Effect of Pre- and Post-Coat Processing on the Fatigue Life of Coated Disk Alloys

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SOA turbine disks can operate up to ~704°C (1300°F) peak rim temperatures



One means of achieving future fuel burn and emission goals is to increase engine and disk temperatures.

As a result, NASA has goals:

- (1) to develop a hybrid polycrystal/single crystal turbine disk to operate at higher temperatures (up to 815°C) and,
- (2) protect those disks from oxidation and hot corrosion attack while maintaining low cycle fatigue (LCF) life.

Oxidation at disk temperatures of interest reduces the fatigue life of disk alloys



"...longer exposures near 704°C or higher temperature exposures can substantially degrade fatigue lives..."

Ref: Gabb, T. P., Sudbrack, C. K., Draper, S. L., MacKay, R. A., and Telesman, J., "Effects of Long Term Exposures on Fatigue of PM Disk Superalloys," *Materials Performance and Characterization*, Vol. 3, No. 2, 2014, pp. 44-67

Hot corrosion at disk temperatures of interest reduces the fatigue life of disk alloys



Ref: Gabb, et al., "The Effects of Hot Corrosion Pits on the Fatigue Resistance of a Disk Superalloy," Journal of Materials Engineering and Performance, Vol. 19, 77, 2010.



Hence, <u>conventional disks</u> operating at SOA temperatures and above (700-760°C), as well as future <u>hybrid disks</u> operating at higher temperatures, <u>will require protective coatings.</u>

Since disks are designed to resist low cycle fatigue cracking, which is dependent on crack initiation, the coating must not enhance crack initiation and degrade LCF life. Last year, results on two similar advanced disk alloys were presented

LSHR (<u>L</u>ow <u>S</u>olvus, <u>H</u>igh <u>R</u>efractory)

• ME3

| Weight | Percent |
|--------|---------|
|--------|---------|

| | Ni | Со | Cr | Al | Ti | W | Nb | Мо | Та | Zr | В | С |
|------|-------|------|------|------|------|------|------|------|------|------|------|------|
| LSHR | 50.14 | 20.4 | 12.3 | 3.49 | 3.48 | 4.24 | 1.51 | 2.72 | 1.59 | 0.05 | 0.03 | 0.05 |
| ME3 | 50.48 | 20.6 | 13.0 | 3.23 | 3.59 | 1.97 | 0.89 | 3.73 | 2.38 | 0.05 | 0.02 | 0.06 |









Both large ME3 bars and small threaded LSHR bars had the same polish in the reduced gauge section.

Coating texture follows longitudinal polishing marks on surface, creates longitudinal "cracks" or gaps.



Coatings applied at SwRI, San Antonio, TX using plasma enhanced magnetron sputtering (PEMS)

Fatigue Life (No Shot Peening, No OX, No HC)



Fatigue life increases with Cr content of coating (Ni-45Cr best). All coated bars have lower LCF life than uncoated bars.

Next follow-on study*

Grit blasting (alumina grit at 85 psi) of the LP surface to eliminate longitudinal polishing marks. Roughened surface was coated using High Power Impulse Magnetron Sputtering (HIPIMS) to deposit a Ni-45Cr-0.1Y coating

DOE Study involving 32 bars

Examine effect of:

Coated vs Uncoated

Low and high shot peening (4N-100% vs 16N-200%) Diffusion anneal in air vs low PO_2 (10^{-17} atm O_2), 8h/760°C With and without oxidation exposure (500h/760°C) With and without hot corrosion exposure (50h/760°C with salt)

* Gabb, et al, The Effectiveness of a NiCrY-Coating on a Powder Metallurgy Disk Superalloy NASA/TM—2018-219885

Grit blasting of the LSG+LP surface (alumina grit at 85 psi) eliminated the longitudinal polishing marks and uniformly roughened the surface.







BSE

Longitudinal polishing lines eliminated.

Roughened surface was coated using HIPIMS with a Ni-45Cr-0.1Y coating.*

* Coatings applied at SwRI, San Antonio, TX.

Alumina grit embedded in substrate



BSE

Need to eliminate polishing marks but selected grit blast was too severe (grit embedded in surface)

What is the ideal pre-coat surface treatment for this application?

Purpose:

Examine pre-coat processing to maximize low cycle fatigue (LCF) life of coated disk alloys

Approach:

Examine four <u>pre-coat processing treatments</u> on the LCF life of a Ni-45Cr-0.1Y coated disk alloy

- (1) Highly polished
- (2) Wetblast
- (3) Shot-peening at 8N-200%
- (4) Shot-peening at 16N-200%

Standardized <u>post-coat processing</u> (identified in the previous study as beneficial: shot peening, diffusion anneal)

Standardized LCF Testing protocol:

- Coated versus uncoated (minimum of 2 bars)
- With and without environmental exposures
 - oxidation (OX)
 - hot corrosion (HC)

- <u>Apply pre-coat treatments</u> (Starting condition of all bars was LSG + LP (longitudinal polish)
 <u>A. Highly polished at SWRI, San Antonio, TX</u>
 - B. Wetblast at Wetblast.com, Franklin, WI (80 psi, 4" from piece, 15% 300 glass bead)
 - C. Shot peening at Metal Improvement Co., Cincinnati, OH
 - 1. 8N-200% (AMS 2432 using conditioned cut
 - 2. 16N-200% stainless steel wire (CCW14))
- 2. Characterize surfaces Surface roughness (Zygo 7200), SEM
- 3. Coating: Ni-45Cr-0.1Y applied at SWRI, San Antonio, TX (HIPIMS)
- 4. Post-coat shot peen the surface (16N-200%) (Metal Improvement)
- 5. Perform low PO₂ diffusion anneal (8 hrs, 760°C, PO₂ of 10⁻¹⁷ atm O₂)
 Flow UHP Ar over Ni foils to diffusion bond coating and substrate and promotes protective Cr₂O₃ formation

- 1. Environmental exposures (half of coated bars)
 - A. Oxidation exposure (500 hrs, 760°C (1400°F) in air)
 - B. Hot corrosion exposure (2 mg/cm² eutectic 72% Na₂SO₄-28% MgSO₄ salt, 50 hrs, 760°C (1400°F) in static air), sonic water clean
- 2. LCF Testing (760°C, 1400°F)
 - A. 841/-427 Mpa, 0.33 hertz

(Same LCF test parameters for previous tests)



Internal initiated crack



Surface initiated crack

One or multiple crack initiation sites





LCF bars after surface treatment

Highly polished

Wet-blast



Shot peened, 8N-200%

Shot peened, 16N-200%



Surface Roughness after Surface Treatments









Surface Roughness after Coating



<u>Highly-polished</u> surface after coating





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<u>Wetblast</u> surface after coating







Shot-peened surface (16N-200%) after coating



Ni45Cr-0.1Y Coating "as Coated"

Coating thickness: 11-14 um (11.4-14.3 um)















Coated Surface (HP) "as-Shot Peened"



Coated Surface (HP) "as-Shot Peened"







Coated Surface "as-Shot Peened"



Coated Surface (SP16N) "as-Shot Peened"



Coated Surface (SP16N) "as-Shot Peened"









Post-Shot Peen





Diffusion anneal, 8h/760°C, Low PO₂ (10⁻¹⁷ atm O₂)







Very thin scale of Cr₂O₃

Diffusion anneal, 8h/760°C, Low PO₂ (10⁻¹⁷ atm O₂)





Comparison of <u>Unexposed Bars</u>, Uncoated and Coated





Comparison of all conditions



Where did the failure crack initiate?





Comparison of Unexposed Bars, Uncoated and Coated 160,000 Uncoated, Unexposed Coated, Unexposed 140,000 L S S 120,000 S Fatigue Life-cycles 100,000 Average (83,347 cycles) 80,000 S 60,000 40,000 20,000 0 Longitudinal polish (LP) **Highly polished** Wet blast **Shot Peen Shot Peen** 8N-200% 16N-200%

Comparison of Exposed Bars, Uncoated and Coated



Post LCF Testing



Stress at and below the surface*



 * T. Gabb et al., "Influences of Processing and Fatigue Cycling on Residual Stresses in a NiCrY-Coated Powder Metallurgy Disk Superalloy," Journal of Materials Engineering and Performance, 2017

Conclusions:

- Without exposures, the coating did not degrade the LCF life of the bars.
- After oxidation and hot corrosion, the LCF lives of the coated bars was 1.7-2X higher than that for the uncoated bars.
- After oxidation and hot corrosion, the primary crack leading to failure always initiated at the surface whether coated or uncoated.
- There was not an obvious "winner" between the four pre-coat treatments; wetblast or shot peen could likely both be adopted for use with disks

Future Directions:

- Explore effect of high and low Cr coating compositions.
- Explore stronger, more crack-resistant coatings with sufficient Cr to maintain corrosion resistance.

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