

# **Oxidation of Alumina-Forming MAX Phases in Turbine Environments**

James Smialek, Anita Garg, Bryan Harder,  
James Nesbitt, Timothy Gabb, Simon Gray\*

NASA Glenn Research Center, Cleveland, OH

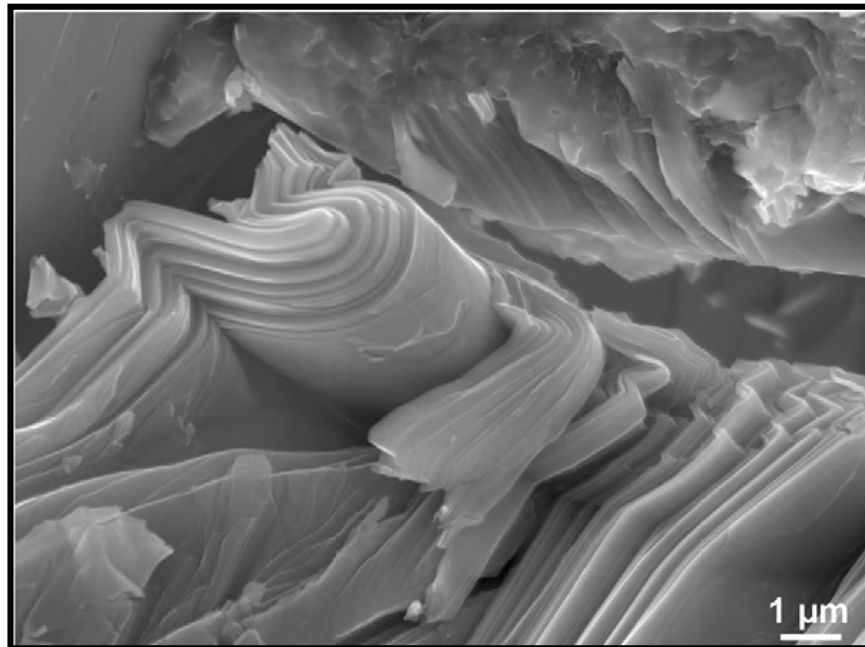
\*Cranfield University, Cranfield, UK

Materials for High Temperature Applications: Next Generation Superalloys and Beyond  
TMS Annual Mtg., San Diego, CA, Feb 26 - Mar 2

# Strain Tolerance: Kinking and Crack Deflection in $\text{Cr}_2\text{Al}(\text{Si})\text{C}$

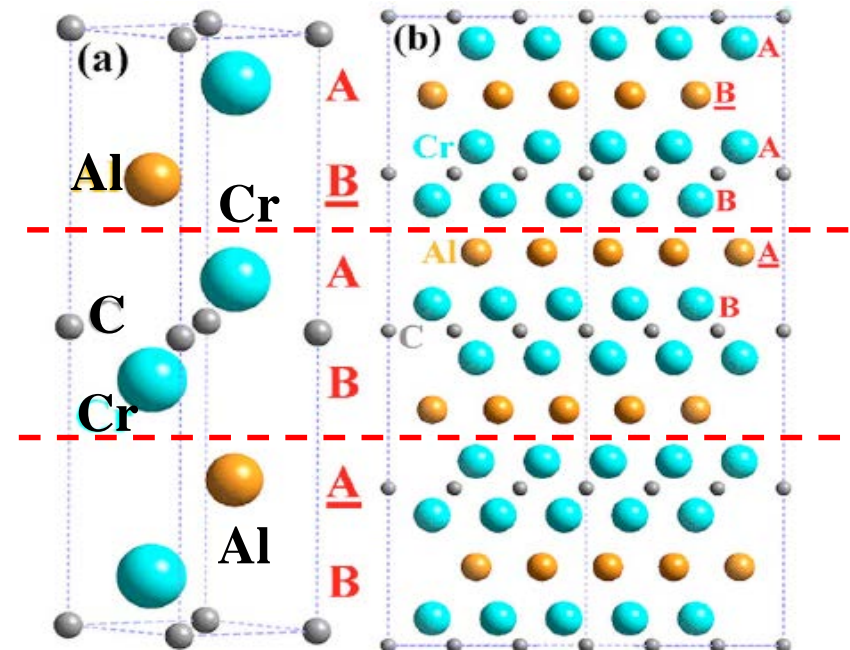
Bend test fracture; basal plane nano-laminate

W. Yu, S. Li, W.G. Sloof, 2007



$$K_{IC} = 6.6 \text{ MPa}/\text{m}^{1/2}$$

Lin et al. 2006



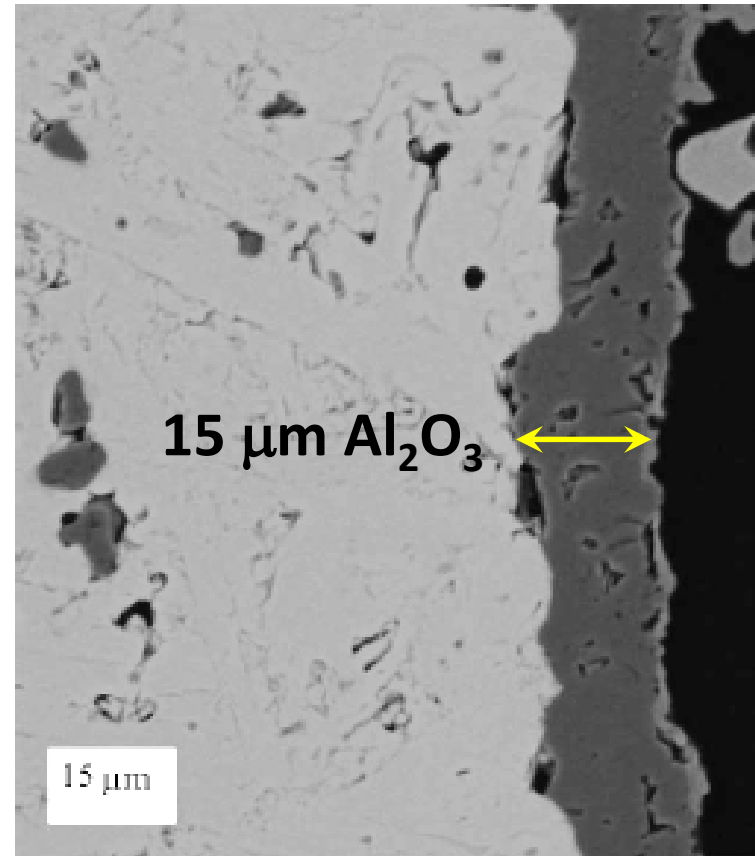
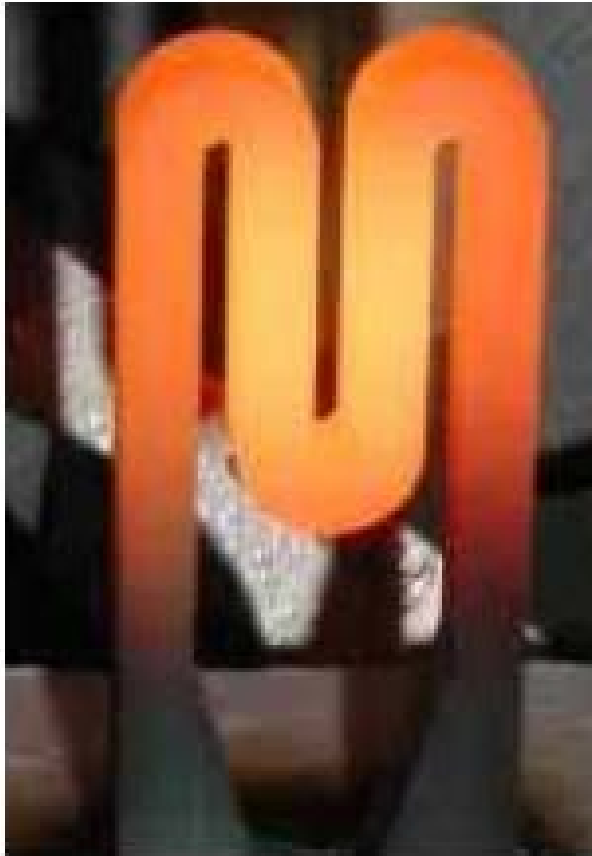
**(0001) Sliding**

$$(\text{Cr-Al})/(\text{Cr-C}) = 0.40 \sigma_{\text{bond}}, 0.43 \gamma_{\text{cleav.}}$$

# Commercial $\text{Ti}_2\text{AlC}$ 211 MAXthal (Sandvik/Kanthal)

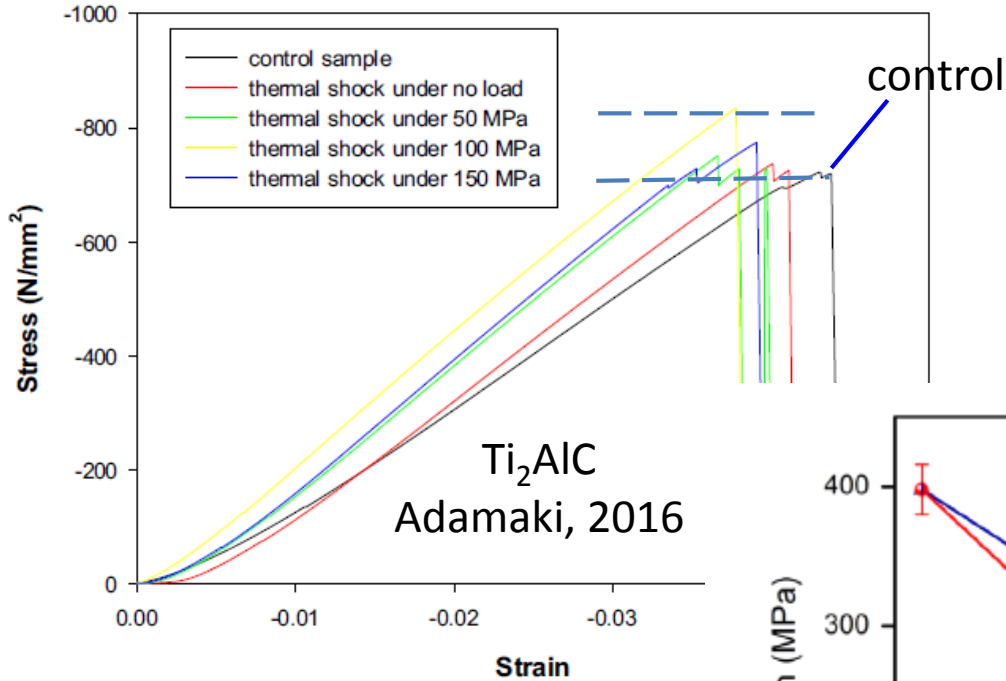
M. Sundberg, G. Malmqvist, A. Magnusson, T. El-Raghy, 2004

8000 cycles to 1350°C!

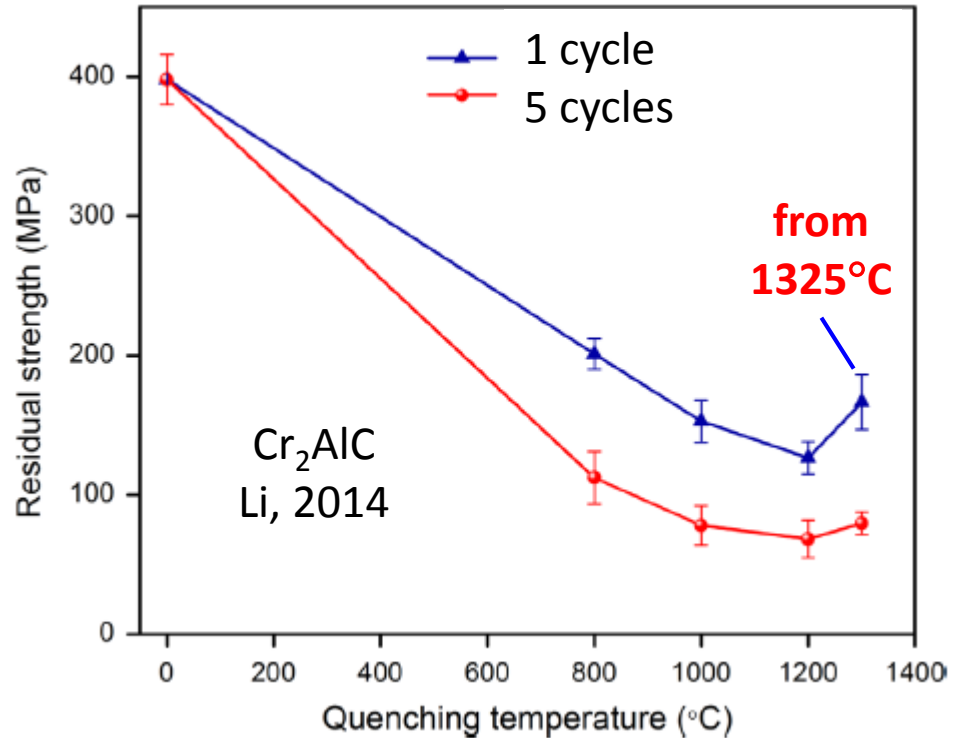


# Thermal Shock of $M_2AlC$

(thermal conductivity x strength) / (CTE x modulus)

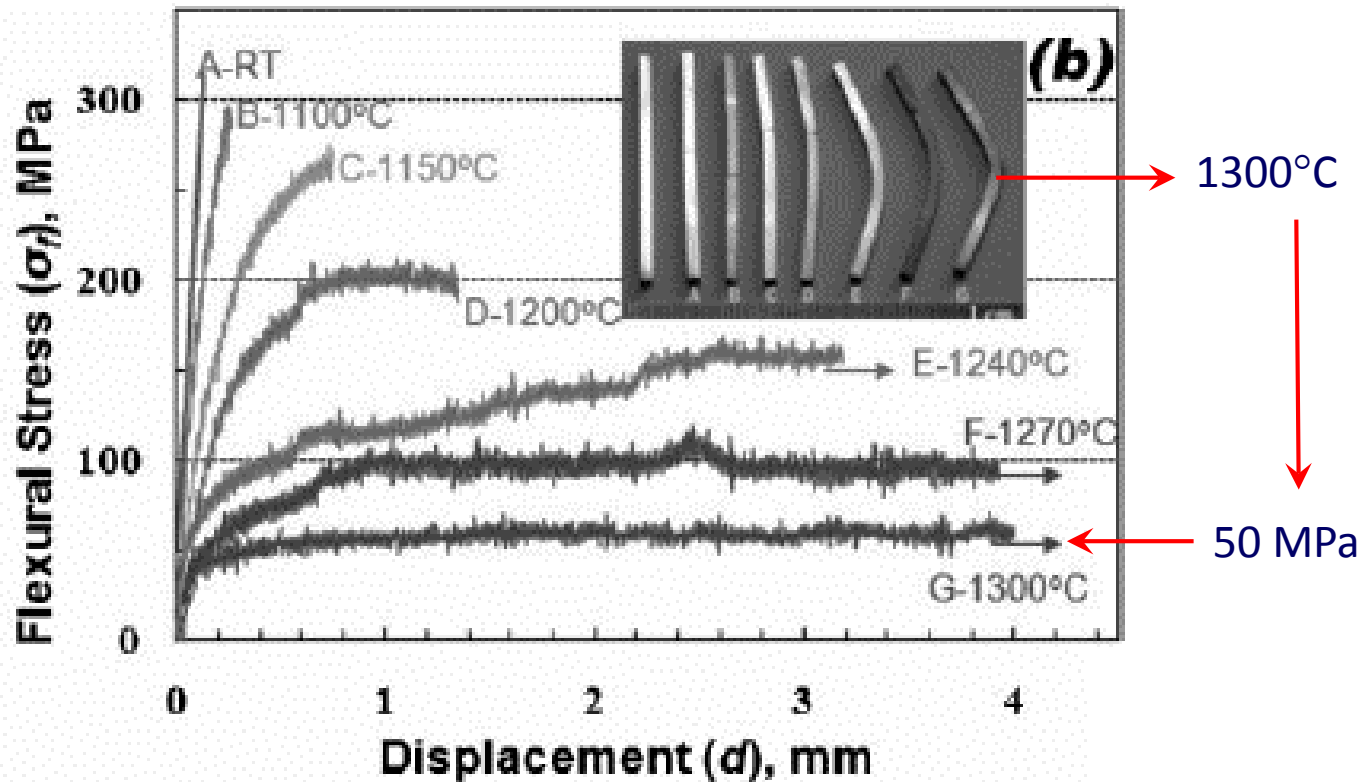


Bend Strength  
after **Water Quench**



700 MPa Compressive strength  
retained (Gleeble test) after  
**-90 $^{\circ}C/s$  quench from 1200 $^{\circ}C$**

# Some ~50 MPa $\text{Ti}_3\text{SiC}_2$ High Temperature Bend Strength (Sun, 2006)



# Motivation and Rationale

## $Ti_3AlC_2$ , $Ti_2AlC$ , $Cr_2AlC$

- $\alpha-Al_2O_3$  formers, (slow, low-volatility)
- $Na_2SO_4$  corrosion resistance
- CTE close to YSZ,  $\alpha-Al_2O_3$
- Damage (Strain ?) tolerance,  
nano-laminate shear, machinable
- Thermal shock resistance:  $\sim 1400^\circ C$  quench
- $K_{IC} \approx 7 \text{ MPa}\cdot\text{m}^{1/2}$

## **Purpose: Relate Al-MAX phases to Coatings and turbine environments**

### **Kanthal $Ti_2AlC$ , $Cr_2AlC$**

1. High temperature  $\alpha-Al_2O_3$  kinetics
2. High pressure burner rig
3. YSZ Thermal Barrier Coatings
4. Superalloy/MAX Phase Hybrids
5. Hot corrosion/LCF

# Al-MAX Phases and Turbine Environments

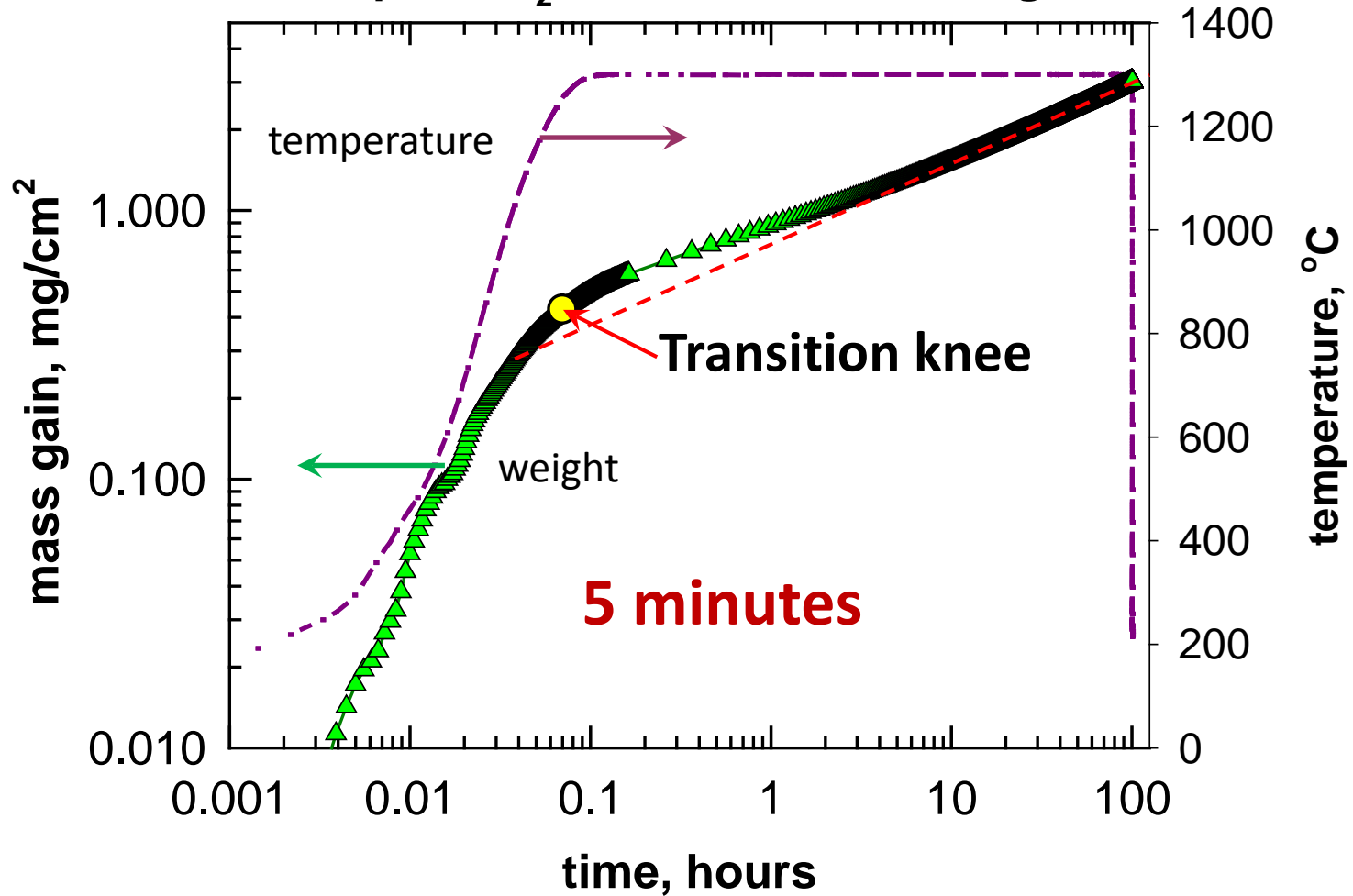
## 1) High temperature $\alpha$ -Al<sub>2</sub>O<sub>3</sub> kinetics

- Al<sub>2</sub>O<sub>3</sub> grain boundary diffusivity
- transient TiO<sub>2</sub> growth
- cubic kinetics

“Oxygen Permeability and Grain-Boundary Diffusion Applied to Alumina Scales,”  
*NASA TM 217855*, pp. 1–14, August 2013.

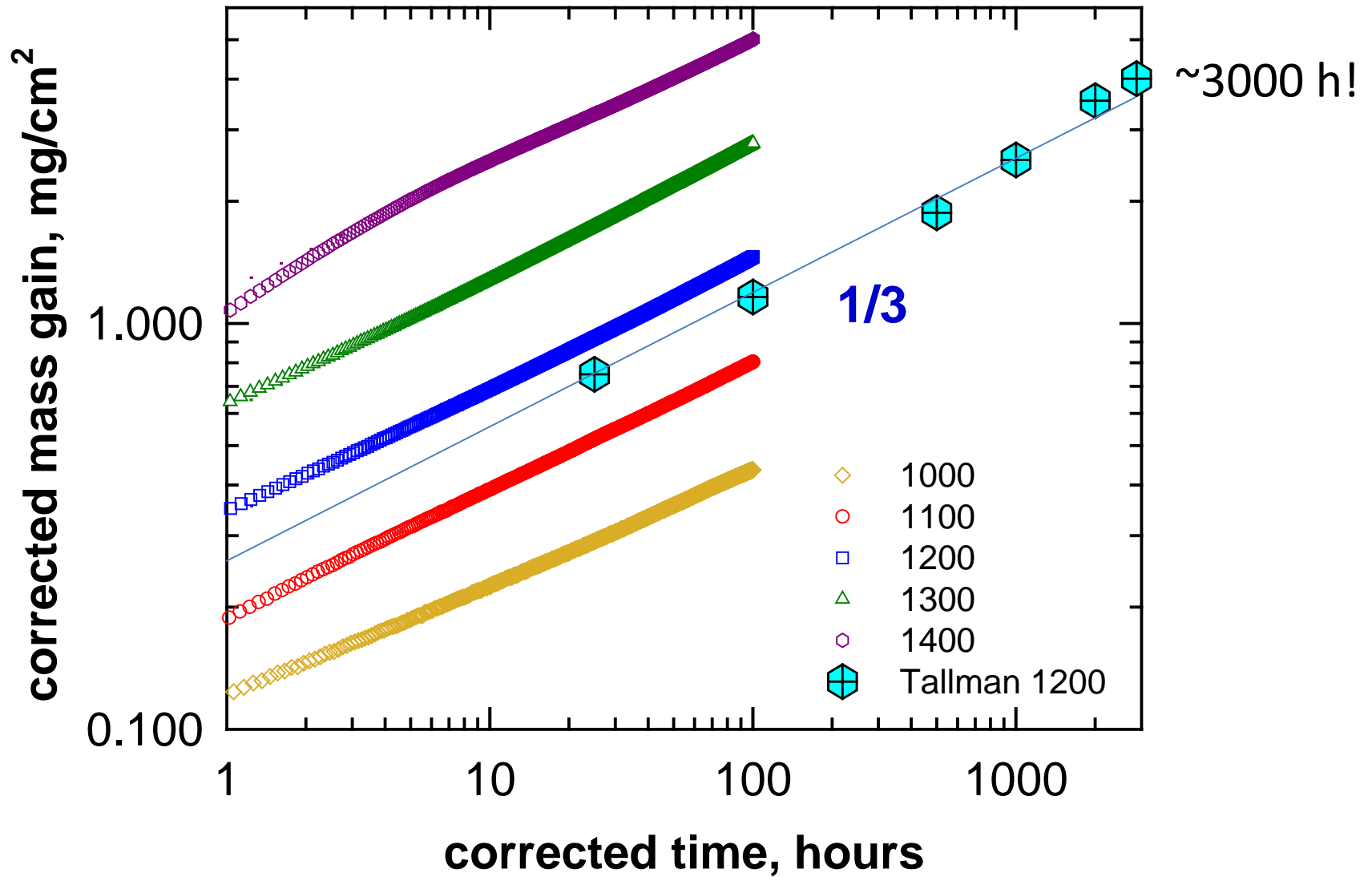


1300°C Ti<sub>2</sub>AlC TGA:  
Rapid TiO<sub>2</sub> Transients on heating

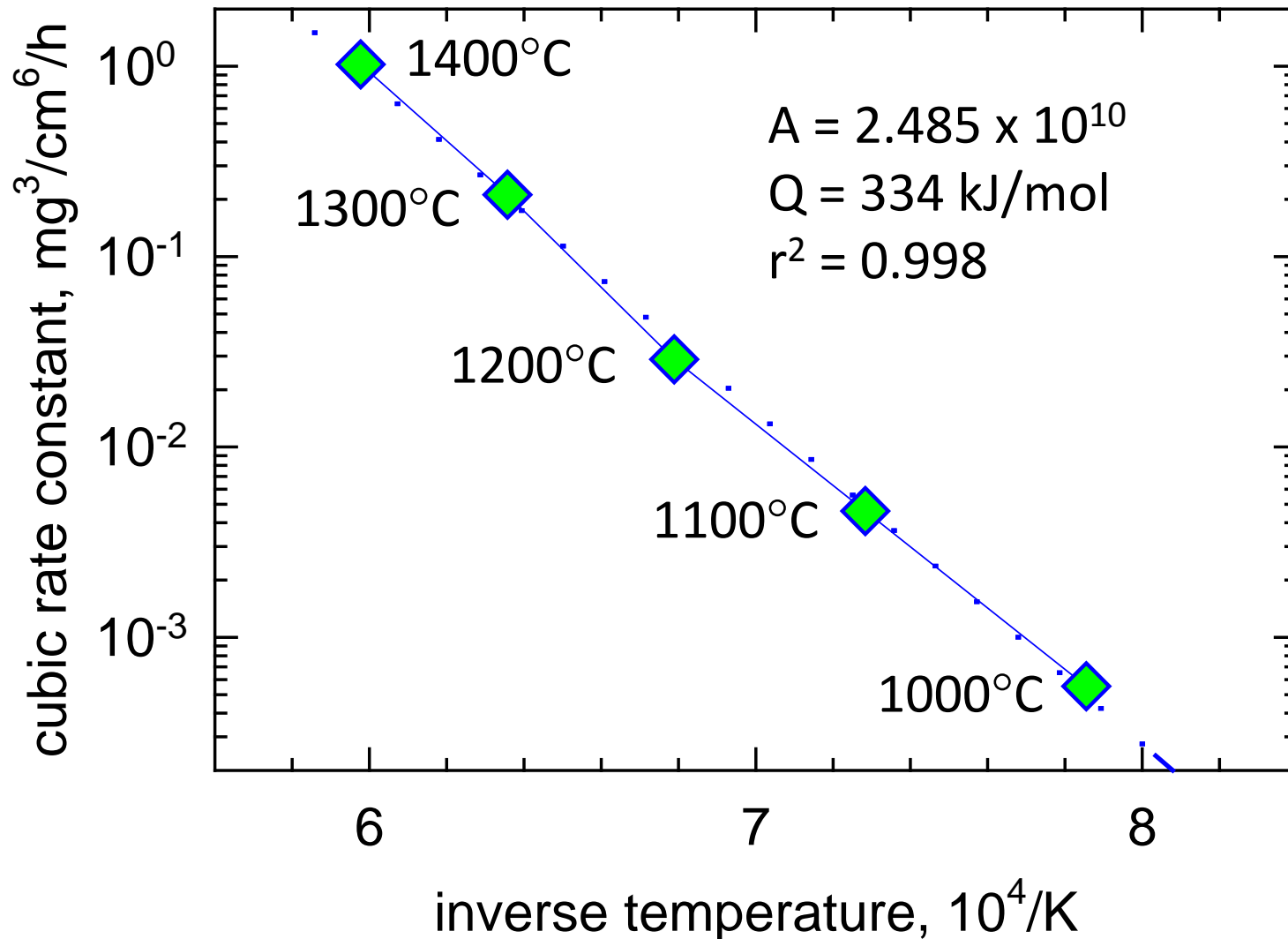


“Kinetic Aspects of Ti<sub>2</sub>AlC Oxidation,” *Oxidation of Metals*, 83 (2015) 351

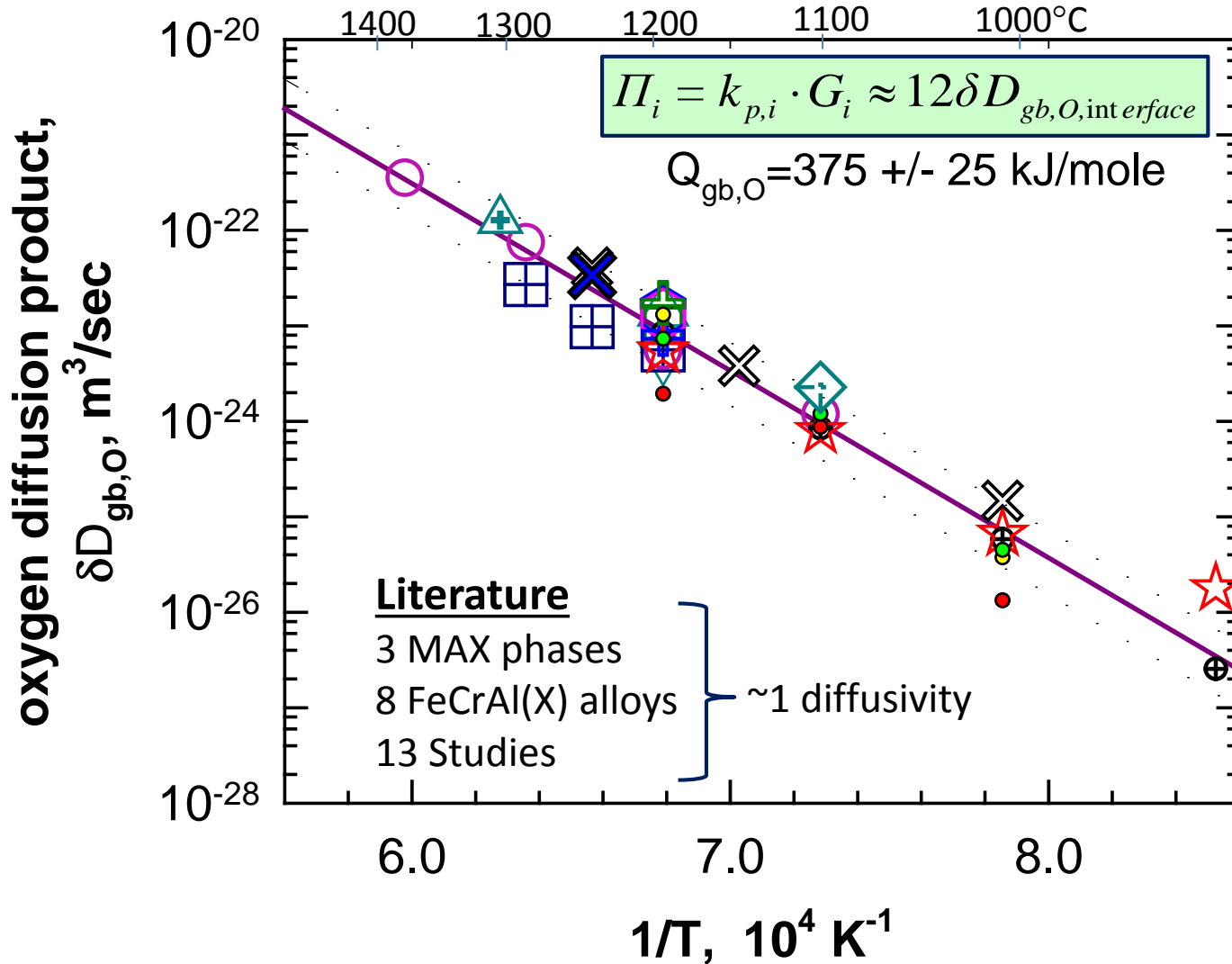
# TGA Oxidation of $Ti_2AlC$ in Air



# Arrhenius Plot of Cubic Oxidation Rate for $Ti_2AlC$



# Scale Grain Boundary Diffusivity MAX Compounds and FeCrAl(X) Alloys



“Oxygen diffusivity in alumina scales grown on Al-MAX phases,”  
*Corrosion Science*, vol. 91, pp. 281–286, Feb. 2015.

## Al-MAX Phases and Turbine Environments

### 2) First high pressure burner rig test of $\text{Ti}_2\text{AlC}$

(1100°-1300°C, Jet A fuel, 25 m/s,

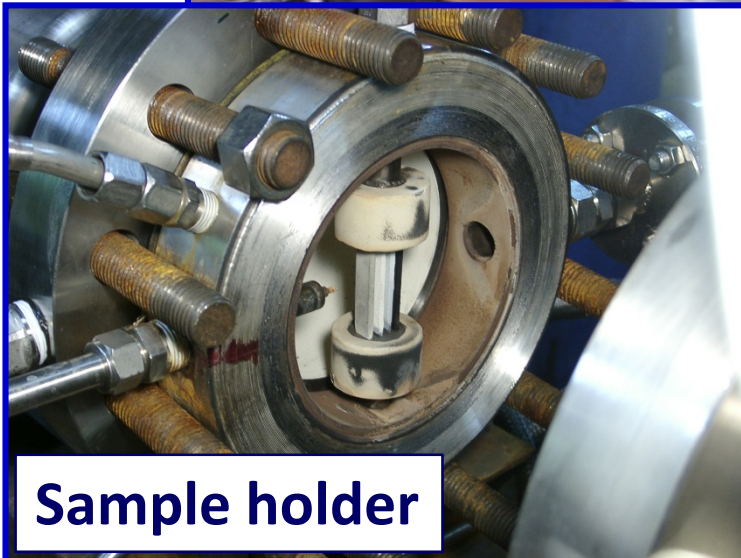
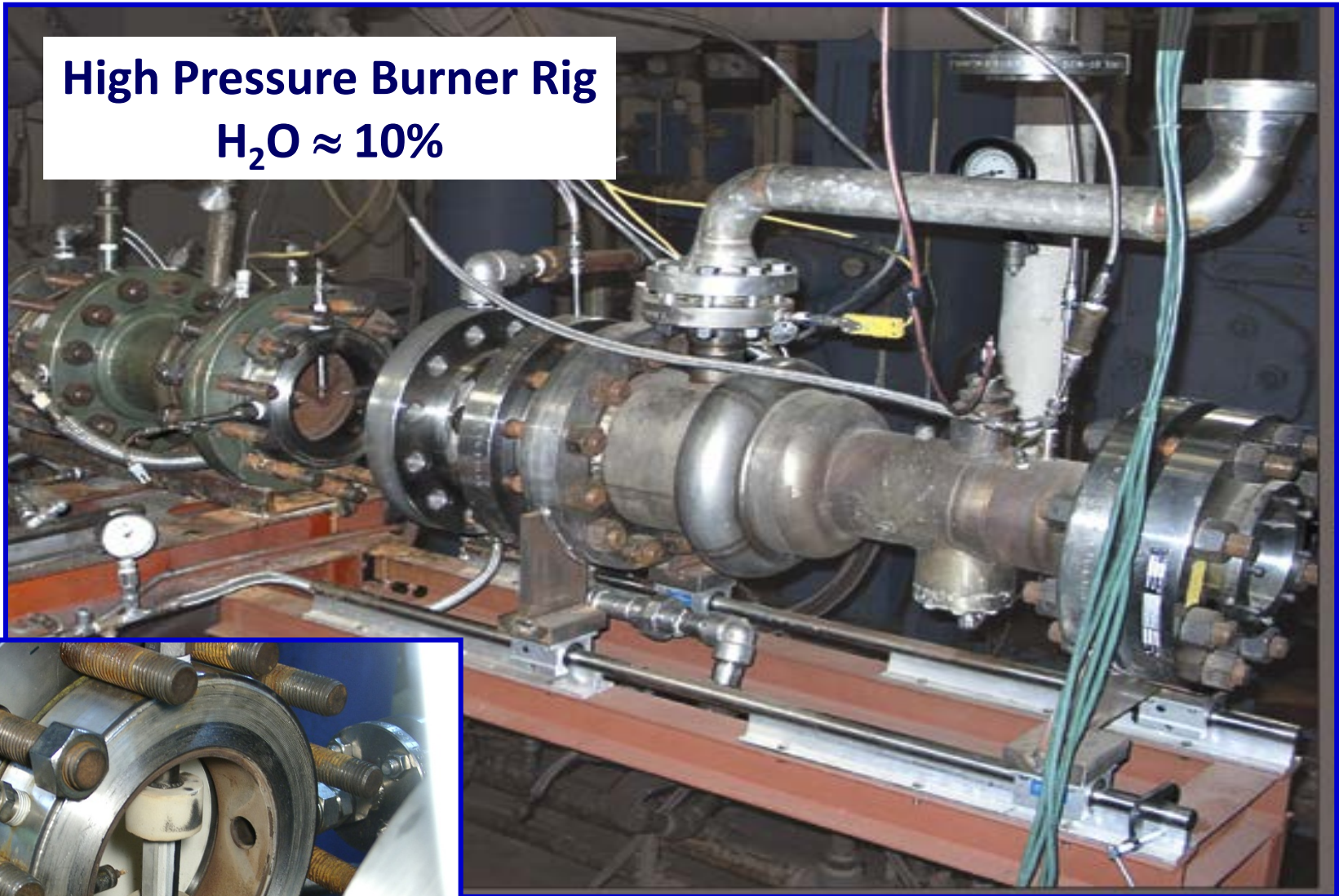
6 atm., 10% water vapor)

- transient  $\text{TiO}_2$  growth
- cubic kinetics
- **$\text{H}_2\text{O}$  - scale volatility issues (?)**

“Environmental resistance of a  $\text{Ti}_2\text{AlC}$ -type MAX phase in a high pressure burner rig,” *J. Eur. Ceram. Soc.* 37 (2017) 23–34.

# High Pressure Burner Rig

$\text{H}_2\text{O} \approx 10\%$



Sample holder

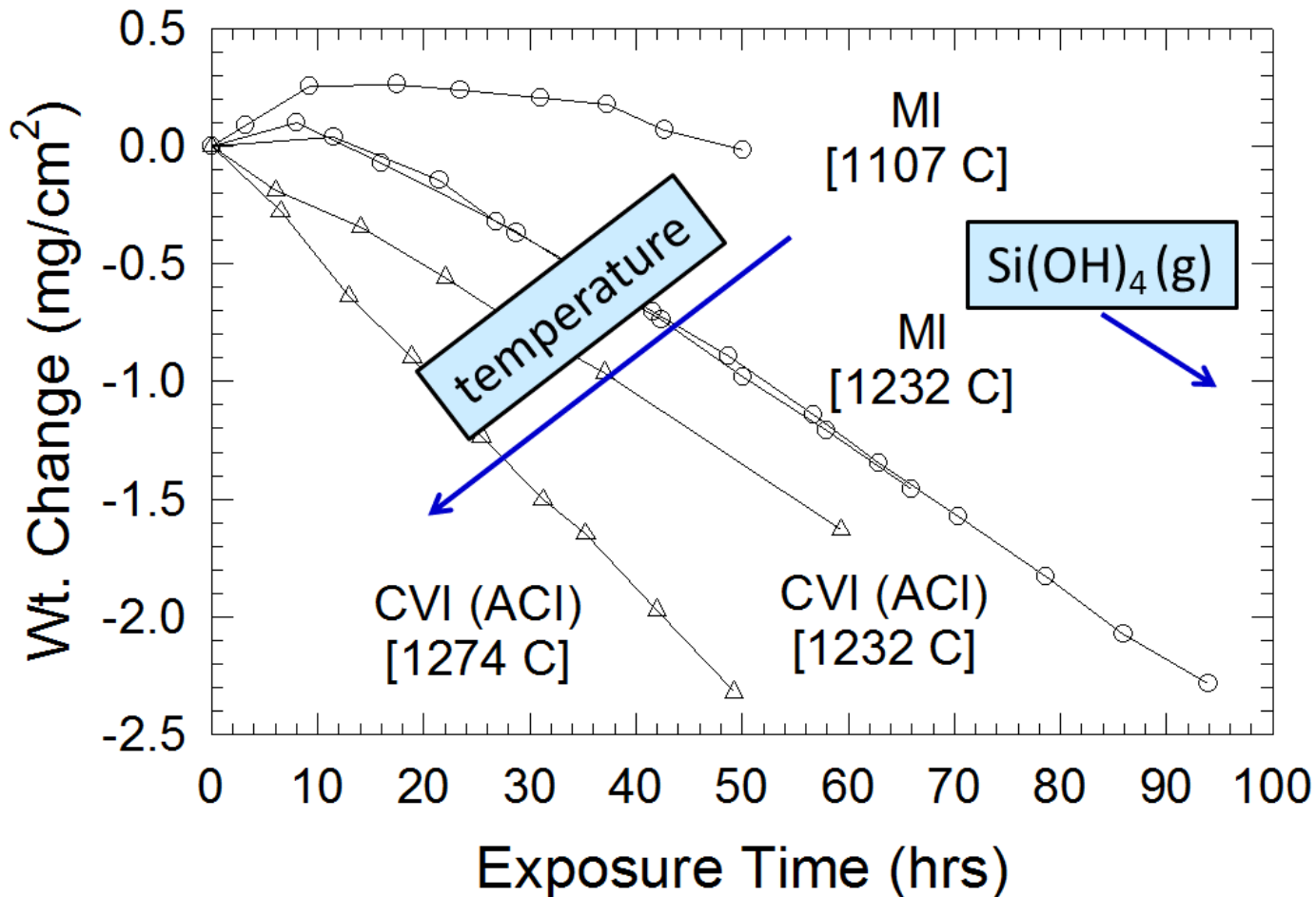
Up to  $\sim 1500^\circ\text{C}$ ,  $\sim 15$  atm,  $\sim 200$  m/s  
(decommissioned 2016)

# HPBR SiC/SiC CMC *Paralinear* Weight Change

(1100 °-1300°C, 6 atm, 25 m/s; Robinson/Smialek 1998)

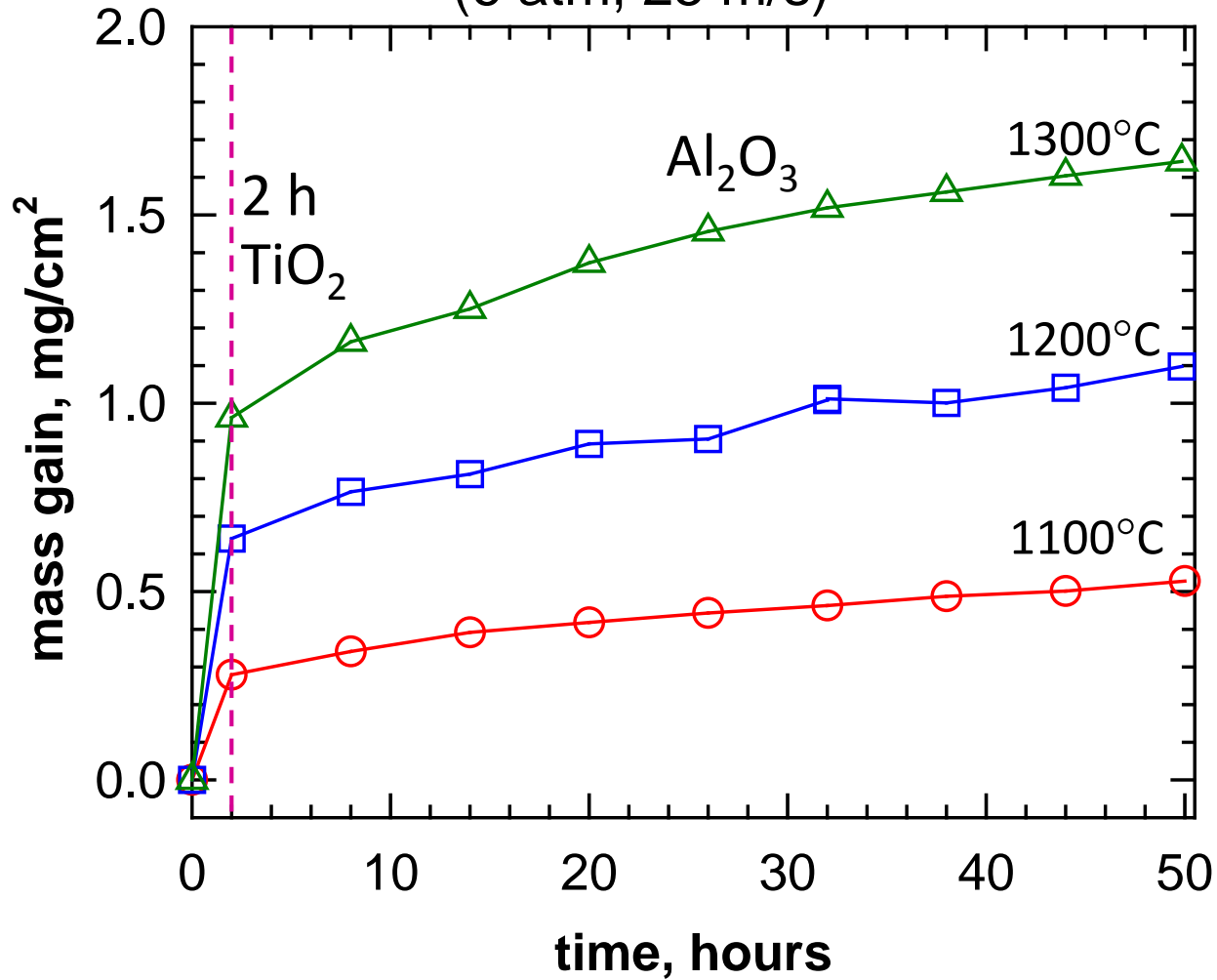


(Opila et al., 1998-2006)



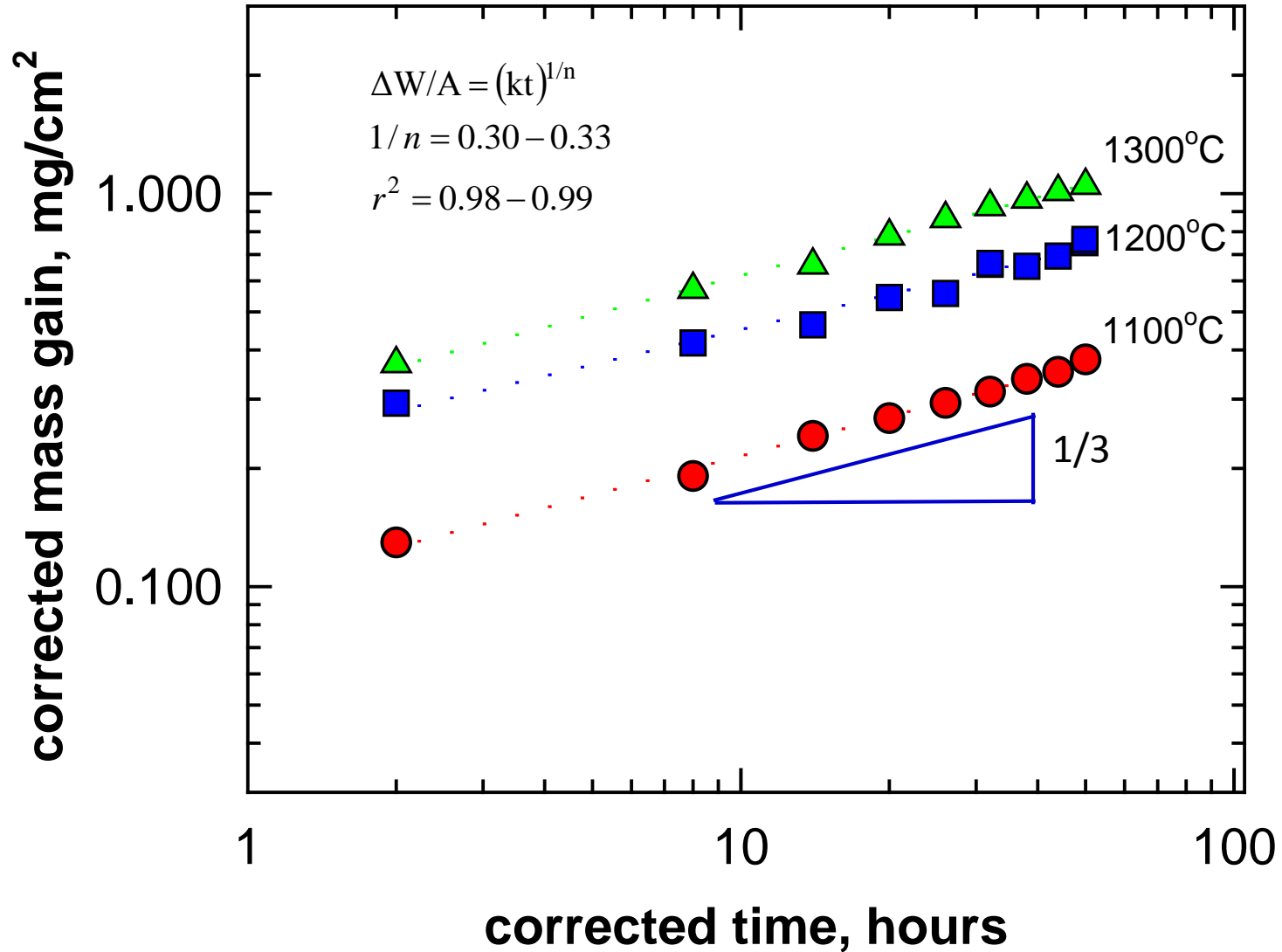
# HPBR Oxidation *Gains* for $\text{Ti}_2\text{AlC}$

(6 atm, 25 m/s)



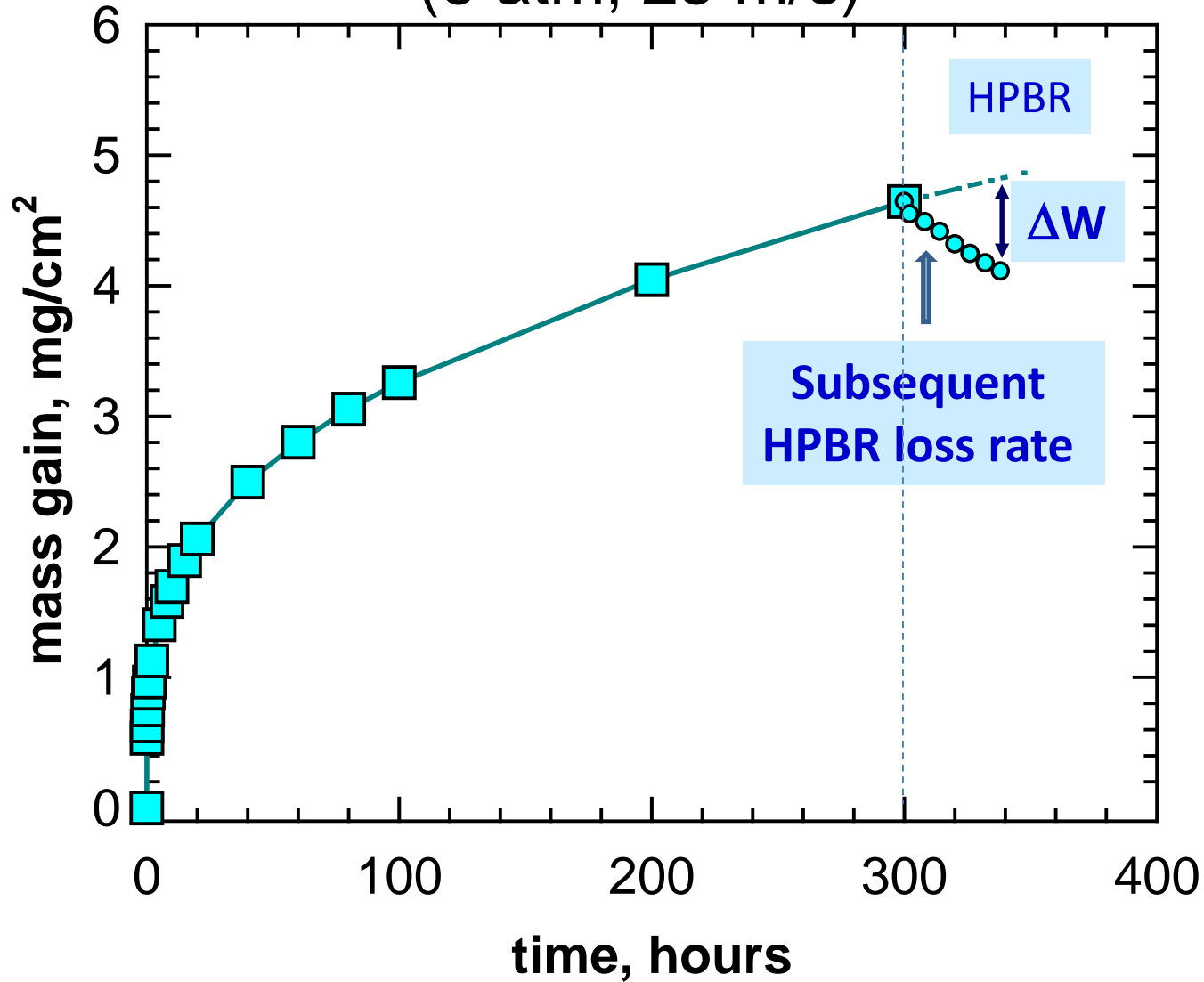


Offset Corrected  $Ti_2AlC$  HPBR Oxidation:  
**Good cubic behavior,  $\sim 1/2$  TGA**



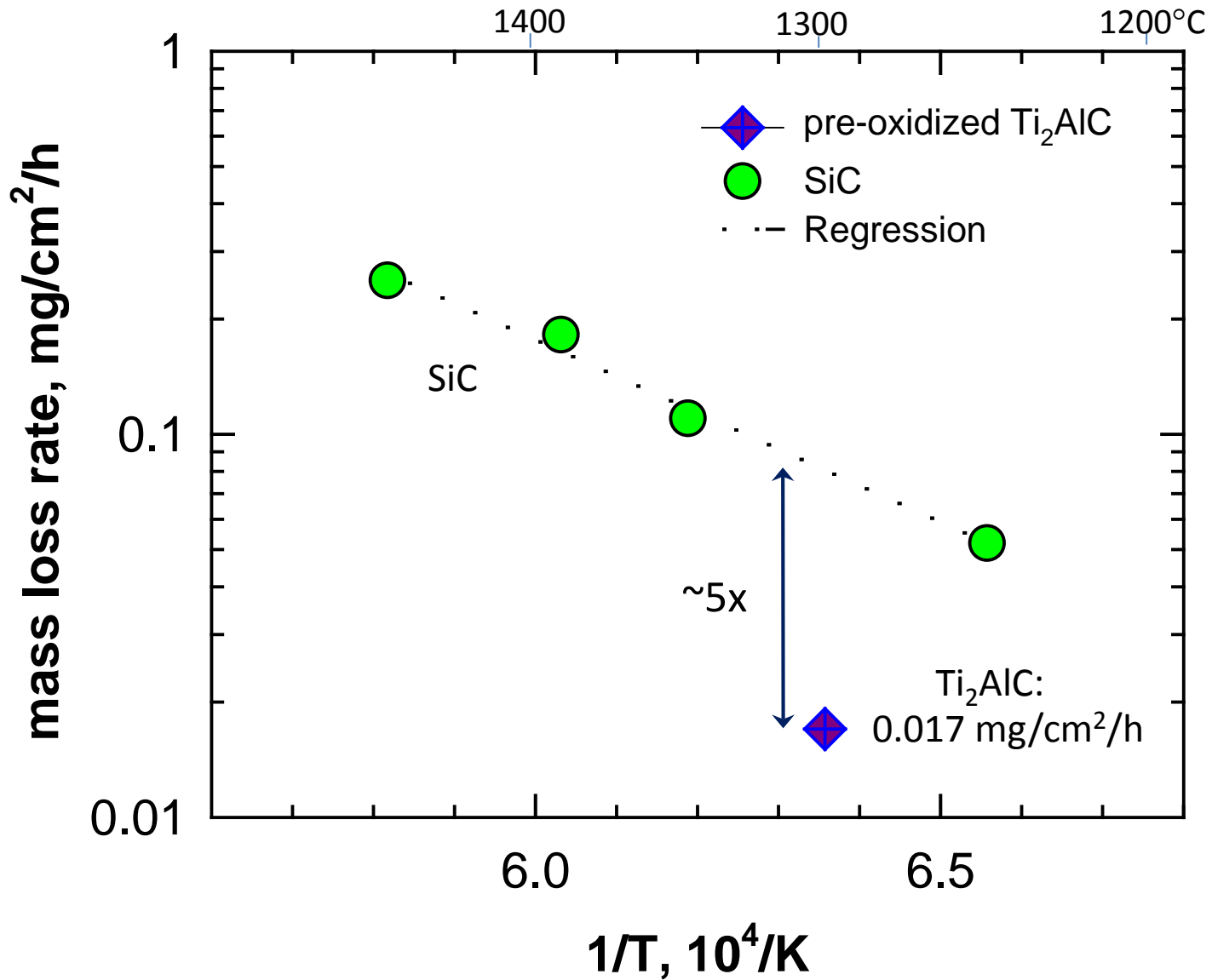
# HPBR after 1300°C Pre-oxidation of Ti<sub>2</sub>AlC

(6 atm, 25 m/s)

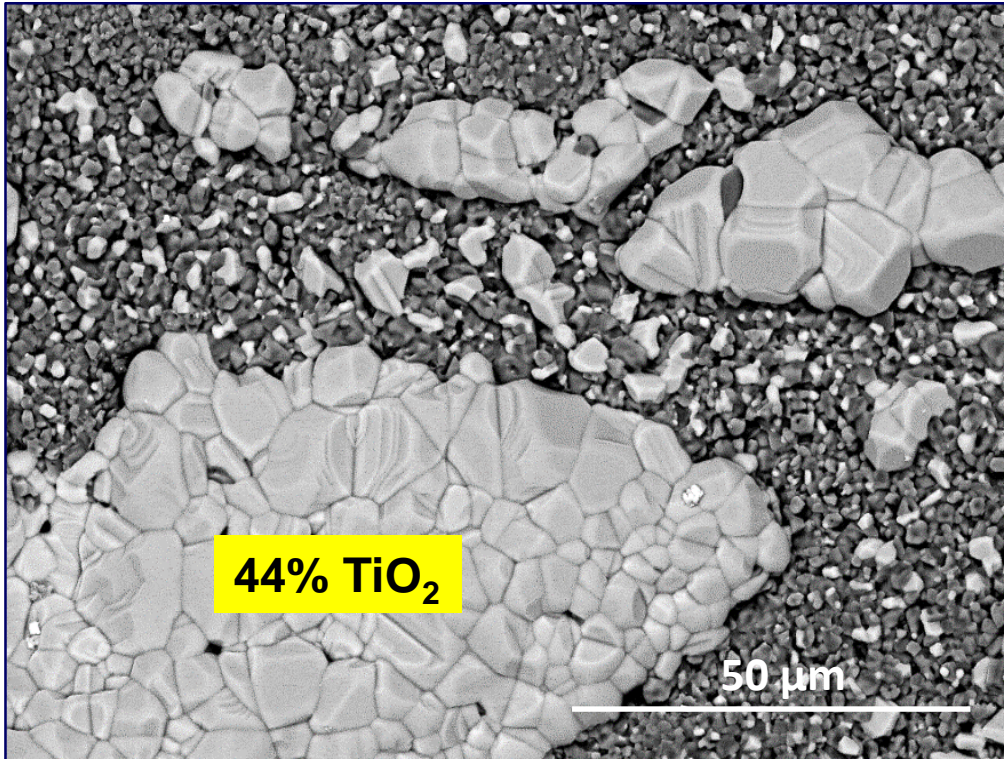


# HPBR Recession Rates for $Ti_2AlC$ and SiC

6 atm., 25 m/s

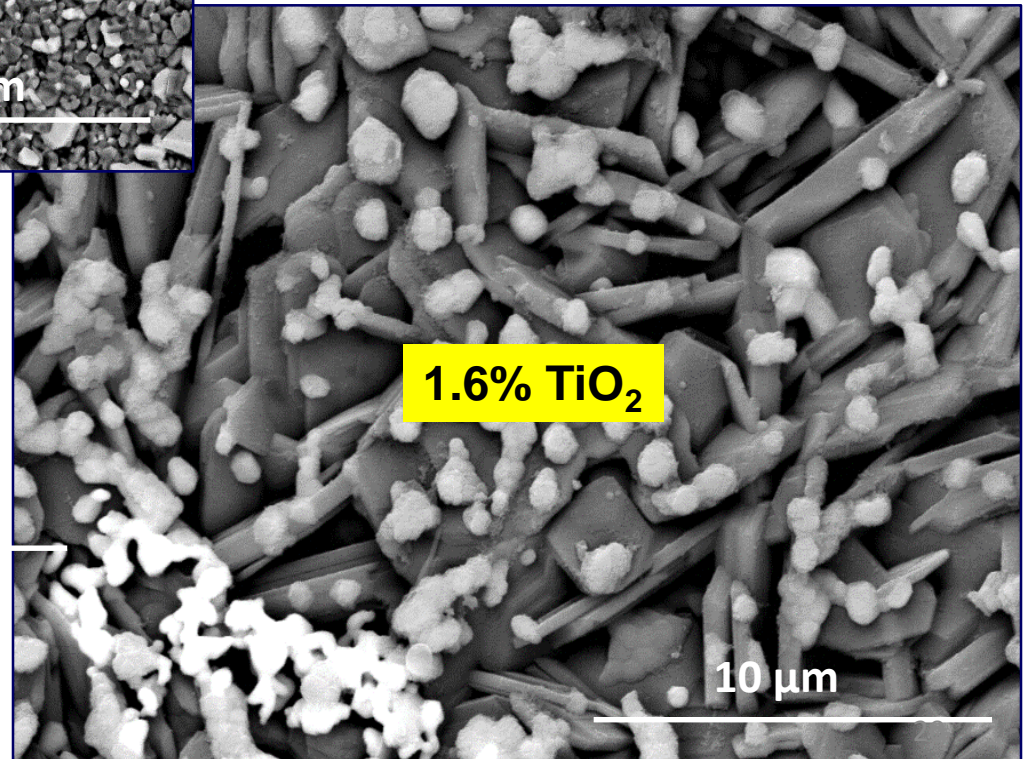


TiO<sub>2</sub> Reduced in  
1200°C HPBR Exposures



HPBR, 50 h

TGA, 100 h



# Hybrid Concepts (EBC/TBC) Enabled by MAX Phases

Intermediate CTE, Strain Tolerance, YSZ Compatibility

Liner, Seals, Bond Coats (?)

CTE ( $10^{-6}/^{\circ}\text{C}$ )

YSZ	10	Top Coat
$\text{Al}_2\text{O}_3$	9	Scale
$\text{Ti}_2\text{AlC}$ $\text{Ti}_3\text{AlC}_2$	8 9	Bond Coat
Rene N5	15	Substrates
SiC CMC	4	
Mo, Nb, Ti	10	



Critical interface

Attachment ?

# Al-MAX Phases and Turbine Environments

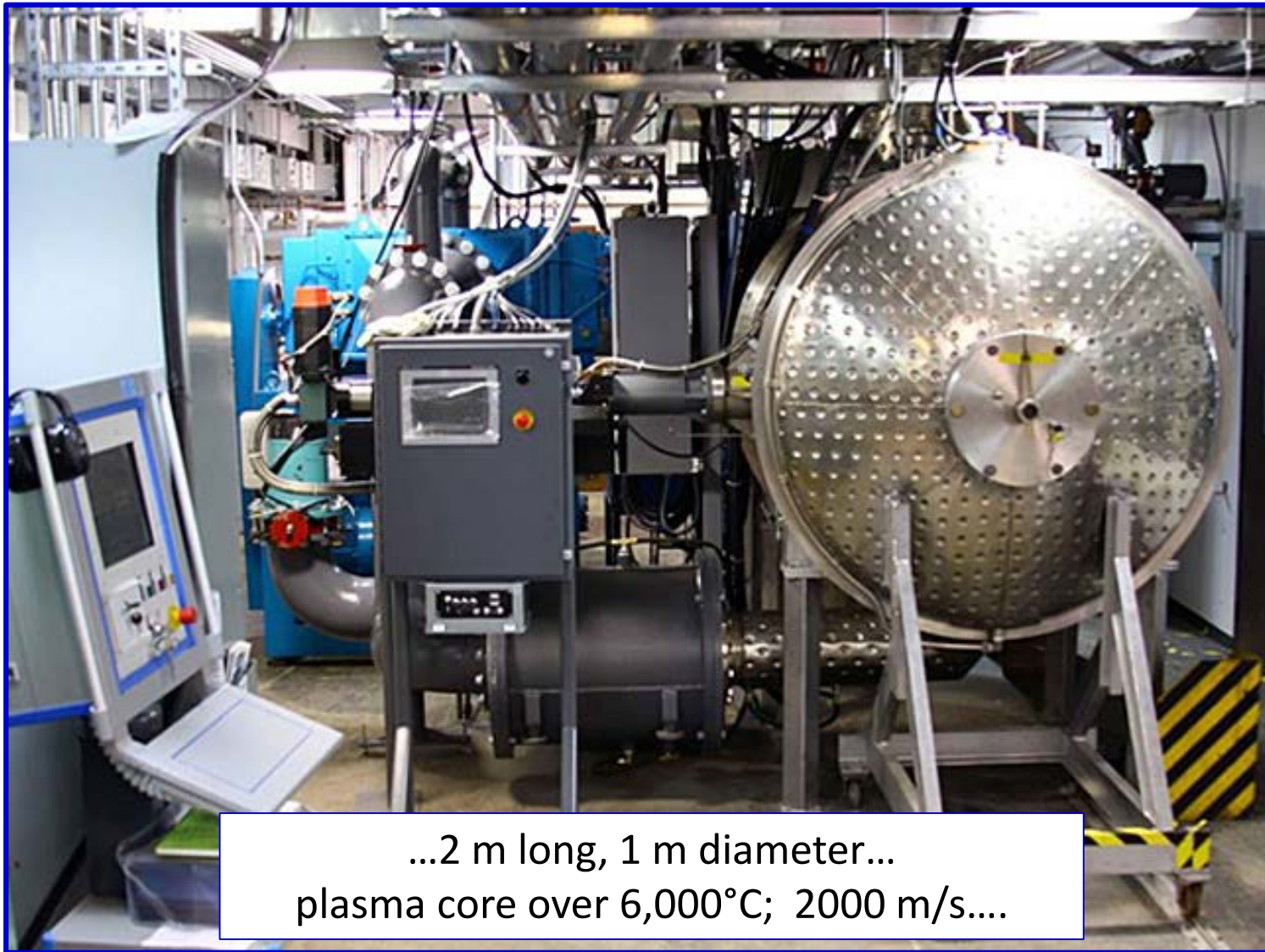
Baseline 1150°C/<1000 h life  
(YSZ/Ni(Pt)Al on CMSX-4)

## 3) YSZ Thermal Barrier Coatings on MAX Phases

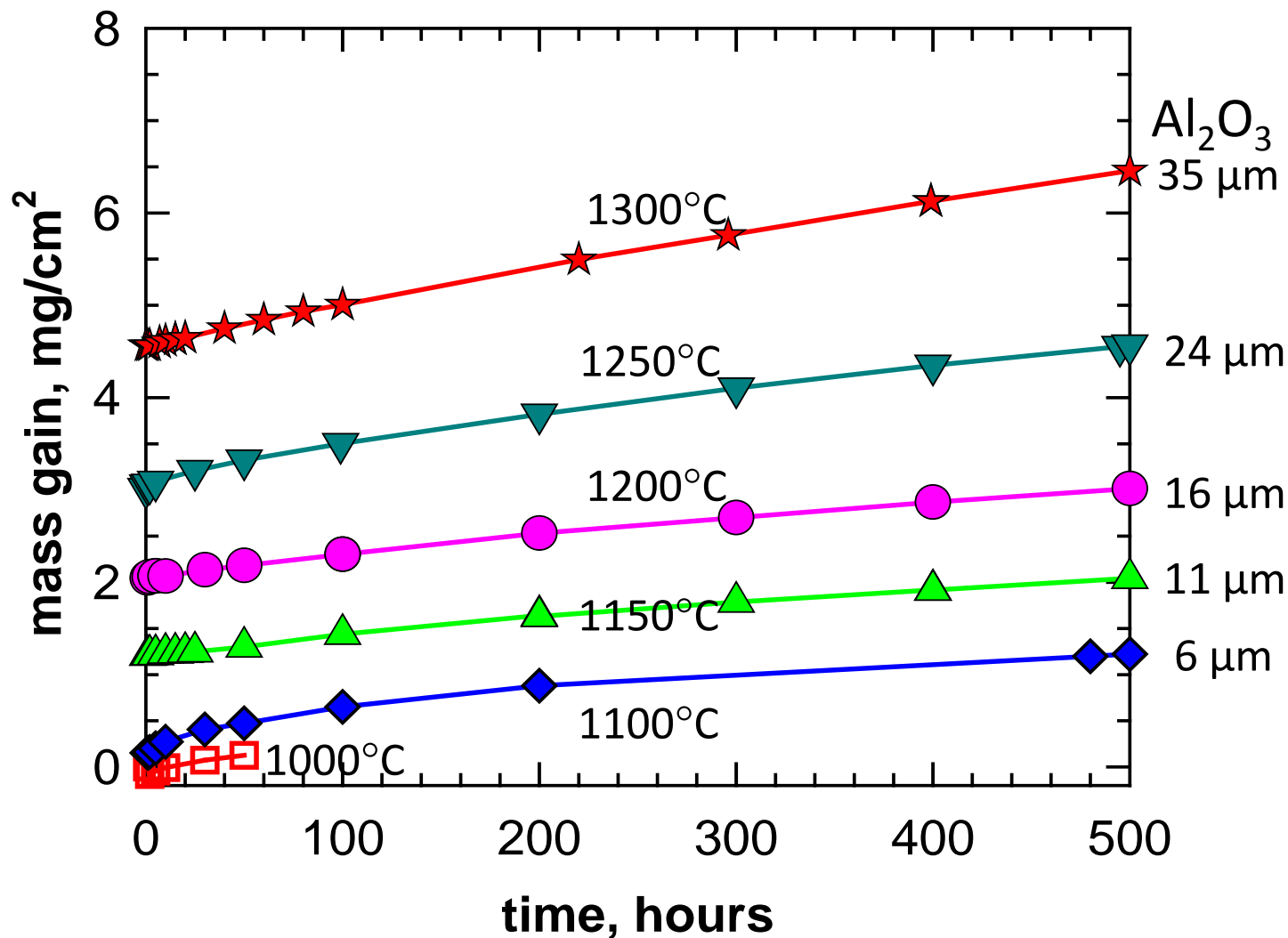
- APS and PS-PVD (~100  $\mu\text{m}$ )
- $\text{Ti}_2\text{AlC}$  (CTE 9);  $\text{Cr}_2\text{AlC}$  (CTE 13)
- Stepped furnace test
  - 1100° - 1300° C
  - 500 h each  $\rightarrow$  2500 h

“Oxidative Durability of TBCs on MAX Phase Substrates,”  
*Surface and Coatings Technology*, [285](#), 15 2016, 77–86.

# Plasma Spray-Physical Vapor Deposition PS-PVD

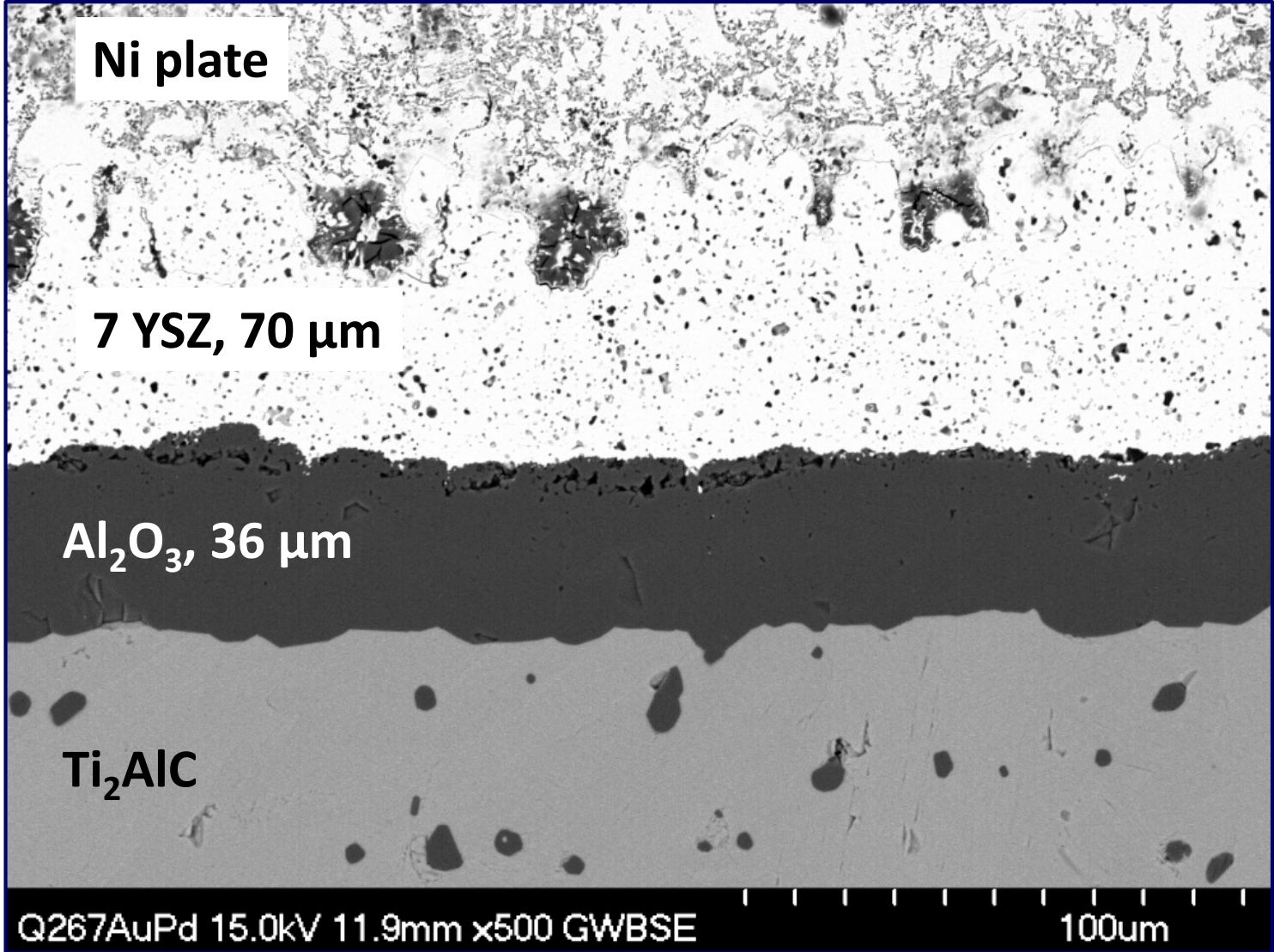


# Interrupted Oxidation of PS-PVD YSZ on $\text{Ti}_2\text{AlC}$



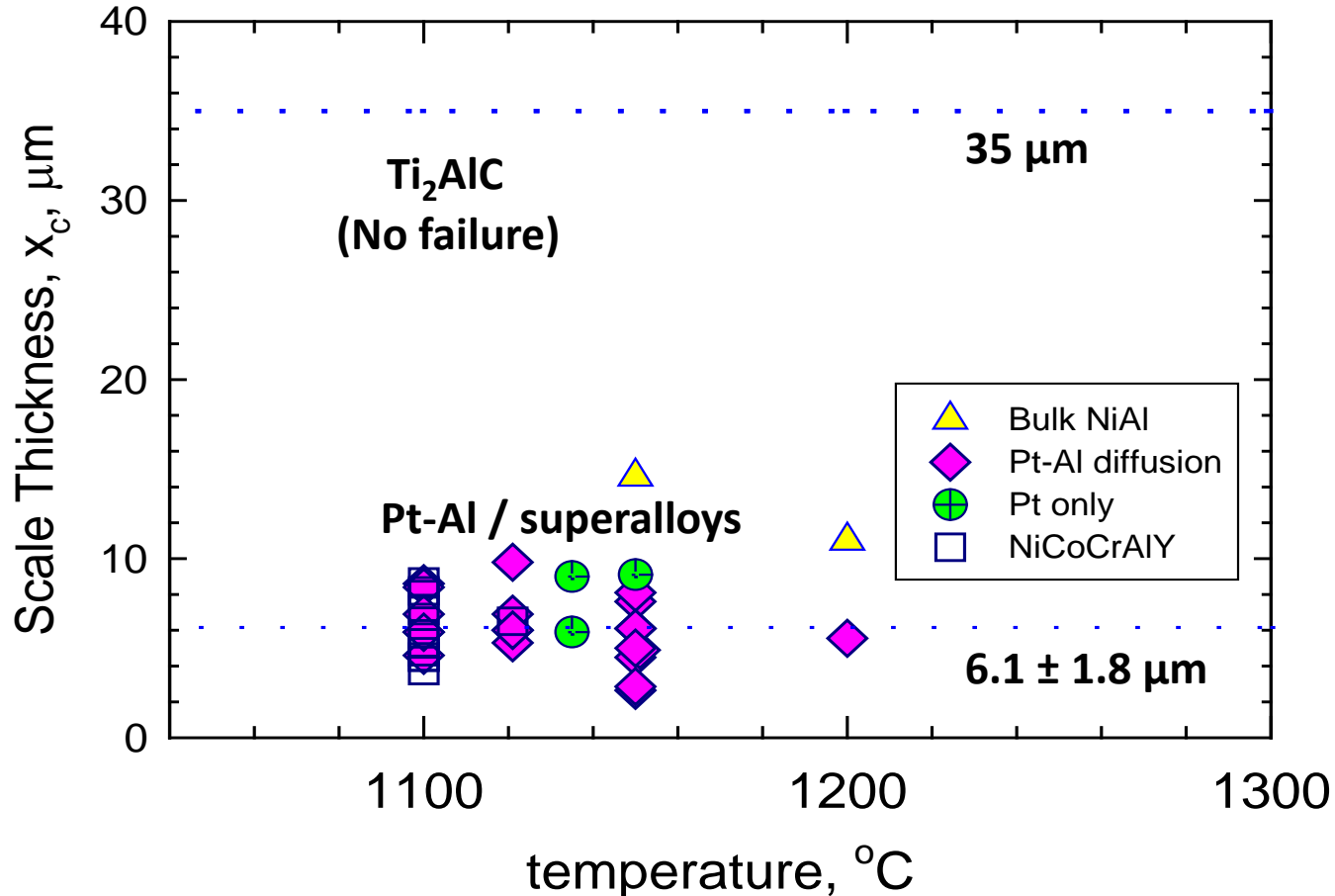


**Intact PS-PVD TBC**  
( $\text{Ti}_2\text{AlC}$ , **1300°C**, 500 h) SEM/BSE



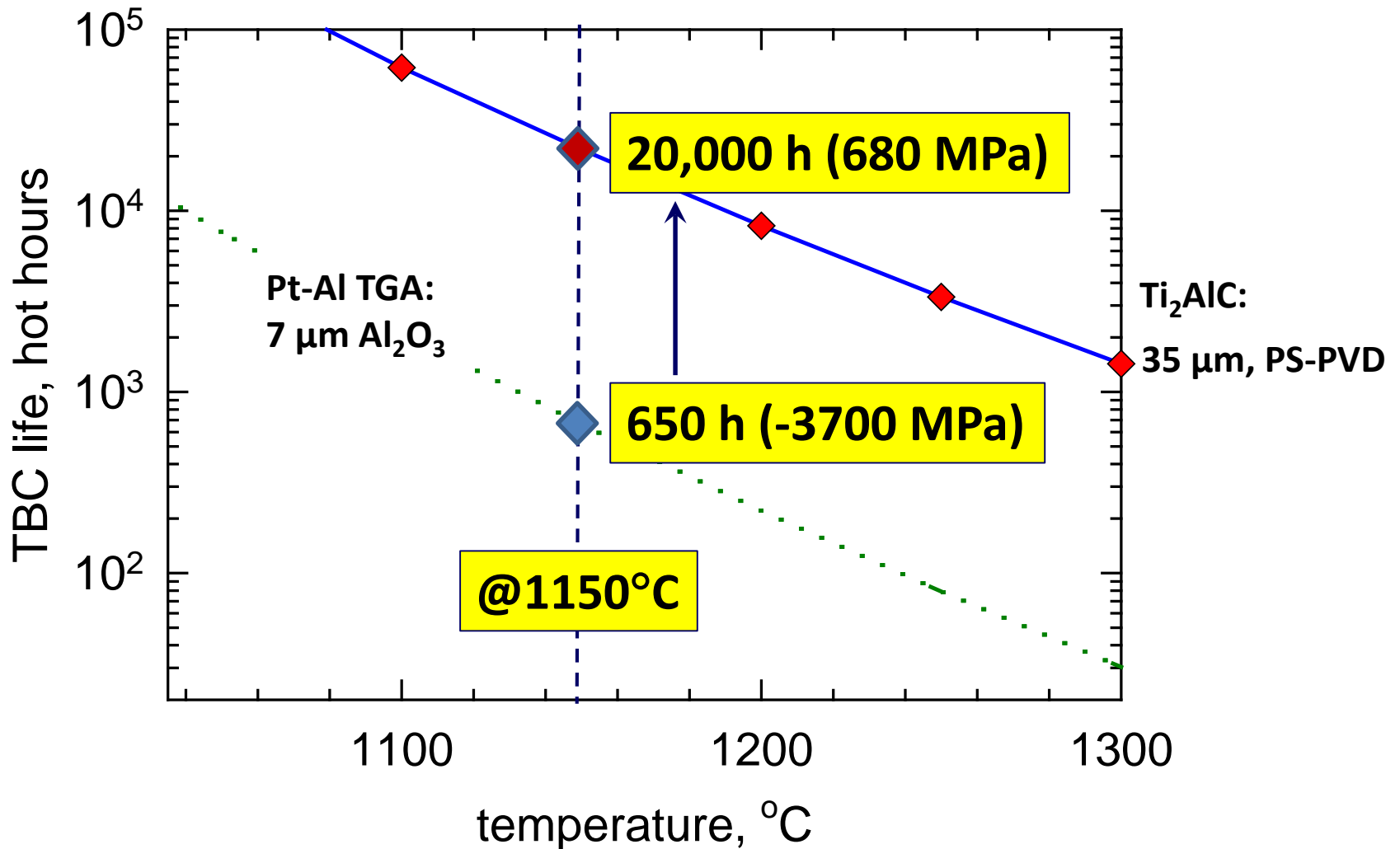
# TBC FCT Life Comparison to Literature: “The 7 $\mu\text{m}$ Rule”

## Alumina Scale Thickness at TBC Failure



“Compiled furnace cyclic lives of EB-PVD thermal barrier coatings,” Surface & Coatings Technology 276 (2015) 31–38.

# EB-PVD TBC FCT Life on Alumina-Forming Systems



# Hybrid Concepts with MAX Phases

Intermediate CTE, Strain Tolerance,  
**LTHC Corrosion Resistance**

CTE,  $10^{-6}/^{\circ}\text{C}$

$\text{Na}_2\text{SO}_4$		Salt
$\text{Al}(\text{Cr})_2\text{O}_3$	9	Scale
$\text{Cr}_2\text{AlC}$	13	Bond Coat
Superalloy	15	Substrates

← Interface stability

## 4) $\text{Cr}_2\text{AlC}$ / Ni-base Superalloys Interfacial Stability:

LTHC Corrosion resistant coating? (92% LCF deficit)

1100°C hot pressed diffusion couples

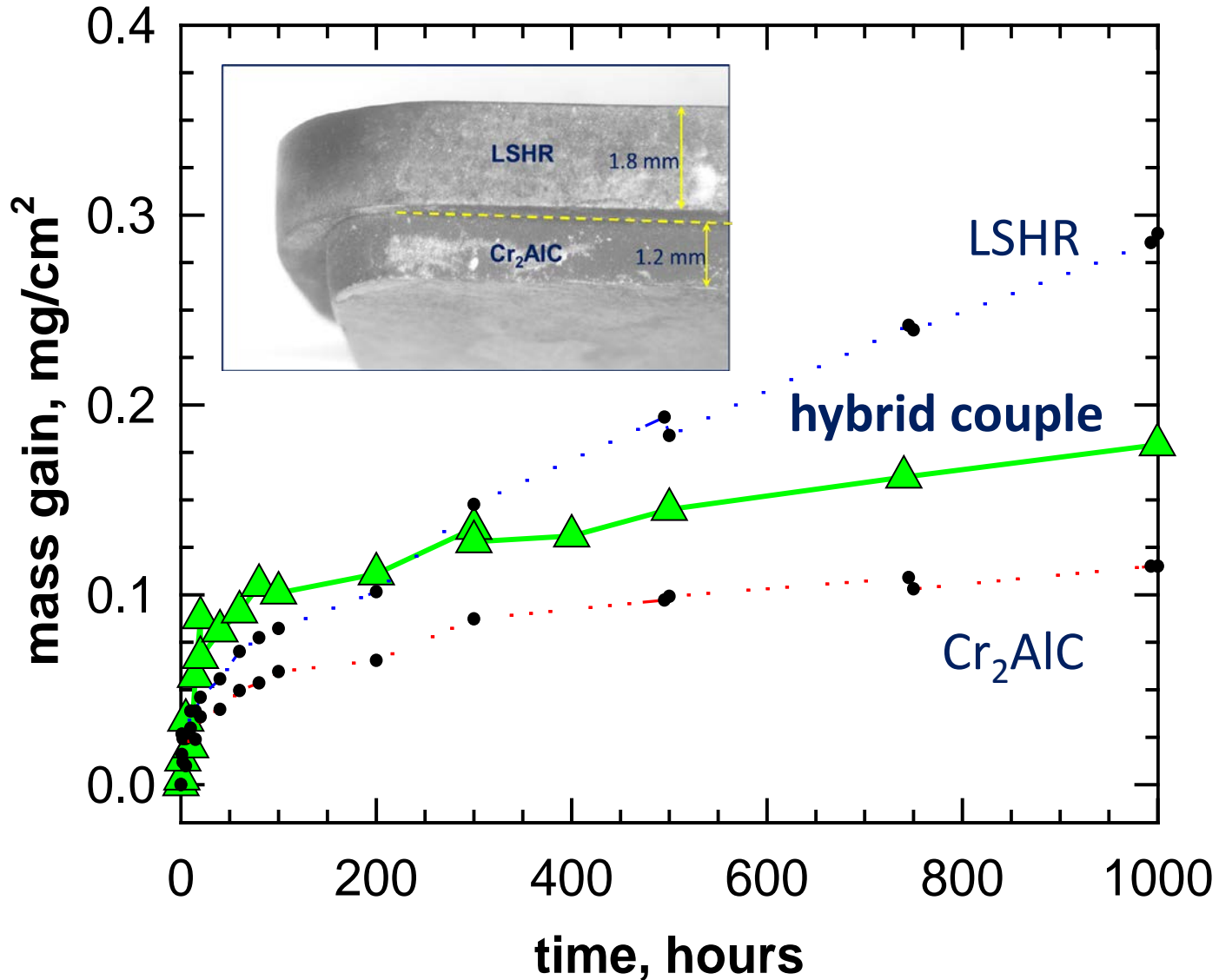
- 0.3 - 3 mm, 2 h @  $10^{-6}$  torr
- $\text{Cr}_2\text{AlC}$  on LSHR disk alloy (or Rene'N5)

800°C (or 1150°C) diffusion/oxidation

- 100 h, 1000 h

“Interfacial Reactions of a MAX Phase-Superalloy Hybrid”  
*Surface and Interface Analysis*, 276 (2015) 31

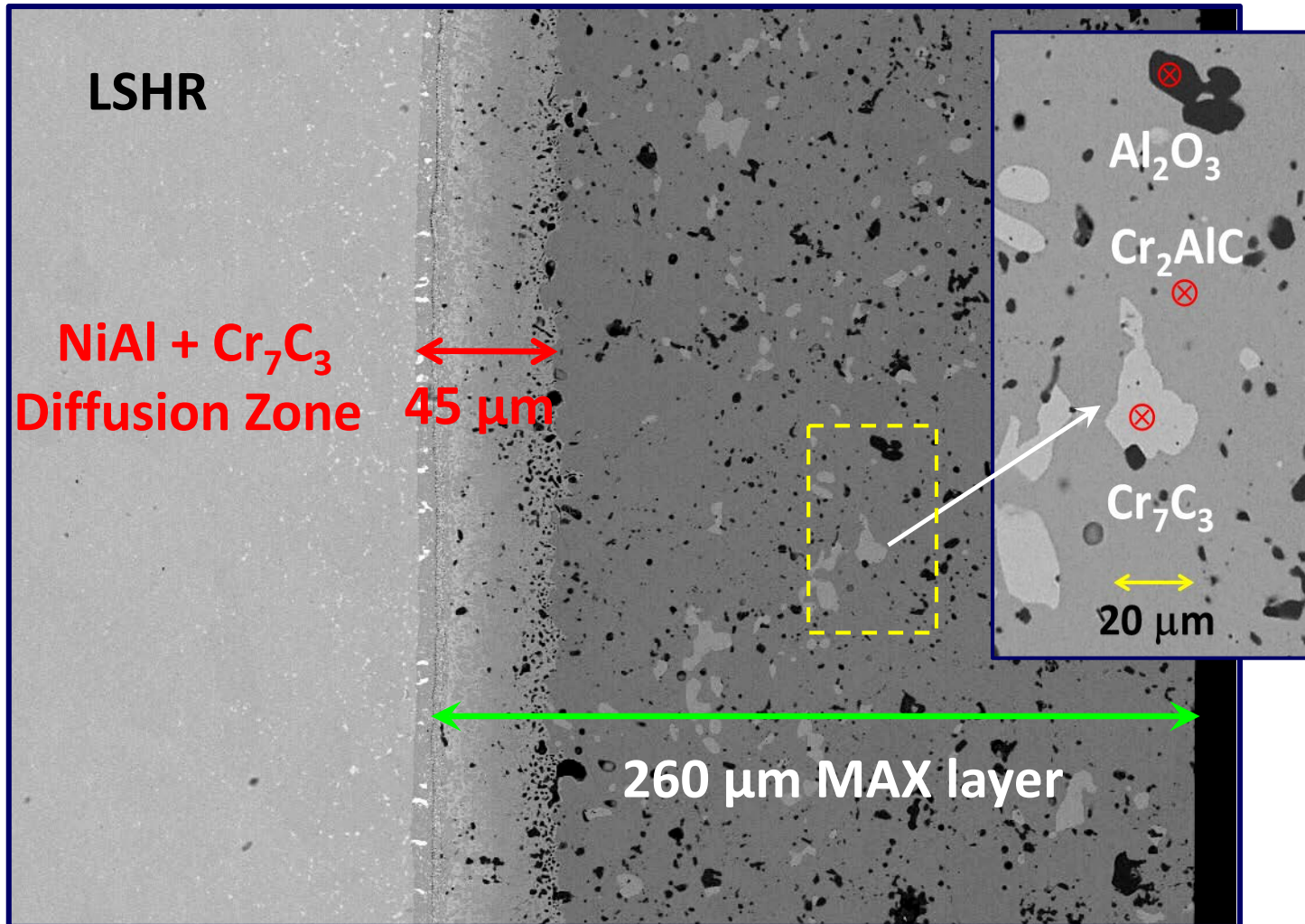
# Survives 1000 h of 800°C Furnace Oxidation LSHR - Cr<sub>2</sub>AlC DC3 Hybrid Couple



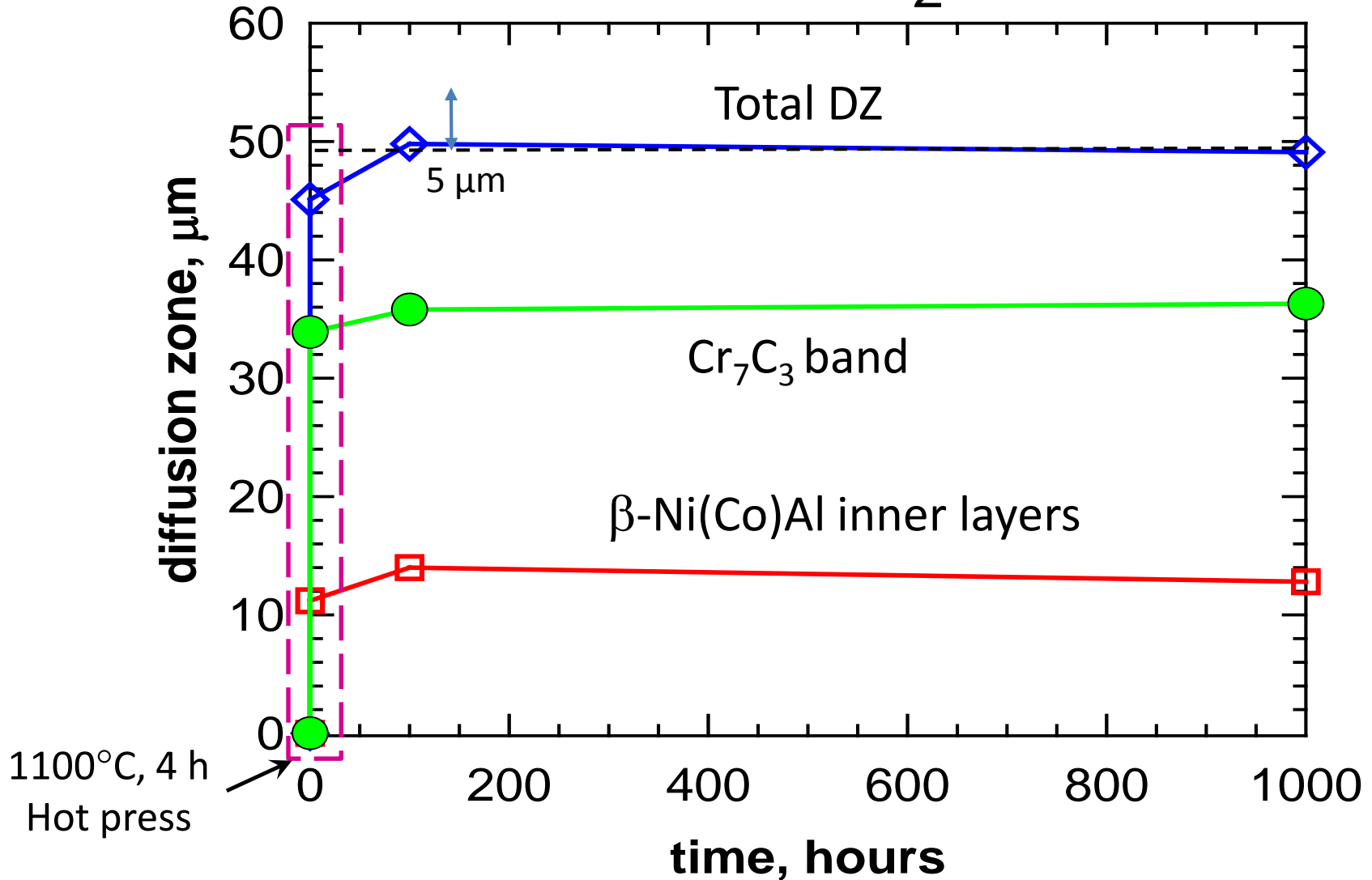
# As-Hot Pressed Cross Section

(2h @ 1100°C produced ~ 50 μm layer)

SEM/BSE

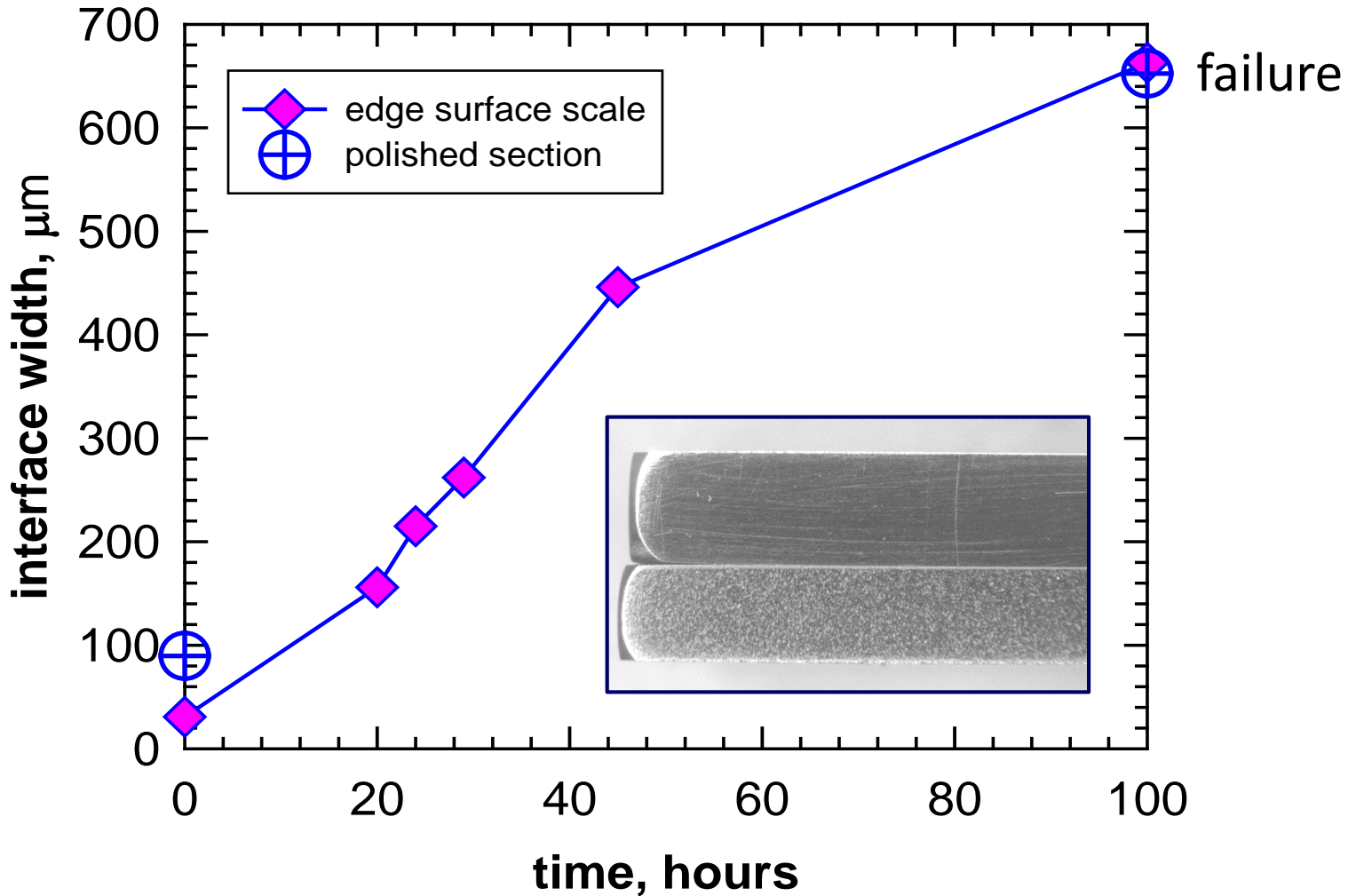


# Minor Diffusion Zone Growth at 800°C LSHR Alloy/Cr<sub>2</sub>AlC



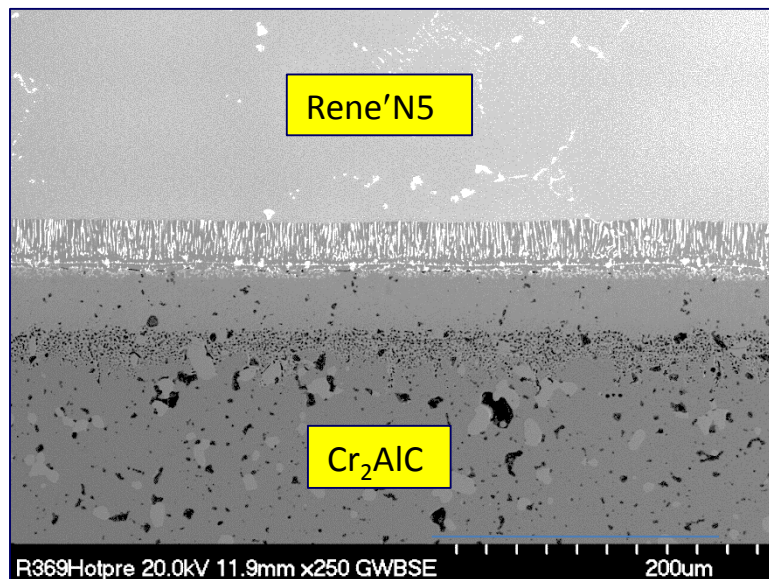


# Cr<sub>2</sub>AlC-Rene'N5 Hybrid Couple: Interdiffusion at 1150°C



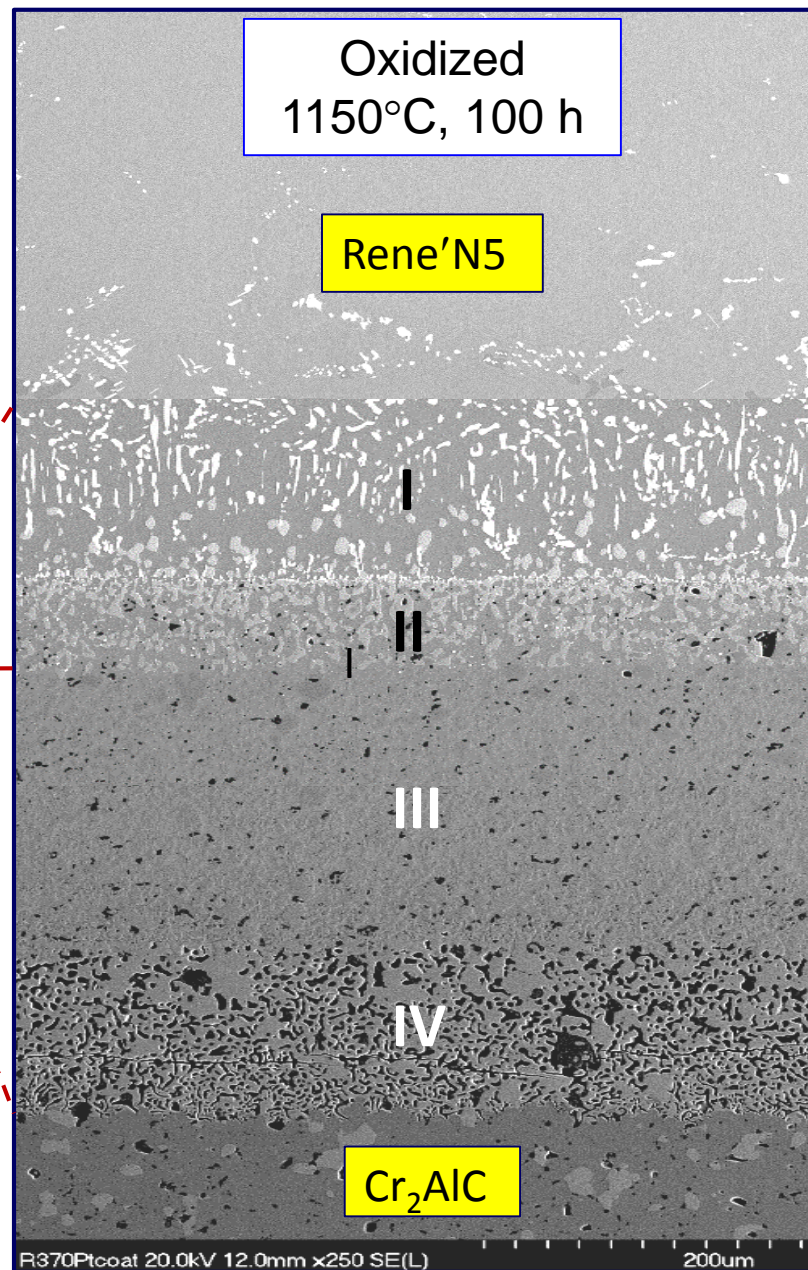
# DC4: Cr<sub>2</sub>AlC/Rene'N5

As Hot-Pressed  
1100°C, 2 h

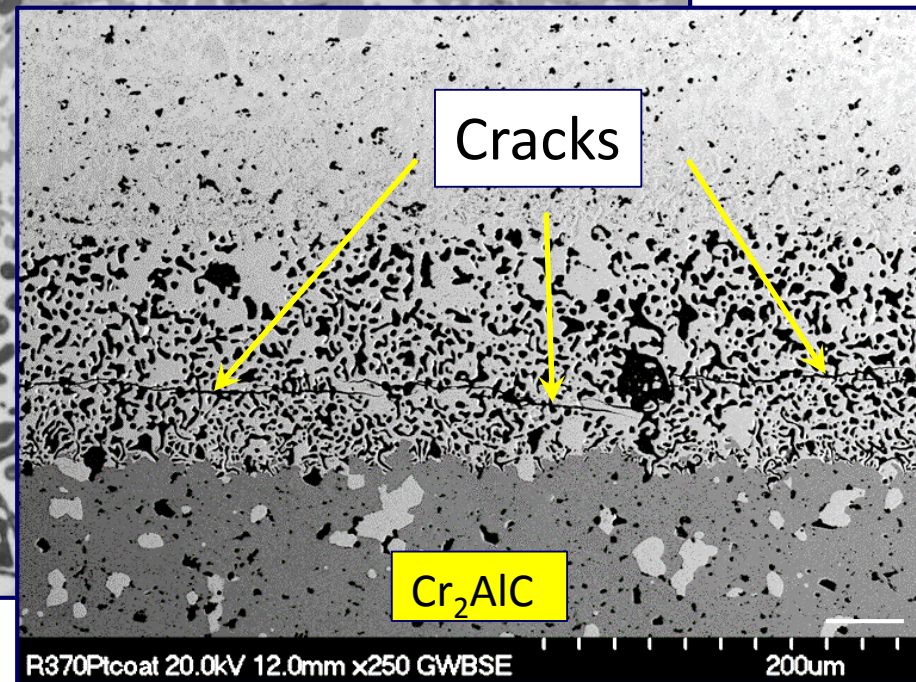
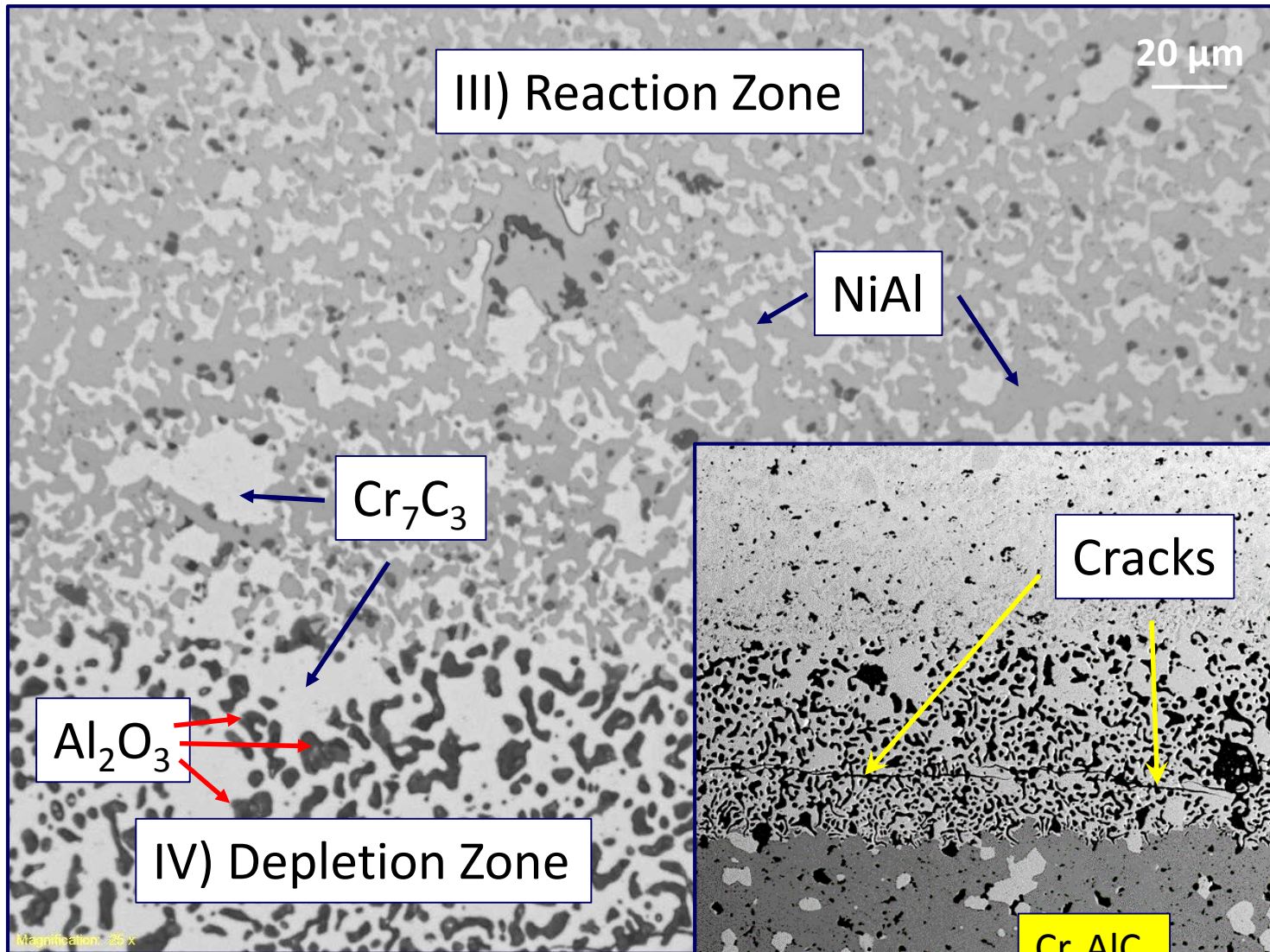


200 μm

Oxidized  
1150°C, 100 h



DC4:  $\text{Cr}_2\text{AlC}/\text{N5}$  (1150°C, 100 h) 200-250 x



# Hybrid Concepts with MAX Phases

Intermediate CTE, Strain Tolerance,  
**LTHC Corrosion Resistance**

CTE,  $10^{-6}/^{\circ}\text{C}$

$\text{Na}_2\text{SO}_4$		Salt
$\text{Al}(\text{Cr})_2\text{O}_3$	9	Scale
$\text{Cr}_2\text{AlC}$	13	Bond Coat
Superalloy	15	Substrates

← Critical chemistry

## Al-MAX Phases and Turbine Environments

### 5) Cr<sub>2</sub>AlC / Disk Alloy Hot Corrosion:

700°C **bulk** LTHC:

500 h, (Na,K)<sub>2</sub>SO<sub>4</sub> (T<sub>e</sub>= 823°C)

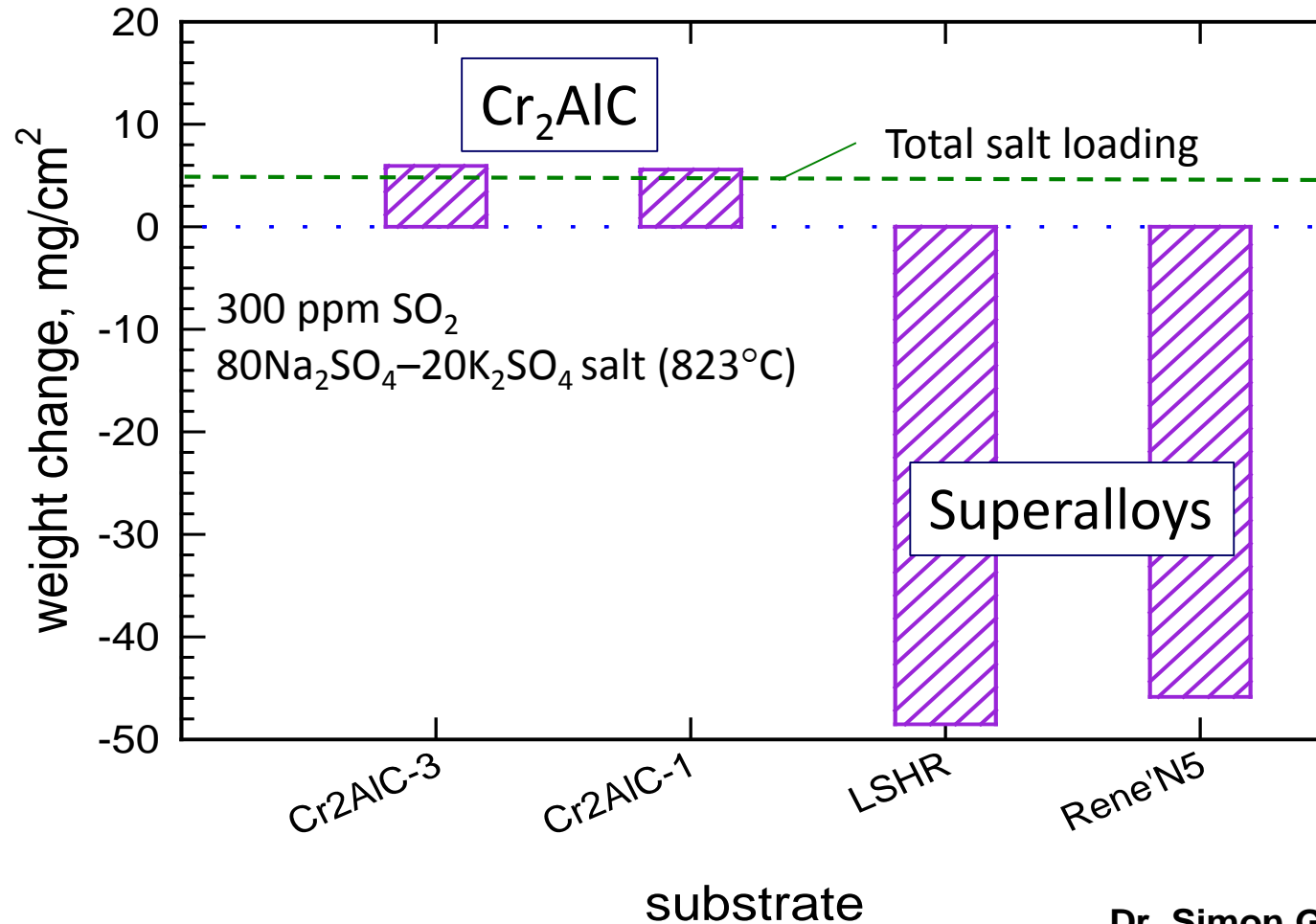
760°C **Coated** LSHR LCF:

840 / -430 MPa at 0.33Hz

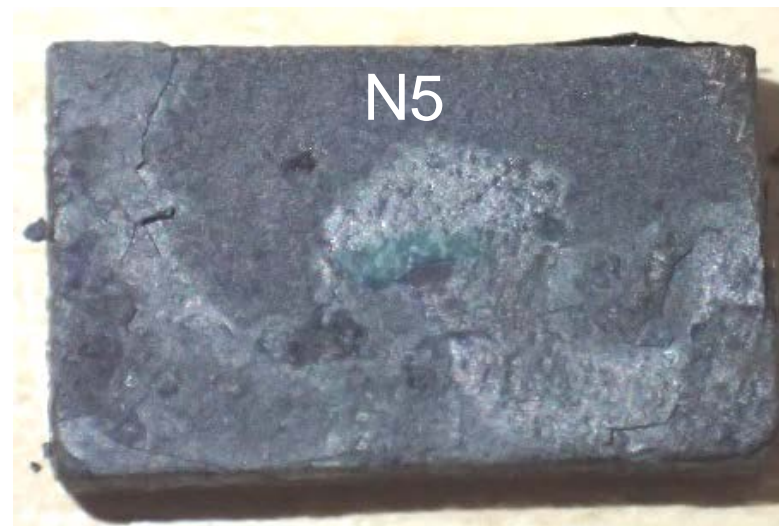
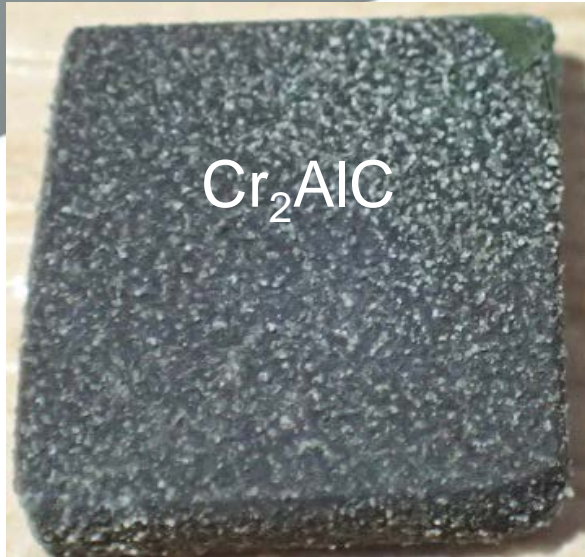
50 h, (Na,Mg)<sub>2</sub>SO<sub>4</sub> (T<sub>e</sub>= 660°C)

# Type II LTHC of Cr<sub>2</sub>AlC (3) and Superalloys (7)

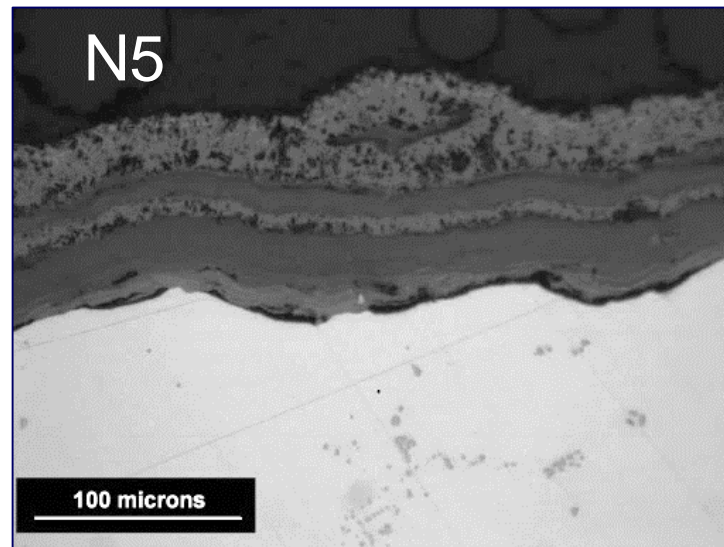
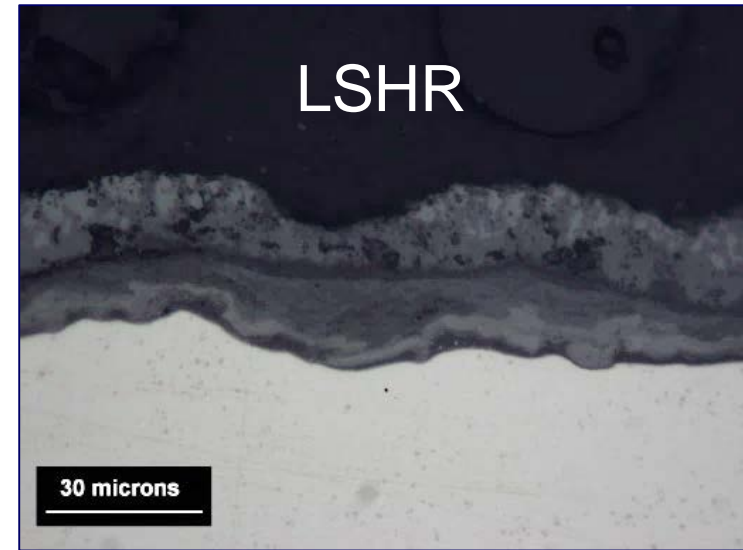
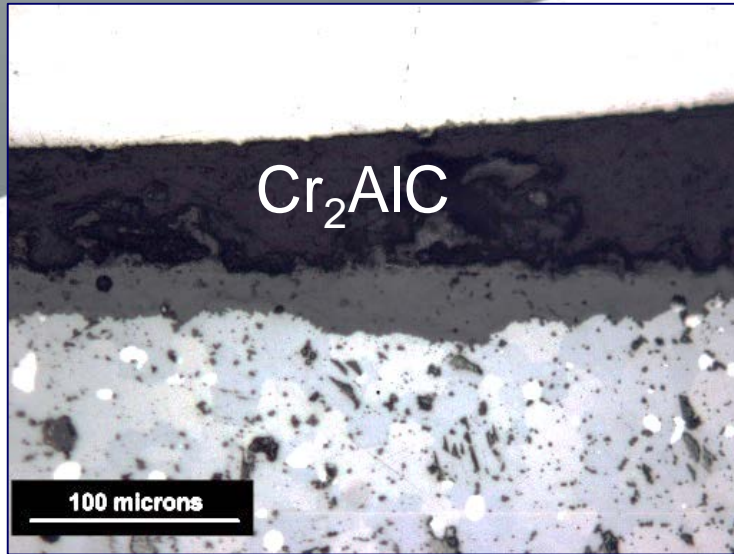
700°C, 500 h, 0.5 mg/cm<sup>2</sup> salt recoat every 50 h



# 500hrs, surface macros



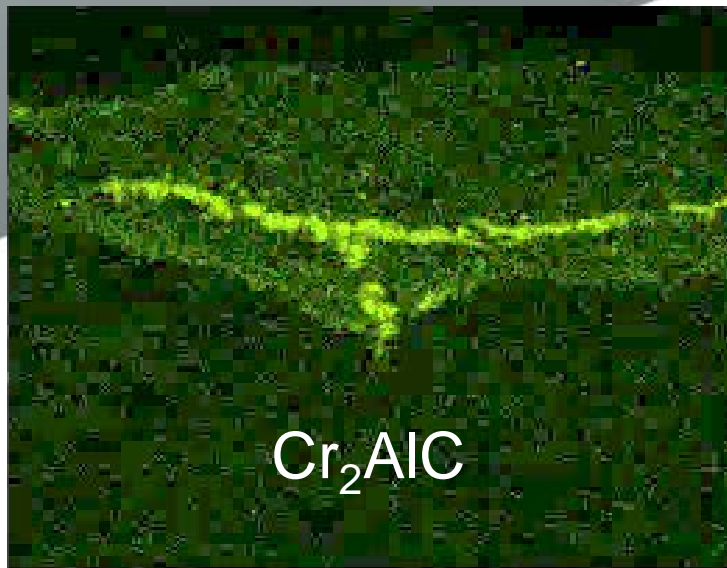
# 500hrs, section SEM





# Sulfur maps

S K $\alpha$ 1

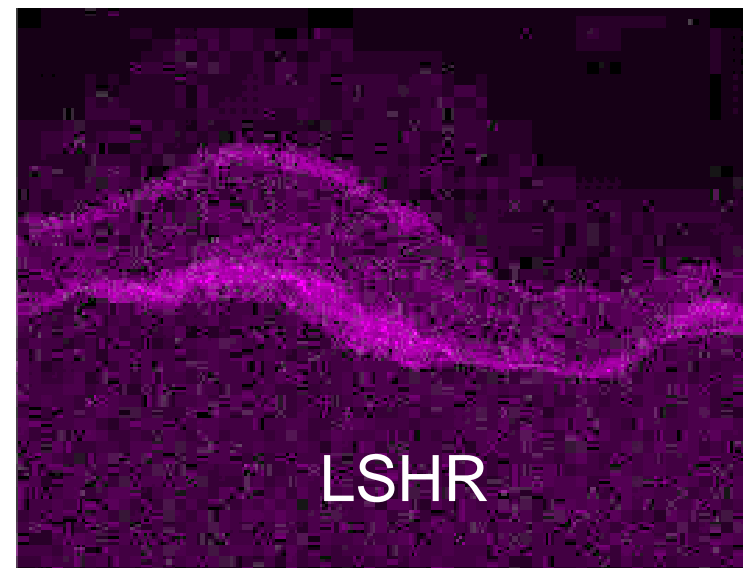


Cr<sub>2</sub>AlC

100 μm

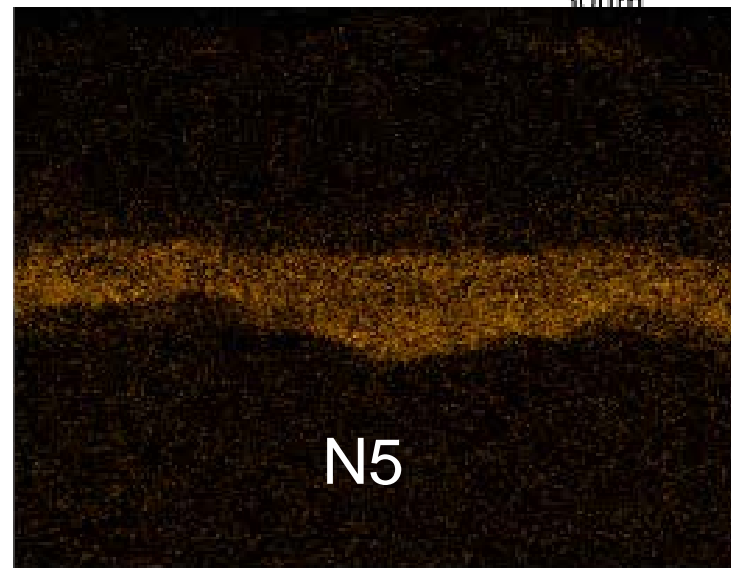
S K $\alpha$ 1

S K $\alpha$ 1



LSHR

50 μm



N5

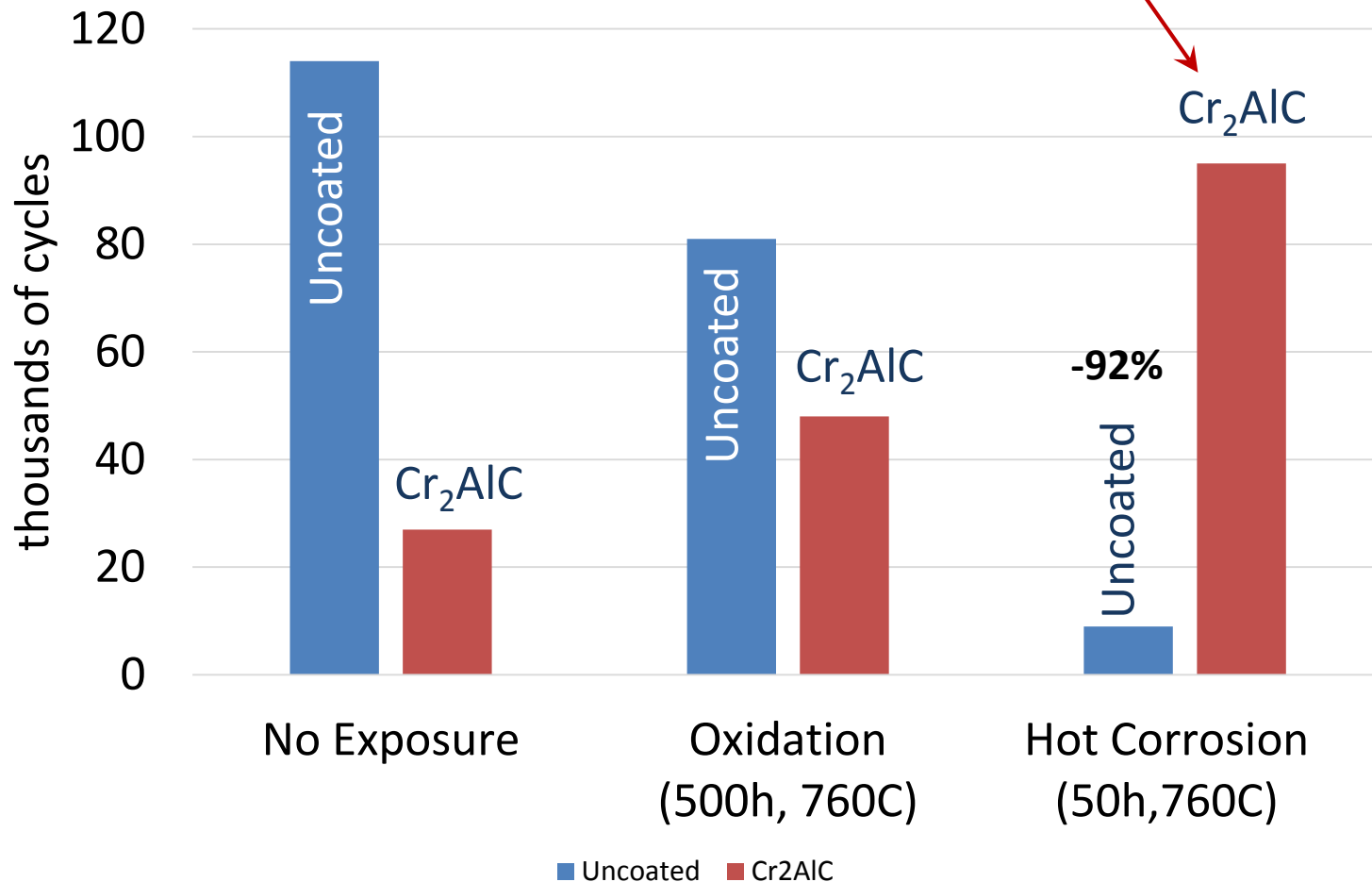
25 μm

Al, Ni, Co
Na, S, Cr

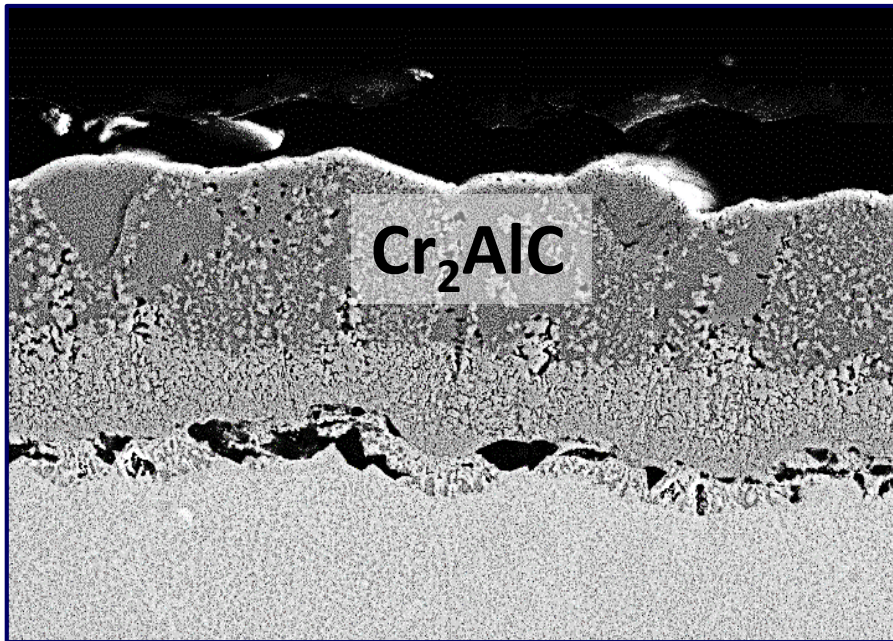
# Cr<sub>2</sub>AlC Max Phase Coatings on LSHR Prevents LTHC Pitting, LCF Debit

(Na,Mg)SO<sub>4</sub> paste; in air

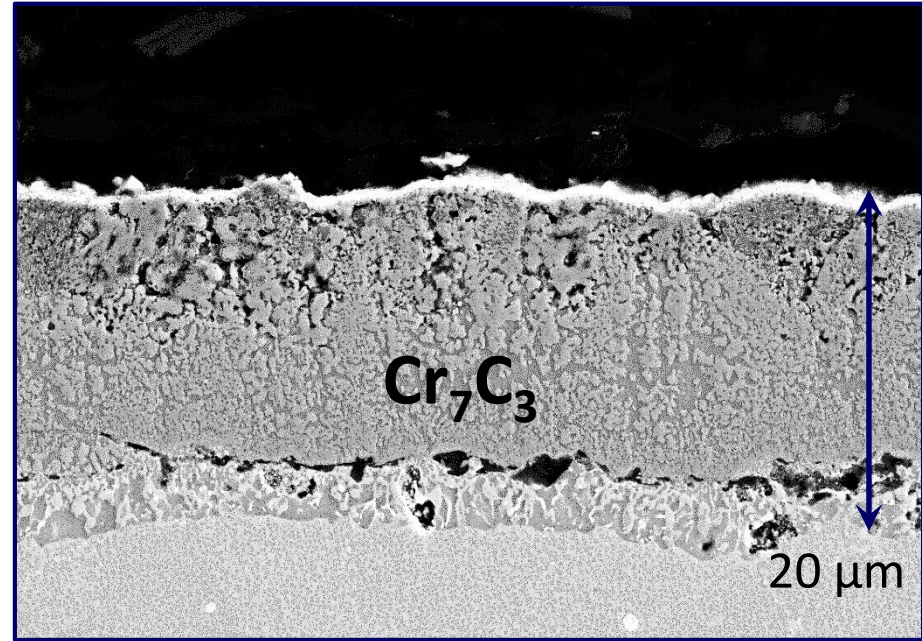
## LCF Life



# $\text{Cr}_2\text{AlC}$ Corrosion Resistance on LSHR Disk Alloy (LCF Samples)



As- processed  
heat treated  $760^\circ\text{C}$ , 8 h



$+760^\circ\text{C}$ , 500 h oxidation  
 $+ 50$  h hot corrosion

## **Ti<sub>2</sub>AlC, Cr<sub>2</sub>AlC MAX Phases: Items of Potential Interest for Turbines**

1. Rate Control by Al<sub>2</sub>O<sub>3</sub> grain boundary diffusion;  
cubic rate protective to 1400°C; Q = 334 kJ/mole
2. Ti<sub>2</sub>AlC stable in 1300°C combustion (HPBR) gas.
3. Extensive TBC furnace life @ 1300°C for Ti<sub>2</sub>AlC.
4. Superalloy / Cr<sub>2</sub>AlC hybrids stable:  
800°C, but not 1150°C.
5. Minimal 700° C Type II LTHC of Cr<sub>2</sub>AlC

### **INTRIGUING ALUMINA FORMERS**

## Ti<sub>3</sub>AlC<sub>2</sub>, Ti<sub>2</sub>AlC, Cr<sub>2</sub>AlC MAX Phase Oxidative Shortcomings

1. Ti<sub>2</sub>AlC breakaway TiO<sub>2</sub> growth:  
< critical Al content; or damaged areas
2. Cr<sub>2</sub>AlC isothermal oxidation: CrO<sub>3</sub> losses (Cr<sub>7</sub>C<sub>3</sub>)
3. Cr<sub>2</sub>AlC cyclic oxidation:  
interfacial spallation; Cr<sub>7</sub>C<sub>3</sub> depletion zones
4. Superalloy / Cr<sub>2</sub>AlC compatibility:  
interdiffusion at, delamination from 1150°C.

## Al-MAX Phases and Turbine Environments Current Engineering Assessment

substrate	MAX phase	top coat	use	advantage	max temp.
none	$Ti_2AlC$	none	gas path	protective $k_c$ , low volatility, adherent $Al_2O_3$	$\sim 1300^\circ C$
none	$Ti_2AlC$	YSZ	TBC	CTE match, adherent $Al_2O_3$	$\sim 1300^\circ C$
superalloy	$Cr_2AlC$	none	hot corrosion	higher CTE, Cr, Al	$\sim 900^\circ C$