Inc. 12345 W. 52nd Ave. Wheat Ridge, CO 80033-1917 http://www.tda.com Refer to MSC-22907.

## Improved BN Coatings on SiC Fibers in SiC Matrices Outside debonding would be favored over inside debonding.

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Modifications of BN-based coatings that are used as interfacial layers between the fibers and matrices of SiCfiber/SiC-matrix composite materials have been investigated to improve the thermomechanical properties of these materials. Such interfacial coating layers, which are also known as interphases (not to be confused with "interphase" in the biological sense), contribute to strength and fracture toughness of a fiber/matrix composite material by providing for limited amounts of fiber/matrix debonding and sliding to absorb some of the energy that would otherwise contribute to the propagation of cracks.

Heretofore, the debonding and sliding have been of a type called "inside debonding" because they have taken place predominantly on the inside surfaces of the BN layers — that is, at the interfaces between the SiC fibers and the interphases. The modifications cause the debonding and sliding to include more of a type, called "outside debonding," that takes place at the outside surfaces of the BN layers — that is, at the interfaces between the interphases and the matrix (see figure).

One of the expected advantages of outside debonding is that unlike in inside debonding, the interphases would remain on the crack-bridging fibers. The interphases thus remaining should afford additional protection against oxidation at high temperature and should delay undesired fiber/fiber fusion and embrittlement of the composite material. A secondary benefit of outside debonding is that the interphase/matrix interfaces could be made more compliant than are the fiber/interphase inter-



Specimens of SiC/SiC Composites that have undergone outside and inside debonding are depicted in scanning electron micrographs at various magnifications.

faces, which necessarily incorporate the roughness of the SiC fibers. By properly engineering BN interphase layers to favor outside debonding, it should be possible, not only to delay embrittlement at intermediate temperatures, but also to reduce the effective interfacial shear strength and increase the failure strain and toughness of the composite material.

Two techniques have been proposed and partially experimentally verified as candidate means to promote outside debonding in state-of-the-art SiC/SiC composites. The first technique is one of application of a weak layer (for example, a layer of C) to the outer surface of the BN interphase. If residual radial tension exists across the interphase (caused, for example, by a thermal-expansion mismatch between the fiber and matrix), then outside debonding could occur during cool down of the composite from its matrix-processing temperature (typically >1,000 °C) to room temperature. If the residual tension is not great enough, outside debonding should nevertheless occur as matrix cracks approach the fibers under some stress conditions.

The second technique for promoting outside debonding is one of heat treatment of the composite at a temperature above that normally used for processing the matrix. During the heat treatment, the BN interphase, which is typically formed in a porous state at temperatures below 1,000 °C, becomes densified by sintering, so that the interphase contracts away from the SiC matrix. This contraction may cause a gap to form between the BN interphase and SiC matrix, or, at the very least, increase the residual radial tension at the BN/matrix interface.

In stress-rupture tests in air at a temperature of 800 °C, the 100-hour failure stress of state-of-the-art SiC/SiC composite modified to promote outside debonding was as much as 50 percent greater than that of the corresponding unmodified (inside debonding only) composite. The room-temperature strain to failure of the outside-debonding composite was found to be 0.5 percent, as compared with 0.4 percent for the inside-debonding composite, and the ultimate tensile strength of the outside-debonding composite was not less than that of the insidedebonding composite.

This work was done by Gregory N. Morscher, Ramakrishna Bhatt, Hee-Mann Yun, and James A. DiCarlo of Glenn Research Center. Further information is contained in a TSP (see page 1). Inquiries concerning rights for the commercial use of this invention should be addressed to NASA Glenn Research Center, Commercial Technology Office, Attn: Steve Fedor, Mail Stop 4-8, 21000 Brookpark Road, Cleveland, Ohio 44135. Refer to LEW-17240-1.