

High-Fidelity Modeling and Materials Characterization of Inconel 718 Component Fabrication by Selective Laser Melting Additive Manufacturing



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Company Overview





Selective Laser Melting Additive Manufacturing

- Promise:
 - Can fabricate complex parts
 - High 'buy-to-fly' for raw materials
 - Rapid prototyping
 - Rapid build of replacement parts and/or repair
 - Design freedom relative to casting and subtractive processes

Challenges:

- Shrinkage and geometrical tolerances ⇔ post-processing
- Material quality and effective properties ⇔ qualification
- Multiple (20 or more) process variables to 'optimize'

Porosity and 'balling'



Weak bonds at

scan interfaces







Modeling Motivation and Goals **GDRG**.

- Problem and Proposed Solution:
 - Problem: Lack of understanding on the process to property relationships is leading to low yield, unacceptable part-topart variability, costly post-processing and long qualification times.
 - Solution: Develop state-of-art modeling tools to increase build success – thereby reducing variability, postprocessing, and qualification time.
- Modeling tools can give guidance on both component design and build design parameters
 - Will residual stress and materials parameters allow for successful build?
 - What defects are likely to affect material quality?
 - What post-processing is needed?







- Approach based on measurement and validation, application-driven code development for useful software and workflow
- Emphasize build analysis: can you make the as-designed part with acceptable residual stress, if so are materials adequate

EBSD Microstructure of As-Fabricated Inconel-718



• X plane: Columnar structure, Weld-bead feature, with random texture







Middle Top



Тор

• Z plane: Regular patch pattern, random texture



Bottom





Microstructure Model Formulation

- Governing equations for phase field and solute concentration, Inconel 718 properties:
 - $\frac{\partial \emptyset}{\partial t} = -M_{\emptyset} \left[\frac{\partial f}{\partial \emptyset} \epsilon_{\emptyset}^2 \nabla^2 \emptyset \right]$

 $\frac{\partial C}{\partial t} = \nabla \cdot \left[M_{c} c (1-c) \nabla \left(\frac{\partial f}{\partial c} - \epsilon_{c}^{2} \nabla^{2} C \right) \right]$

Liquidus slope, m _l (K wt. % -1)	-10.0
Liquidus temperature, T _I (K)	1528
Solidus temperature, T _s (K)	1610
Latent heat, L (J/Kg)	227,000
Specific heat, Cp (J/Kg/K)	600
D ₁ (m ² s ⁻¹)	3.0E-9
D _s (m ² s ⁻¹)	1.0E-12
Anisotropic factor, ε	0.03
Mesh size, dx (µm)	0.2
Interface thickness, λ (μm)	1.77

• Anisotropic, dimensionless form in model:

$$\begin{split} & \left[1 + (1 - k)U\right]a_{s}(\hat{n})^{2}\frac{\partial\emptyset}{\partial t} \\ & = \vec{\nabla}.\left(a_{s}(\hat{n})^{2}\vec{\nabla}\emptyset\right) - \frac{\partial}{\partial x}\left[a_{s}(\hat{n})a_{s}(\hat{n})'\frac{\partial\emptyset}{\partial y}\right] + \frac{\partial}{\partial y}\left[a_{s}(\hat{n})a_{s}(\hat{n})'\frac{\partial\emptyset}{\partial x}\right] + \emptyset - \emptyset^{3} \\ & -\lambda(1 - \emptyset^{2})^{2}(U + \theta) \end{split}$$

$$\left(\frac{1+k}{2}\right)\frac{\partial U}{\partial t} = \vec{\nabla} \cdot \left(D\frac{1-\emptyset}{2}\vec{\nabla}U + \vec{J}_{at}\right) + \frac{1}{2}\frac{\partial}{\partial t}\{\emptyset[1+(1-k)U]\}$$





0.5

Microstructure Characterization & Modeling



- Micrographs of as-fabricated Inconel 718 consistent with EBSD:
 - Axial 50-100 mm grains in growth direction; equiaxed from above; $\gamma \gamma''$ and Metal Carbide with Laves phase







• Phase field models of Inconel 718 solidification:



Cooling Rate Effects on Grain Size



Nuclei Effects on Grain Morphology

Microstructure Link to Melt Pool **GPRG**.

2D Cartesian SLM thermal model for dimensionless temperature gradient



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- $X T_P$ B r A Liquid T_L
- Phase field models of Inconel 718 solidification:
 - Cooling rate range determined from thermal model



Mesoscale Model Formulation

• Conservation of Total Mass:

$$\frac{\partial}{\partial t}(\rho) + \nabla \cdot \left(\rho \vec{V}\right) = S$$

- Conservation of Secondary Fluid Mass:
 - $\frac{\partial F}{\partial t} + \vec{\nabla} \cdot \left(\vec{V}F\right) = \frac{\dot{m}}{\rho_l}$
- Conservation of Mass-Averaged Momentum $\frac{\partial}{\partial t} \left(\rho \vec{V} \right) + \nabla \cdot \left(\rho \vec{V} \vec{V} - \mu \nabla \vec{V} \right) = \left(-\nabla P + \rho \vec{g} \right) + \vec{B}$
- Conservation of Mass-Averaged Enthalpy $\frac{\partial}{\partial t}(\rho h) + \vec{\nabla} \cdot (\rho \vec{V} h) = \vec{\nabla} \cdot \vec{\mathbf{q}} + \tau : \vec{\nabla} \vec{V} + \frac{\mathrm{d}p}{\mathrm{d}t} + \dot{Q}$



Surface tension as a body force



Mesoscale Laser-Particle Model



- Simulate melting and reflow of metal powder particles to identify conditions causing voids or incomplete melting
 - VOF interface tracking routine coupled with laser source, heat transfer, phase change thermodynamics, fluid dynamics, surface tension
 - Implemented $\mu(T)$ for particles, laser source, phase change thermo



Melting of discrete particles and void formation as laser scans



Static solid Inconel 718 via high-viscosity fluid verified for 3D simulation

Thermal Model Formulation

- Conservation of Energy (Heat equation):
 - The phase change is accounted for by increasing Cp_{eff} between the liquidus and solidus temperatures so that total enthalpy is conserved:

$$h = \int_{0}^{T} C_{p,eff} dT = \int_{0}^{T} C_{p} dT + \Delta H_{f} f$$

$$\frac{\partial}{\partial t} \left[\rho C_p T \right] = \nabla \cdot k \nabla T + \dot{Q}$$

 $\frac{\partial}{\partial t} \left[\rho C_p T \right] = -\nabla \cdot q + \dot{Q}$

 $\vec{q} = -k\nabla T$

with *f* being the solid fraction

- Moving beam (laser, electron) heating source term
 - Local heating dependent on beam width and penetration depth

$$\dot{Q} = P_{abs}q(r(t))f(z)$$

CFDRG

Thermomechanical Model: Demonstrations



- Base Capabilities
 - Mesh Adaptation for speedup - 3x faster than fixed grid
 - Extrusion to address new powder layer
- Temperature Fixed Mesh Solution Adapted 3316 3000 3000-2500-₽ 2000 2000 Surfai 1500-20 b 1000-1000 500-297.4 -0.4 -0.2 0.2 0 Distance from Beam Center, mm heat_temp 9048 900-800-700-600-500-400-Layer 1 Layer 2 300-273 s0 stressYY 1.159E+07 Temperature Temperature 600 600--5E+07-550-550--1E+08 500-500-Scan 1 Scan 3 -1.5E+08 450-450--2E+08-400-400-Transverse-2-5E+08-350-350--3E+08-300 300 - 3.5<mark>E+08</mark>-Stresses 298 298 - 3.983E+08 2 cm long x 2mm deep
- Demonstrations:
 - Hatch scan for 2 cm line build
 - Thermomechanical Coupling

EBAM Temperature and Melt Pool **CFDRC**.

- Good agreement of peak temperature
 - Sensitive to effective conductivity in melt
 - Slightly longer plateau in liquidsolid 2-phase region





Benchmark, agrees with experiment

Temperature, K



2.114mm x 0.555mm x 0.140mm

Thermomechanical Application



- Demonstration applications by UA to analyze hatch scan patter effects on T uniformity, stress
 - 6x6 mm scanned region, 8x8 mm base
 - Beam diameter 400 μm , power/area into beam consistent with Concept 3 typical operating conditions





Thermomechanical Application





- Line scan patterns gave lowest max T during scanning due to small part size, long beam line
 - Resulted in lowest distortion for those cases

Thermal Model Accuracy/Efficiency



 Resolving the laser spot energy input in position and time imposes a CFL constraint – very small simulation time steps



T profile after hatch scan, 'brute force'



T profile after hatch scan, large Δt

 Alternative: follow beam path and integrate heat input over longer time step



T profile after hatch scan, adaptive quadrature with large Δt

Thermal Model Accuracy/Efficiency

- Current approaches for more practical application:
 - Others: Overset grids, move a high-resolution 'block' with the spot

CFDRC: Adaptive meshing,





- **CFDRC Adaptive integration:**
 - User-specified tolerance for total heat input accuracy
 - Captures T history during cooling and solidification



Order 5x speedup in clock time

Method	Time step,	CPU / Sim	Relative	Accuracy
	Δt (s)	time (s/s)	Speed-up	
Point Input	5E-5	9.3E5		$\checkmark\checkmark$
	1E-3	1.1E5	8.90	×
Adaptive	5E-5	1.7E6	0.56	$\checkmark\checkmark$
Integration	1E-3	1.8E5	5.25	 ✓

10x speedup possible when combined with adaptive meshing



Overhang First Layer Effects





Phase (blue=powder, purple=solidified)



Temperature

- Manufacturing an overhang adds significant challenges
- Powder does not conduct heat well
 - Increases temperature of melt pool
 - A support may be required to serve as a heat sink

Overhang First Layer Effects

- High stresses in the overhang
 - Limited support from powder
 - High thermal gradients
 - Thermal expansion
- Result in large deflections









- Parts fabricated by ASRC and MSFC
 - ≈1% tolerances after stress relief heat treatment









Summary



- The CFDRC-UA team is developing modeling tools to address SLM process performance at each critical level:
 - Microstructure: material properties
 - Mesoscale: material quality
 - Component: manufacturability and dimensional tolerances



Test structure for material characterization vs. build height



- Companion experimental efforts in process monitoring and material characterization provide learning for material qualification and data for model validation
- Next step is to put the pieces together and apply to test builds
- Outcome will be advanced tools to inform design and process

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