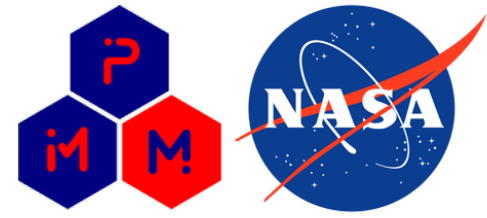


# PuMA and Multiscale Modeling

Presenters: Federico Semeraro  
Krishnan S. Gopalan

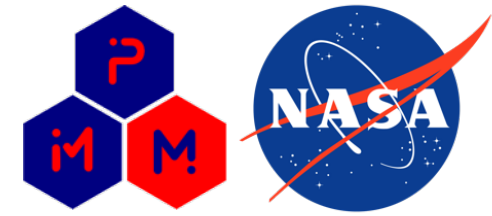
Project contributors: Joseph C. Ferguson, Marcos Acin,  
Arnaud Borner, John Thornton  
Francesco Panerai, Nagi N. Mansour

# Content



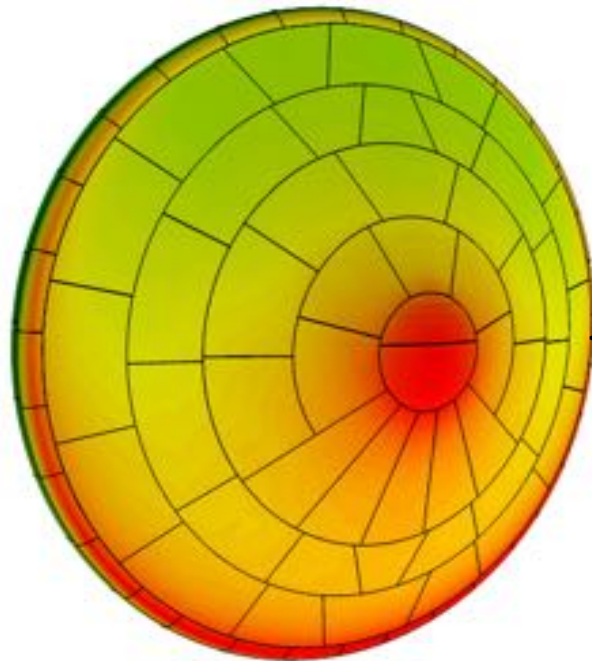
- Motivation
- Overview of PuMA
- Effective properties for fibrous media
  - ❖ Fiber orientation
  - ❖ Thermal conductivity
  - ❖ Elasticity
- Surface chemistry
  - ❖ Molecular beam
  - ❖ Oxidation model

# Modeling Thermal Protection Systems



## Macroscale Modeling

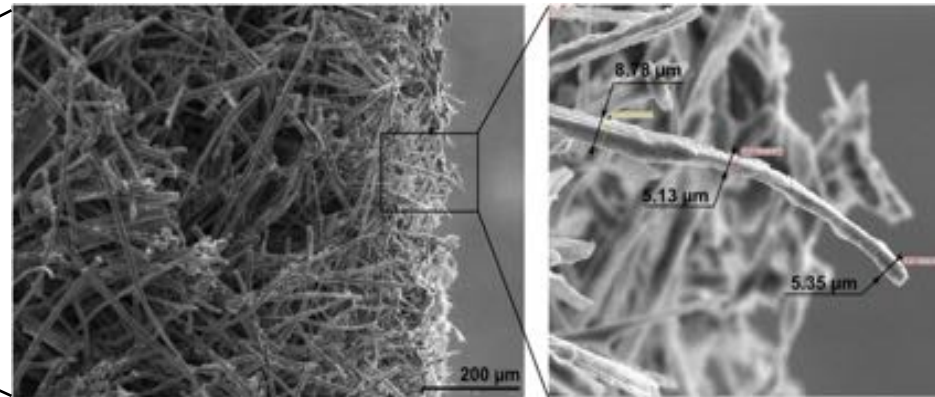
Full scale material response solvers, using volume-averaged techniques to solve conservation equations for ablation



Simulation of surface temperature for MSL heatshield

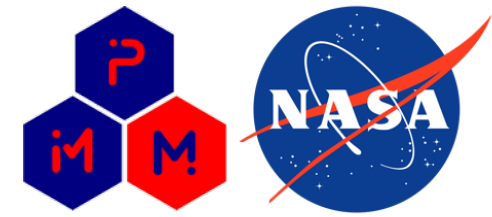
## Microscale Modeling

Used to inform material properties and material response parameters used in macro-scale modeling



Lachaud and Mansour, *JTHT* 2013

# X-Ray Microtomography



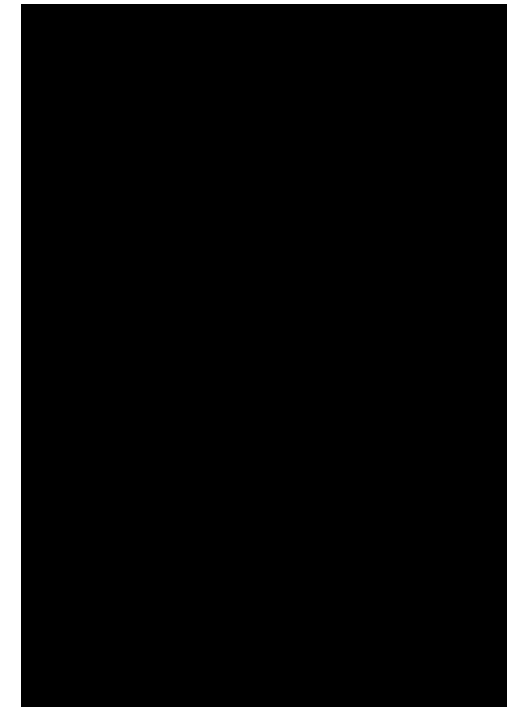
Collect X-ray images of the sample as you rotate it through  $180^\circ$

Use this series of images to reconstruct the 3D object



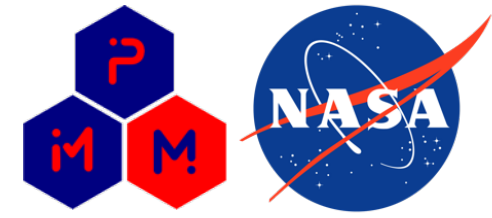
Penetrating power

Multiple angles



Courtesy of D. Parkinson (ALS)

# Porous Microstructure Analysis (PuMA)



## Material Properties

- Volume Fraction and Surface Area
- Pore Diameter Estimation and Filling
- Tortuosity (continuum and rarefied)
- Computation of Material Orientation
- Conductivity (isotropic and anisotropic)
  - Elasticity and Stress Analysis

## Domain Generation

- Artificial Materials:
  - Weaves
  - Random Fibers
  - Foams
  - Analytical shapes
- Micro-CT Import and Filtering

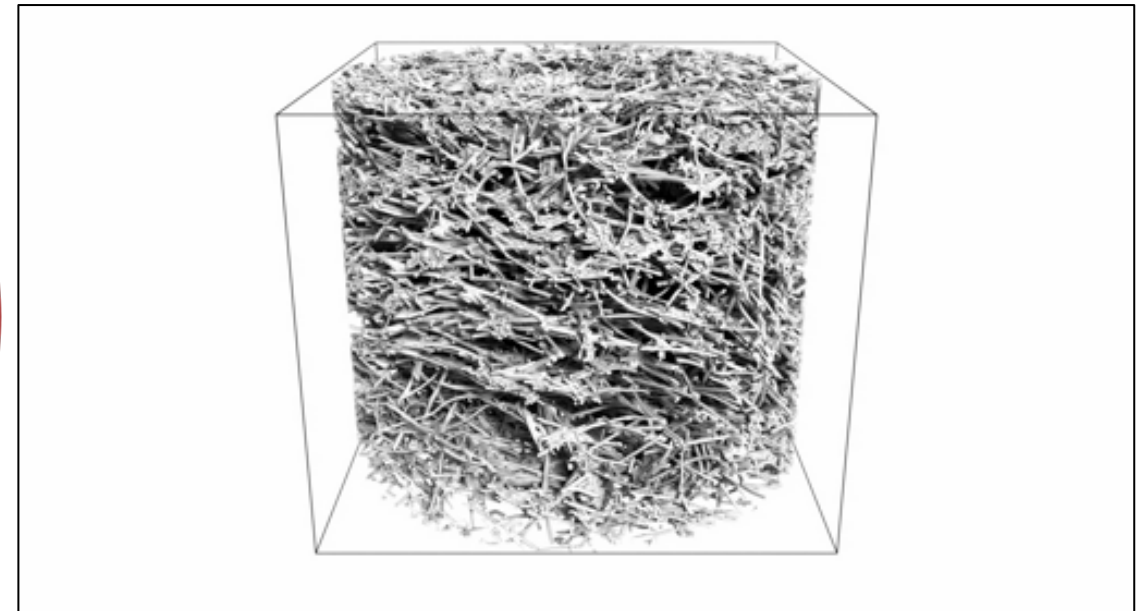


## Shared Utilities

- Matrix and Vector data structures/operations
  - Linear solvers
- Iso-surface extraction
- Import/Export to binary, TIFF, STL, VTK formats

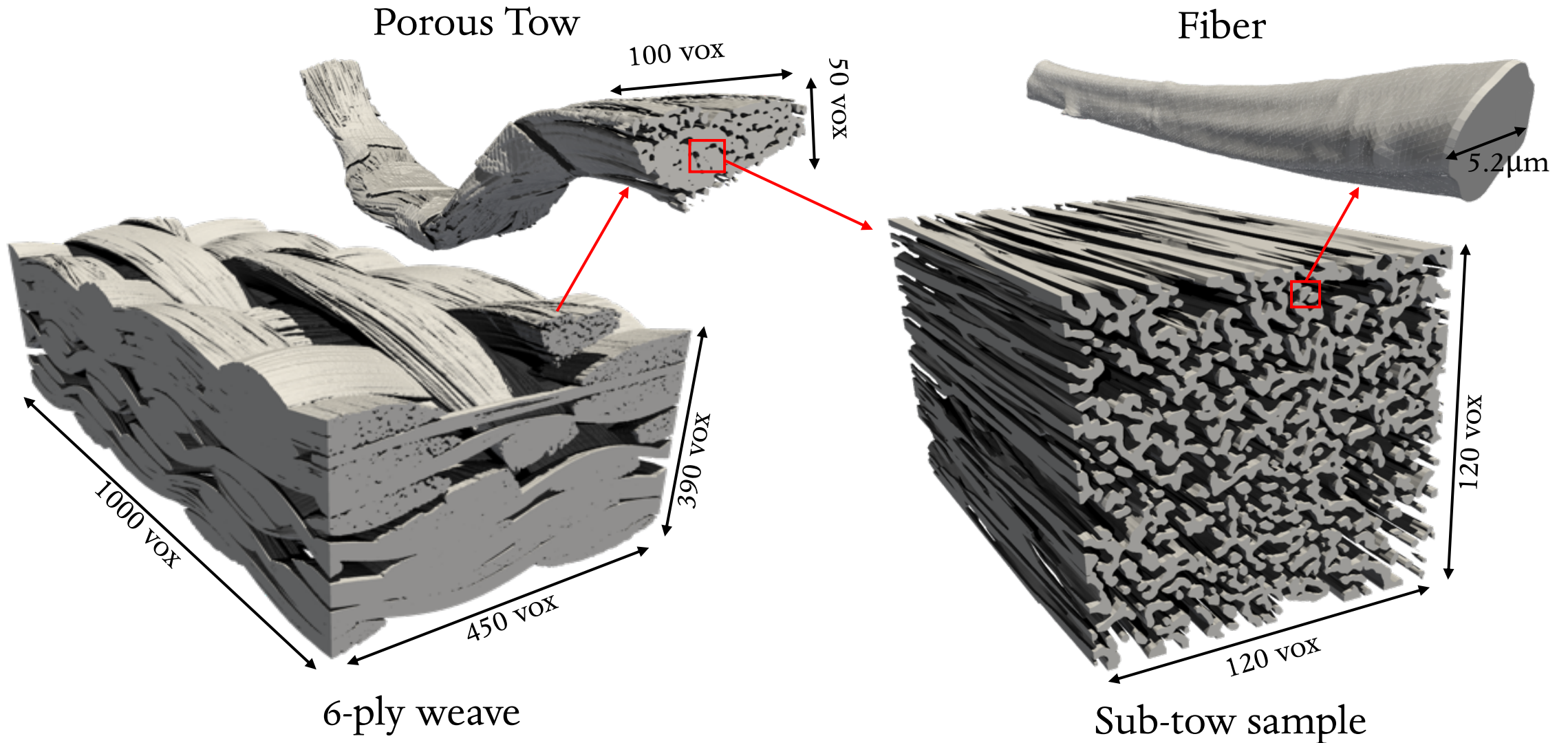
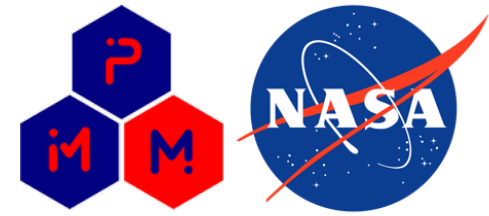
New features in Version 3:

- Modular C++ library
- Redesigned GUI
- Wrapped as Python module
- Linux and Mac compatibility

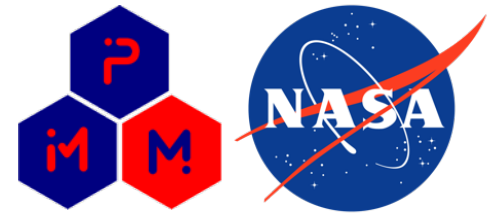


CT Reconstruction of FiberForm

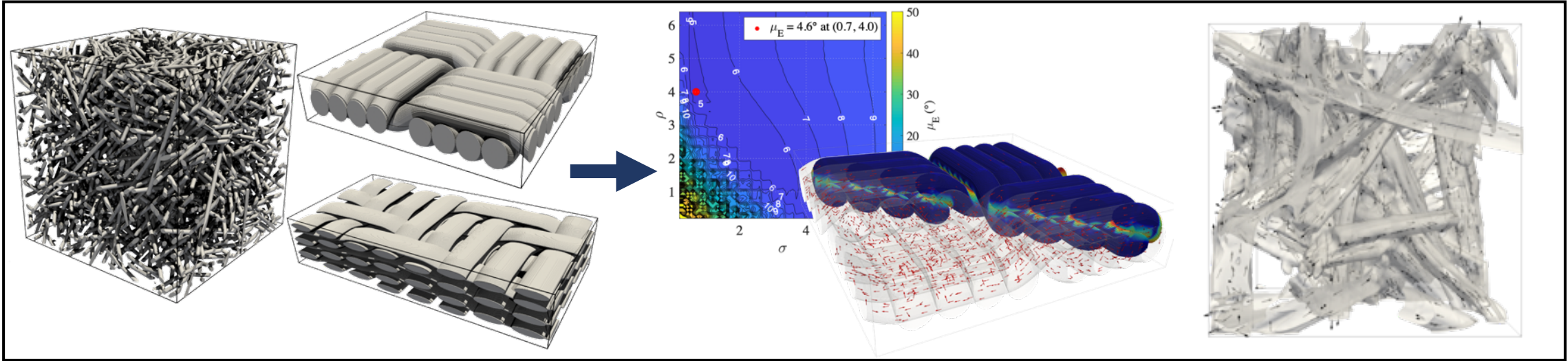
# A Multiscale Problem



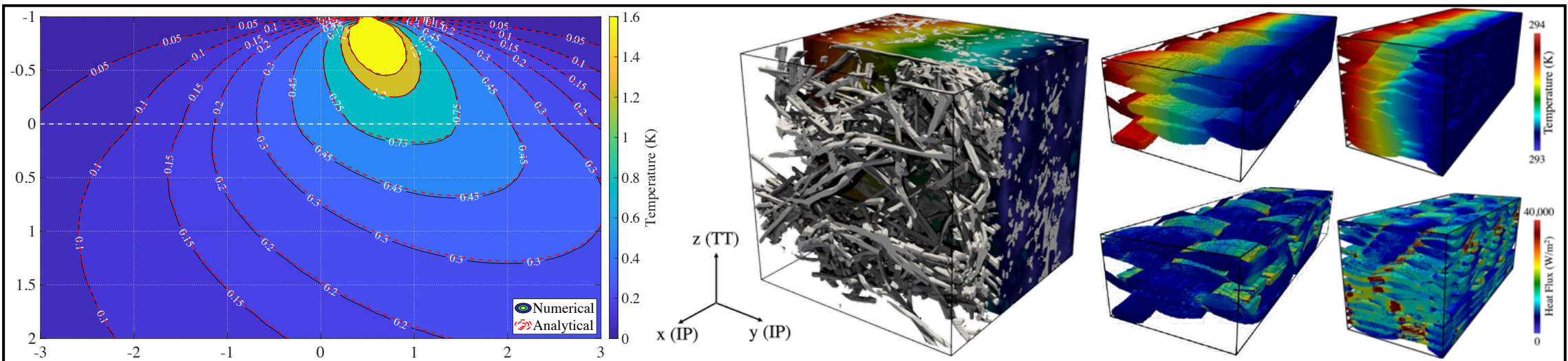
# Effective properties for fibrous media



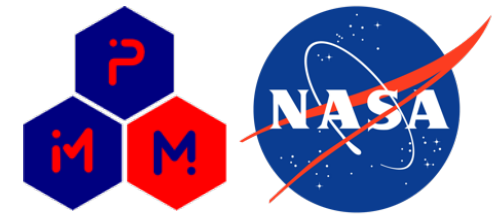
Semeraro, F., Ferguson, J.C., Panerai, F., King, R.J. and Mansour, N.N., Anisotropic analysis of fibrous and woven materials part 1: Estimation of local orientation. *Computational Materials Science*, 178, p.109631. (2020)



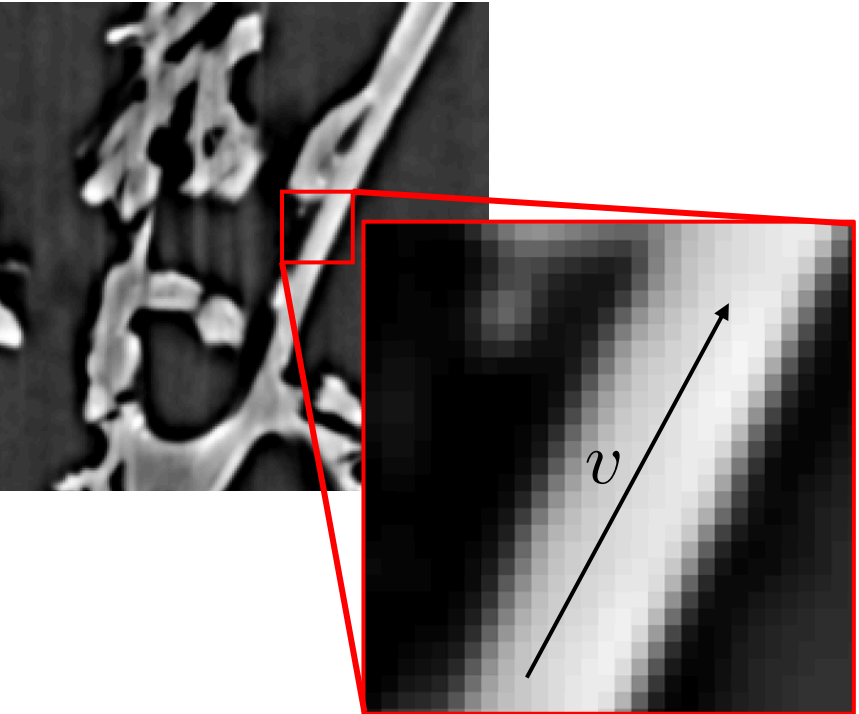
Semeraro, F., Ferguson, J.C., Acin, M., Panerai, F. and Mansour, N.N., Anisotropic analysis of fibrous and woven materials part 2: Computation of effective conductivity. *Computational Materials Science*, 186, p.109956. (2021)



# Fiber Orientation Methods

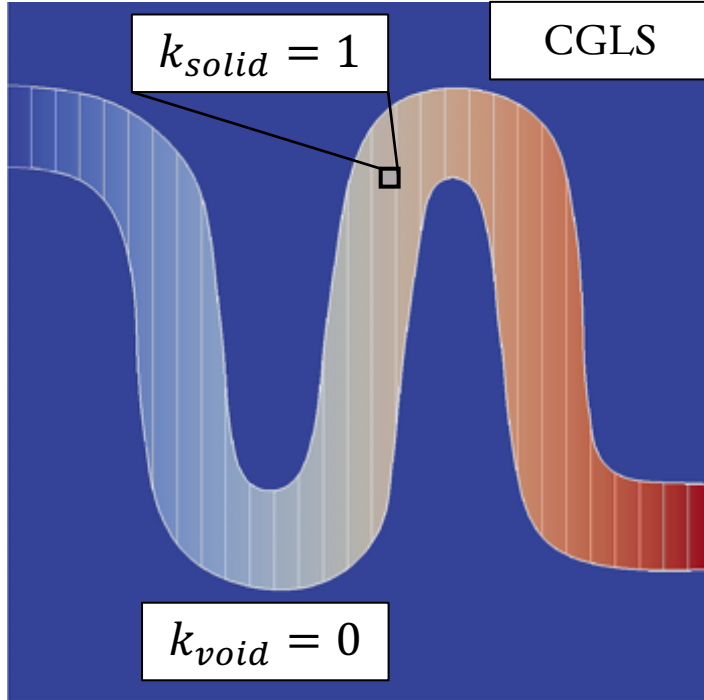


Structure Tensor

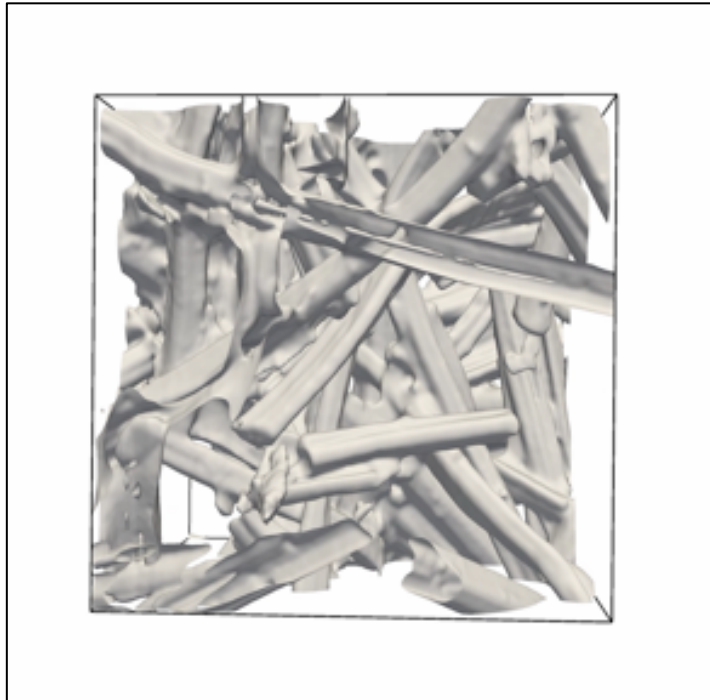


$$(I(x + v) - I(x))^2 \approx 0$$

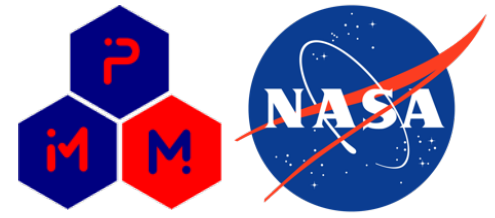
Artificial Flux



Ray Casting

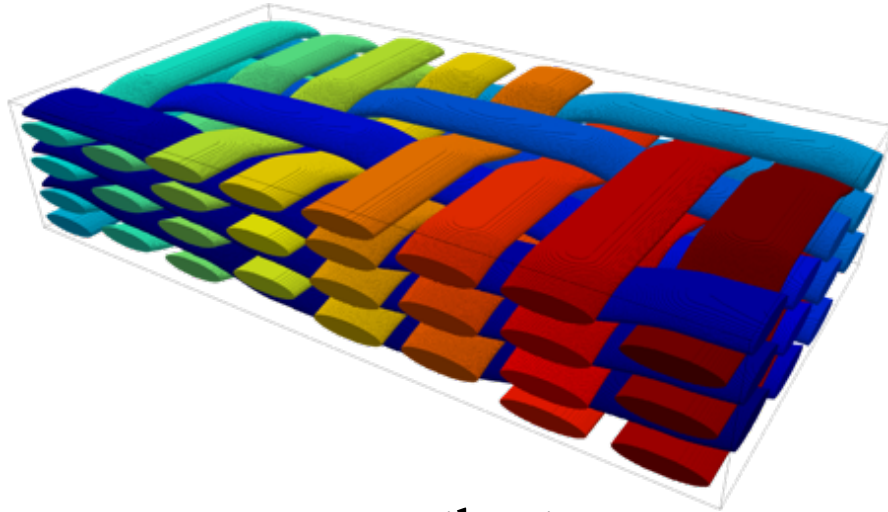




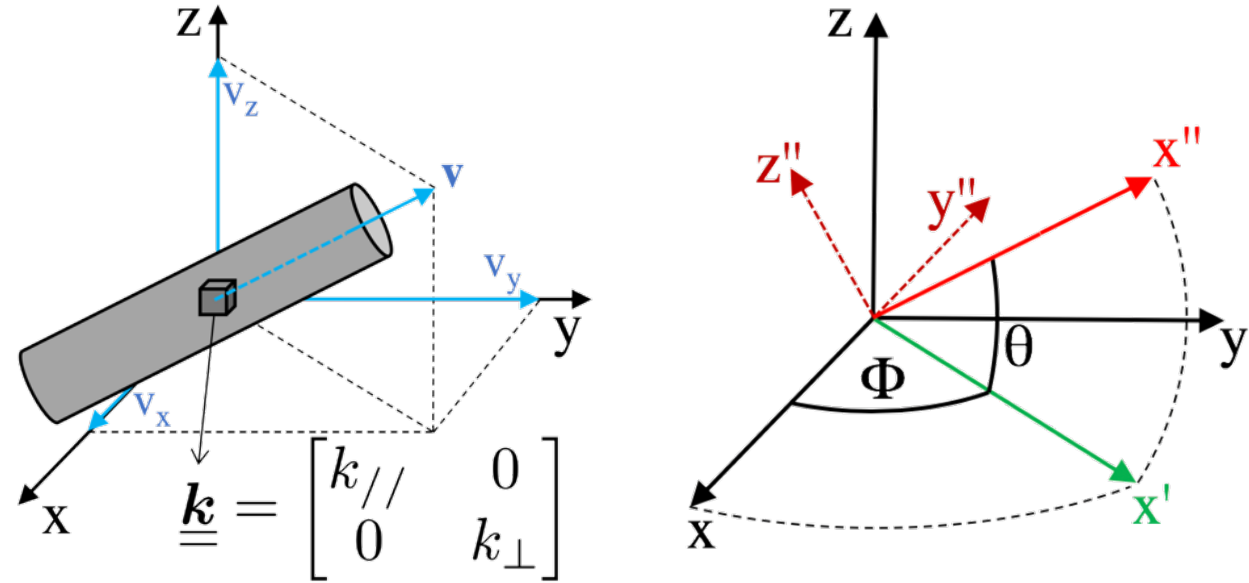
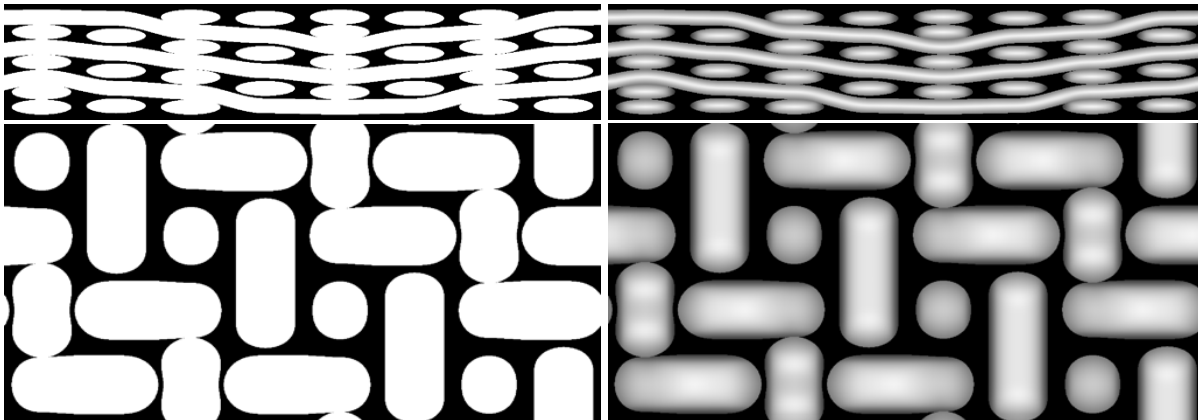


# Workflow for Weave Orientation

Individual tow segmentation



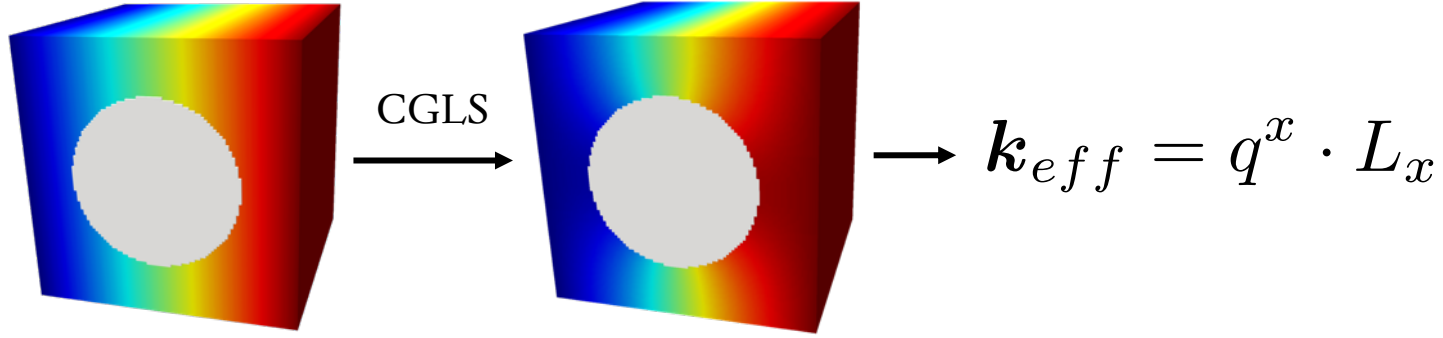
Mean Filtering



$$q = \underbrace{[R^{-1} k'' R]}_k \nabla T$$

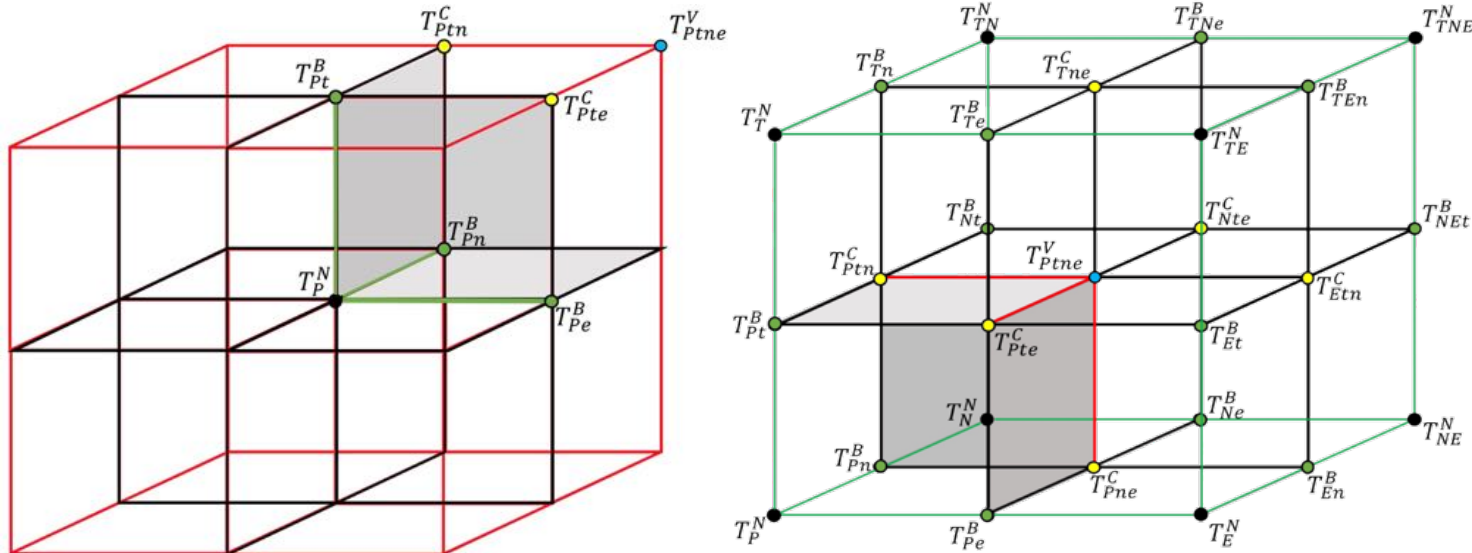
$$R = \begin{bmatrix} \cos \theta & 0 & -\sin \theta \\ 0 & 1 & 0 \\ \sin \theta & 0 & \cos \theta \end{bmatrix} \begin{bmatrix} \cos \phi & \sin \phi & 0 \\ -\sin \phi & \cos \phi & 0 \\ 0 & 0 & 1 \end{bmatrix}^9$$

# Effective Thermal Conductivity

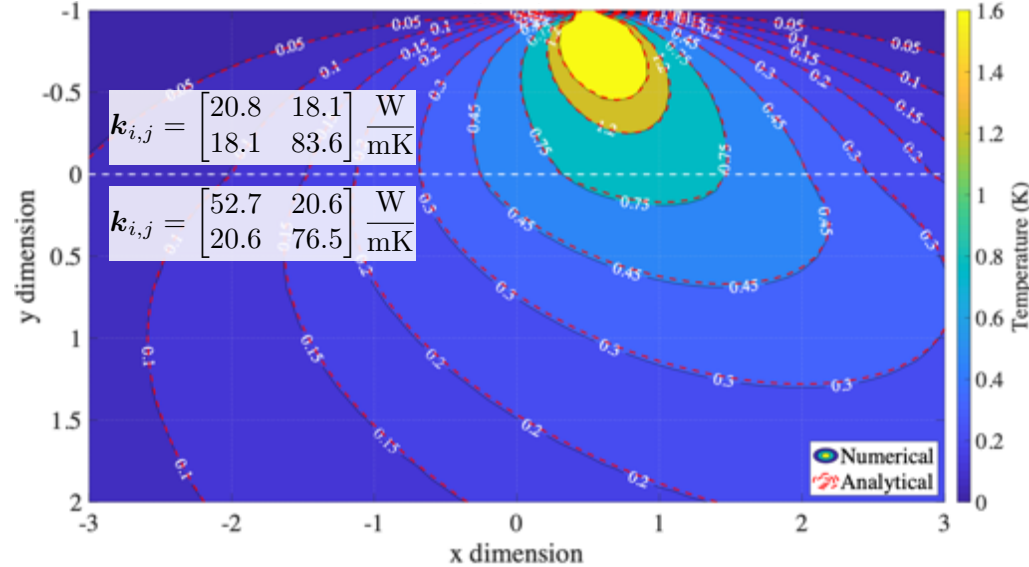
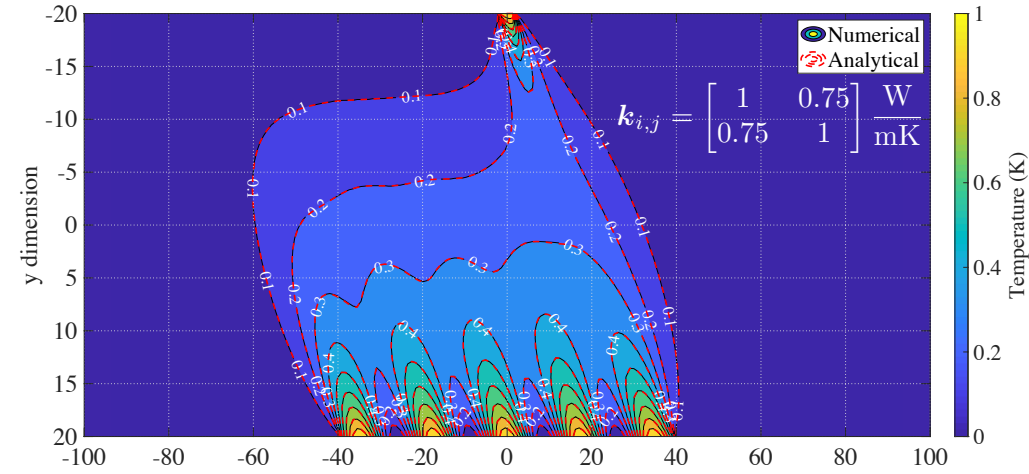


$$\nabla \cdot \mathbf{q} = 0 \longrightarrow \mathbf{q}(x, t) = \mathbf{E}(x) \mathbf{T}^N(x, t)$$

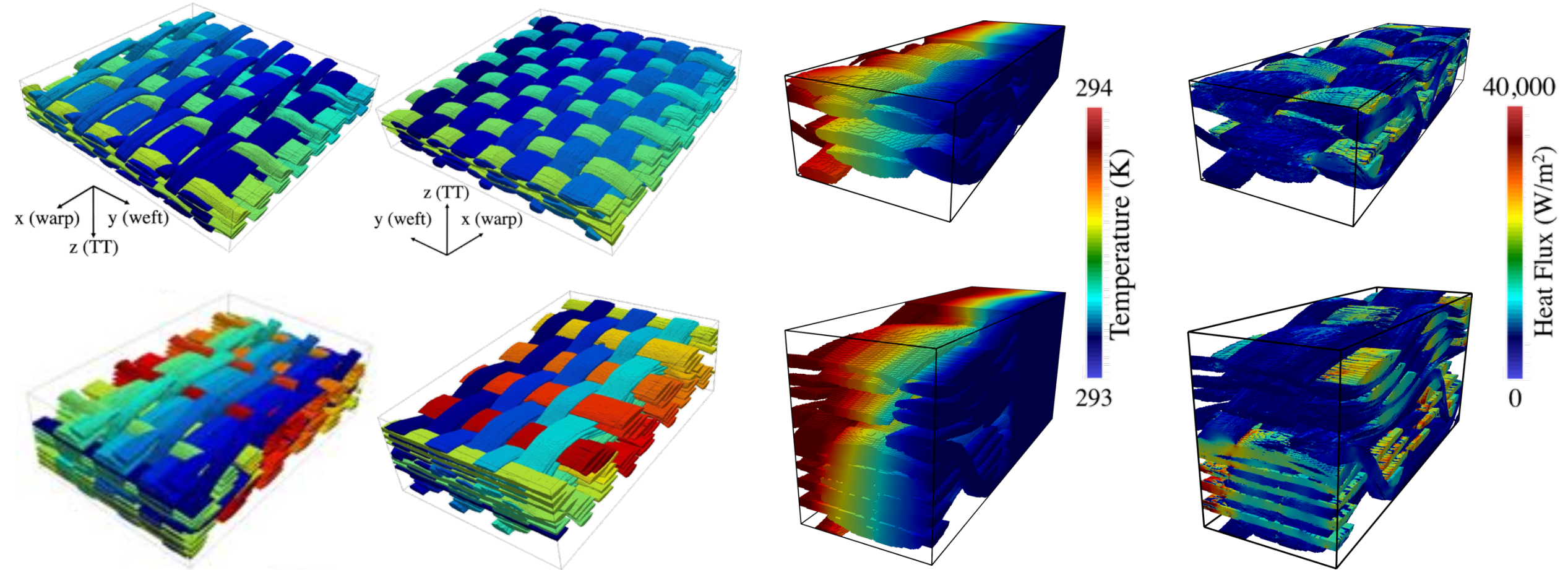
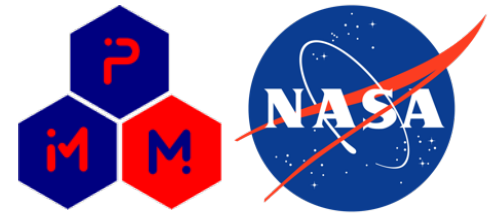
Multi-Point Flux Approximation (MPFA):



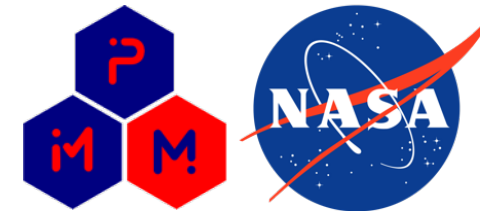
Verification against analytical solutions



# ADEPT Validation



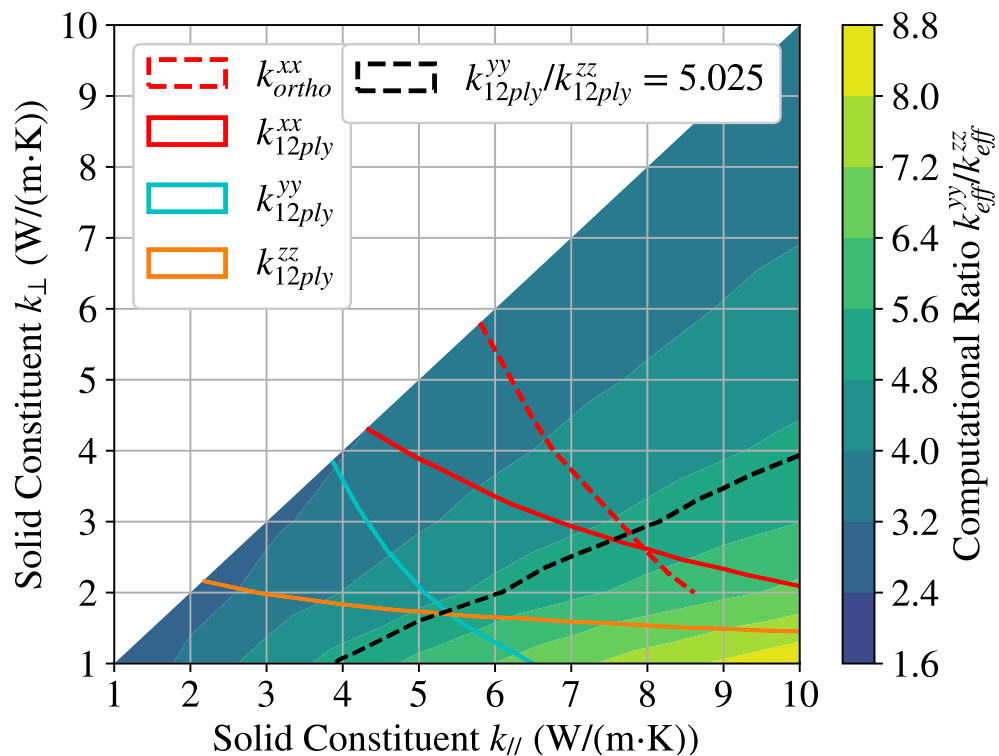
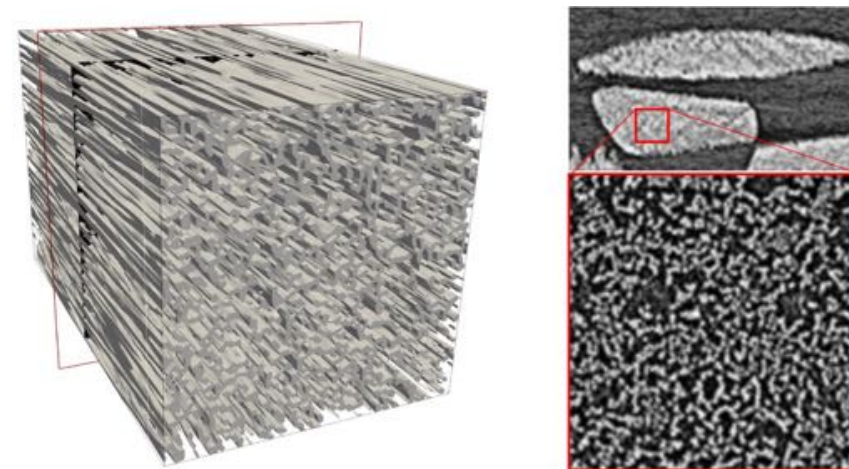
# Single Fiber Conductivity Estimation



Two experimental values at Room Temperature:

$$\mathbf{k}_{eff}^{ortho} = \begin{bmatrix} 2.342 & - & - \\ - & 1.766 & - \\ - & - & 0.305 \end{bmatrix} \quad \mathbf{k}_{eff}^{12ply} = \begin{bmatrix} 2.184 & - & - \\ - & 1.980 & - \\ - & - & 0.394 \end{bmatrix}$$

$$[k_{//}^{weft}, k_{\perp}^{weft}] = [5.4, 1.8] \quad \text{and} \quad [k_{//}^{warp}, k_{\perp}^{warp}] = [7.7, 2.7] \frac{\text{W}}{\text{mK}}$$



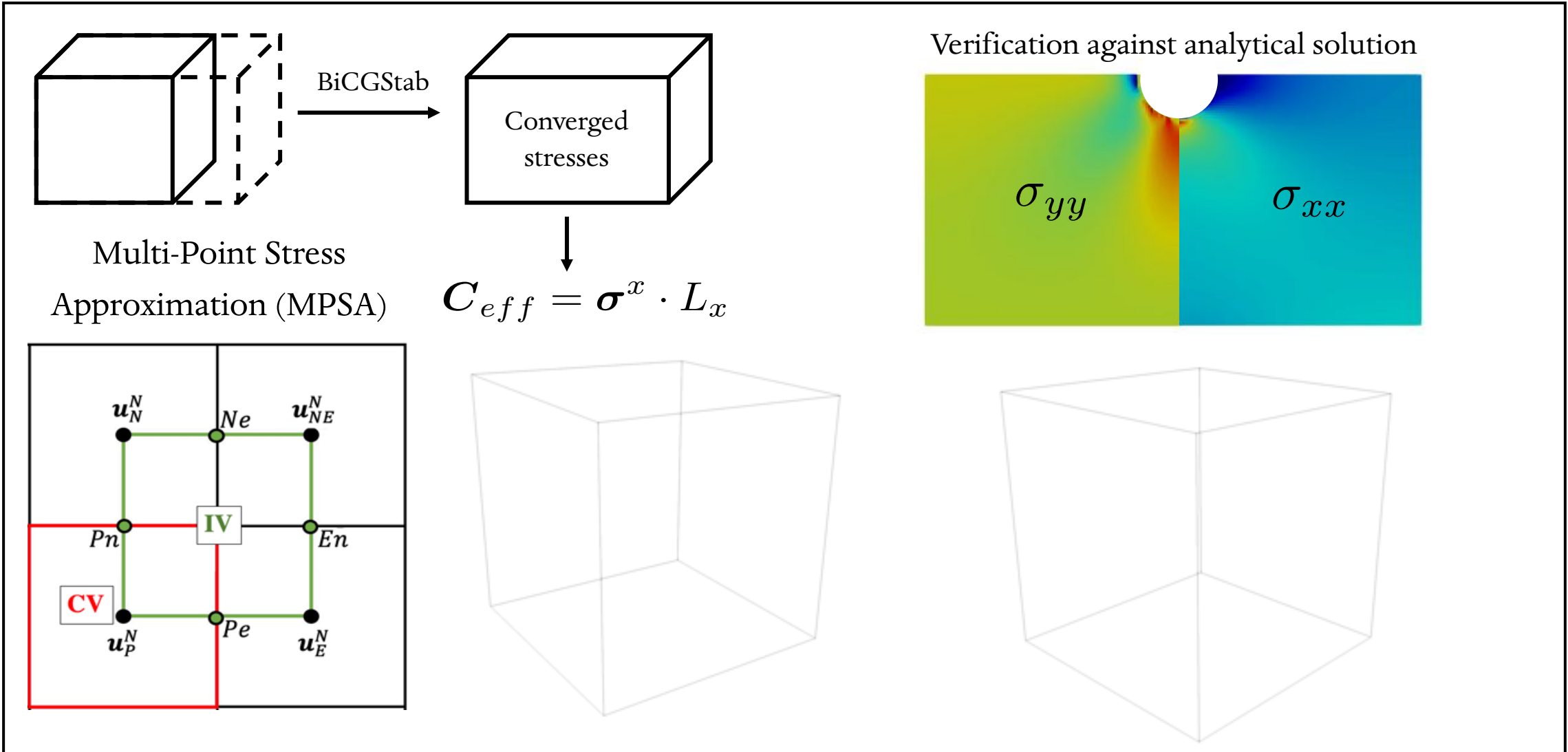
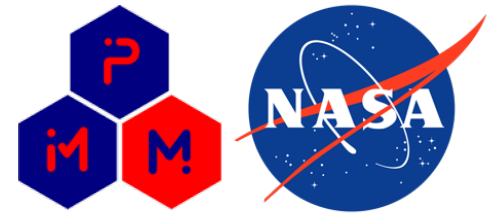
$$[k_{//}, k_{\perp}] = [9.7, 5.5] \frac{\text{W}}{\text{mK}}$$

$$\mathbf{k}_{eff}^{4ply} = \begin{bmatrix} 1.750 & -0.032 & -0.024 \\ -0.093 & 1.490 & 0.002 \\ 0.039 & 0.001 & 0.193 \end{bmatrix} \quad \mathbf{k}_{eff}^{6ply} = \begin{bmatrix} 1.740 & -0.141 & -0.020 \\ -0.105 & 1.710 & -0.005 \\ 0.091 & -0.023 & 0.200 \end{bmatrix}$$

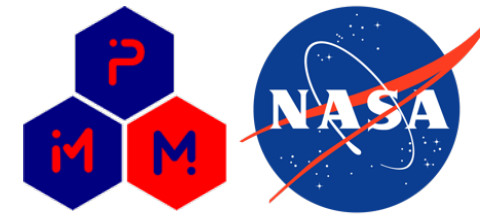
$$\mathbf{k}_{eff}^{8ply} = \begin{bmatrix} 1.830 & -0.244 & 0.016 \\ -0.192 & 1.510 & 0.005 \\ 0.039 & 0.000 & 0.231 \end{bmatrix} \quad \mathbf{k}_{eff}^{12ply} = \begin{bmatrix} 2.310 & -0.414 & 0.000 \\ -0.524 & 2.030 & 0.071 \\ 0.007 & 0.005 & 0.504 \end{bmatrix}^{12}$$

# Effective Elasticity

Semeraro, F., Acin, M., Panerai, and Mansour, N.N., Anisotropic analysis of fibrous and woven materials  
 part 3: Computation of effective elasticity. *Computational Materials Science*, (work in progress)

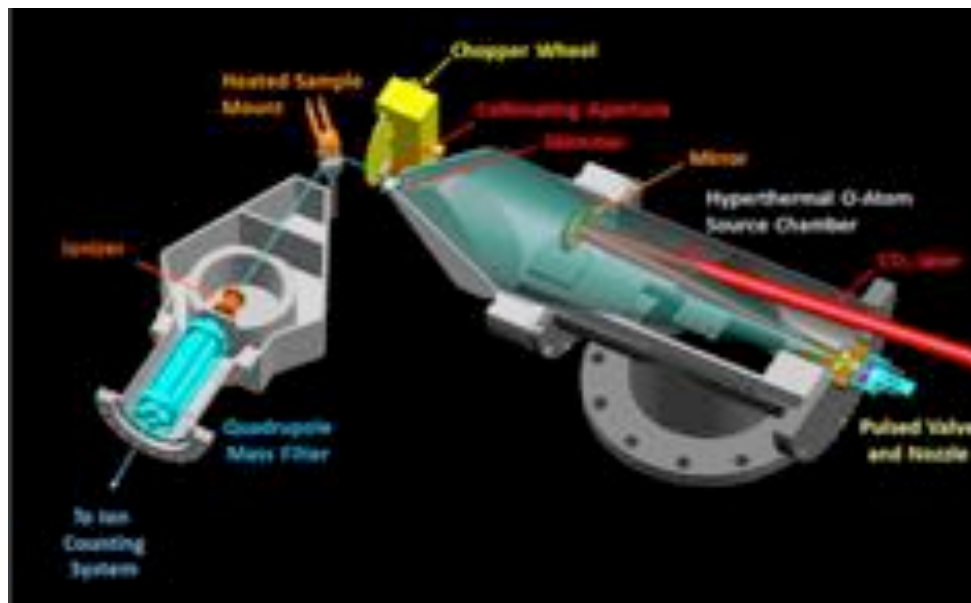


# Application motivation

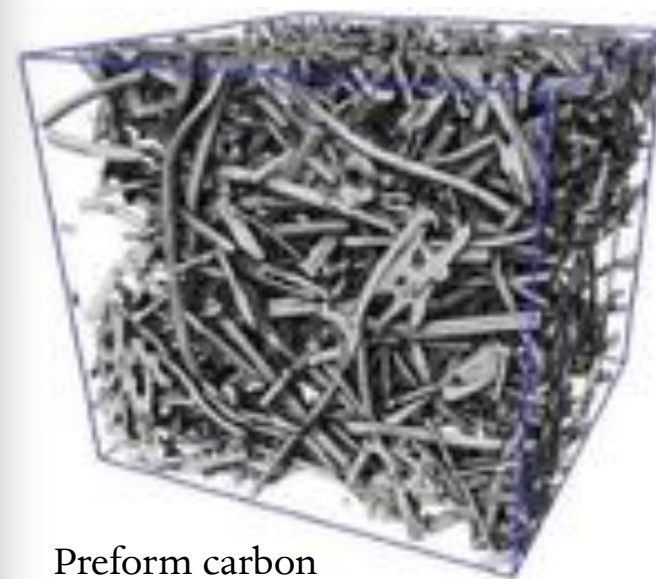


## Objectives:

- Characterize recession/ablation of carbon surface coated with NuSil due to oxidation
- Develop a predictive model for use in microscopic (DSMC, PuMA) and macroscopic (CFD, MRC) solvers

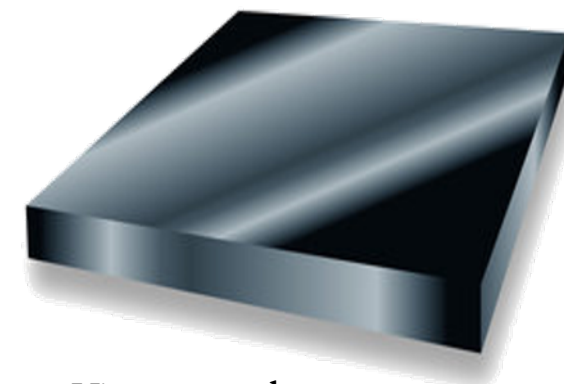


Murray *et al*



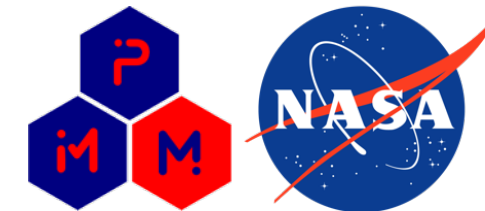
Preform carbon

Image courtesy: Prof. Panerai



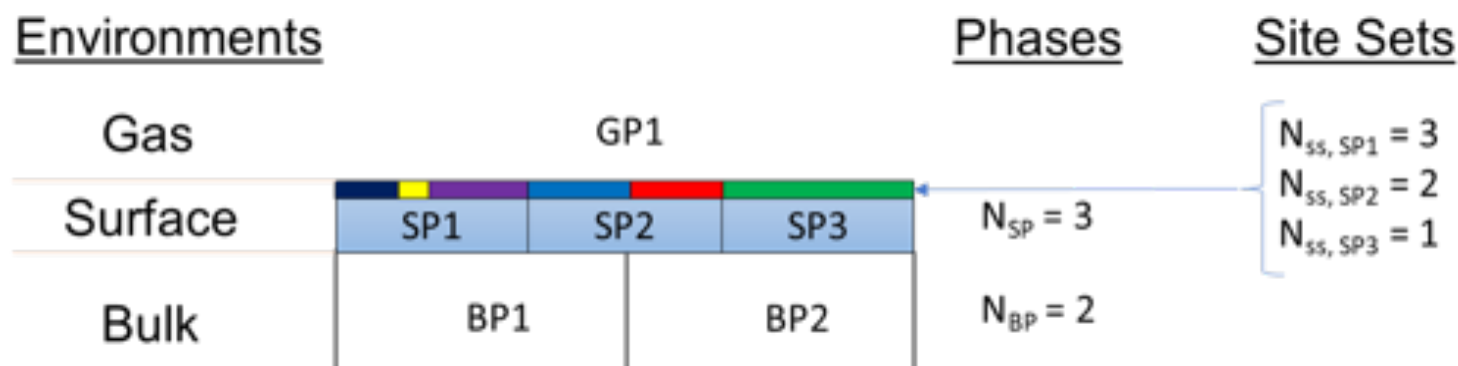
Vitreous carbon

SPI Supplies



# Surface chemistry framework

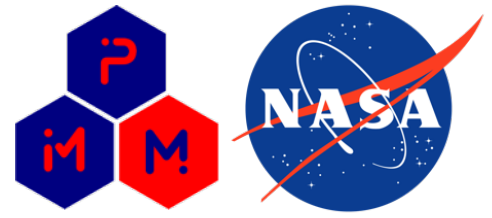
- Methodology to represent surface sites similar to Marschall, Maclean and Driver [4] for CFD.
- Particles adsorbed (deleted) and desorbed (created), surface element stores adsorbed particle concentration.
- Surface reactions based on concentration within surface element.
- Multiple triangulated elements (like cells) on surfaces
- Langmuir model for surface sites.



taken from Marschall and Maclean.

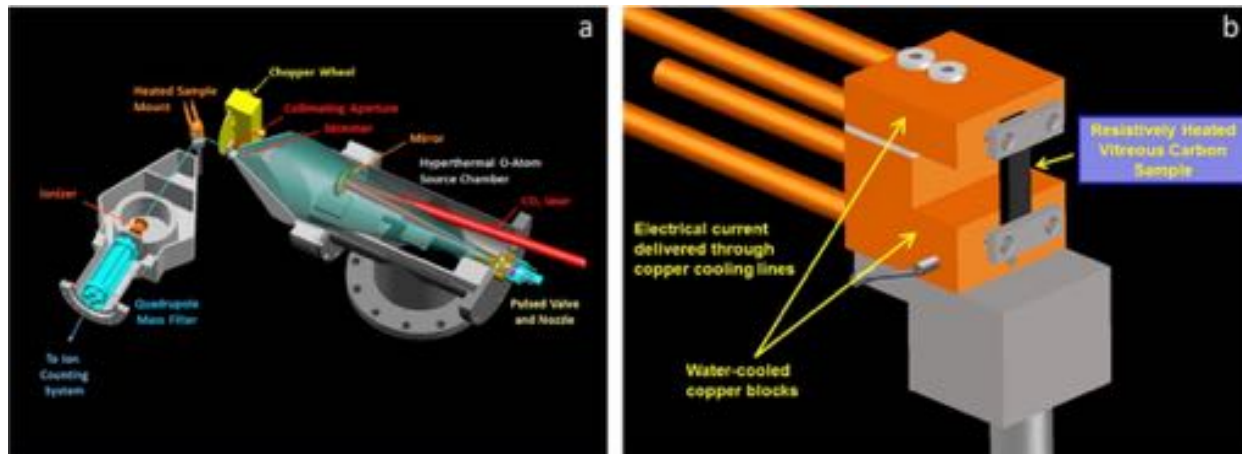
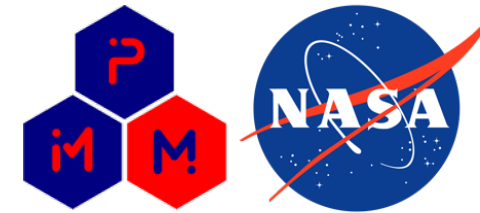


# PuMA simulation

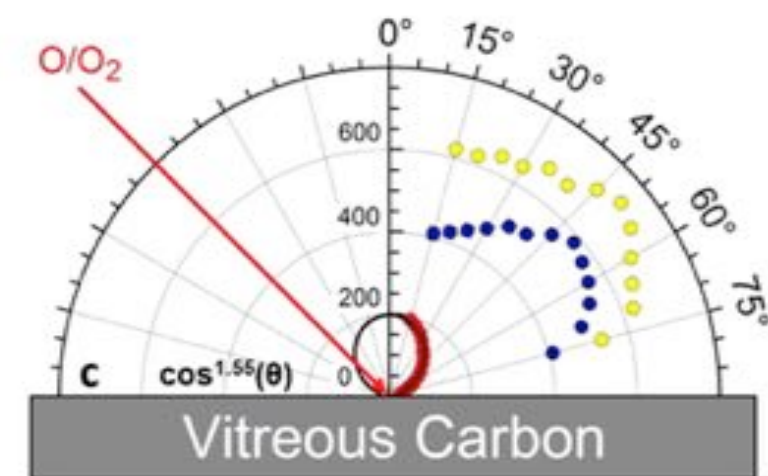
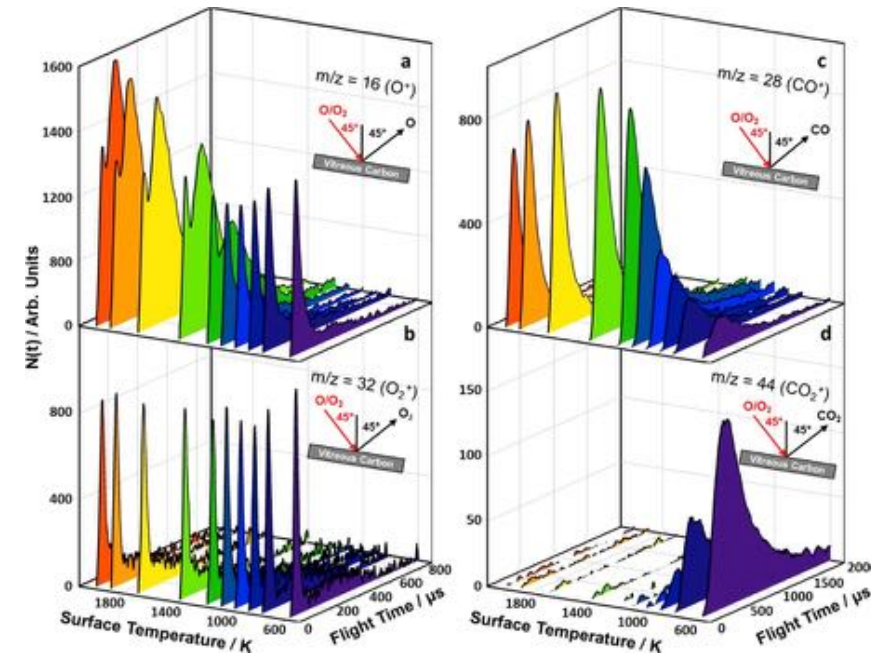
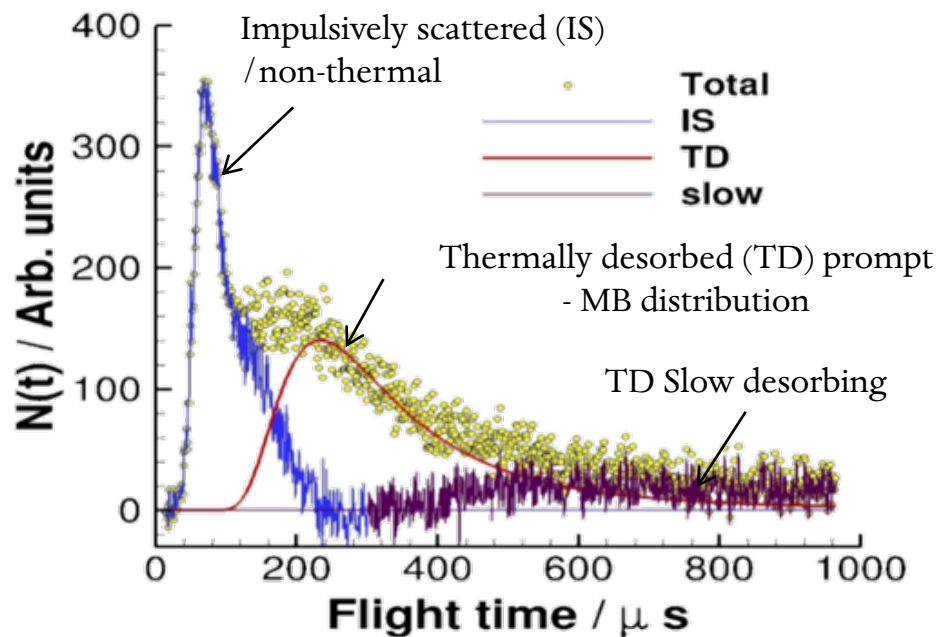


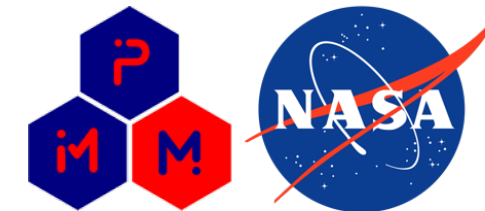


# Molecular Beam Experimental Setup



Murray *et al*



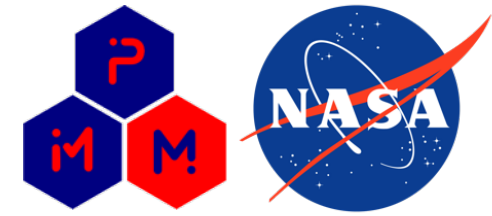


# Carbon Oxidation Model

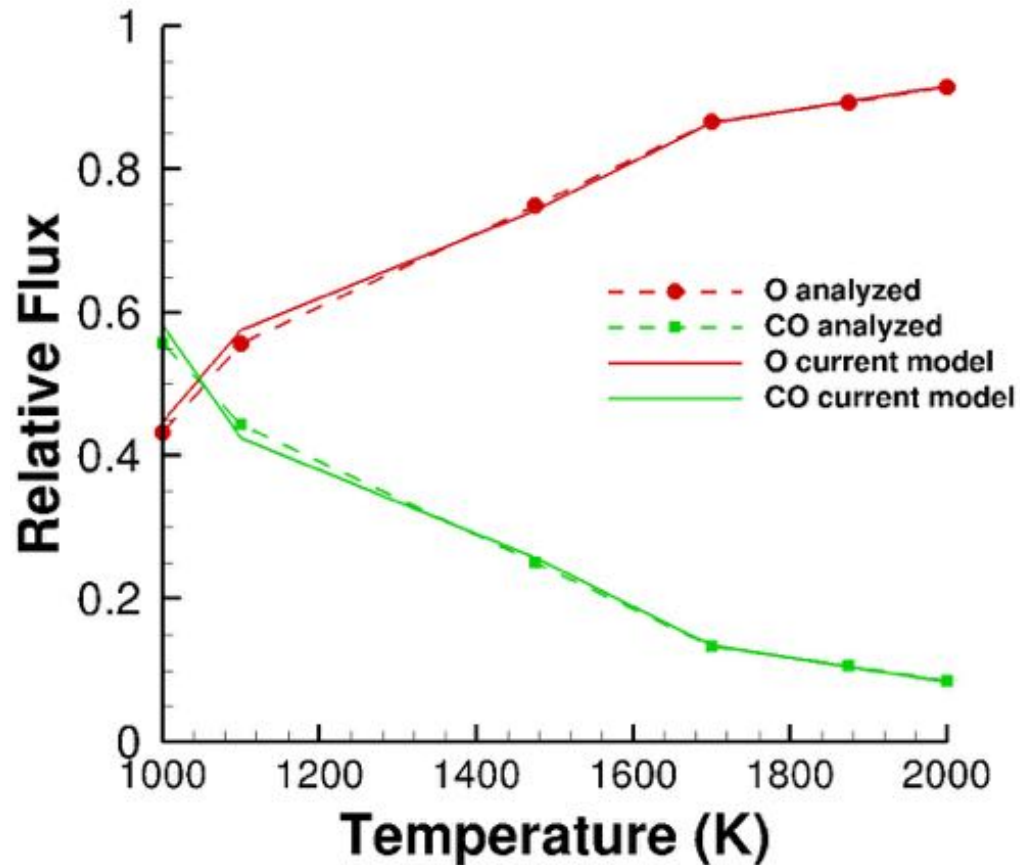
Type	Mechanisms	Reaction	
Adsorption	Adsorption	$O(g) + (s) \rightarrow O(ads)$	
Adsorption-mediated	LH1 O formation	$O(ads) \rightarrow O(TD)(g) + (s)$	Fast TD reactions
	LH1 CO formation	$O(ads) + C(b) + O'(ads) \rightarrow CO(g) + (s) + O'(ads)$	
	LH1 CO <sub>2</sub> formation	$O(ads) + O(s) + C(b) + 4O'(ads) \rightarrow CO_2(g) + 2(s) + 4O'(ads)$	
GS reactions	LH3 O{a} formation	$O(ads) \rightarrow O\{a\}(s)$	Slow formation reactions
	LH3 CO{a} formation	$O(ads) + C(b) + O'(ads) \rightarrow CO\{a\}(s) + O'(ads)$	
	LH3 CO{b} formation	$O(ads) + C(b) + O'(ads) \rightarrow CO\{b\}(s) + O'(ads)$	
PS reactions	LH3 O{a} desorption	$O\{a\}(s) \rightarrow O(g) + (s)$	Desorption reactions
	LH3 CO{a} desorption	$CO\{a\}(s) \rightarrow CO(g) + (s)$	
	LH3 CO{b} desorption	$CO\{b\}(s) \rightarrow CO(g) + (s)$	

K. Swaminathan-Gopalan *et al.*, "*Development and validation of a finite-rate model for carbon oxidation by atomic oxygen.*" Carbon 137 (2018): 313-332.

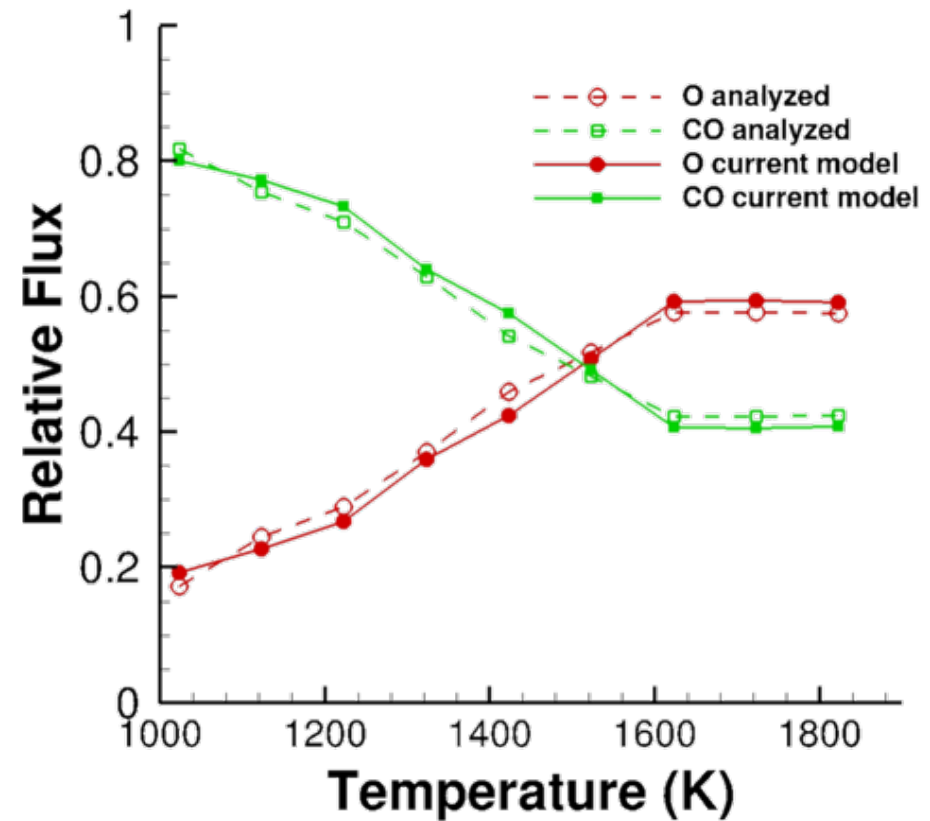
# Vitreous and Preform Carbon model



### Vitreous carbon model



### Preform carbon model



# Effective model for macroscopic solvers

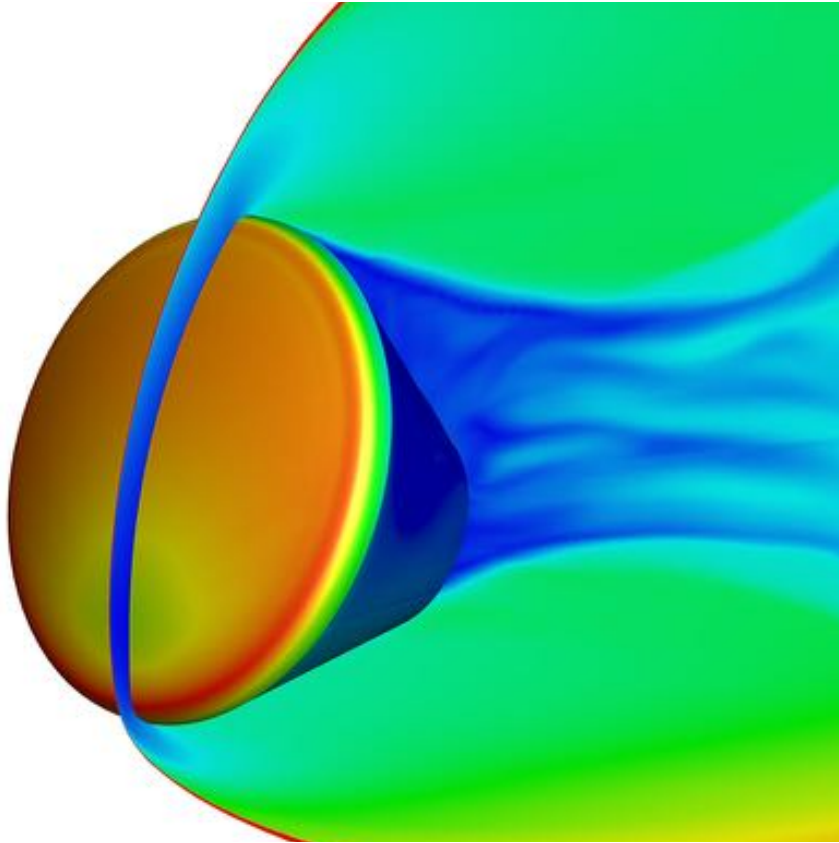
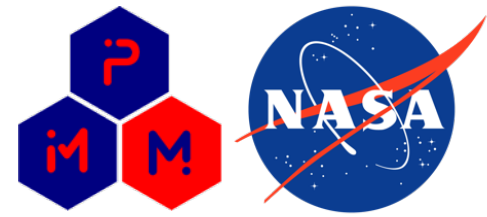
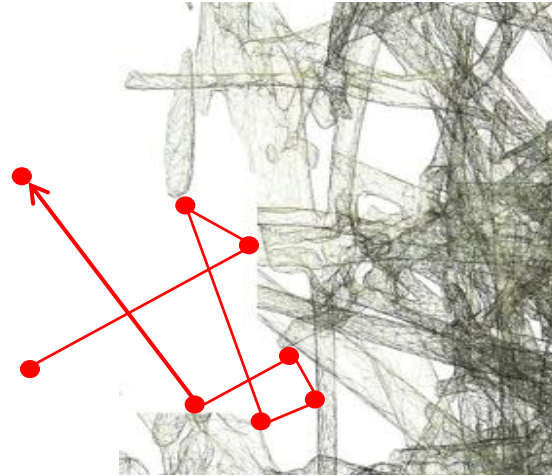
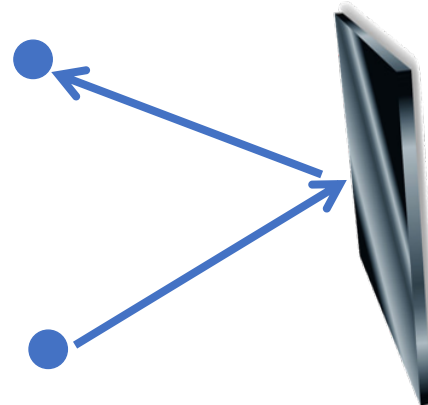


Image credit: NASA

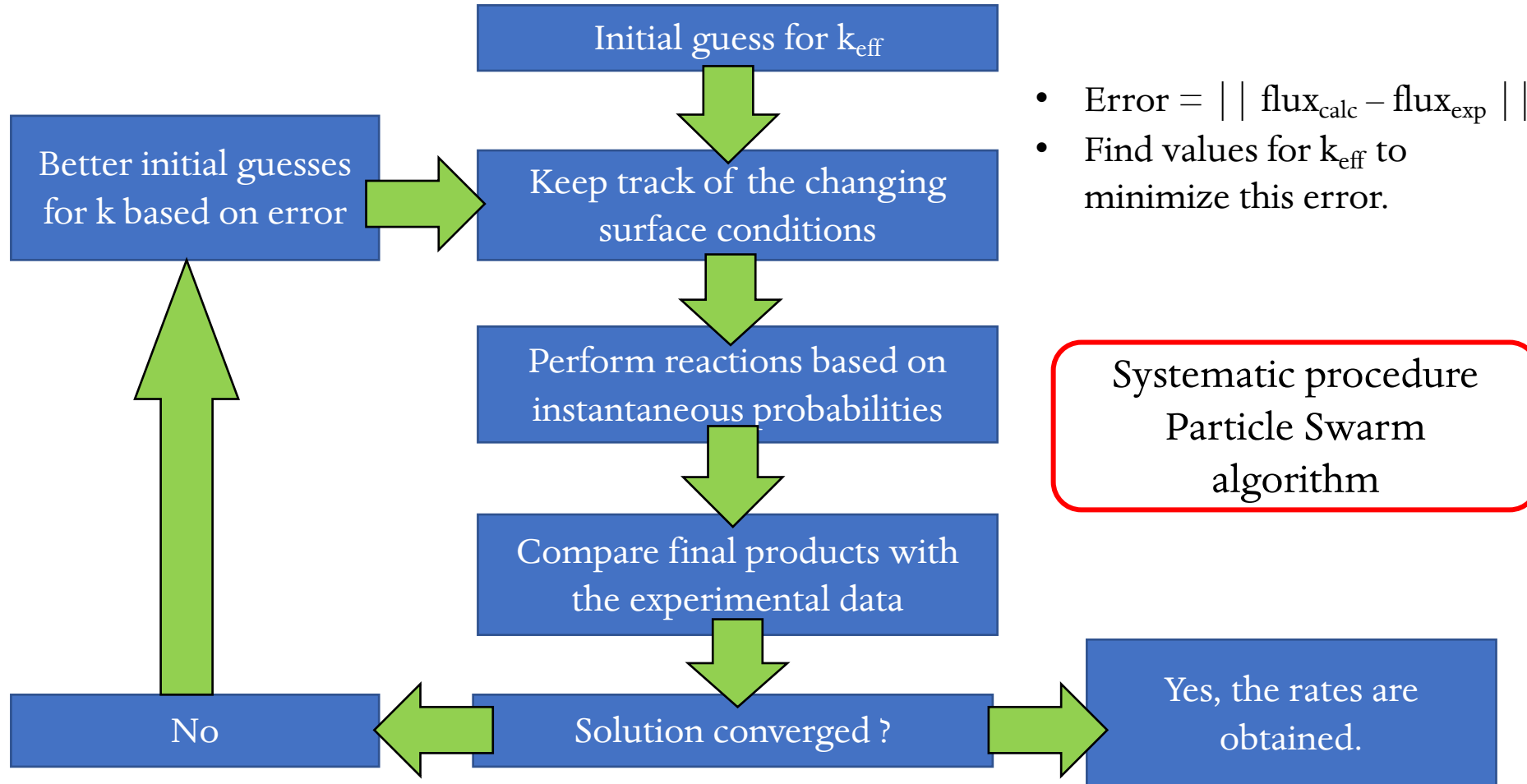
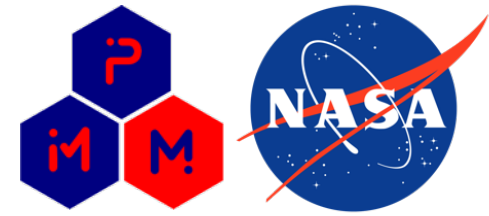


Real model  
Rates  $\rightarrow k$

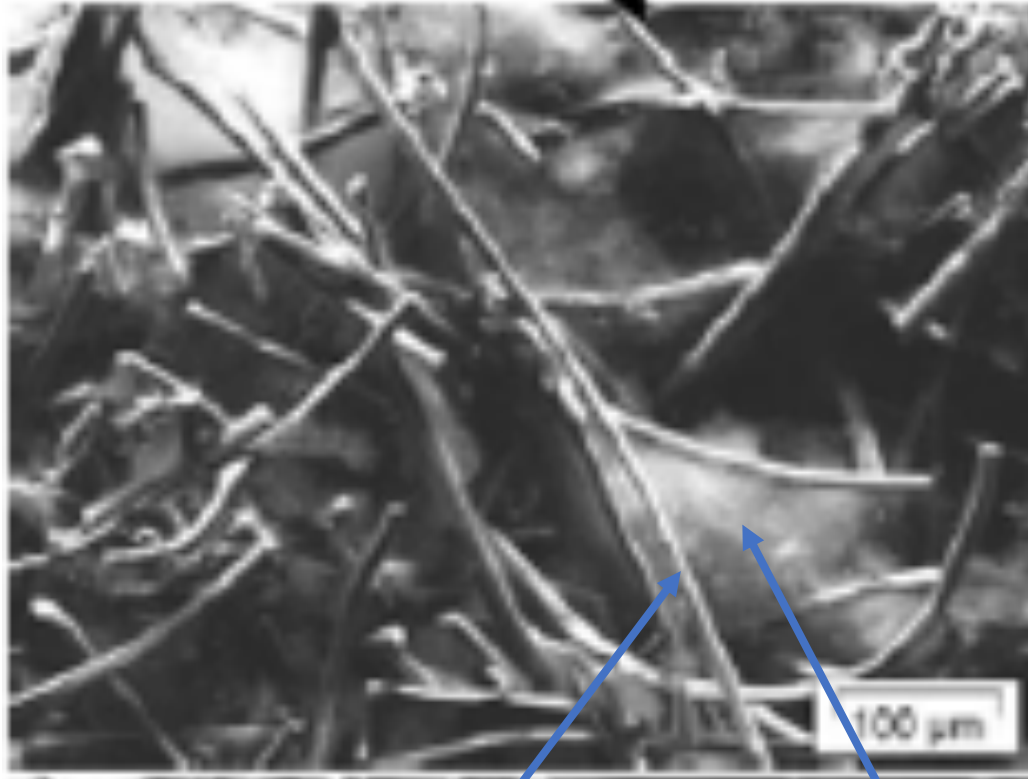
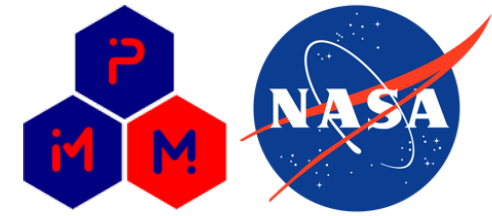


Effective model  
Rates  $\rightarrow k_{\text{eff}}$

# Reaction Rate Constants Fitting Method



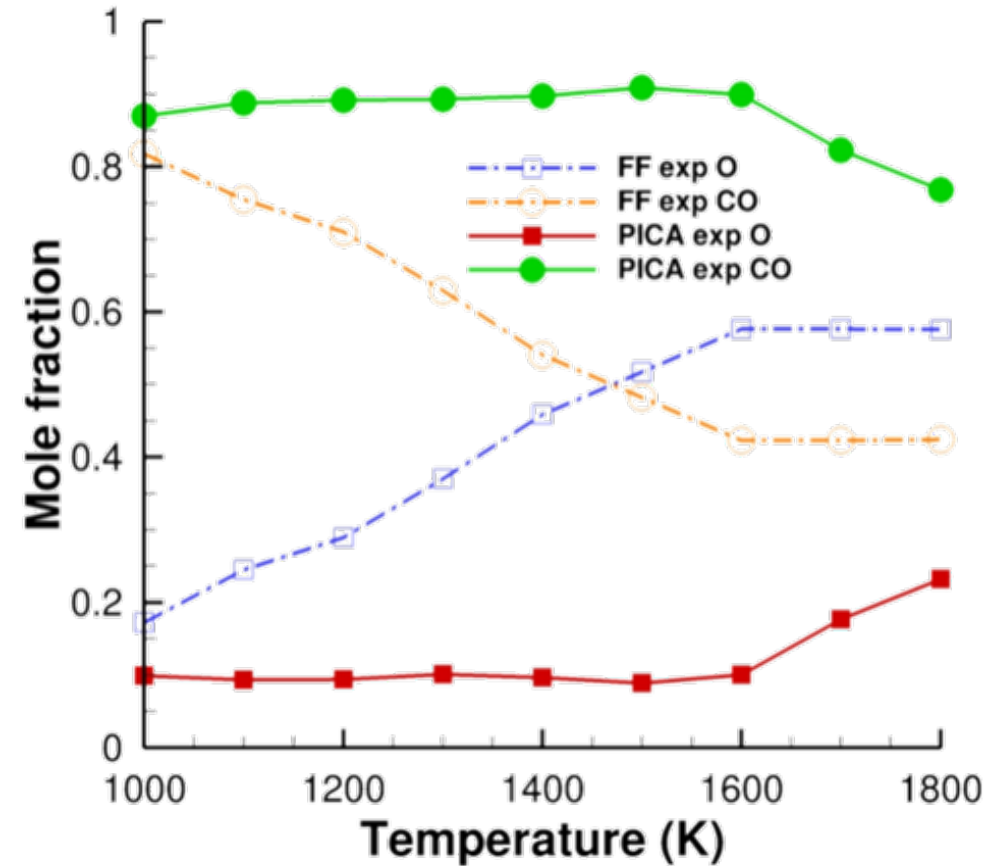
# PICA oxidation model



Preform carbon

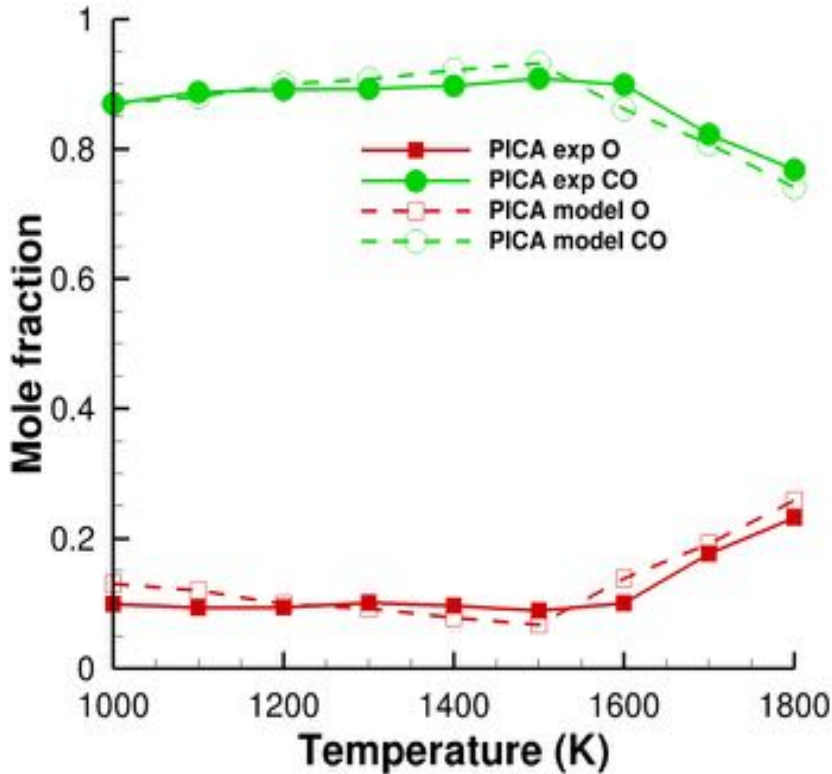
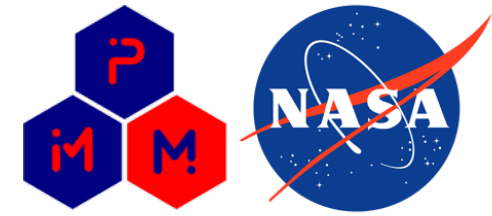
PICA  
CHAR

2 phase model



- PICA and FF trends are qualitatively and quantitatively different.
  - Difference in reactivity.
  - Difference in porosity.

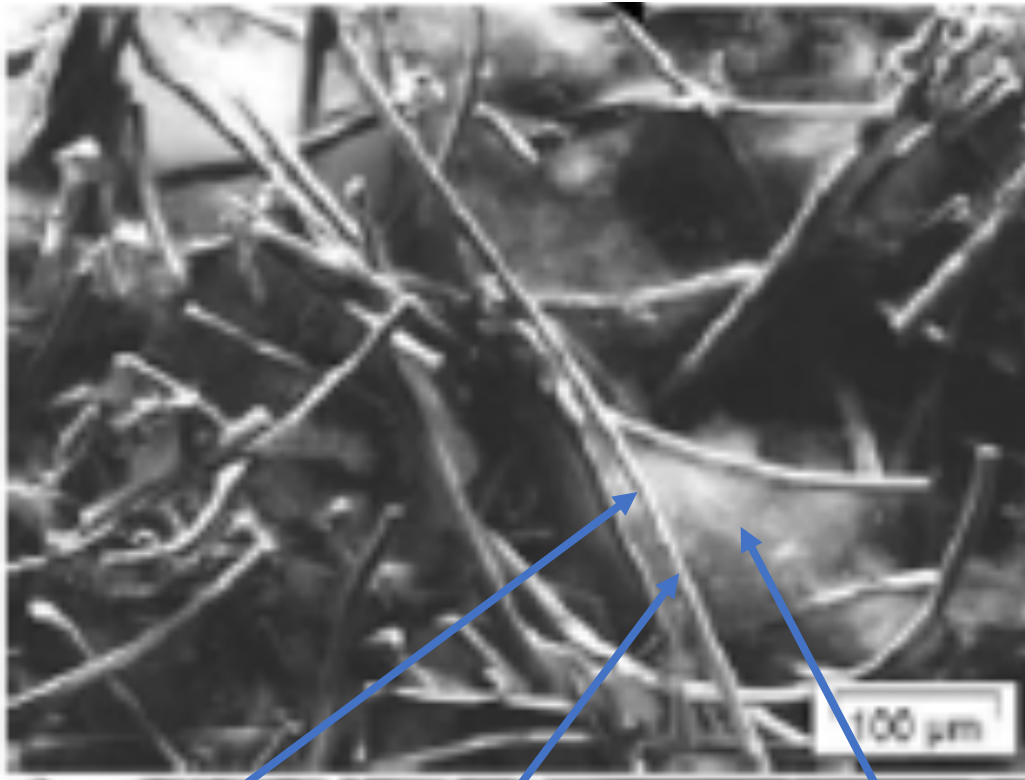
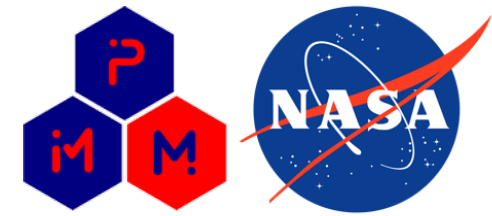
# PICA oxidation model



Mechanisms	Reaction	Rate constant (k)
Adsorption	$O(g) + (s) \rightarrow O(ads)$	$\frac{1}{\phi} * \frac{1}{4} \sqrt{\frac{8k_B T_s}{\pi m}} * 1$
TD O formation	$O(ads) \rightarrow O(TD)(g) + (s)$	$0.946 \exp\left(-\frac{4686.98}{T_s}\right)$
TD CO formation	$O(ads) + C(b) + O'(ads) \rightarrow CO(g) + (s) + O'(ads)$	$\frac{1}{\phi} * 4262.96 \exp\left(-\frac{7669.43}{T_s}\right)$
LH O{a} formation	$O(ads) \rightarrow O\{a\}(s)$	1
LH CO{a} formation	$O(ads) + C(b) + O'(ads) \rightarrow CO\{a\}(s) + O'(ads)$	$\frac{1}{\phi} * 1226.66 \exp\left(-\frac{5992.25}{T_s}\right)$
LH CO{b} formation	$O(ads) + C(b) + O'(ads) \rightarrow CO\{b\}(s) + O'(ads)$	$\frac{1}{\phi} * 388.87 \exp\left(-\frac{2174.87}{T_s}\right)$
LH O{a} desorption	$O\{a\}(s) \rightarrow O(g) + (s)$	$0.05 T^2 \exp\left(-\frac{3177.2}{T_s}\right)$
LH CO{a} desorption	$CO\{a\}(s) \rightarrow CO(g) + (s)$	$4485.5 \exp\left(-\frac{1581.4}{T_s}\right)$
LH CO{b} desorption	$CO\{b\}(s) \rightarrow CO(g) + (s)$	$1.20 \exp\left(-\frac{2251.60}{T_s}\right)$

- New 2 phase model for PICA
  - Phase 1: FiberForm – same rates as preform carbon model.
  - Phase 2: PICA char – new rates for same reactions.
- Newly developed PICA model agrees well with the experiments.

# PICA-NuSil oxidation model

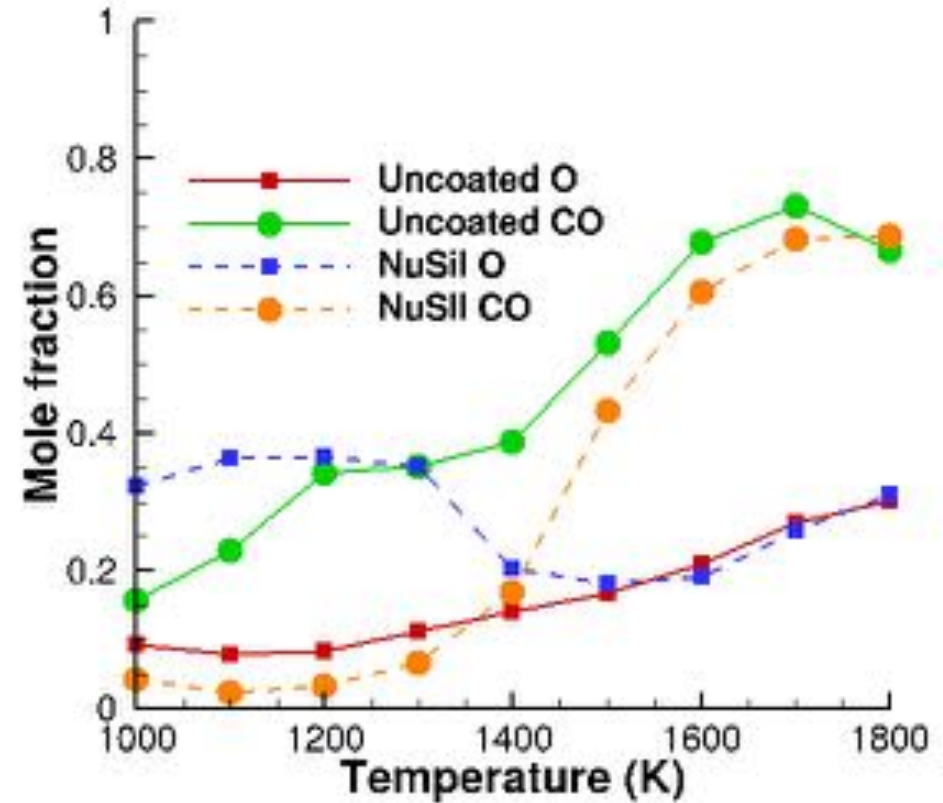


NuSil

Preform carbon

PICA  
CHAR

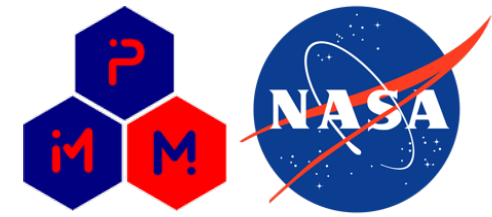
3 phase model



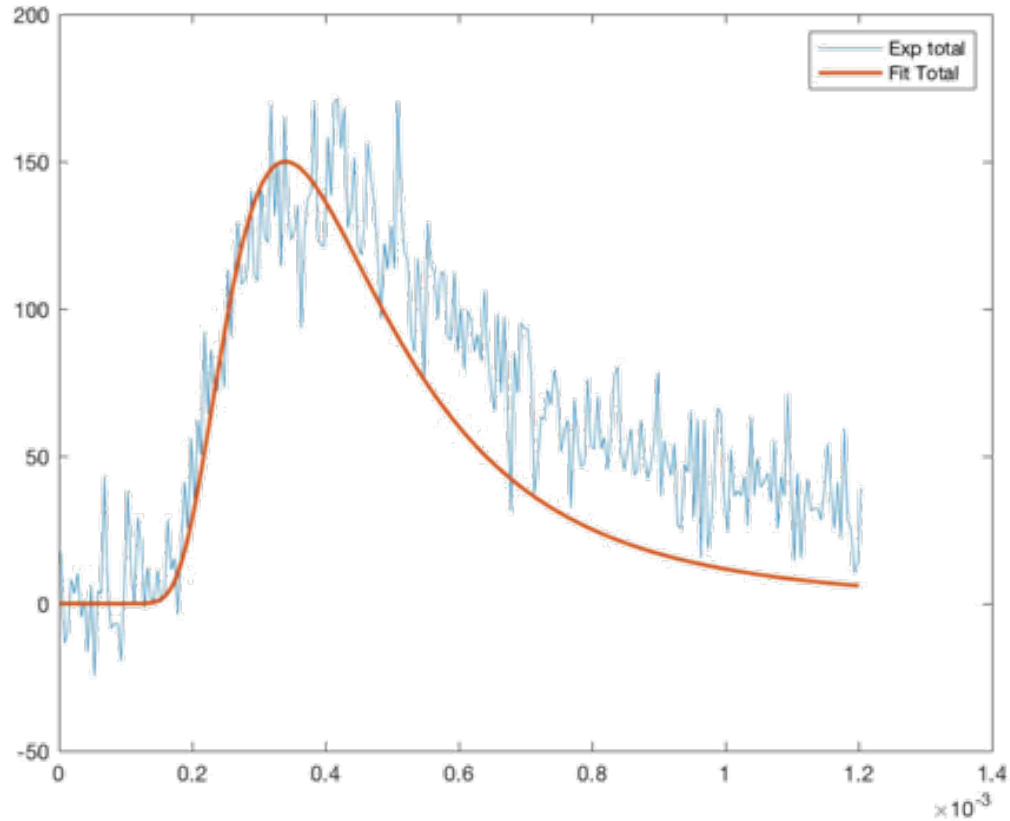
- PICA and FF trends are qualitatively and quantitatively different.
  - Difference in reactivity.
  - Difference in porosity.



# SiO reactions

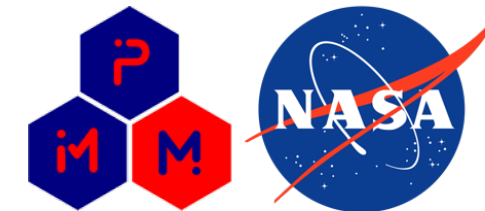


PICA-NuSil T = 1800K SiO TOF

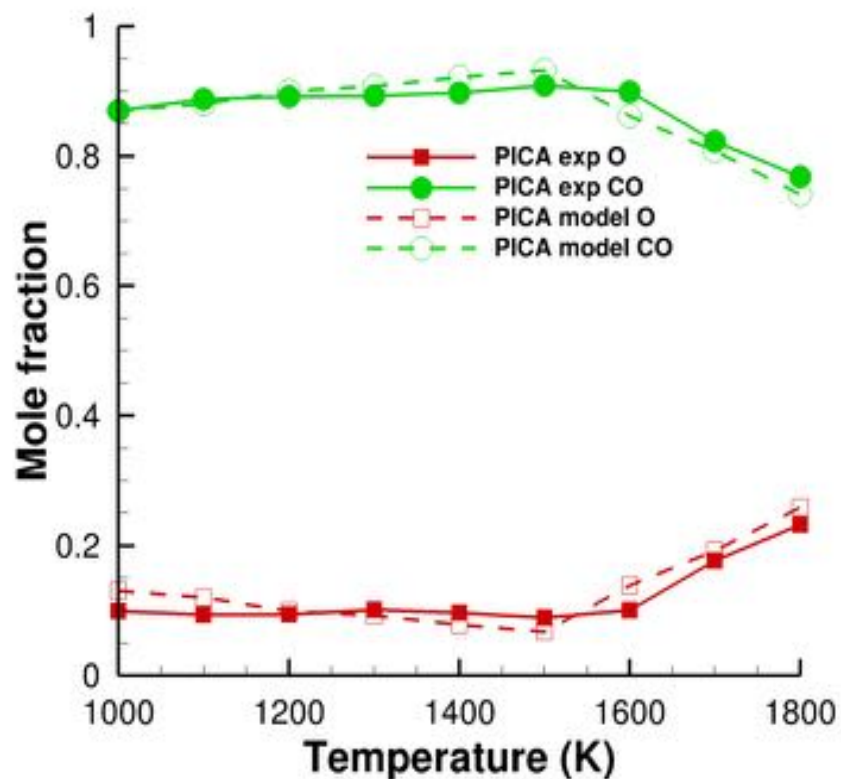


Mechanisms	Reaction
Scattering	$O(g) + Si(s) \longrightarrow O(g) + Si(s)$
TD SiO formation	$O(g) + Si(s) \longrightarrow SiO(g) + (s)$
LH SiO formation	$O(g) + Si(s) \longrightarrow SiO(g) + (s)$

- SiO TOF has a fast TD and slow desorption component.



# PICA-NuSil oxidation model



Mechanisms	Reaction
Scattering	$O(g) + Si(s) \longrightarrow O(g) + Si(s)$
TD SiO formation	$O(g) + Si(s) \longrightarrow SiO(g) + (s)$
LH SiO formation	$O(g) + Si(s) \longrightarrow SiO(g) + (s)$

- New 3 phase model for PICA-NuSil
  - Phase 1: FiberForm – same rates as preform carbon model.
  - Phase 2: PICA char – same rates as PICA 2 phase model.
  - Phase 3: NuSil – new rates for same reactions.
- Newly developed PICA-NuSil model agrees well with the experiments.