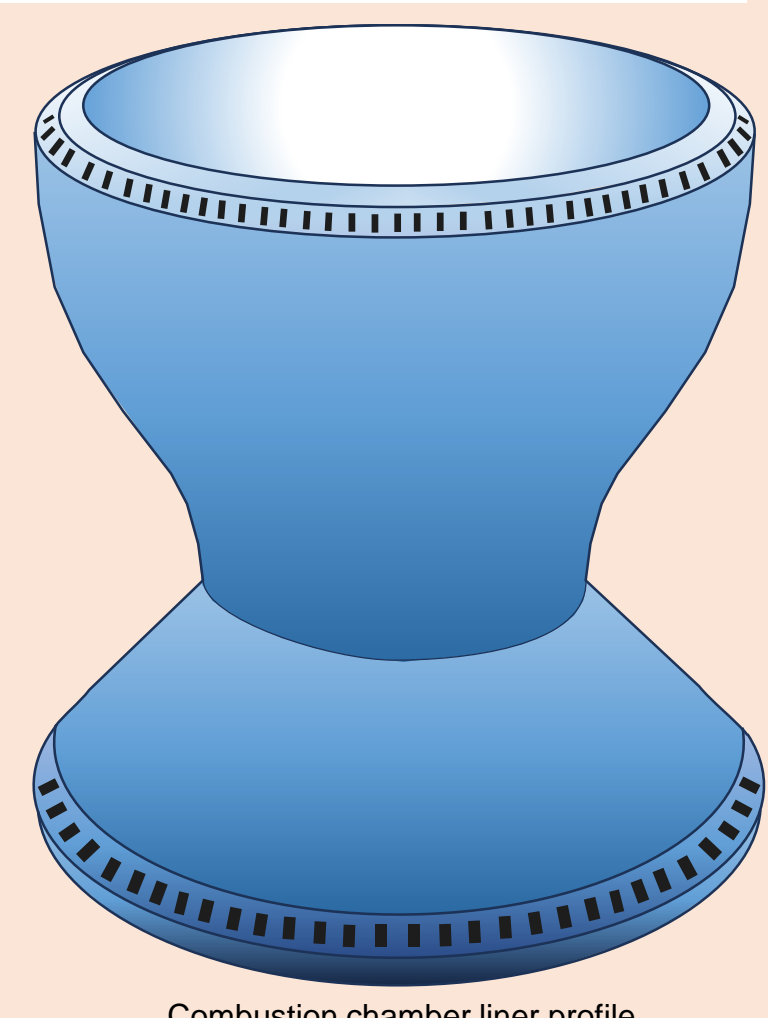


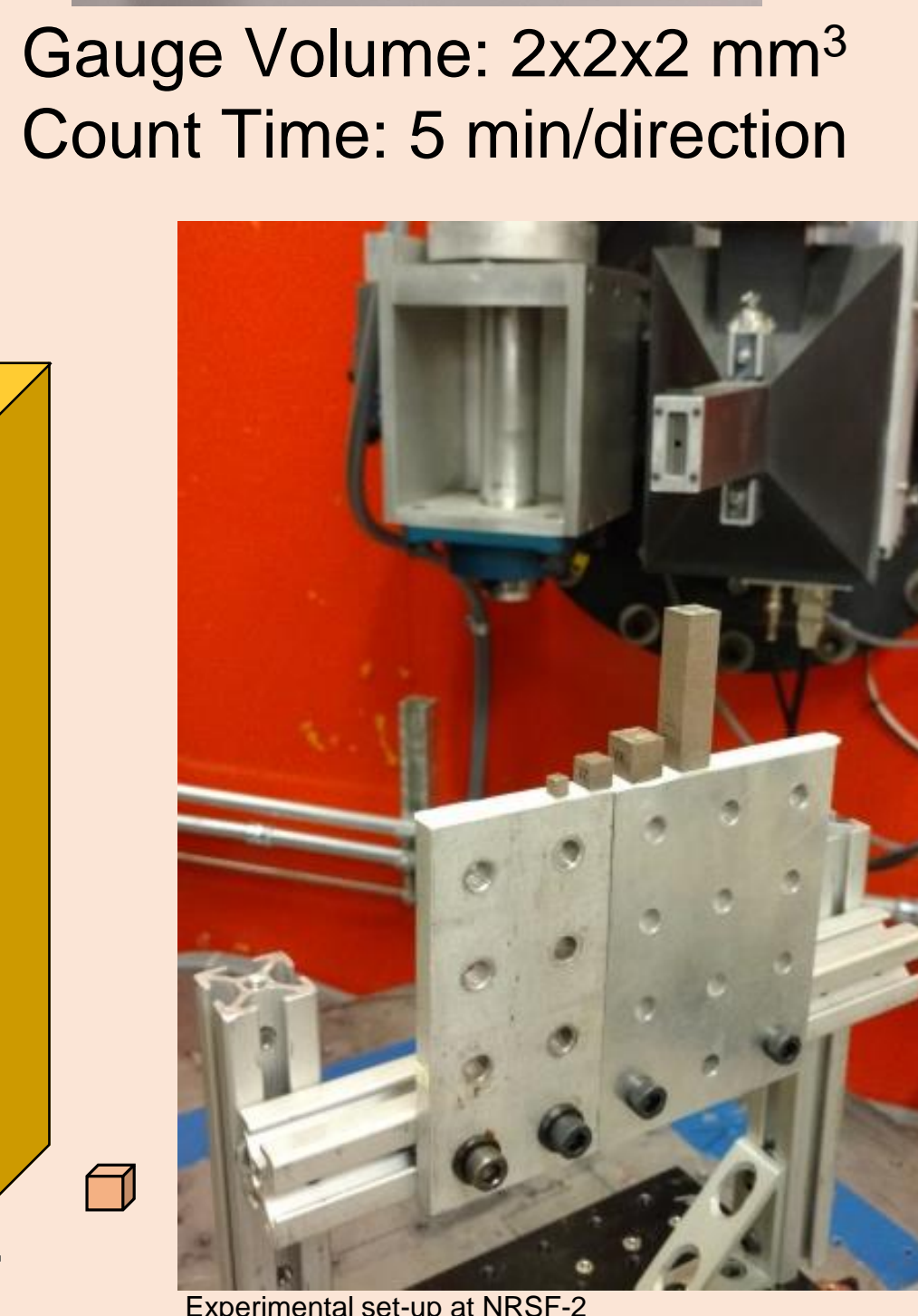
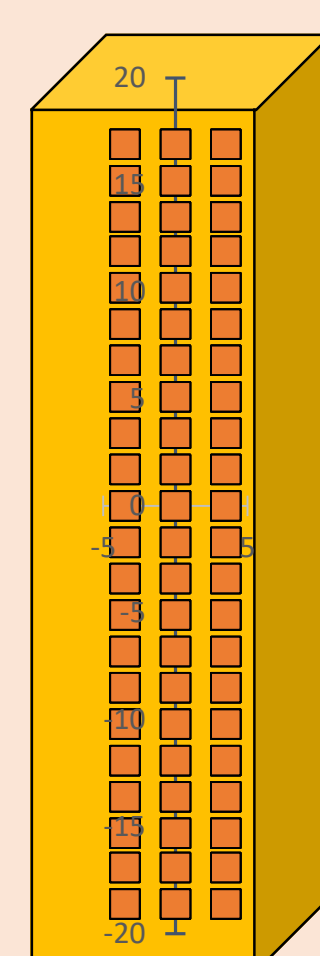
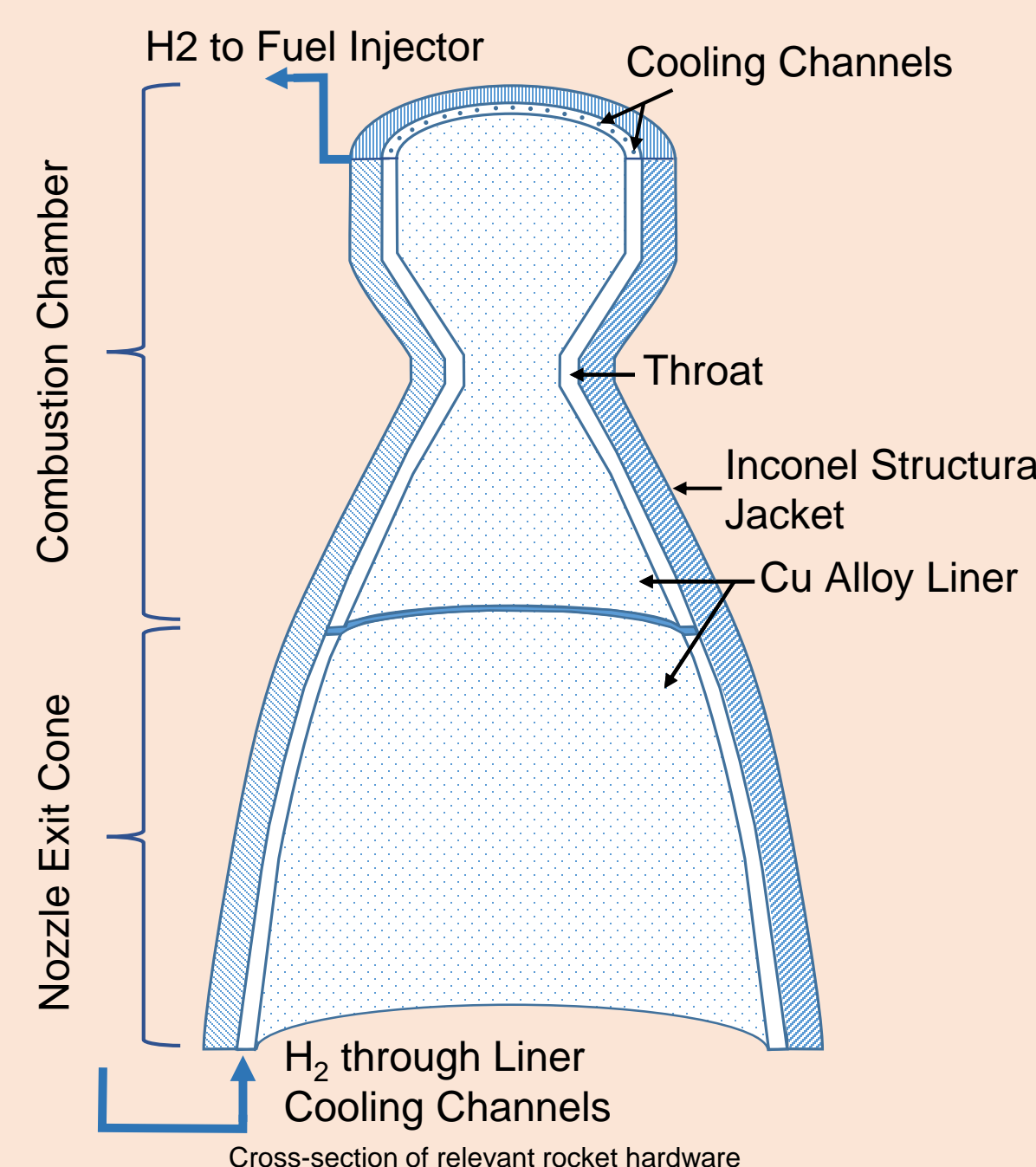
Background

GRCop-84 is a precipitation strengthened alloy composed of Cu-8 Cr-4 Nb at% with Cr₂Nb precipitates that provide dispersion and precipitation strengthening characteristics and limited solubility in the Cu matrix. The particle role of Cr₂Nb is unusual only contributing 1/3 of strengthening at high temperatures while the matrix provides the remainder. The particles mechanically and thermally stabilize the matrix retaining purity, preventing coarsening, and loss of strength. At high temperatures (50-85% T_{m,Cu}), GRCop-84 provides the best thermal and mechanical properties of available alloys.

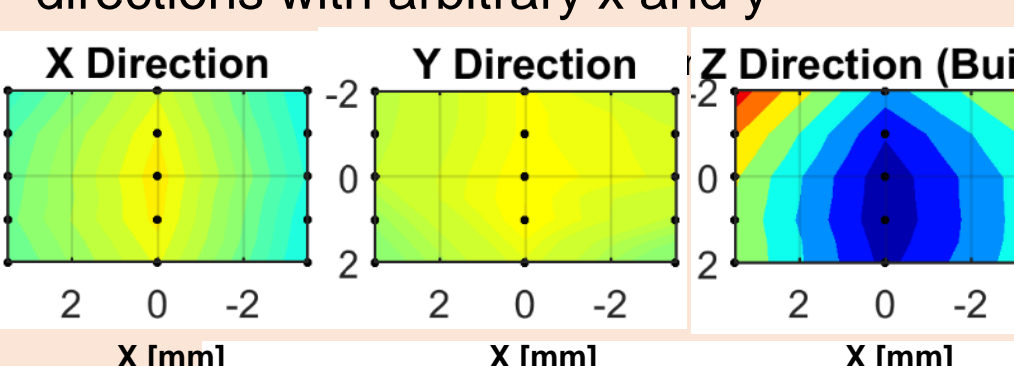
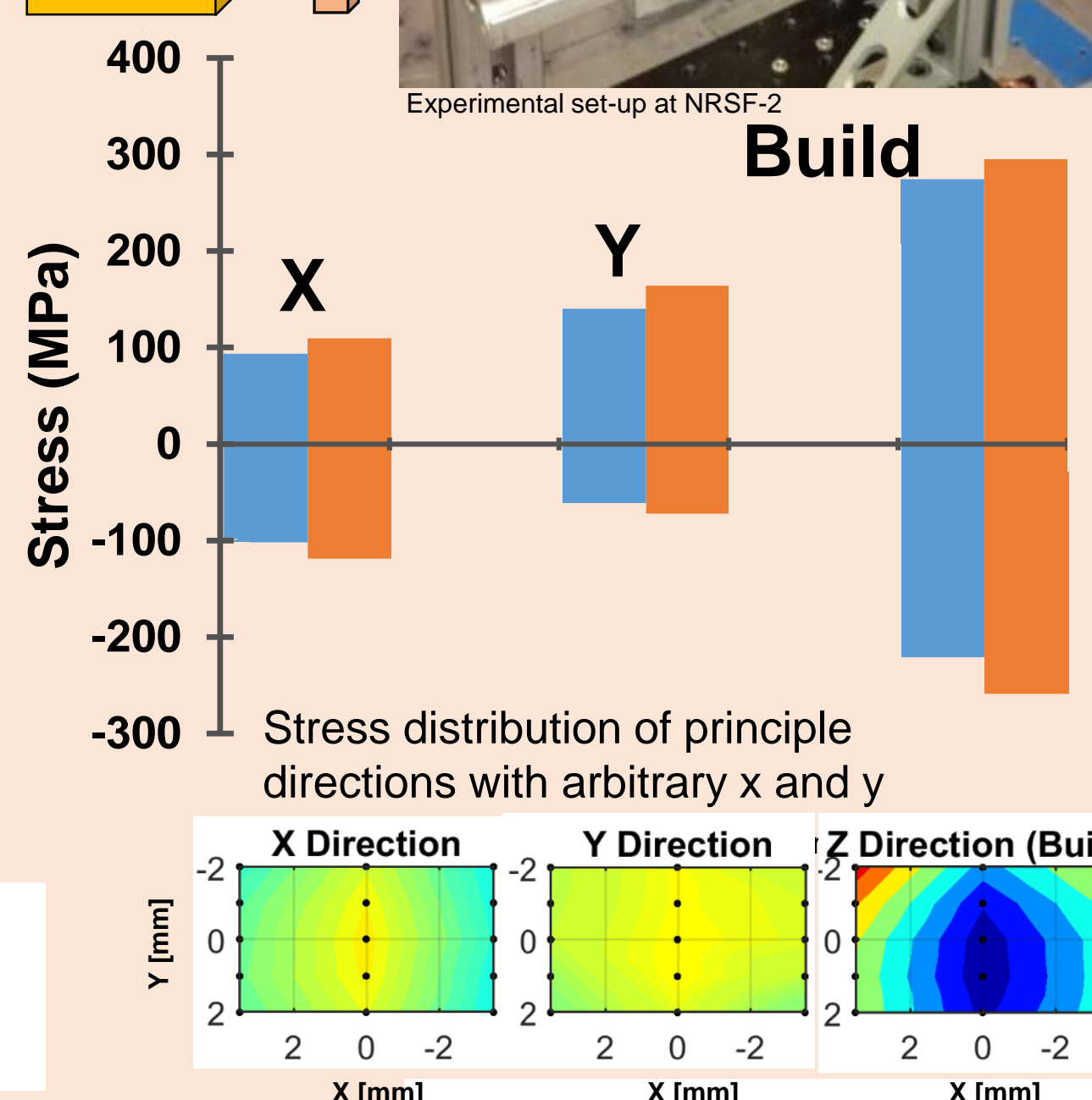


Combustion chamber liner profile

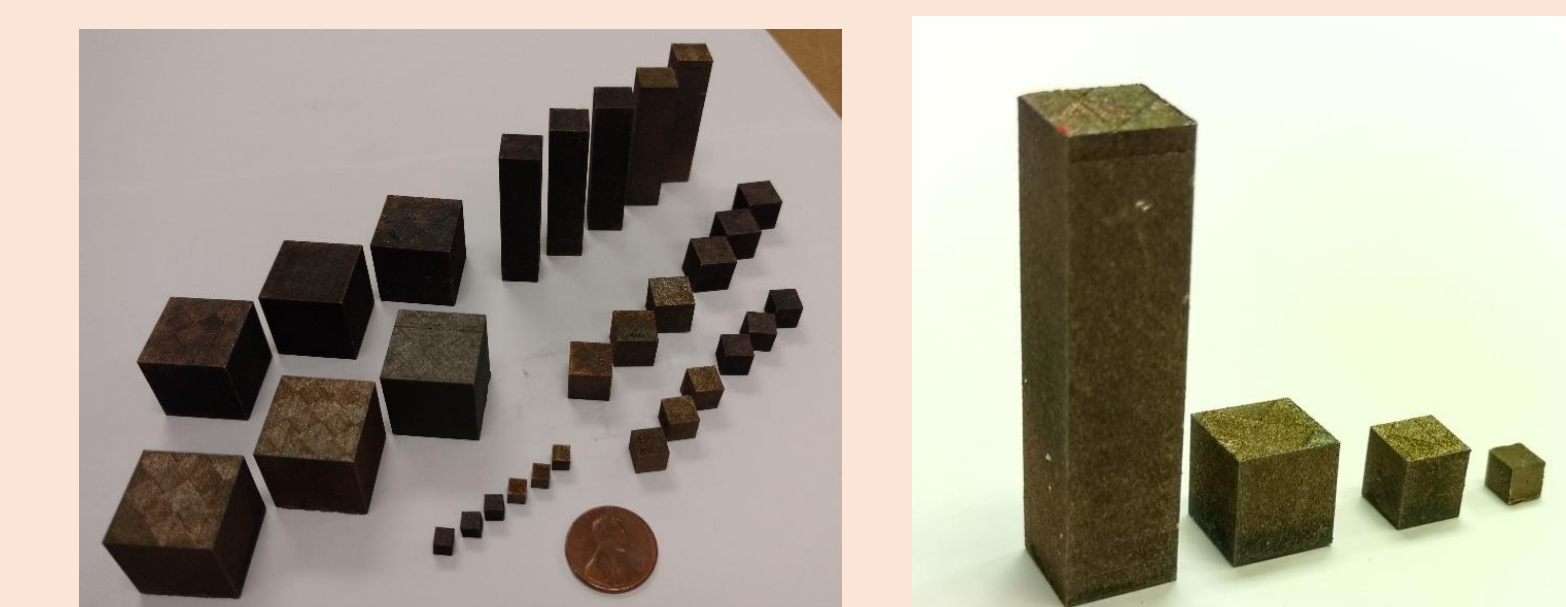
GRCop-84 is currently developed for reusable launch vehicles, including the Space Launch System (SLS), with a focus on fabrication via additive manufacturing (AM) techniques. GRCop-84 is an optimal material for consolidating with AM. The base material is costly, the production times are long, and geometry control can considerably improve cooling efficiency. Development of AM GRCop-84 with selective laser melting (SLM) has rapidly progressed due to ease of printing and limited operator adjustment between builds, but the necessary knowledge-base of thermal history and stress state during consolidation is still under development. During typical SLM, high thermal energy transferred by the laser develops into thermal strain between volumes cooling at different rates. If the stress exceeds yield, the part plastically deforms. The success of a build is often limited by the final cooling phase of the system after a part has been fully formed and before annealing. Residual thermal strain after heat treatments can interfere with additional fabrication or end properties, so a thorough understanding of the development of stress is vital future progress of building functional hardware with GRCop-84.



Build



Strain Mapping Data Collected at HFIR's NRSF-2 HB2B beamline

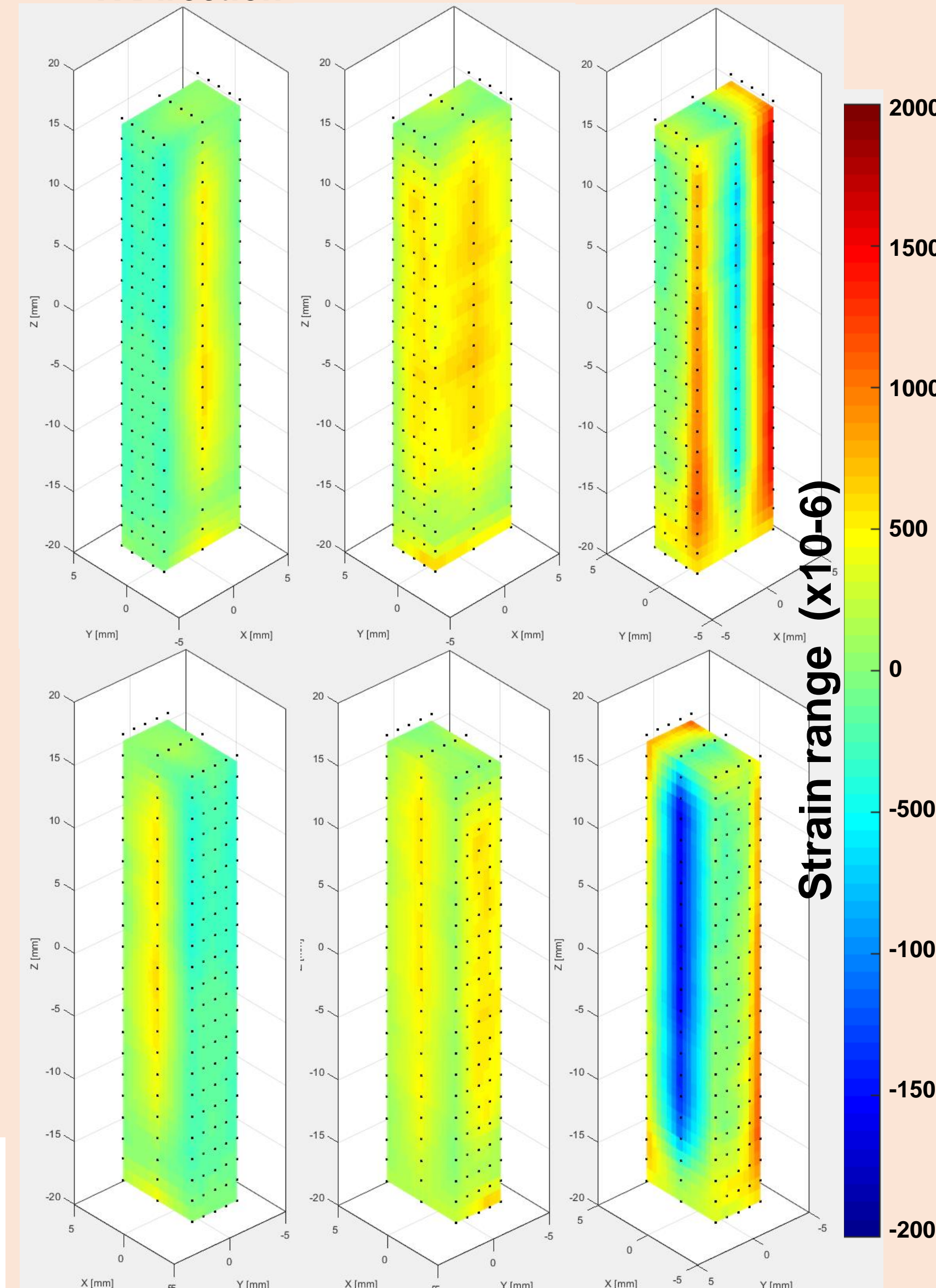


Gauge Volume: 2x2x2 mm³
Count Time: 5 min/direction

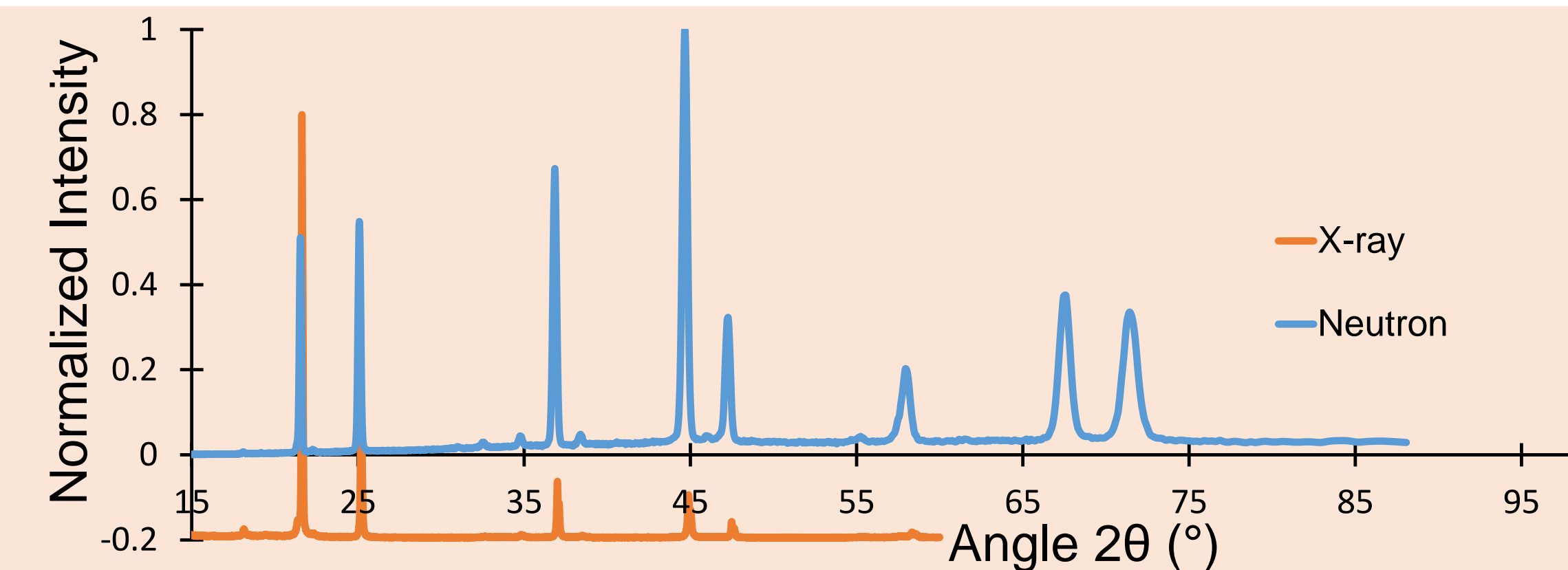
Sample Types

- **Cubes**
 - 4x4x4 mm³
 - 7x7x7 mm³
 - 10x10x10 mm³
- **Pillars**
 - 10x10x40 mm³

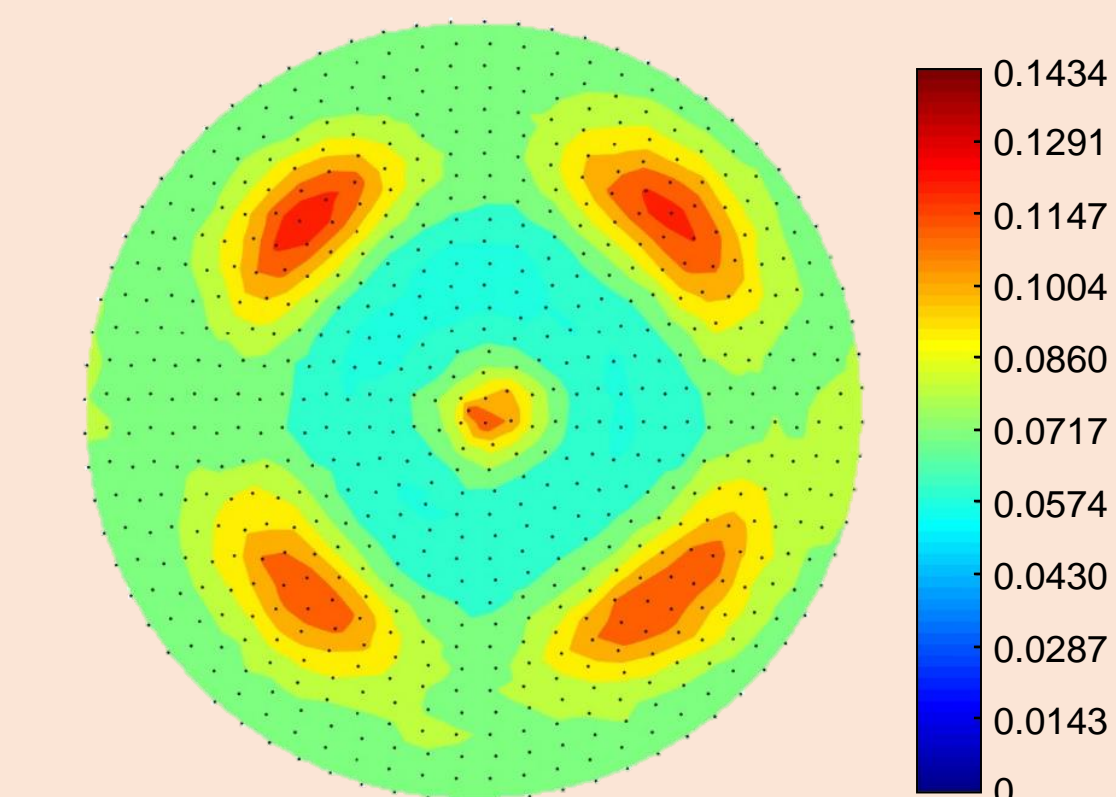
As-Built Pillar Strain (rotated views)



Neutron Diffraction



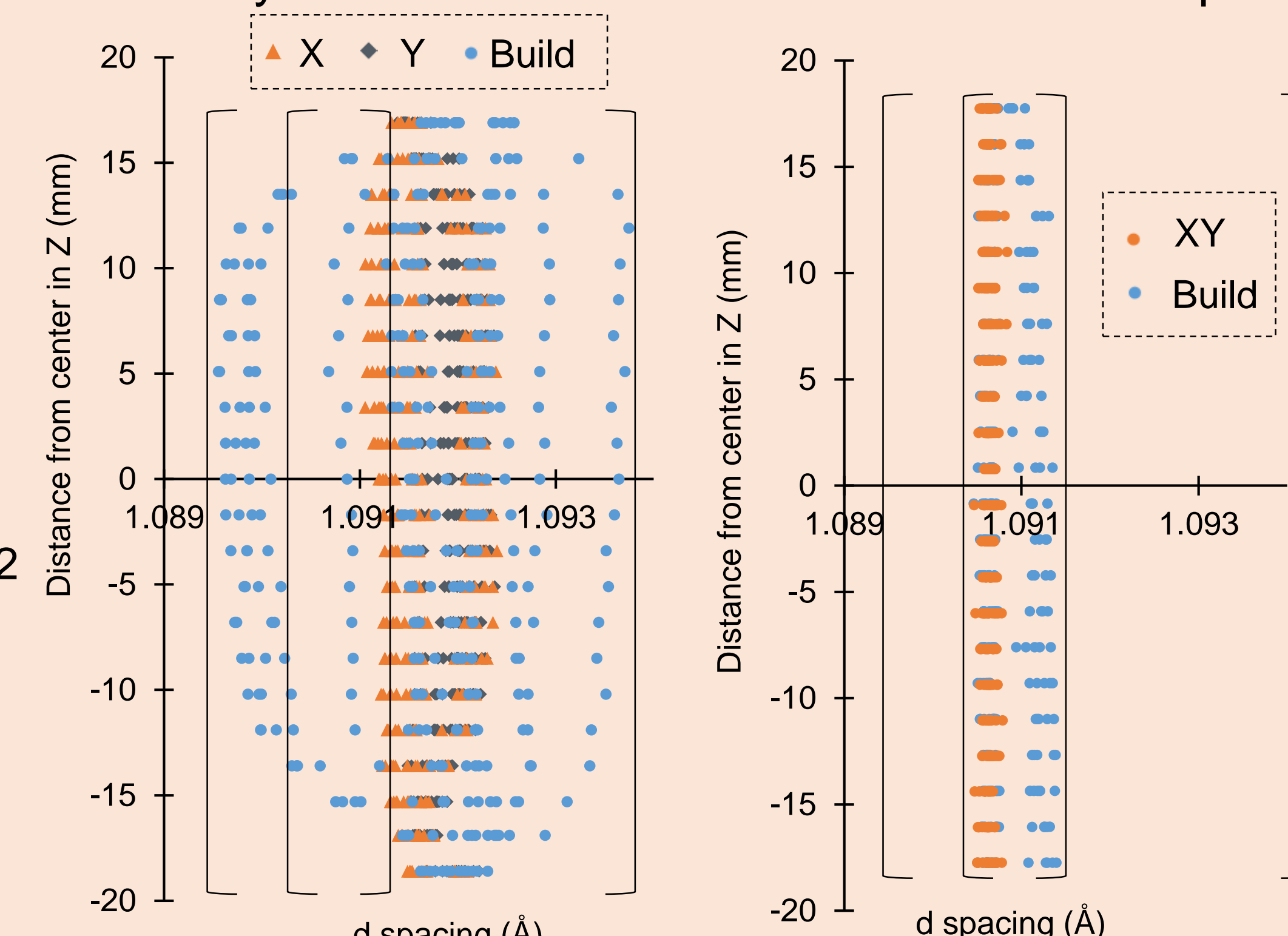
Normalized X-ray and neutron diffraction data of GRCop-84 with XRD offset by for visualization



Texture Data (220) as-built GRCop-84

Effects of HIP

- High levels of strain present in as-built parts exceeding nominal yield strength limits
- HIP envelope of strain compared to as-built is greatly reduced
- HIP visibly introduces a reduction to the overall d-spacing



As-built Strain Distribution

HIP Strain Distribution

Motivation

Neutron diffraction provides an accurate non-destructive method of quantifying stresses in the volume of a part through the highly penetrating nature of neutrons. Specialized instruments like ORNL's NRSF-2 or VULCAN can be utilized to characterize and map the stresses generated in the AM process by measuring the interplanar atomic spacing of a single reflection or full diffraction pattern. Stress can be calculated from interplanar strain.

- Type-1 – Macro-stress over several grains
- Type-2 – Micro-stress developed within one grain
- Type-3 – Sub-micro stress over several atomic distances

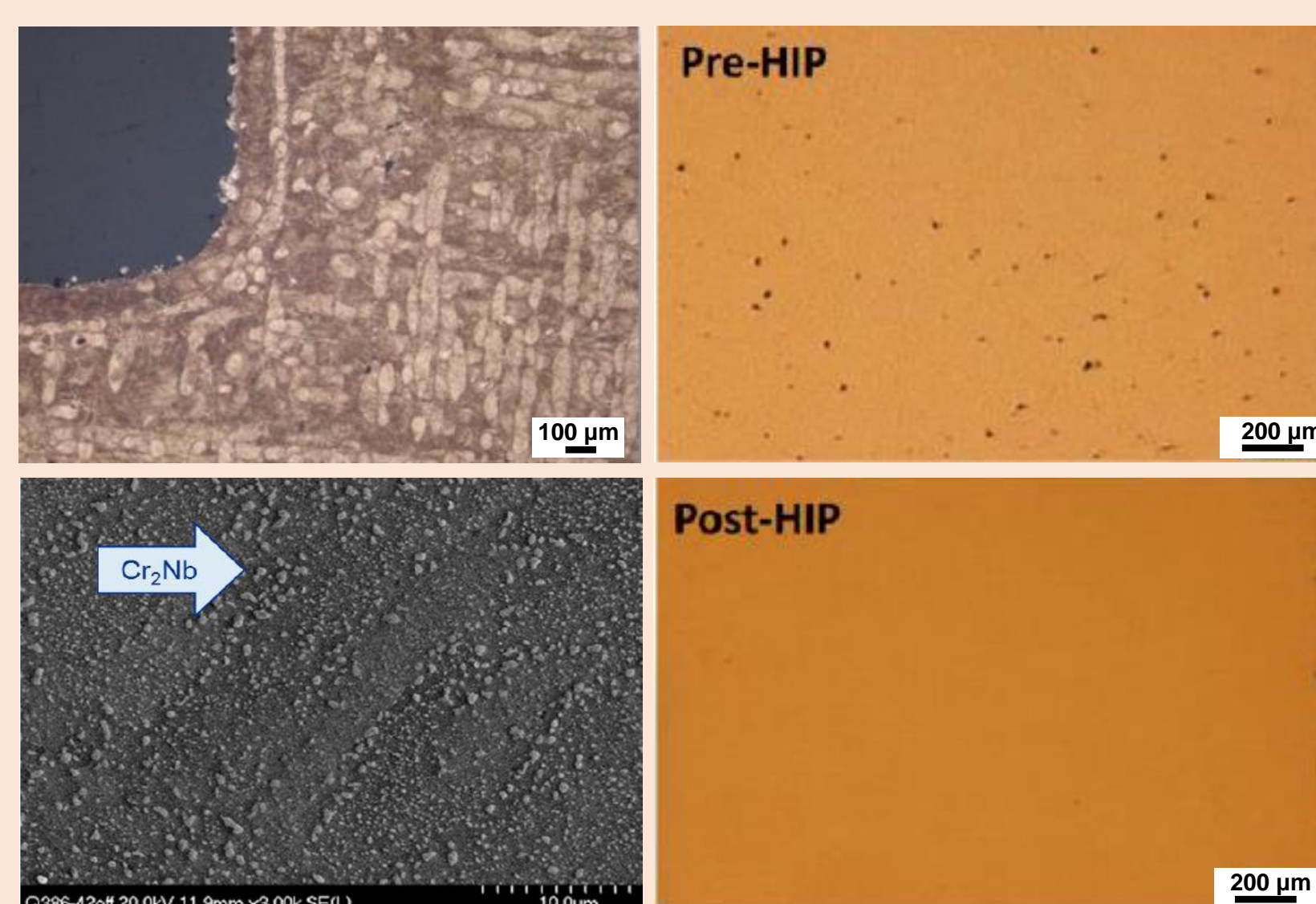
$$\epsilon_{hkl} = \frac{d_{hkl} - d_{hkl}^0}{d_{hkl}^0}$$

Stress/Strain Estimates

- Young's modulus, E, 111-130 GPa
- Poisson's ratio, ν , of 0.29
- d_0 of 1.09152 Å

$$\sigma_{ij} = \frac{E}{1 + \nu} \left(\epsilon_{ij} + \frac{\nu}{1 - 2\nu} (\epsilon_{11}^{hkl} + \epsilon_{22}^{hkl} + \epsilon_{33}^{hkl}) \right)$$

Microstructure Evolution



Images reproduced from: Gradl, Paul R., et al. [1]

- **Electrical Discharge Machining (EDM)**
 - Removes part from build plate
 - Relieves stress at interface
 - Excessive stress can result in deformation
- **Hot Isostatic Pressing (HIP)**
 - Parts are annealed prior to HIP
 - HIP is conducted well above annealing temperature and high pressure
 - Reduces porosity
 - Homogenizes microstructure
 - Parts are assumed to be fully stress relieved



Conclusions

Strain maps of pillar sample indicate large compressive strain in the center balanced by tensile strains towards the edges in the build direction, which is typical of metal expansion and cooling. Literature values and a recently determined elastic modulus obtained from compression tests are higher than originally anticipated, and using this higher modulus results in pillar stresses that would substantially exceed the yield stress. The material has not yielded despite the strain, suggesting that the microstructure might be responsible for this behavior.

Previously, it was assumed that the HIP process effectively eliminates the residual strain within the parts, but significant, albeit small, stresses remain. Additionally, the HIP process appears to interfere further with the microstructure as the d_0 and even elastic modulus seem to change slightly. While the elastic modulus may be statistical variance, the d_0 has been consistently reduced between samples.

Future work includes obtaining an accurate elastic modulus for principle build directions to further refine stress values and investigate modulus change as a function of processing. Metallography, EBSD, SEM, are planned for samples exposed to the beam to provide direct supplemental microstructure information.

Acknowledgements

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