



A Machine Learning-Derived Atomistic Potential for $Y_2Si_2O_7$

Cameron J. Bodenchatz¹, Wissam A. Saidi^{2,3}, and Jamesa L. Stokes¹

¹NASA Glenn Research Center, Cleveland, OH

²National Energy Technology Laboratory, Pittsburgh, PA

³University of Pittsburgh, Pittsburgh, PA

Funding provided by the NASA Transformational Tools & Technologies (TTT) Project

Materials Science & Technology 2022

Pittsburgh, PA

October 10th, 2022

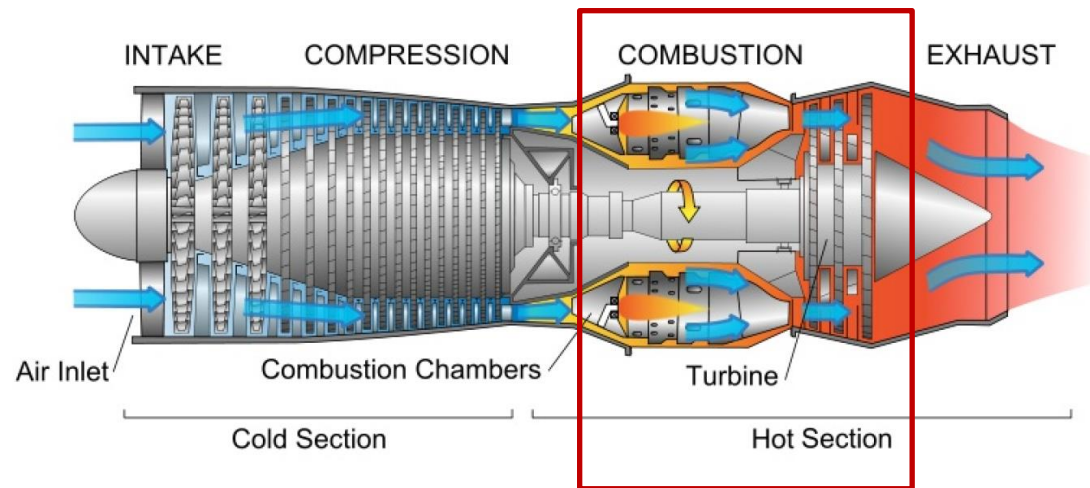


Outline

- Background
- Simulation Methods – Machine Learning-based Interatomic Potential for Molecular Dynamics (MLMD)
- Crystal Cell Optimization Calculations
- Thermochemical Property Calculations
- Thermomechanical Property Calculations – CTE
- Conclusions

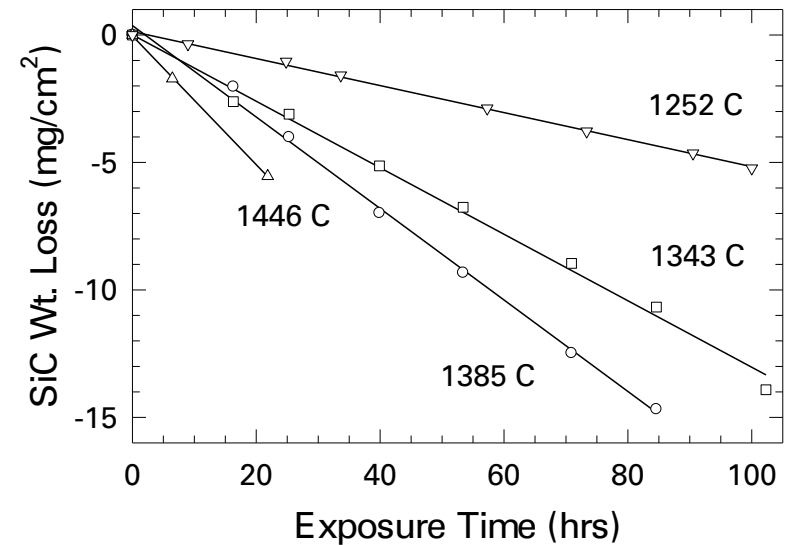
SiC/SiC Ceramic Matrix Composite (CMC) Components

- Replace metal alloy engine components with SiC/SiC ceramic matrix composites (CMCs)
- Increased efficiency and cost savings
 - Higher temperature stability
 - Lower density
- CMCs can degrade under O_2 and H_2O environments at high temperature ($>800^\circ C$)



Target: 1482°C

Weight loss of SiC in High-Pressure Jet Fuel Burner (6 atm, 20 m/s)



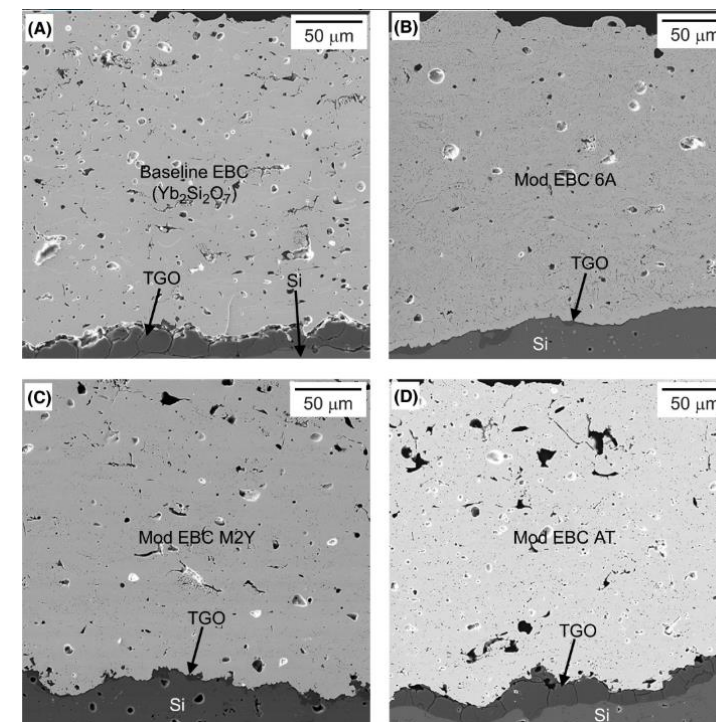
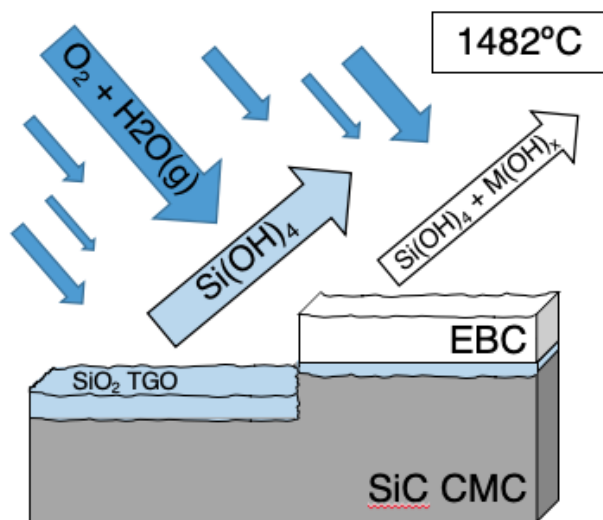
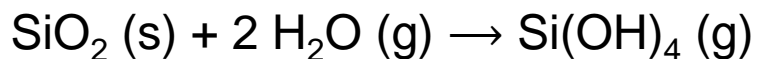
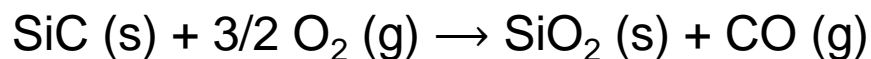
Environmental Barrier Coatings (EBCs) for SiC/SiC CMCs

- CMCs can recess under O₂ and H₂O environments at high temperature (>800°C)
- Rare-Earth (RE) disilicates (RE₂Si₂O₇) are a promising class of EBCs
- EBCs can prevent H₂O diffusion
- Some formation of SiO₂ TGO still occurs with EBC



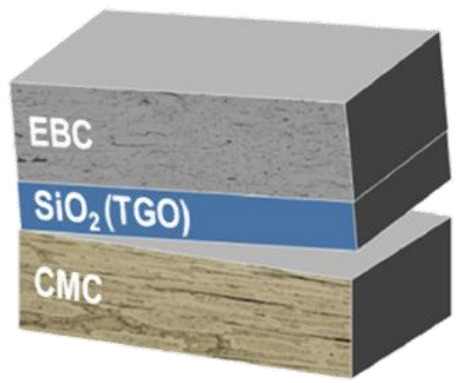
EBC-coated turbine vanes

EBC chemistry affects TGO growth and CMC recession

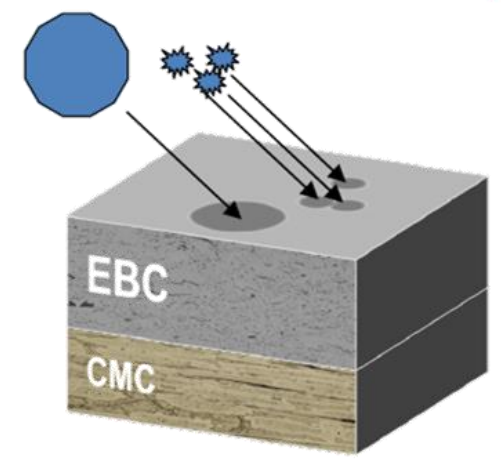


1000 h/1000 cycles at 1316°C in 90% H₂O + 10% O₂

EBC Failure Modes

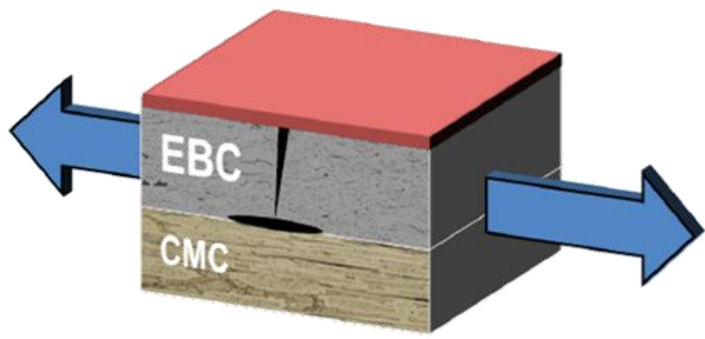
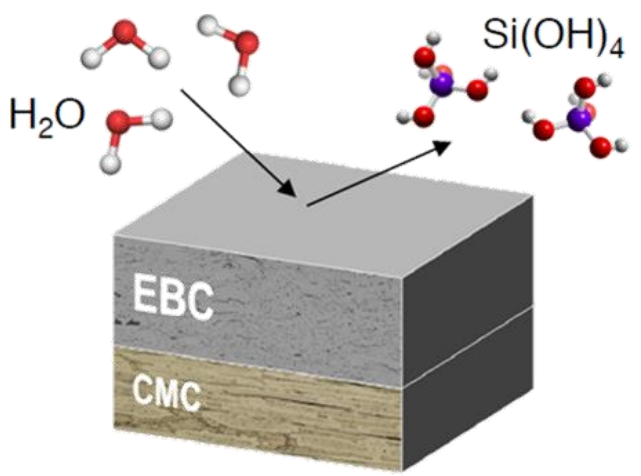


EBC lifetime and design requirements determined by combination of extrinsic failure modes

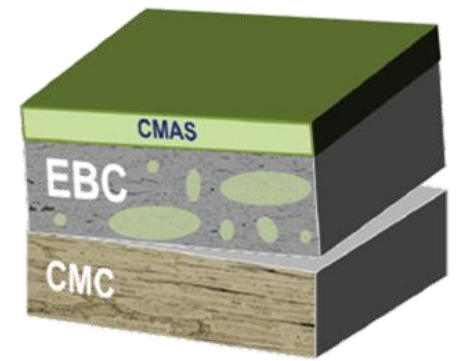


Steam Oxidation

Erosion and FOD



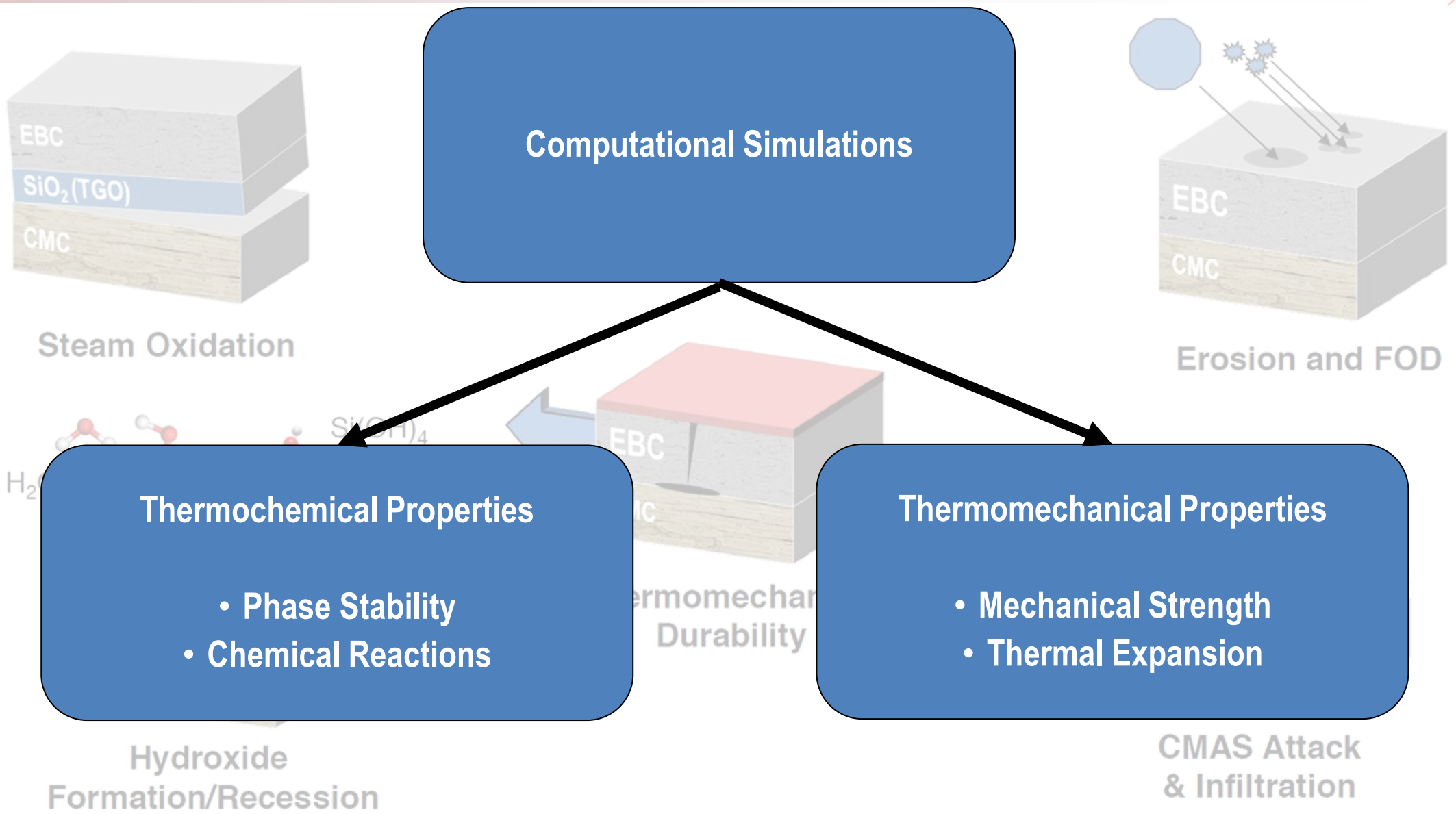
Thermomechanical Durability



CMAS Attack & Infiltration

Hydroxide Formation/Recession

EBC Failure Modes



Atomic-scale simulation methods for property calculations



DFT

- Electron-level theory
 - Cost scales with number of electrons
- Smaller simulation cells
 - Self-interaction errors?

Classical Molecular Dynamics

- Atom-level theory
 - Cost scales with number of atoms
- Bonding based on empirical trend fits
 - Generally, no bond breaking/formation

	DFT	Classical MD
Pros	<ul style="list-style-type: none">• Higher accuracy• Generalizability	<ul style="list-style-type: none">• Time<ul style="list-style-type: none">• ~Hours• System Size<ul style="list-style-type: none">• Larger cells possible (~1,000s-10,000s of atoms)
Cons	<ul style="list-style-type: none">• Time<ul style="list-style-type: none">• ~Days-Weeks• System Size<ul style="list-style-type: none">• Smaller unit cells (~100s of atoms)	<ul style="list-style-type: none">• Requires pre-parameterized potential<ul style="list-style-type: none">• Not as generalizable



Atomic-scale simulation methods

DFT

- Electronic structure calculations
 - Costly (scales as N^3 electrons)
- Smaller simulation cells
 - Self-interaction errors?

Accuracy

Classical Molecular Dynamics

- Atom-level simulation
 - Costly (scales as N^2 atoms)
- Bonding based on empirical trend fits
 - Generally no bond breaking/formation

Cost

Property Calculations

	DFT	Classical MD
Pros	<ul style="list-style-type: none"> • Higher accuracy • Generalizability 	<ul style="list-style-type: none"> • Time <ul style="list-style-type: none"> • ~Hours • System Size <ul style="list-style-type: none"> • Larger unit cells possible (~1,000s-10,000s of atoms)
Cons	<ul style="list-style-type: none"> • Time <ul style="list-style-type: none"> • ~Days-Weeks • System Size <ul style="list-style-type: none"> • Smaller unit cells (~100s of atoms) 	<ul style="list-style-type: none"> • Requires pre-parameterized potential <ul style="list-style-type: none"> • Not as generalizable



Machine learning approaches for crystalline materials

Direct Property Calculations

- Train ML model to directly predict property
- ML algorithm suitable for problem
 - Regression vs. Clustering
- Descriptors to capture system information
 - Crystal structure information
 - Atomic composition
 - Environmental variables
 - Temperature, Pressure, etc.
- Example: Neural network to predict CTE for rare-earth disilicates¹

Indirect Property Calculations

- Train ML model to run simulations
 - Interatomic potentials
- Regression algorithms, typically
 - Almost always neural networks
- Descriptors often include atomic neighborhood information
 - Nearest-neighbor atoms within cutoff
 - Bonding information
- Example: NN-based interatomic potential for HfO₂

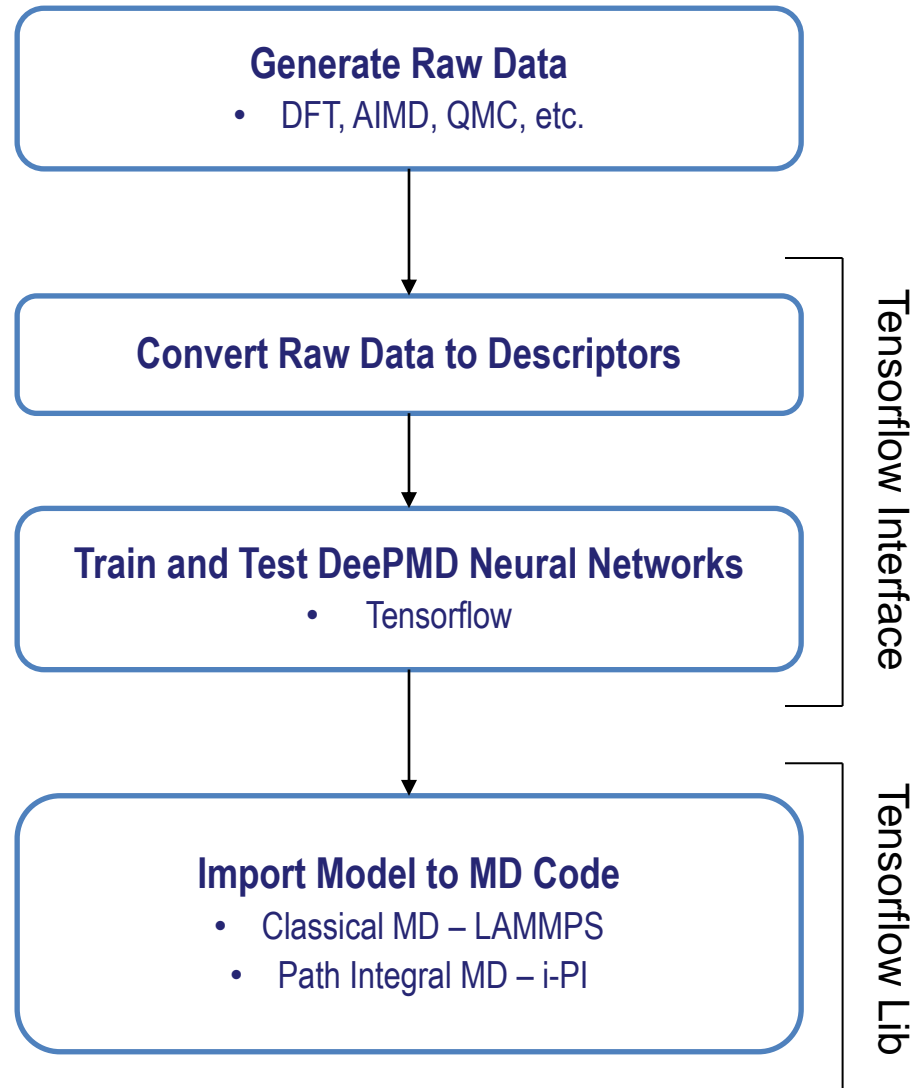
[1] Ayyasamy, et al. *J Am Ceram Soc.* (2020)

[2] Wu, et al. *Phys Rev B.* (2021)



DeePMD theory

- Deep Neural Network using descriptors to preserve translational, rotational, and permutational symmetries
- Descriptors dependent on atomic neighbor environment
 - Full information (radial and angular) included for first- and second-nearest neighbors
 - Radial information only for other atoms inside the user-defined cutoff radius
- Training data obtained from ab initio molecular dynamics (AIMD) simulations using DFT
- DNN used to calculate atomic energies; total energy is the sum of atomic energies.
- Forces and virial calculated using gradient of energy w.r.t. positions





Molecular Dynamics: Minimization and Lattice Constants

Phase	Theory	a	b	c	α	β	γ
β C2/m	DFT (PBE)	6.91	9.06	4.78	90	101.97	90
	DFT (PBEsol)	6.85	8.97	4.74	90	101.79	90
	MLMD (PBE*)	6.90	9.05	4.77	90	101.95	90
	Expt ¹	6.88	8.97	4.72	90	101.70	90
γ P2 ₁ /c	DFT (PBE)	4.75	10.90	5.63	90	96.18	90
	DFT (PBEsol)	4.71	10.81	5.57	90	95.98	90
	MLMD (PBE*)	4.74	10.88	5.62	90	96.14	90
	Expt ²	4.69	10.84	5.58	90	96.03	90
	Expt ³	4.69	10.86	5.59	90	96.01	90
	Expt ⁴	4.66	10.78	5.54	90	96.06	90
δ Pna2 ₁	DFT (PBE)	13.80	5.09	8.20	90	90	90
	DFT (PBEsol)	13.62	5.03	8.12	90	90	90
	MLMD (PBE*)	13.77	5.07	8.19	90	90	90
	Expt ⁵	13.81	5.02	8.30	90	90	90
	Expt ⁴	13.69	5.02	8.17	90	90	90

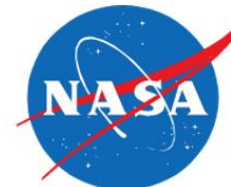
[1] Liddell, et al. *J Br Ceram Soc.* (1986)

[2] Dolan, et al. *Powder Diff.* (2008)

[3] Leonyuk, et al. *J Cryst Growth.* (1999)

[4] Smolin, et al. *Acta Cryst B: Struct Cryst Cryst Chem.* (1970)

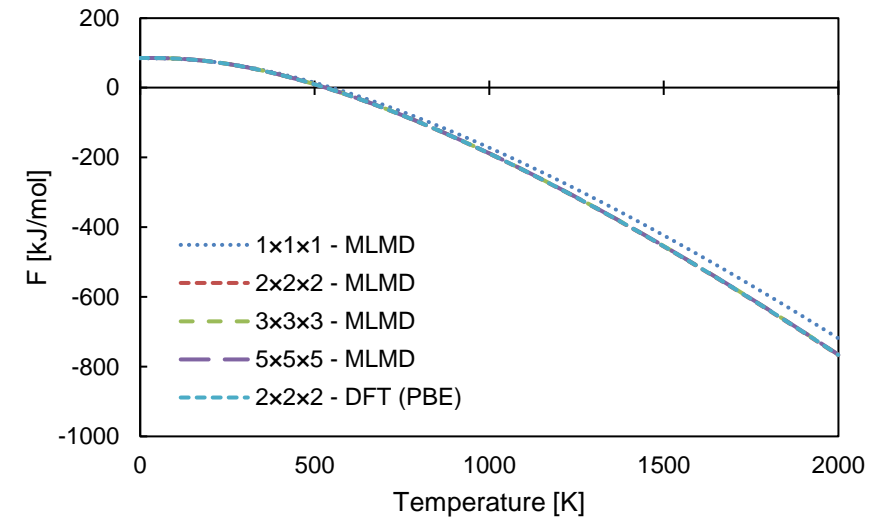
[5] Dias, et al. *Zeitschrift für Krist.* (1990)



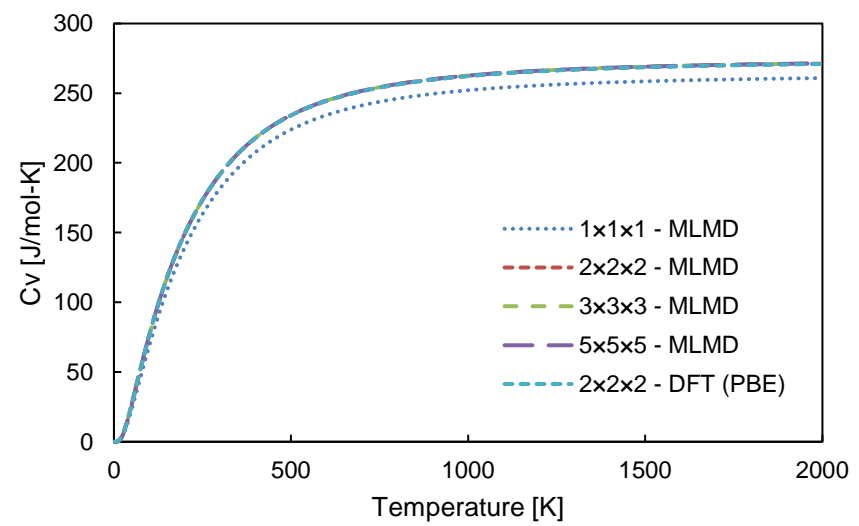
Finite-Difference Phonon Calculations

- Consistent results with DFT across supercell sizes
 - $1 \times 1 \times 1$ (22 atoms)
 - $2 \times 2 \times 2$ (176 atoms)
 - $3 \times 3 \times 3$ (594 atoms)
 - $5 \times 5 \times 5$ (2,750 atoms)
- Slight deviation between $1 \times 1 \times 1$ cell in MLMD compared to DFT

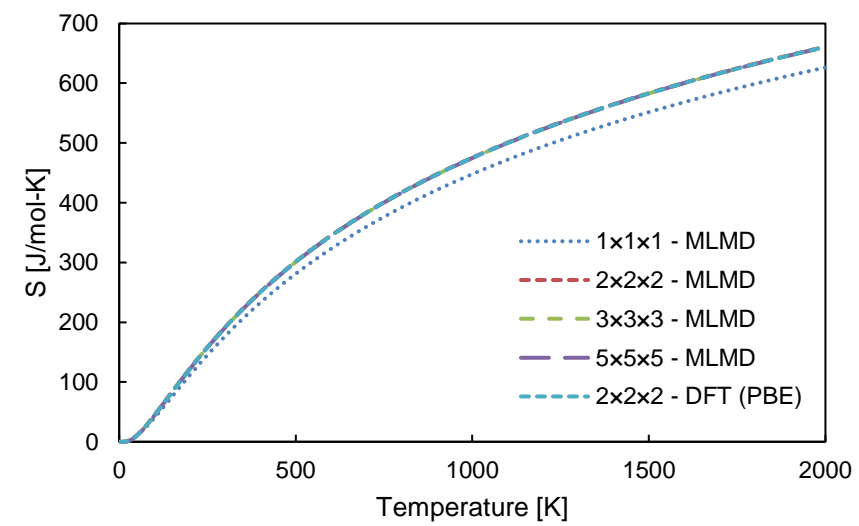
β -Y₂Si₂O₇ Free Energy



β -Y₂Si₂O₇ Heat Capacity



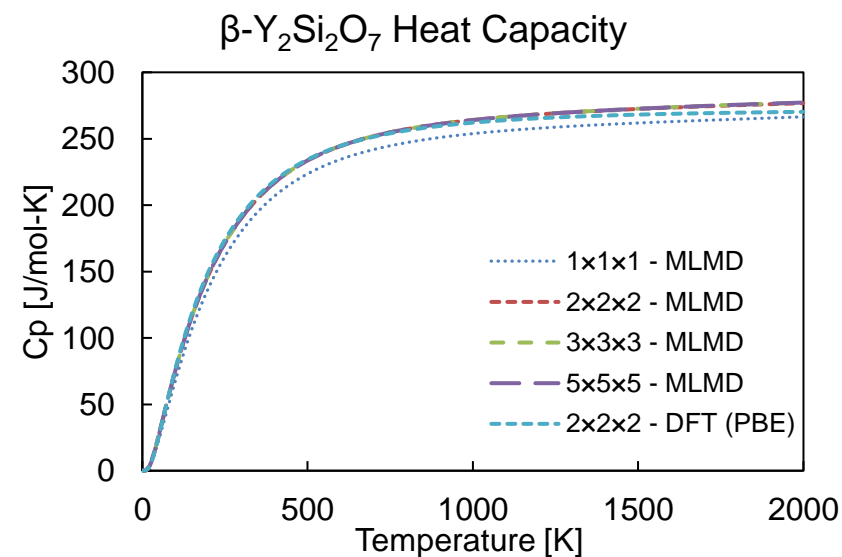
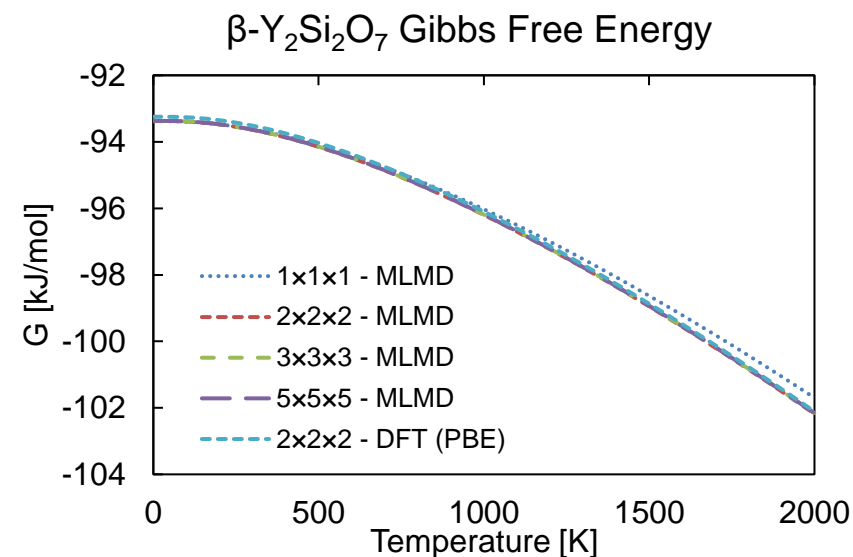
β -Y₂Si₂O₇ Entropy





Quasi-Harmonic Approximation Phonon Calculations

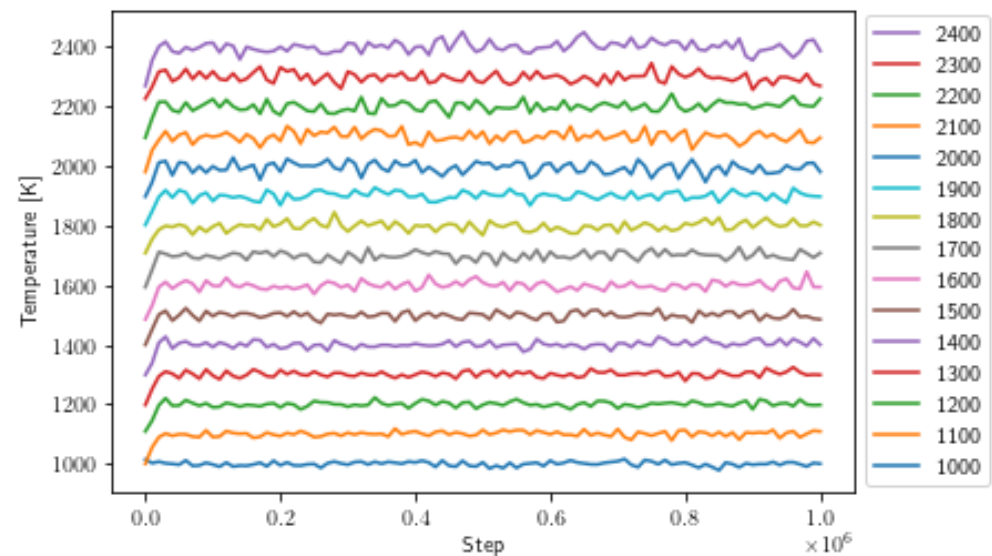
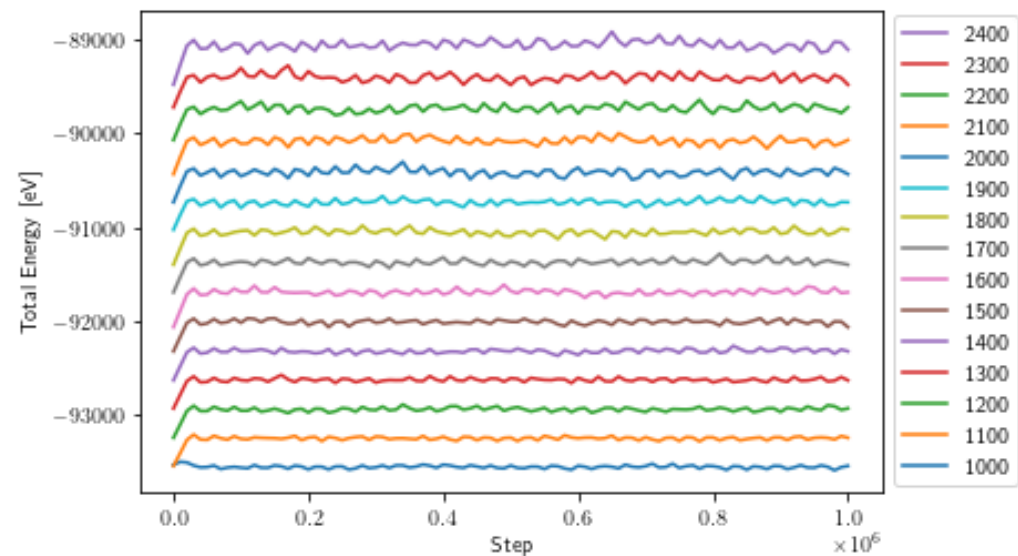
- Quasi-harmonic approximation considers non-equilibrium volume contributions
 - Span of volumes from 94% to 106% of equilibrium
- Slight deviation between $1 \times 1 \times 1$ cell in MLMD compared to DFT
- Can provide thermochemical properties required for phase stability and formation calculations
 - CALPHAD via Thermo-Calc software
- Larger supercell capabilities could enable studies of more complicated properties, simulation techniques, etc.





Long-duration and large-cell molecular dynamics

- Supercell size increased to $8 \times 8 \times 8$ conventional unit cells
 - 11,264 atoms
- Total energy and temperature stable over a 2 μ s simulation
- Cost-prohibitive cell size and timeframe for DFT



Timestep = 2 ps, Total Simulated Time = 2 μ

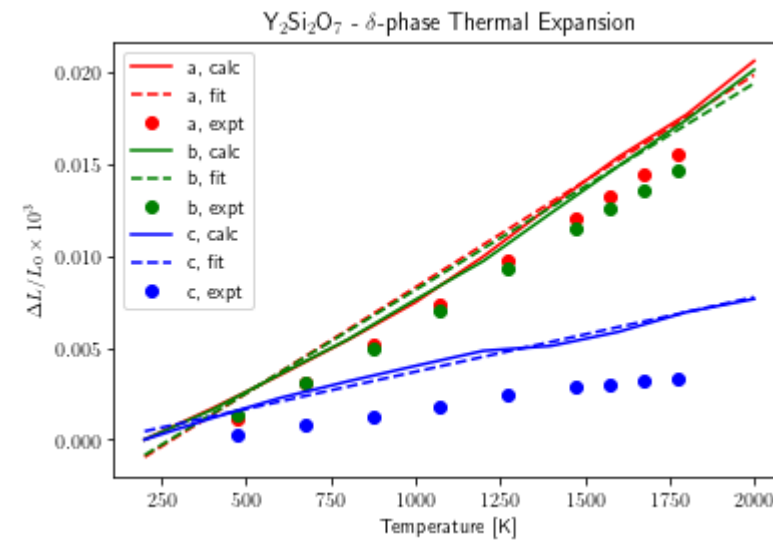
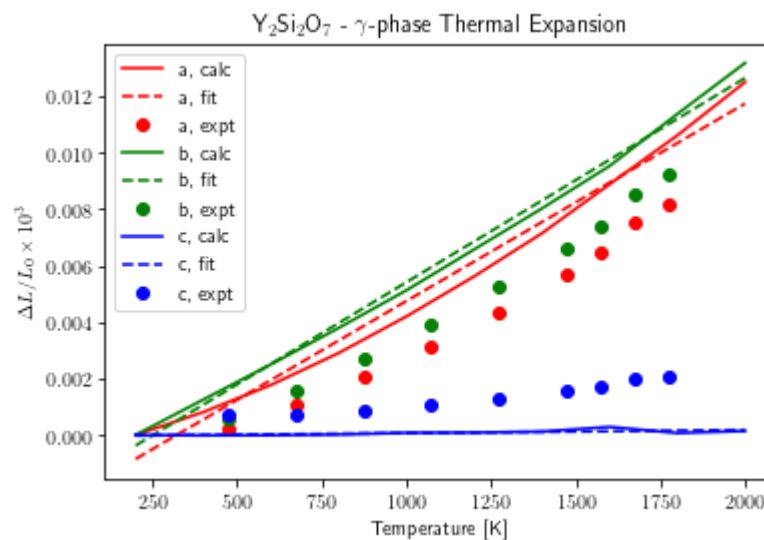
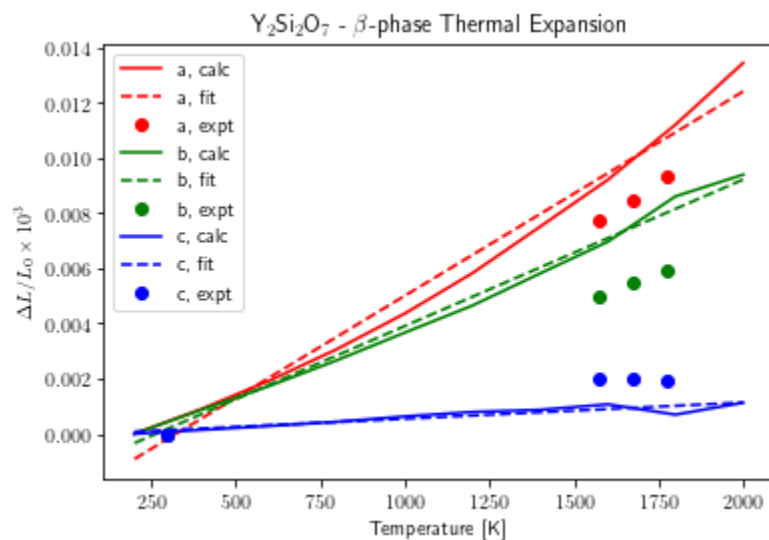


CTE calculations as function of temperature

- Good agreement between MLMD simulations and experiment¹
- Overprediction of *a* and *b* CTE, underprediction of *c* CTE

	CTE ($\times 10^6 \text{ K}^{-1}$)		
	β	γ	δ
<i>a</i>	7.4	0.1	4.1
<i>b</i>	5.3	7.2	11.5
<i>c</i>	0.6	7.0	11.2
Avg Bulk	4.4	4.8	8.9

Table 1: Linear lattice CTE and average bulk CTE of $\text{Y}_2\text{Si}_2\text{O}_7$ crystal phases



[1] Dolan, et al. Powder Diff. (2008)



Conclusions

- Machine learning was used to develop an interatomic potential for YDS
 - Potential was successfully used to calculate various properties of YDS including CTE
 - Results from MD match well with results from DFT
- MLMD enables calculation of phonon vibrational frequencies at similar accuracy but at substantial faster timeframes as compared to DFT (~hours vs. ~weeks)
- MLMD enables long-duration and/or large cell calculations that would be cost prohibitive with DFT
 - Simulations on the order of 10-100 ps were successfully completed
 - Simulations including $8 \times 8 \times 8$ conventional unit cells (11,264 atoms) of β -YDS were successfully completed
- Training of MLMD potentials is resource intensive, and training of potentials for additional materials is required. However, training is a one-time expense (per material), and potentials can be flexibly used in additional calculations.



Acknowledgements

- Environmental Effects & Coatings Branch at NASA Glenn Research Center
 - Dr. Bryan Harder – Wednesday, 8:00 AM in 335 David L. Lawrence Convention Center
 - Oxidation and Erosion Implications of CMAS on Environmental Barrier Coatings

- NASA Pleiades Supercomputer Cluster

- Funding
 - NASA Transformational Tools and Technologies (TTT) project
 - NASA Transformative Aeronautics Concepts Program (TACP)
 - NASA Aeronautics Research Mission Directorate (ARMD)