# Twin nucleation and growth mechanism in Ni-based superalloys

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# **Deformation mechanisms in Ni-based superalloys**



# **Micro-twinning in Ni-based superalloys**

- Micro-twinning is a dominant creep deformation mechanism in Ni-based superalloys at intermediate temperatures.
- ➤ Many aspects of twin nucleation and growth remain unexplored.
- The Kolbe mechanism for micro-twinning, based on thermally activated reordering, is currently widely accepted in the community to explain these processes.



We propose a qualitatively different mechanism for nucleation and growth of twins.
The proposed mechanism is demonstrated via molecular dynamics simulations.

# **Micro-twinning in Ni-based superalloys**

#### Simulation geometry and procedure:

- Composite simulation system (Ni-Al), containing  $\gamma$  phase (matrix) and  $\gamma$  ' phase (precipitate) regions and two edge dislocation dipoles.
- LAMMPS package; Ni-Al interatomic potential by Mendelev.
- Simulation cell size:  $\sim 100 \times 2.5 \times 29 \text{ nm}^3$  ( $\sim 700,000 \text{ atoms}$ ).
- > PBCs in all directions.
- > The system was relaxed at T = 1000K, using hybrid MC/MD prior to introduction of dipoles.
- The dipoles were positioned in such a way that individual dislocations of upper and lower dislocation pairs would glide on adjacent {111} planes when a  $\sigma_{xz}$  shear stress was applied.
- > The MD deformation simulations were carried out under applied shear stress  $\sigma_{xz} = 800$  MPa.





> The first step of twin nucleation and growth is formation of SISF inside precipitate.

SISF formation (discovered via MD simulation); colored according to lattice structure; Ni atoms are not shown



> The first step of twin nucleation and growth is formation of SISF inside precipitate.



- > The second step of twin nucleation and growth is formation of SESF inside precipitate.
- > Recurring arrival of additional lattice dislocations from  $\gamma$  matrix will lead to growth of twin.

Υ <u>I. ISF(δB) I</u> <u>I. ISF(δB)</u>	$\mathbf{L} \qquad \qquad \mathbf{A} \qquad \mathbf{P} $	a) Arrival of the 1 <sup>st</sup> BC dislocation to the interface
γ	γ'	b) The 2 <sup>nd</sup> BC dislocation, arriving to the interface (on the plane below) "pushes" the 1 <sup>st</sup> one inside t precipitate (as a result, APB is produced).
	<u>APB</u> <u>CSF(<i>bB</i>)</u> <u>b</u> -	
γ	γ' <u>SISF</u> <u>APB</u> <u>CSF(δB)</u> C)	c) In addition, the arrival of the $2^{nd}$ BC dislocation triggers nucleation and emission of A $\delta$ Shockley partial (at the interface, on the glide plane of the $1^{st}$ BC dislocation). This converts APB left by the $1^{st}$ into SISF.
γ 	γ' <u>CSF(δB)</u> <u>SISF</u> <u>APB</u> <u>CSF(δB)</u> <u>d</u>	d) The A $\delta$ Shockley quickly propagates towards the 1 <sup>st</sup> BC dislocation, extending SISF into the precipita

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e) Following the same logic, arrival of another BC dislocation one plane above the configuration depicted in d) will trigger the repetition of the processes described in c)-d) on the corresponding planes.

f) In particular, the arrival of the  $3^{rd}$  BC dislocation will trigger nucleation of yet another A $\delta$  Shockley leading to formation of SESF embryo.

g) Propagation of this A $\delta$  Shockley into precipitate will remove APB left by the 2<sup>nd</sup> BC dislocation and extend the SESF into precipitate.

h) Recursive arrival of additional BC dislocations on the planes above the configuration depicted in g) will lead to formation and growth of the twin.

#### **Comparison with the Kolbe mechanism**

In the Kolbe mechanism twin can grow only due to passage of two Shockley partials that create 2-layer CSF that can be reordered. The process adds two layers to the twinned structure. The reordering is the rate limiting process.



matrix ( $\gamma$ ) strengthening precipitate ( $\gamma'$  phase) matrix ( $\gamma$ ) strengthening precipitate ( $\gamma'$  phase) In the proposed mechanism reordering is not involved. The rate limiting process is nucleation of Shockley partial. Single step adds one layer.



matrix ( $\gamma$ ) strengthening precipitate ( $\gamma'$  phase)

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# Conclusions

- We propose a new twin nucleation and growth mechanism in Ni-superalloys.
- It is qualitatively different from the Kolbe mechanism.
- The rate-limiting process in the Kolbe mechanism is thermally activated reordering (via vacancy diffusion).
- The rate-limiting process of the new mechanism is nucleation of the Shockley partial.
- Studying rate-limiting processes of both mechanisms will allow us to propose changes in alloy composition, that would make the material stronger.



The first step of new twin growth mechanism: formation of super intrinsic stacking fault (SISF)



Schematics of the discovered twin growth mechanism