

# Thermochemistry of CaO-MgO-Al<sub>2</sub>O<sub>3</sub>-SiO<sub>2</sub> (CMAS) and Advanced Thermal and Environmental Barrier Coating Systems

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# **Outline of Presentation**

- Thermal and Environmental Barrier Coating Systems
- Experimental
- Sample preparation and reaction with CMAS
- Results
- Thermodynamic modeling of YSZ-CMAS system
- Characterization:
- 1 Pristine NASA composition CMAS by XRD, ICP-OAS and DSC
- 2 CMAS reacted with the hollow tube coating specimens by SEM-EDS and XRD
- Summary



Thermal and Environment Barrier Coating Developments

### Baseline ZrO<sub>2</sub>-(7-8)wt%Y<sub>2</sub>O<sub>3</sub> and Rare Earth Doped-Low Conductivity Thermal Barrier Coating Systems - Continued

### Baseline ZrO<sub>2</sub>-(7-8) wt%Y<sub>2</sub>O<sub>3</sub>:

- Relatively low intrinsic thermal conductivity ~2.5 W/m-K
- High thermal expansion to better match superalloy substrates
- Good high temperature stability and mechanical properties
- Additional conductivity reduction by micro-porosity

### Low Conductivity Defect Cluster Thermal Barrier Coatings

Multi-component oxide defect clustering approach

e.g.:  $ZrO_2/HfO_2-Y_2O_3-Nd_2O_3(Gd_2O_3,Sm_2O_3)-Yb_2O_3(Sc_2O_3)$  systems

Serial Stabilizer Primary stabilizer

Oxide cluster dopants with distinctive ionic sizes

- Defect clusters associated with dopant segregation
- The 5 to 100 nm size defect clusters for significantly reduced thermal conductivity (0.5-1.2 W/m-K) and improved stability
- Advanced TEBC systems for Ceramic Matrix Composites use the low k based compositions

TEBCs-CMAS Degradation is of Concern with Increasing Operating Temperatures







Plasma-sprayed ZrO<sub>2</sub>-(Y, Nd,Yb)<sub>2</sub>O<sub>3</sub>

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### Experimental:sample preparation and heat treatment



- Air plasma sprayed coating (0.030" thickness) specimens on to 1/8" diameter graphite bar substrates then 1500 °C, 5 h sintering, resulting hollow tubes.
- NASA composition CMAS used for reaction at 1300 ° C for 5h.



\*(ρgeometric\*100/ρHe). \*\*ρgeometric-ρHe.

(1:10 CMAS to sample mass ratio, concentration of 70-150 mg/cm<sup>2</sup>)



(A) (B) (C) Hollow 12YSZ tube samples: (A) pristine; (B) before heat treatment in which it was half filled with CMAS powder, wrapped and sealed with Pt foil; (C) after heat treatment at 1310 °C for 30 min and unwrapped.

### Results: characterization of NASA composition CMAS (as processed) before reaction







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### Results: SEM cross-section images at low magnification (lower cut section)







ZrO<sub>2</sub>-9.0Y<sub>2</sub>O<sub>3</sub>-4.5Gd<sub>2</sub>O<sub>3</sub>-4.5Yb<sub>2</sub>O<sub>3</sub> ZrO<sub>2</sub>-9.6Y<sub>2</sub>O<sub>3</sub>-2.2Gd<sub>2</sub>O<sub>3</sub>-2.1Yb<sub>2</sub>O<sub>3</sub>

 $ZrO_2 \text{--} 3.0Y_2O_3 \text{--} 1.5Sm_2O_3 \text{--} 1.5Yb_2O_3$ 

 $ZrO_2$ -3.0 $Y_2O_3$ -1.5Nd $_2O_3$ -1.5Yb $_2O_3$ -0.3Sc $_2O_3$ 

SEM cross – sectional electron images of the lower section of the ceramic hollow tube samples reacted with CMAS at 1300  $^{\circ}$ C for 5 h.

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### Results: 12YSZ lower section of the hollow tube reacted with CMAS.



SEM image of (reacted region) at high magnification.





Elemental content from EDS.

### National Aeronautics and Space Administration Results: 18YSZ lower section of the hollow tube reacted with CMAS.





#### National Aeronautics and Space Administration Results: 7DySH lower section of the hollow tube reacted with CMAS.



SEM image at high magnification.

200 -









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Results Rare Earth Content *versus* apatite phase formation.





 $ZrO_2$ -18RE<sub>2</sub>O<sub>3</sub> (RE = Y, Gd and Yb)

## $ZrO_{2}-18Y_{2}O_{3}$

 $ZrO_2$ -13.9RE<sub>2</sub>O<sub>3</sub> (RE = Y, Gd and Yb)

 $ZrO_{2}-12Y_{2}O_{3}$ 

 $HfO_{2}-6.3Dy_{2}O_{3}$ 

 $ZrO_2$ -6.3RE<sub>2</sub>O<sub>3</sub> (RE = Y, Nd, Yb and Sc)

 $ZrO_2$ -6.0RE<sub>2</sub>O<sub>3</sub> (RE = Y, Sm and Yb)

XRD patterns of the ground hollow tubes reacted with CMAS at 1310 °C for 5 h (lower cut section).

### Results: content of the Rare-earth in the glass/silicate phase.





Depedence of the Rare-earth content in the glass/silicate phase versus Rare-earth content in the coating.

### Results: content of the Rare-earth in the glass/silicate phase.





ZrO<sub>2</sub>-3.0Y<sub>2</sub>O<sub>3</sub>-1.5Sm<sub>2</sub>O<sub>3</sub>-1.5Yb<sub>2</sub>O<sub>3</sub>

2.5 -

2.0

(%) Mole (%) 1.5

0.5

0.0

ZrO<sub>2</sub>-9.6Y<sub>2</sub>O<sub>3</sub>-2.2Gd<sub>2</sub>O<sub>3</sub>-2.1Yb<sub>2</sub>O<sub>3</sub>

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# Summary

- Thermochemical reactions between CMAS and EBC and TBC materials were studied at 1310  $^{\circ}\mathrm{C}$  for 5h.
- CMAS penetrated the samples at the grain boundaries and dissolved the EBC/TBC material to form silicate glassy and orthosilicate crystalline phases containing the rare-earth elements.
- Apatite crystalline phase was formed in the samples with rare-earth content higher than 12 mole (%) total of Rare-earths in the reaction zone.
- 18YSZ, 7DySH and  $ZrO_2$ -9.5 $Y_2O_3$ -2.2Gd<sub>2</sub>O<sub>3</sub>-2.1 $Yb_2O_3$  samples have lower reactivity or more resistance to CMAS than the other coating compositions of this work.

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