

# Understanding thermal stability in zirconia-based aerogels

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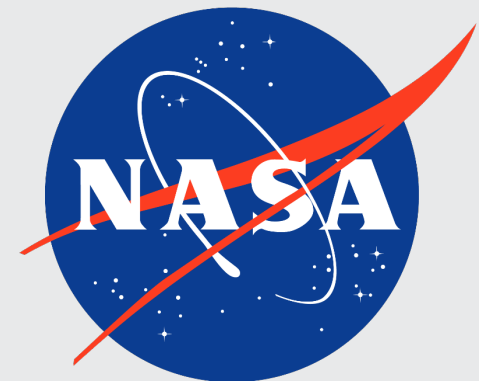
\* Retired

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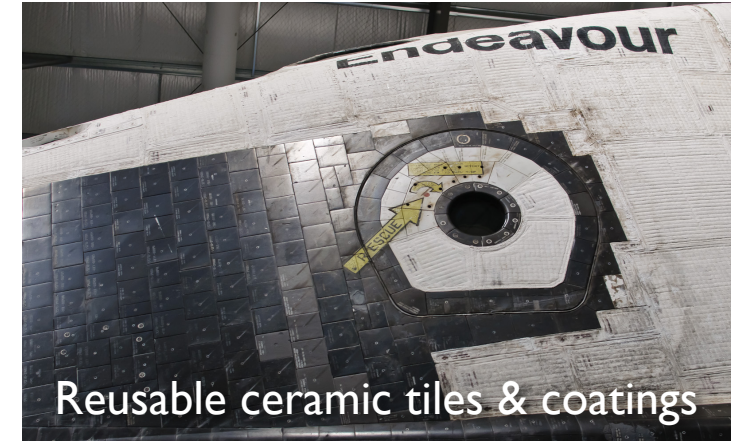
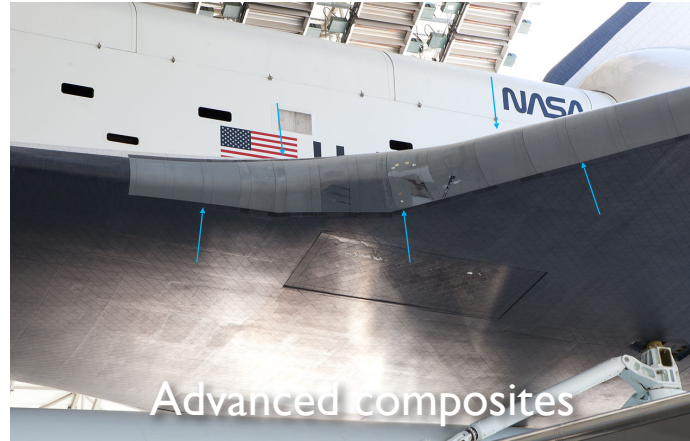
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# Developing lightweight, high-performance aerospace thermal protection systems (TPS)



## TPS Needs:

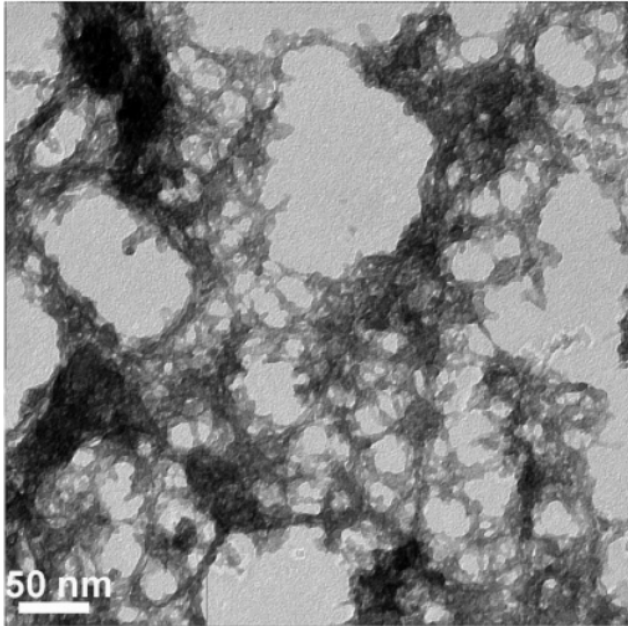
- Manage heat loads
- Withstand mechanical loads
- Lightweight
- Reusable when possible

## Our Aims:

Reduce **thermal conductivity** to improve performance.

Reduce **mass/volume** to lower costs.

# Aerogels are highly insulating and lightweight materials



Highly porous structure of aerogel is responsible for its extremely low thermal conductivity.

Low density = Low solid conductivity

Pore sizes  $\leq$  mean free path of gas  
= Low gas convection

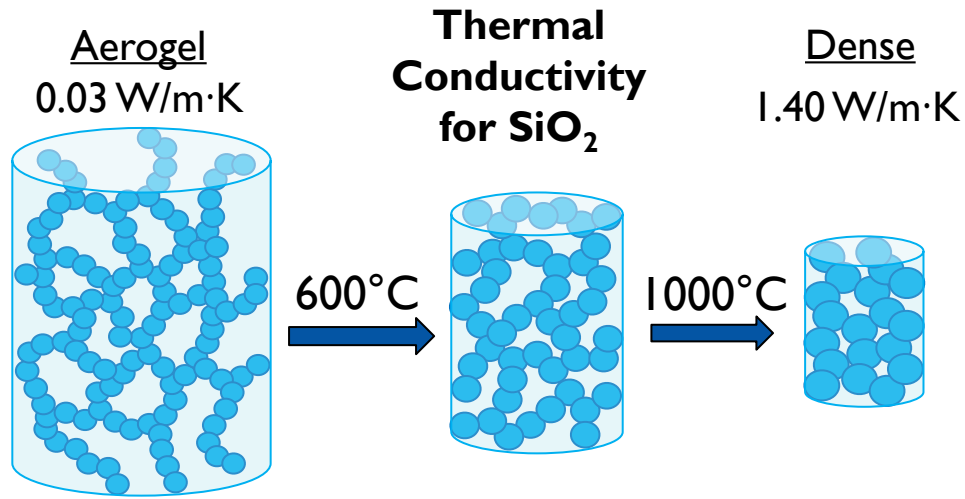
**High SSA:** 200 to 1000 m<sup>2</sup>/g

**High Porosity:** 90 to 99.9%

**Low Density:** 0.2 to 0.05 g/cm<sup>3</sup>

**Low thermal conductivity:**  
0.009 W/(m•K) in atmosphere  
and 0.003 W/(m•K) under vacuum

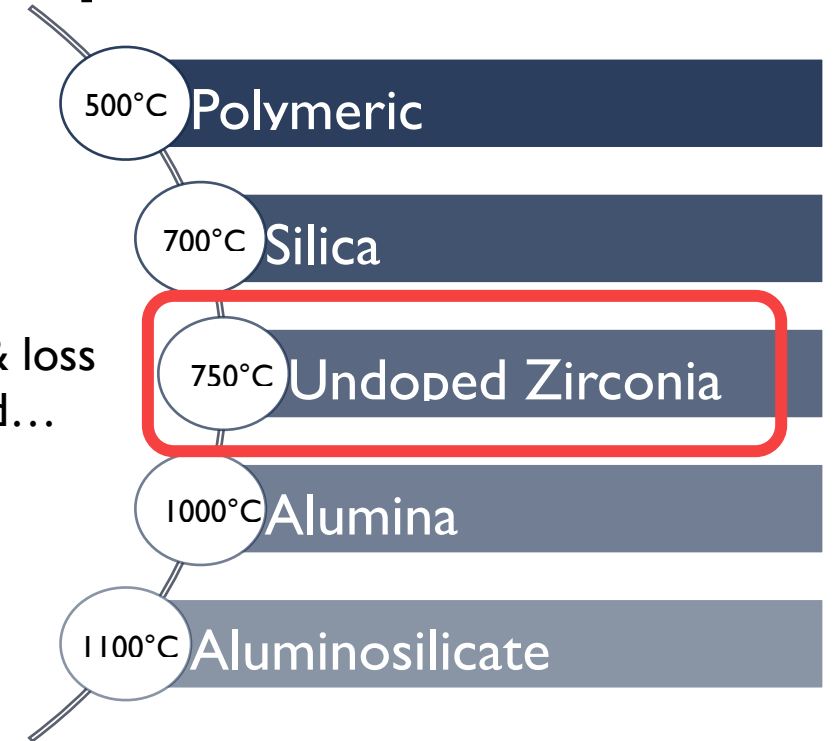
# Collapse of pore structure and loss of favorable properties occurs upon thermal exposure



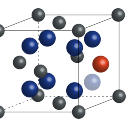
Loss of SSA, porosity  
↑ thermal conductivity,  
cracking, and shrinkage

Large SSA & porosity contribute  
to driving force for densification

Rapid densification & loss  
of porosity beyond...

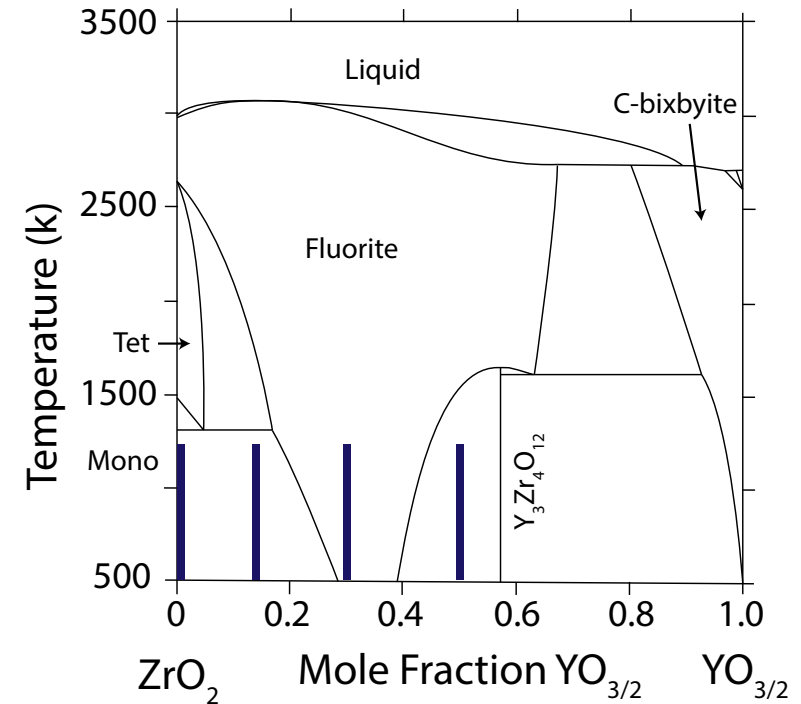


Develop aerogel to maintain **porosity** at high  
temperatures ( $\geq 1200^{\circ}\text{C}$ ) for use as  
insulation in next-gen aerospace applications



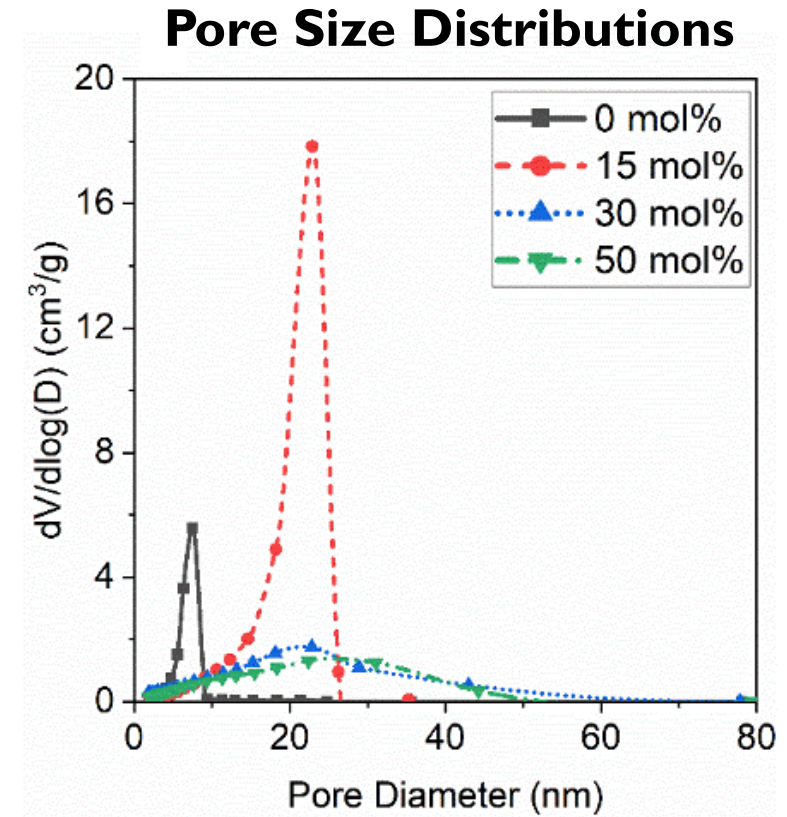
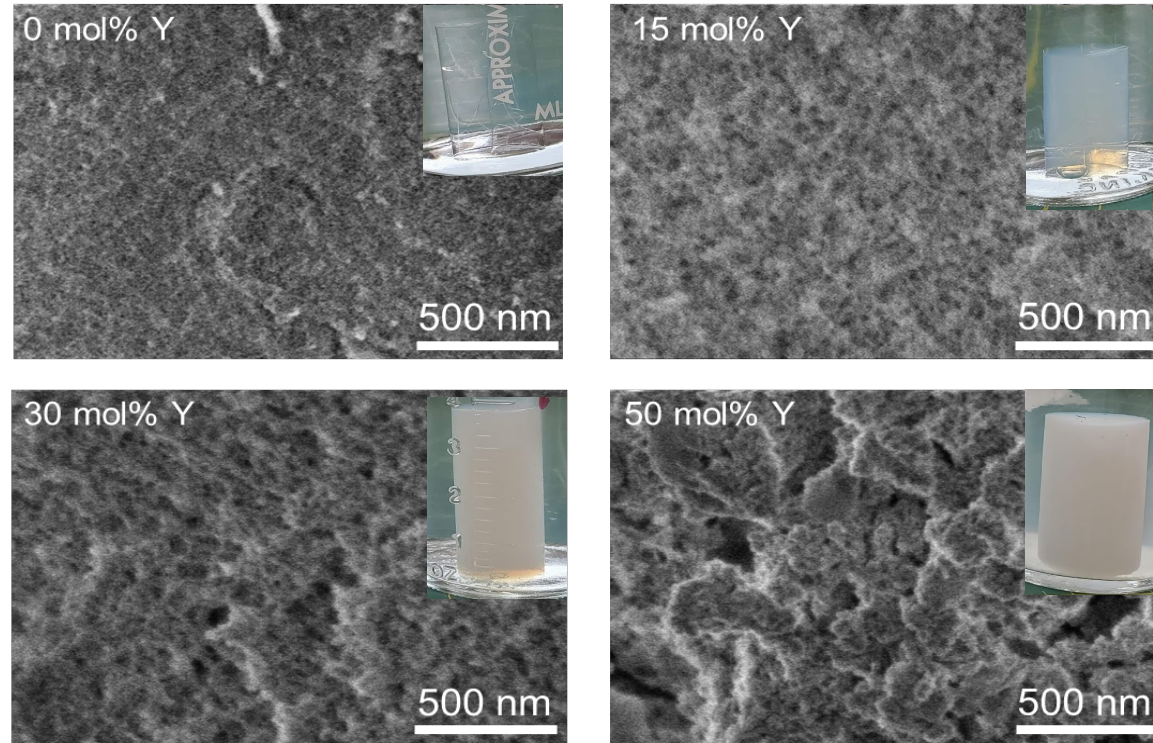
# Yttria-stabilized zirconia as a candidate composition for a thermally stable aerogel

- YSZ well-known for use as a thermal barrier coating for superalloys in turbine engines
    - *Low thermal conductivity* of 0.8 to 2.9 W/(m•K)
    - $Y_2O_3$  doping *inhibits phase transformations*
- (1) How/why does yttria change the **as-dried structure**?
  - (2) How does yttria influence phase behavior & **structural evolution**?
  - (3) Does yttria improve the **thermal stability** of YSZ aerogels? If so, how?



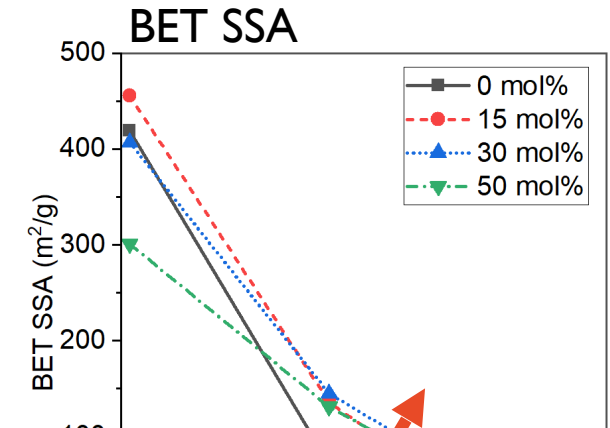
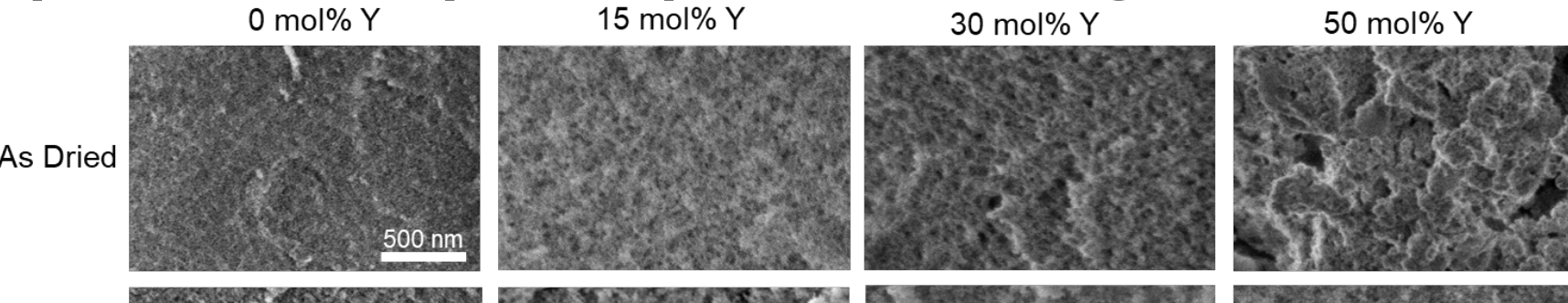
**Selected compositions for study:**  
0, 15, 30, and 50 mol% YO<sub>1.5</sub>

# As dried aerogels: yttria increases pore size and distribution breadth

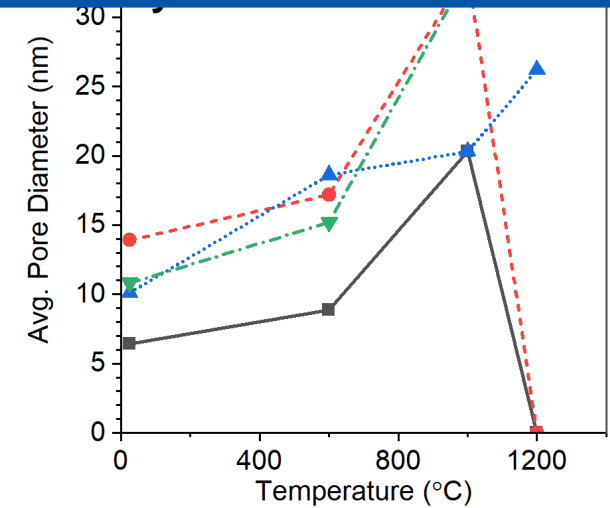
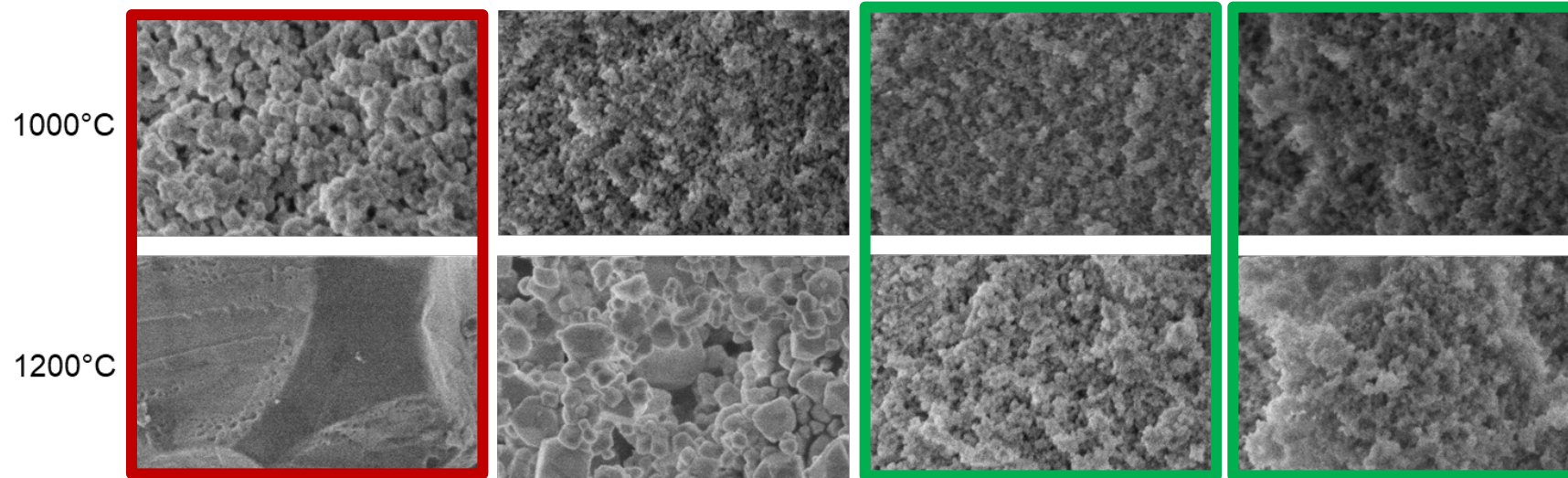


mol % $YO_{1.5}$	SSA ( $m^2/g$ )	Pore Volume ( $cm^3/g$ )	$D_{peak}$ (nm)	Shrinkage (%)	Bulk Density ( $g/cm^3$ )
0	419	0.986	7.5	-25.6	0.292
15	456	1.950	22.9	-21.1	0.250
30	407	1.190	22.8	-13.2	0.193
50	301	0.997	28.5	-	-

# Increased yttria content reduces densification & pore collapse upon heating to 1200°C



6 Is the improved resistance to densification a result of yttria concentration or the as dried structure?



# As dried structure is tunable via synthetic parameters of SL and WC



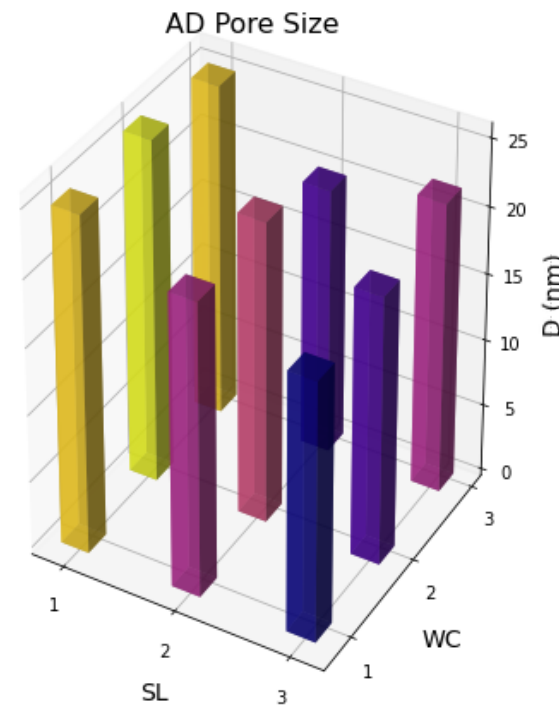
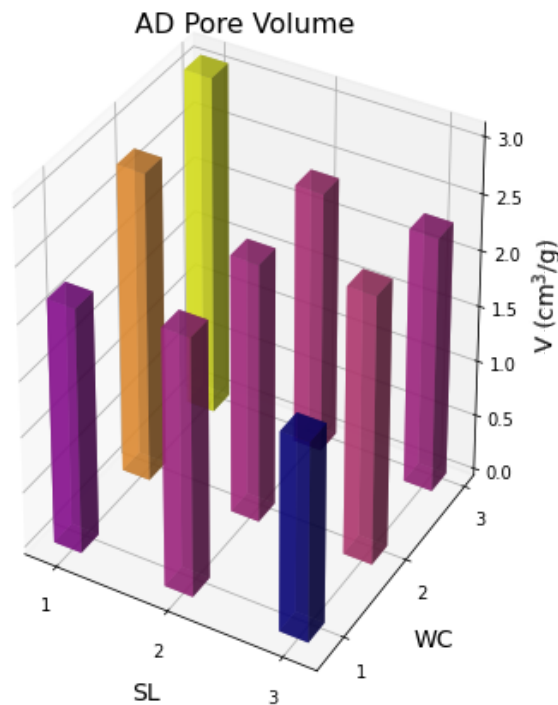
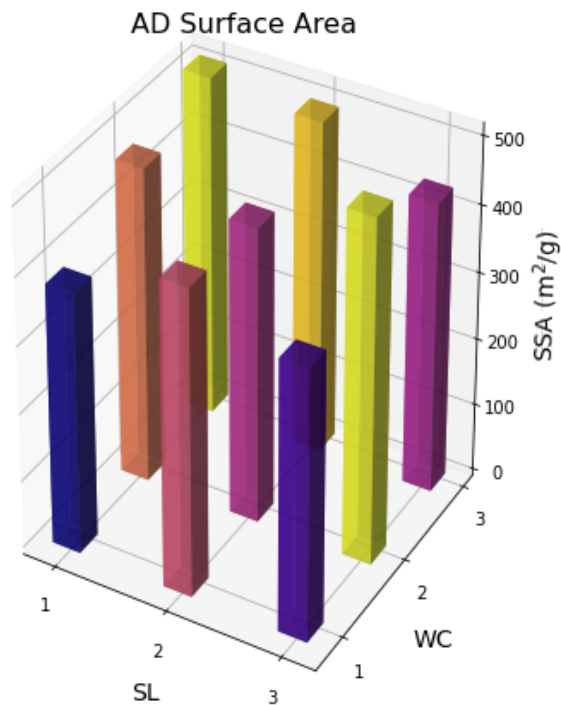
Finely adjust as dried structure independently of composition

All 30 mol%  $\text{YO}_{1.5}$  in  $\text{ZrO}_2$

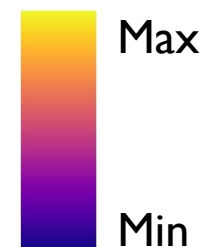
Water Content

	L	M	H
Solids Loading	LS-LW	LS-MW	LS-HW
	MS-LW	MS-MW	MS-HW
	HS-LW	HS-MW	HS-HW

Water content = mmol water added / mmol metal  
Solids loading = mmol metal / mL solvent



In general, **smaller SSA, pore size and pore volume** with:  
increased solids loading  
*and*  
decreased water content

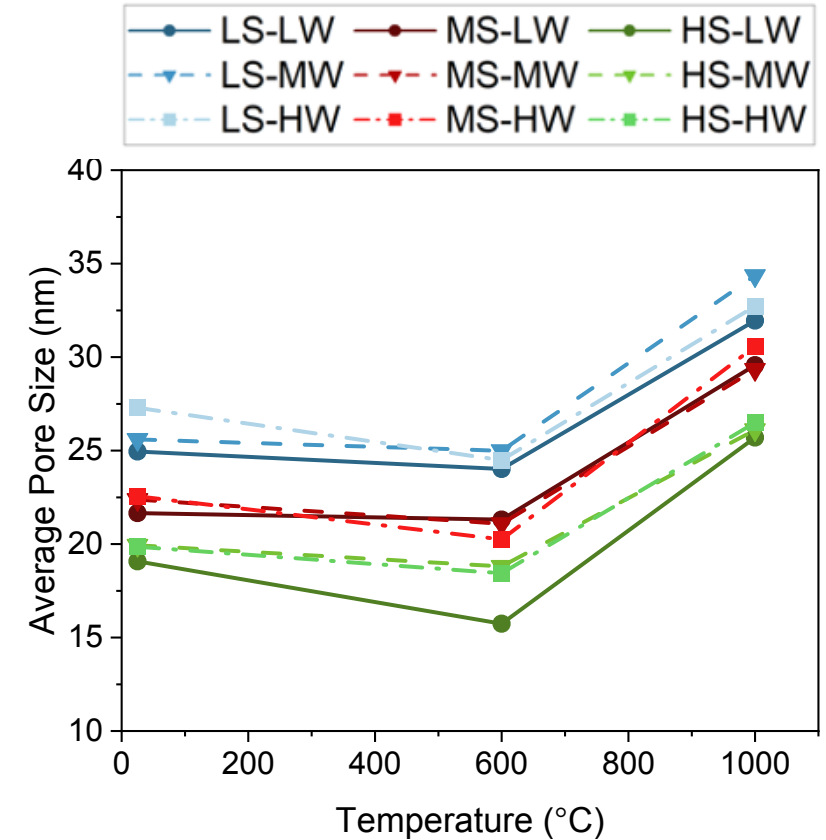
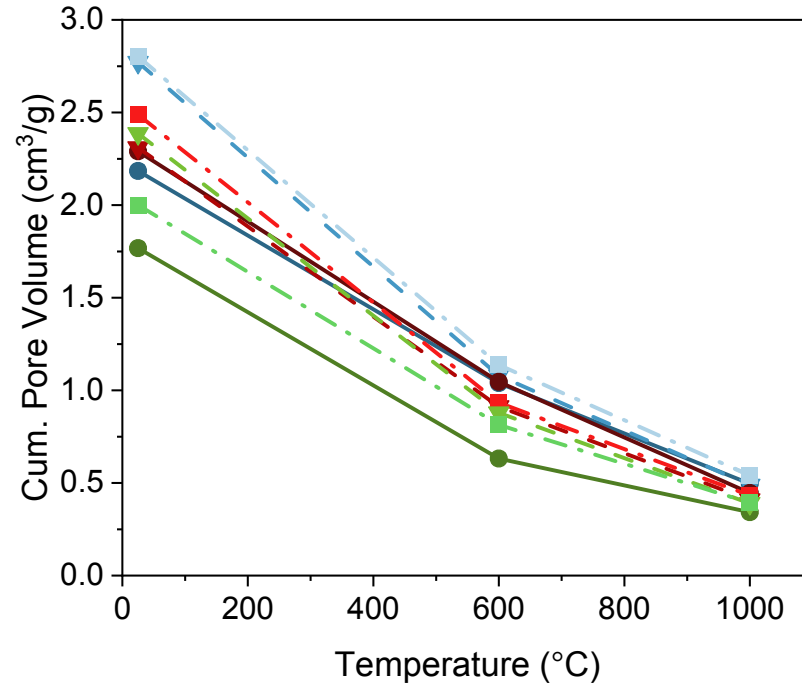
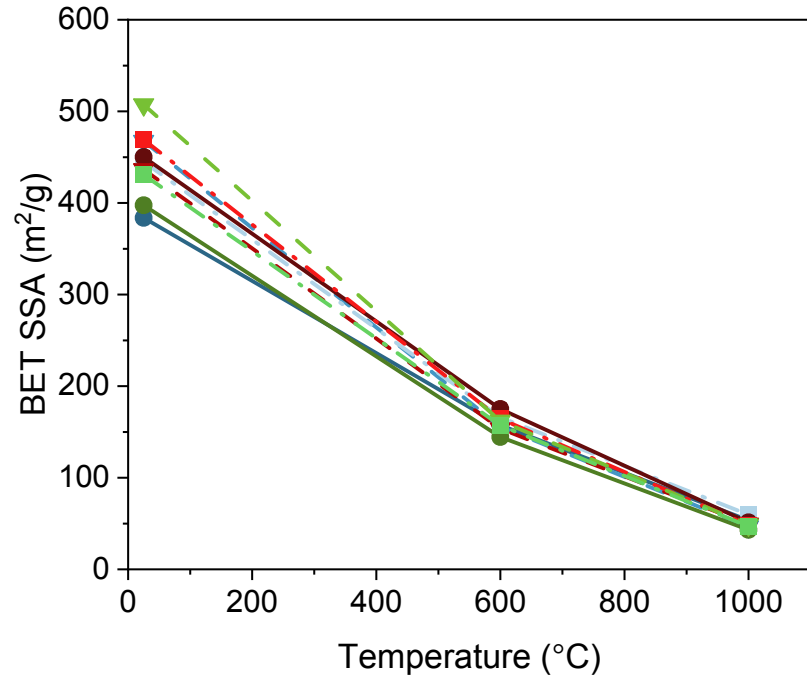


Same color = same solids loading  
Same line style = same water content

Work performed by Jordan Meyer (UIUC MatSE U-Grad)  
Meyer, J., Olson, N.S., et al. (2023). Under review at *Journal of the American Ceramic Society*.



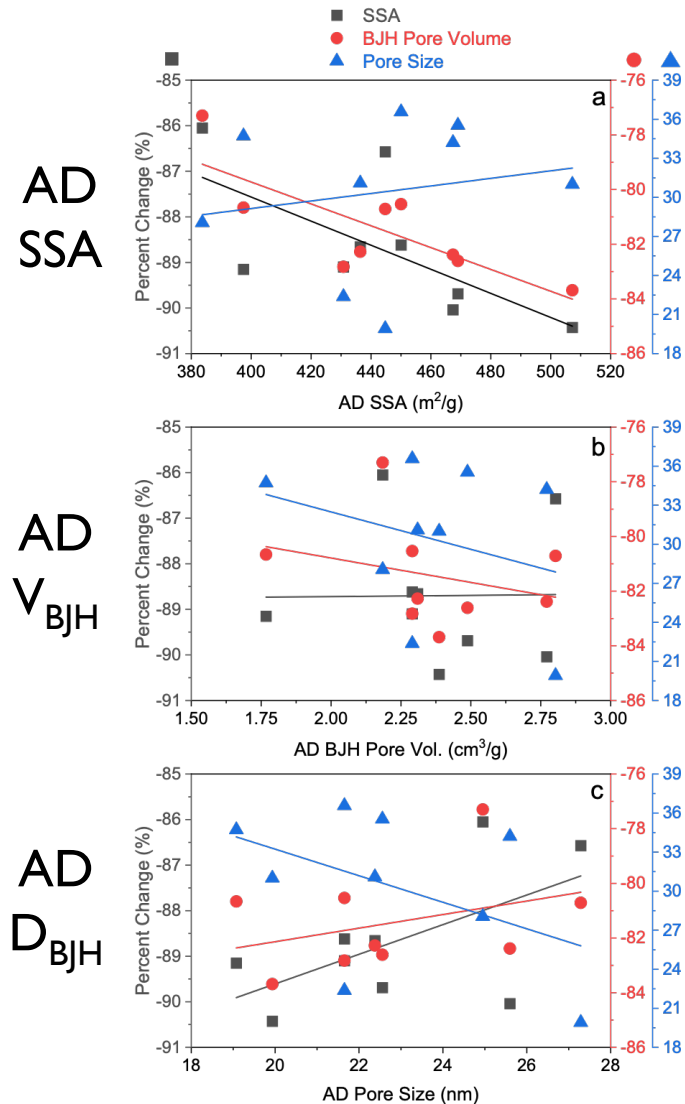
# Impact of starting structure on subsequent evolution at high temperatures



# Exploration of relationships between the as dried structure & thermal stability

To evaluate thermal stability, we switch the **independent variable** to the **as dried properties (starting structure)**.

Here, the **dependent variable** is the **percent change in SSA,  $V_{BJH}$ , and  $D_{BJH}$**  over various temperature ranges.

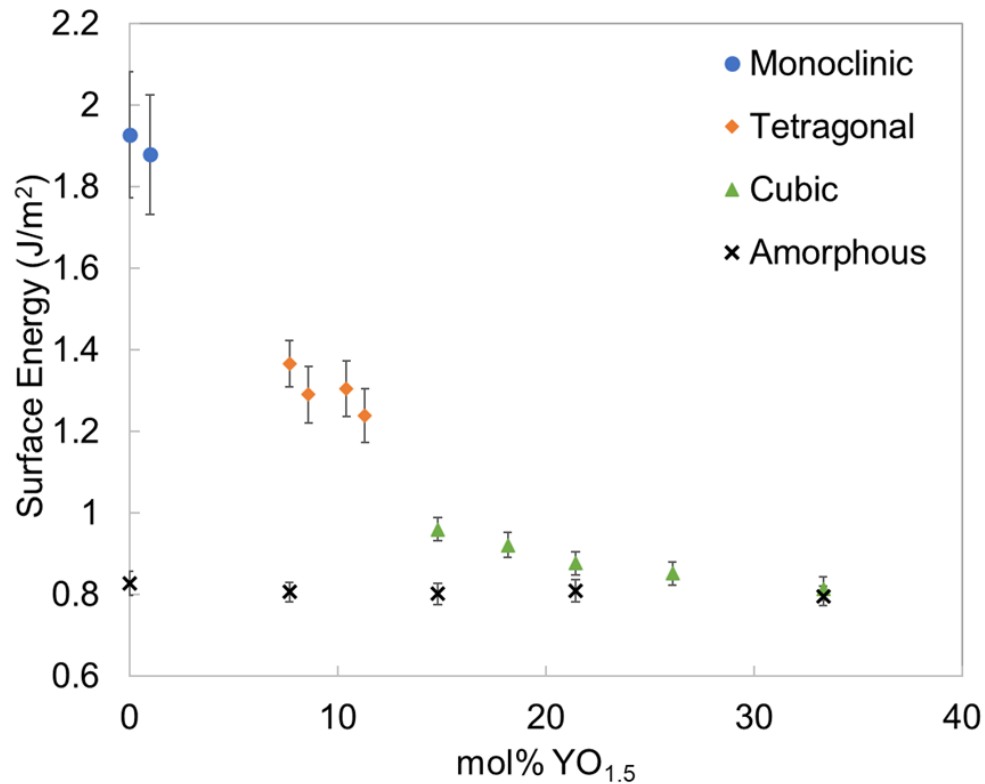


AD Property (x)	% Change - AD to 1000°C (y)	Slope	Intercept	p-value
SSA	SSA	-0.03	-77	0.049
SSA	V	-0.04	-64	0.012
SSA	D	0.03	18	0.635
D	SSA	0.33	-96	0.251
D	V	0.25	-87	0.357
D	D	-1.02	54	0.209
V	SSA	0.05	-89	0.977
V	V	-1.79	-77	0.446
V	D	-5.75	44	0.427

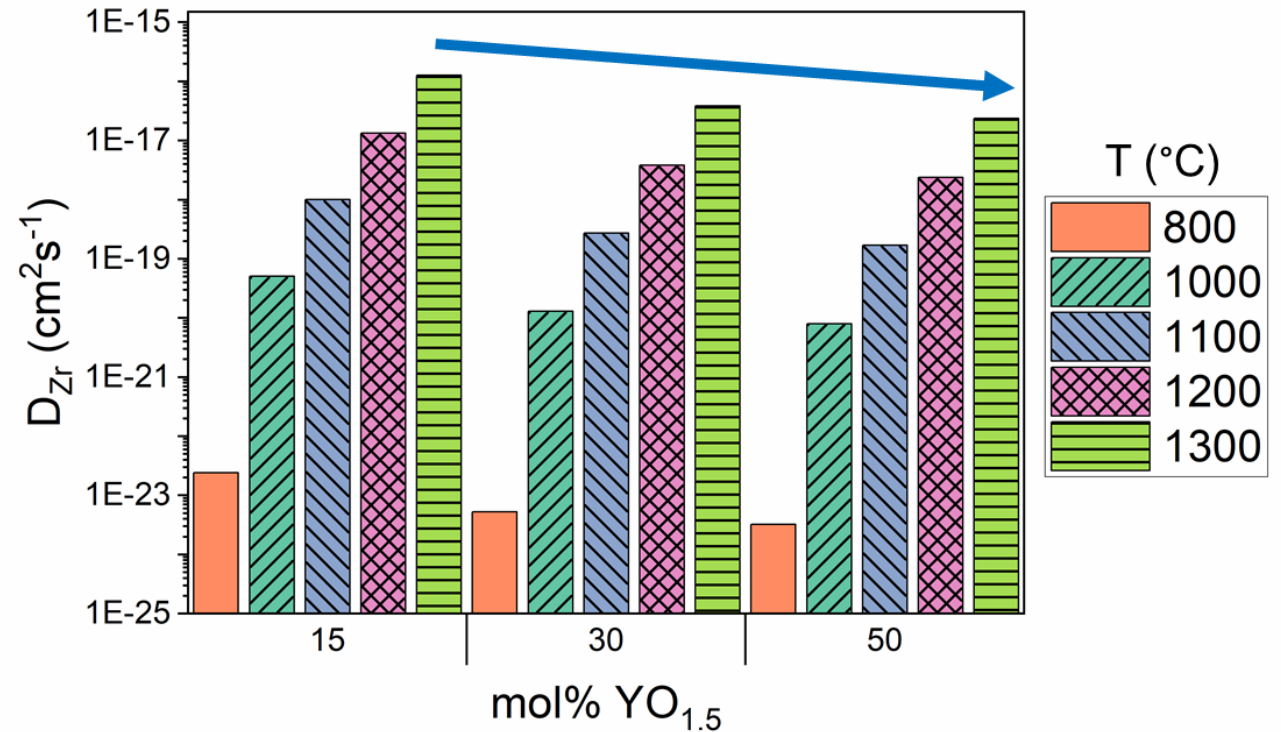
Coarser structures (lower SSA) sinter and densify less rapidly, as expected by thermodynamics.

Is the improved resistance to densification observed in the YSZ aerogels a result of yttria concentration or the as dried structure?

# Improved thermal stability in context of thermodynamic ( $\gamma$ ) and kinetic ( $D_{Zr}$ ) factors



**Reduced surface energy will reduce the driving force for elimination of surface area.<sup>1</sup>**

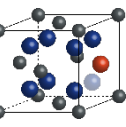


**Reduced diffusivity with increased yttria content may slow kinetics of densification & crystallite growth<sup>2-4</sup>**

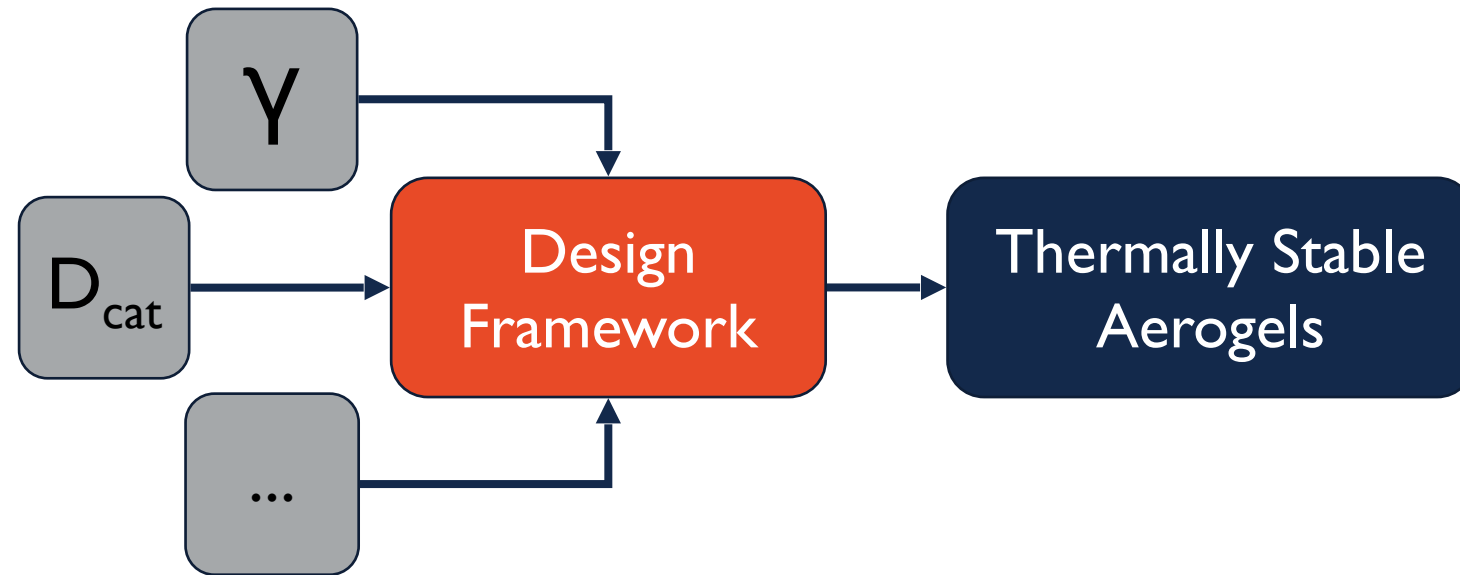
1. Drazin, J.W., & Castro, R. H. (2015). *Journal of the American Ceramic Society*, 98(4), 1377-1384.  
 2. Kilo, M., et al. (2000). *Journal of the European Ceramic Society*, 20(12), 2069-2077.

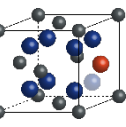
3. Kilo, M., et al. (1997). *Berichte der Bunsen-Gesellschaft*, 101(9), 1361-1365.  
 4. Kilo, M., et al. (2003). *Journal of applied physics*, 94(12), 7547-7552.

# A design framework for thermally stable aerogels informed by kinetics and thermodynamics



1. **Decreased cation diffusivity** of  $Zr^{4+}$ ,  $Y^{3+}$  in YSZ with increased  $Y_2O_3$  doping responsible for decreased mass flow
  - *Less densification AND crystallite/grain growth* (reduced rate of mass flow)
2. **Lower surface energy** with increased yttria content ( $\gamma_c < \gamma_t < \gamma_m$ ) leads to:
  - *Improved stability of pore structure* (lower driving force for densification)



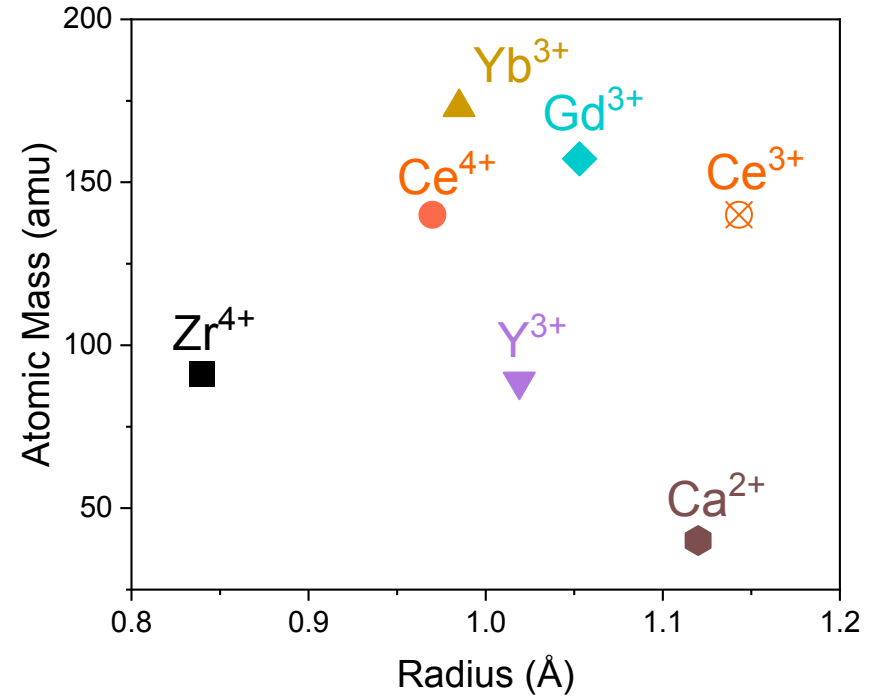


# Study of other dopants (Y, Yb, Gd, Ca, Ce) in zirconia aerogels at 15 and 30 mol% M/(M+Zr)

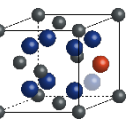
Further exploration of dopant properties (size, mass, charge) on aerogel thermal stability

Periodic Table of the Elements

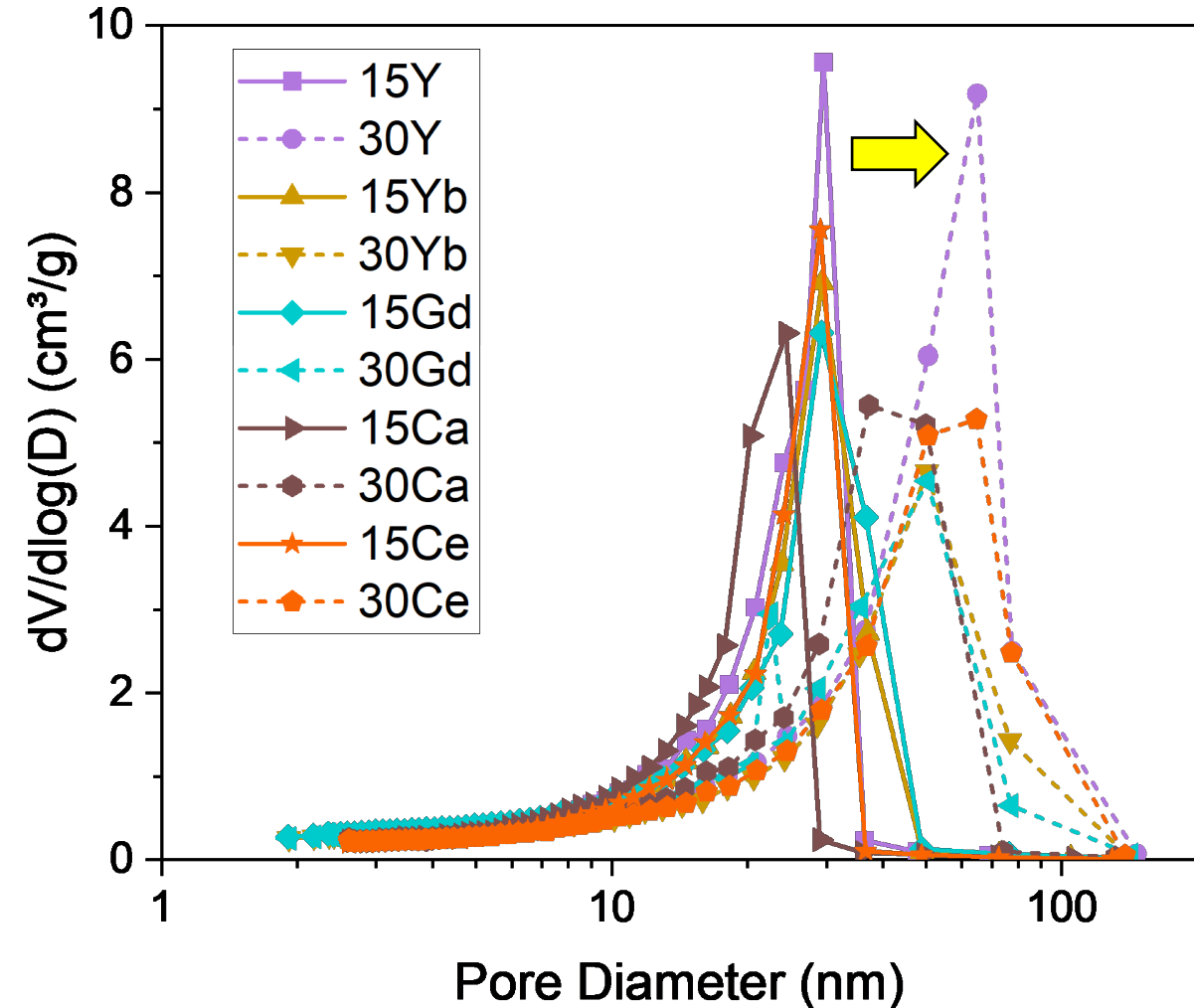
1 IA 1A	2 IIA 2A	3-10																11 IB 1B	12 IIB 2B	13 IIIA 3A	14 IVA 4A	15 VA 5A	16 VIA 6A	17 VIIA 7A	18 VIIIA 8A
1 H Hydrogen 1.008	4 Be Beryllium 9.012																	11 Na Sodium 22.990	12 Mg Magnesium 24.305	13 Al Aluminum 26.982	14 Si Silicon 28.086	15 P Phosphorus 30.974	16 S Sulfur 32.066	17 Cl Chlorine 35.453	18 Ar Argon 39.948
19 K Potassium 39.098	20 Ca Calcium 40.078	21 Sc Scandium 44.956	22 Ti Titanium 47.867	23 V Vanadium 50.942	24 Cr Chromium 51.996	25 Mn Manganese 54.938	26 Fe Iron 55.845	27 Co Cobalt 58.933	28 Ni Nickel 58.693	29 Cu Copper 63.546	30 Zn Zinc 65.38	31 Ga Gallium 69.723	32 Ge Germanium 72.631	33 As Arsenic 74.922	34 Se Selenium 78.971	35 Br Bromine 79.904	36 Kr Krypton 83.798								
37 Rb Rubidium 85.468	38 Sr Strontium 87.62	39 Y Yttrium 88.906	40 Zr Zirconium 91.224	41 Nb Niobium 92.906	42 Mo Molybdenum 95.95	43 Tc Technetium 98.907	44 Ru Ruthenium 101.07	45 Rh Rhodium 102.906	46 Pd Palladium 106.42	47 Ag Silver 107.868	48 Cd Cadmium 112.414	49 In Indium 114.818	50 Sn Tin 118.711	51 Sb Antimony 121.760	52 Te Tellurium 127.6	53 I Iodine 126.904	54 Xe Xenon 131.294								
55 Cs Cesium 132.905	56 Ba Barium 137.328	57-71 Lanthanide Series	72 Hf Hafnium 178.49	73 Ta Tantalum 180.948	74 W Tungsten 183.84	75 Re Rhenium 186.207	76 Os Osmium 190.23	77 Ir Iridium 192.227	78 Pt Platinum 195.085	79 Au Gold 196.967	80 Hg Mercury 200.592	81 Tl Thallium 204.383	82 Pb Lead 207.2	83 Bi Bismuth 208.980	84 Po Polonium (208.982)	85 At Astatine 209.987	86 Rn Radon 222.018								
87 Fr Francium 223.020	88 Ra Radium 226.025	89-103 Actinide Series	104 Rf Rutherfordium (261)	105 Db Dubnium (262)	106 Sg Seaborgium (266)	107 Bh Bohrium (264)	108 Hs Hassium (265)	109 Mt Meitnerium (278)	110 Ds Darmstadtium (281)	111 Rg Roentgenium (280)	112 Cn Copernicium (285)	113 Nh Nihonium (289)	114 Fl Flerovium (289)	115 Mc Moscovium (289)	116 Lv Livermorium (293)	117 Ts Tennessine (294)	118 Og Oganesson (294)								
Lanthanide Series		57 La Lanthanum 138.905	58 Ce Cerium 140.116	59 Pr Praseodymium 140.908	60 Nd Neodymium 144.243	61 Pm Promethium 144.913	62 Sm Samarium 150.36	63 Eu Europium 151.964	64 Gd Gadolinium 157.25	65 Tb Terbium 158.925	66 Dy Dysprosium 162.500	67 Ho Holmium 164.930	68 Er Erbium 167.259	69 Tm Thulium 168.934	70 Yb Ytterbium 173.055	71 Lu Lutetium 174.967									
Actinide Series		89 Ac Actinium 227.028	90 Th Thorium 232.038	91 Pa Protactinium 231.036	92 U Uranium 238.029	93 Np Neptunium 237.048	94 Pu Plutonium 244.064	95 Am Americium 243.061	96 Cm Curium 247.070	97 Bk Berkelium 247.070	98 Cf Californium 251.080	99 Es Einsteinium (254)	100 Fm Fermium 257.095	101 Md Mendelevium 258.1	102 No Nobelium 259.101	103 Lr Lawrencium (262)									



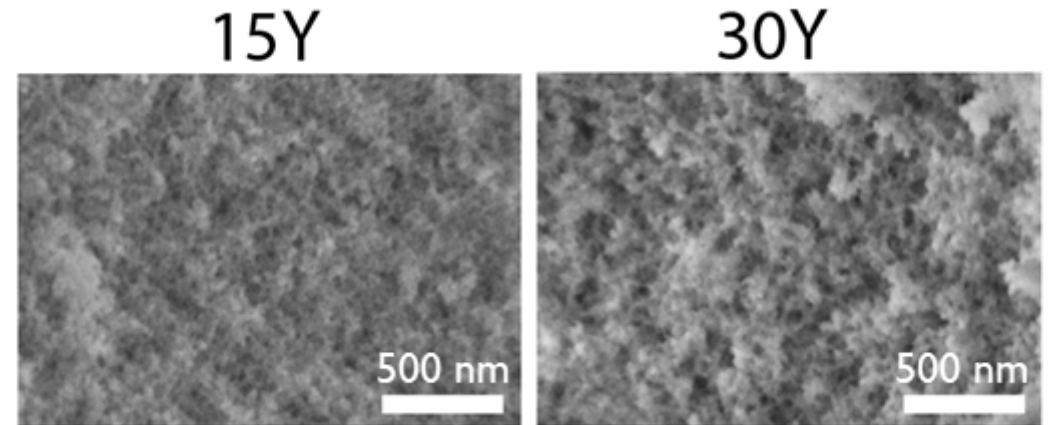
→ Modify thermal conductivity, surface energy and cation diffusivity  
→ Connect material properties to changes in structural evolution

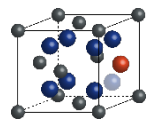


# As dried structure characterized with nitrogen physisorption and SEM

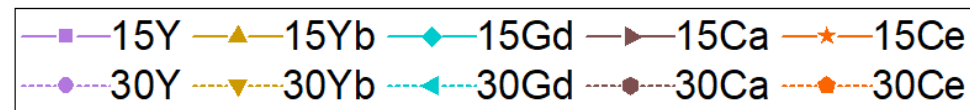
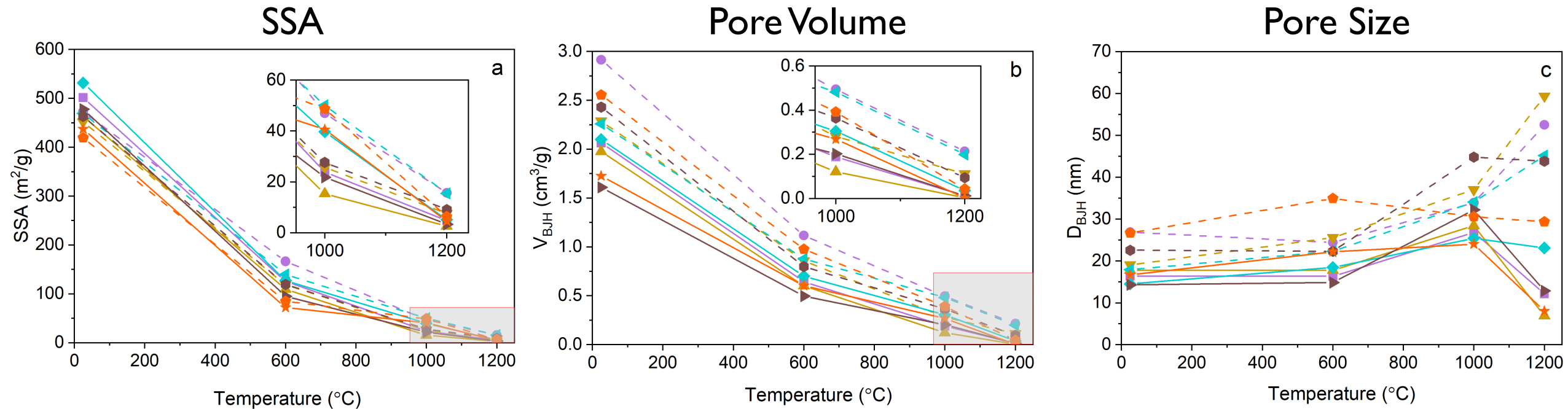


Increased dopant concentration from 15 to 30 mol% increases average pore size and distribution breadth for *all* dopants.





# Evaluation of pore structure with nitrogen physisorption quantifies change in performance



## Best Performers\*

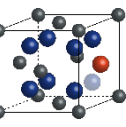
1000 °C: 30Y, 30Gd, 30Ce

1200 °C: 30Y, 30Gd

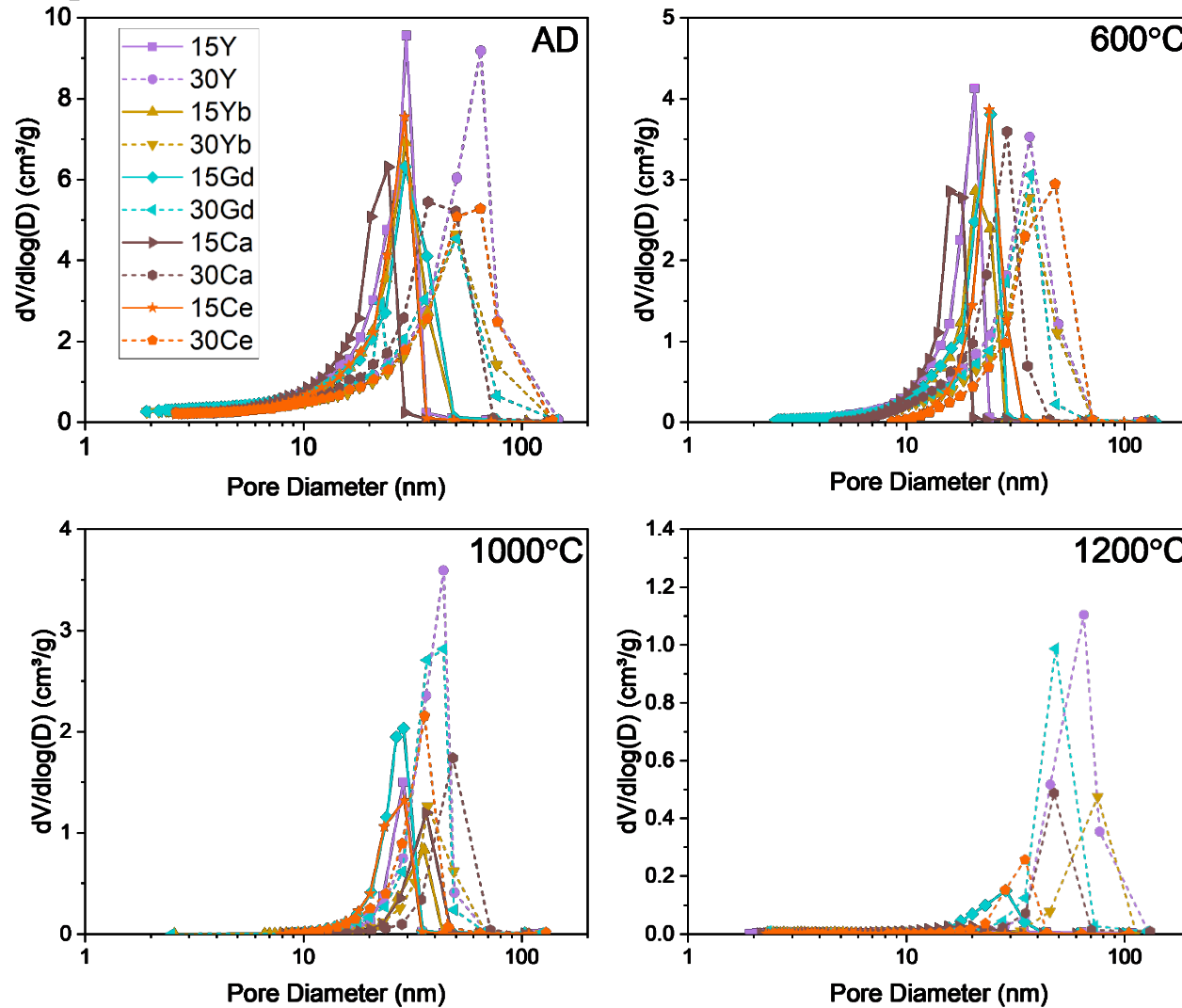
Increased dopant concentration leads to reduced densification.

\*Best performance dictated by maximum SSA and pore volume at a given temperature



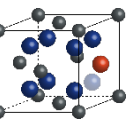


# Increased dopant content improves stability of pore structure to 1200 °C



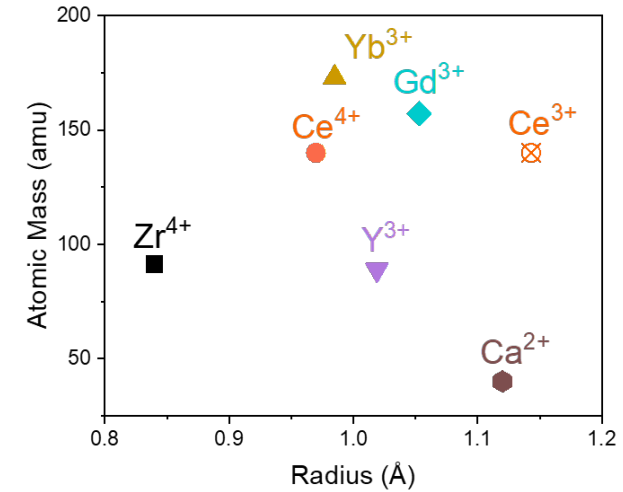
At 600 °C, no visible difference between 15 and 30 mol% MO<sub>x</sub>

At 1000 and 1200 °C, it becomes apparent 30 mol% MO<sub>x</sub> maintains more porosity. Gd and Y perform the best.



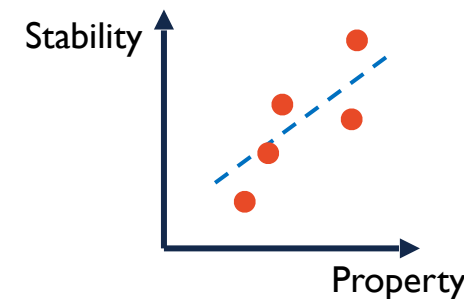
# Connecting material properties to thermal stability

Given this absence, we turned to something readily available: cation properties (mass, radius, charge)



$$\text{Weighted Property} = x_{\text{Zr}}P_{\text{Zr}} + x_{\text{M}}P_{\text{M}}$$

$x_{\text{M}}$  = mole fraction  $\text{MO}_y$   
 $P_{\text{M}}$  = property of dopant  $\text{M}^{2y+}$



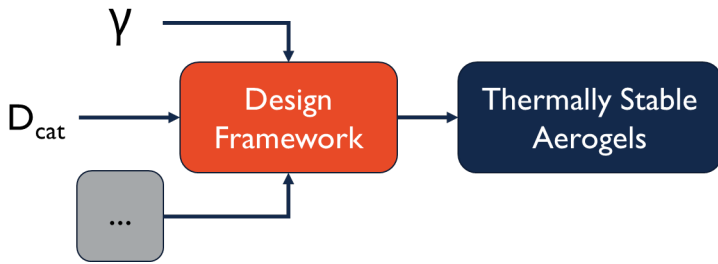
$$\% \text{ Change} = \frac{x_f - x_i}{x_i}$$

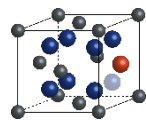
Next, we calculated a weighted average for each material

We then performed linear regression on the **absolute** (property at a given temperature) and **relative** (percent change) thermal stability.

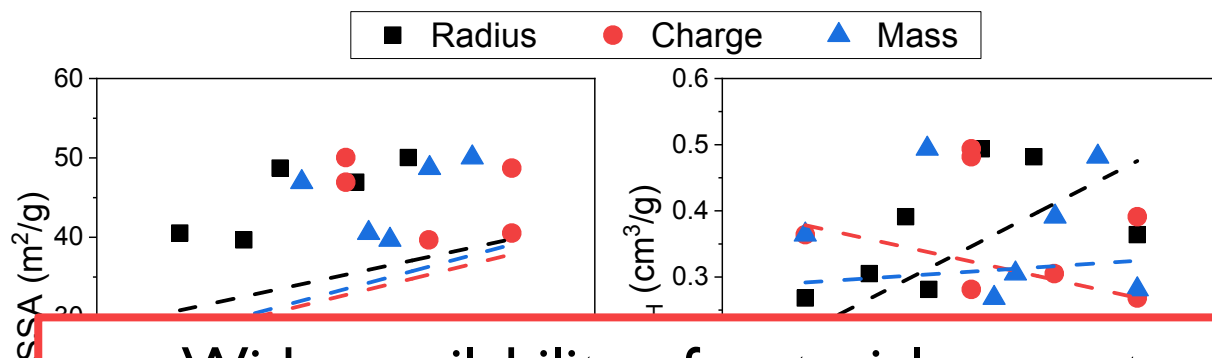
From our work on YSZ, we were able to connect our results to others' measurements of surface energy and cation diffusivity.

But... neither those properties nor others are available for wider ranges of dopants and concentrations





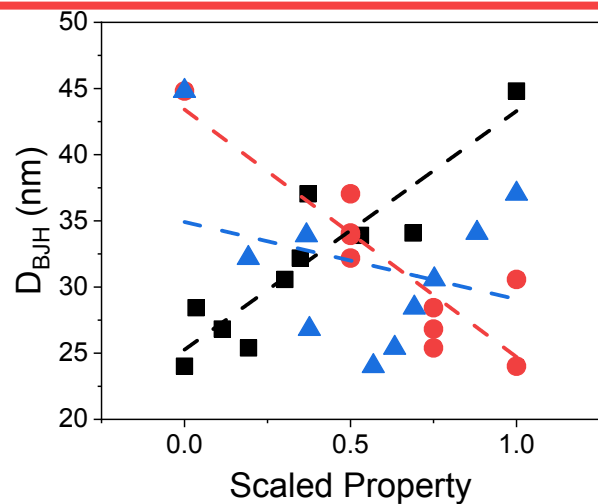
# Cation properties are not clearly related to thermal stability (absolute or relative)



Property	Response	p-value
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Radius		0.54
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Wider availability of material property data (such as surface energy, cation diffusivity, etc.) would enable more thorough analysis of property-stability relationships.

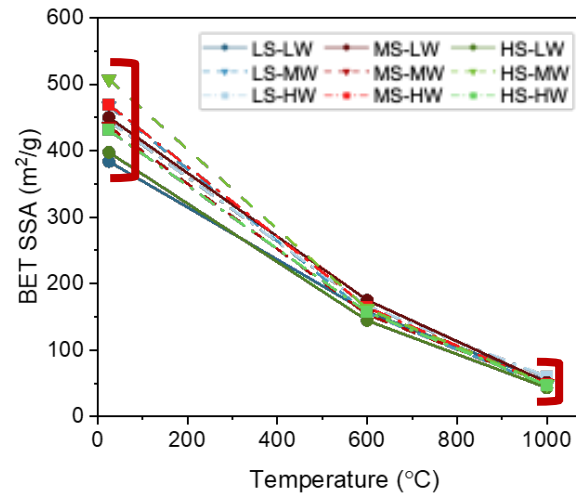


Charge	$V_{BJH}$	1000	0.47
Mass			0.8
Radius			3.0E-04
Charge	$D_{BJH}$	1000	5.9E-04
Mass			0.41

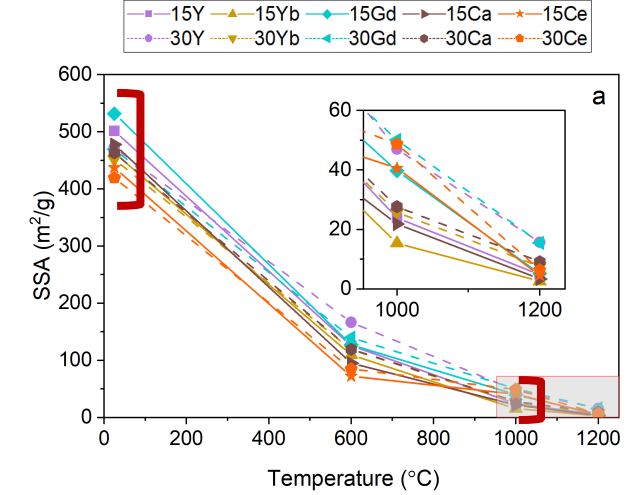
# Synthetic parameter study can serve as a “control” for studies on composition

- For both YSZ and dopant study, the amount of dopant influenced the starting structure & thermal stability
- Presumably, lower starting SSA and coarser structure should densify less rapidly, but by how much?

**Synthetic Parameters**



**Dopants**

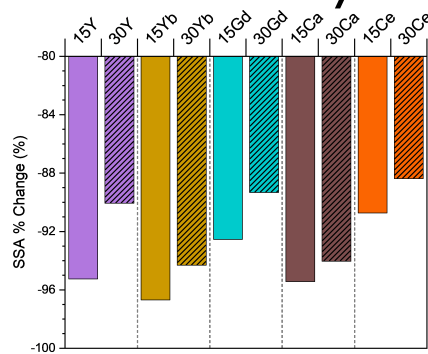


Study	$\Delta$ SSA as dried
YSZ	118
Dopants	112
Synthetic Parameters	123

Improvement in thermal stability likely result of change in material properties and *not* the change in as dried structure.

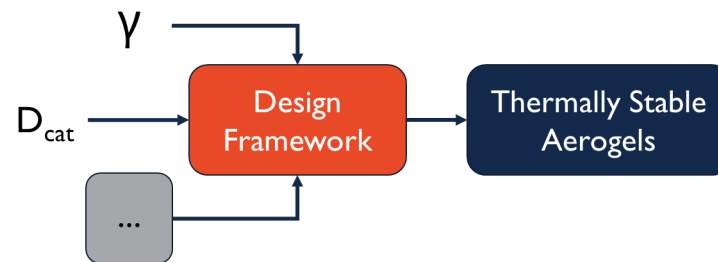
# Summary

1. Aerogels are promising candidates for lightweight, highly insulating materials, but the pore structure must be preserved to  $T \geq 1200\text{ }^\circ\text{C}$
2. Reduced surface energy and cation diffusivity are hypothesized to improve aerogel thermal stability.
3. Increased dopant concentration from 15 to 30 mol% M/(M+Zr) reduces densification of the pore structure (Gd, Y perform best).



# Looking Forward

1. Wider availability of material property data (surface energy, cation diffusivity, etc.) may help understand source(s) of variability in aerogel thermal stability.



2. Alternative strategies for zirconia-based aerogels synthesis need to be identified including the potential of nucleating agents and post-synthetic modifications such as surface capping strategies

# Thank you for your attention! Special thanks to...

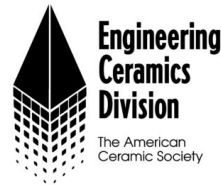
- PhD Student: Nathaniel Olson (UIUC, soon to be NASA JSC)
- Technical Collaborator: Dr. Jamesa Stokes (NASA GRC)
- Dr. Frances Hurwitz (NASA GRC, retired)
- Jordan Meyer (UIUC MatSE U-Grad)
- Krogstad Group members
- Others at NASA GRC: Dr. Haiquan (Heidi) Guo, Dr. Richard Rogers, Jessica Cashman

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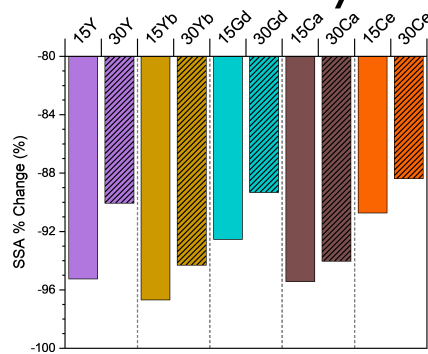
## Facilities:

- Materials Research Laboratory, UIUC
- SCS Microanalysis Laboratory, UIUC
- NASA Glenn Research Center



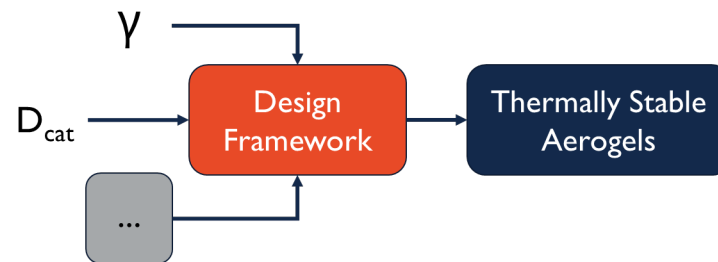
# Summary

1. Aerogels are promising candidates for lightweight, highly insulating materials, but the pore structure must be preserved to  $T \geq 1200\text{ }^{\circ}\text{C}$
2. Reduced surface energy and cation diffusivity are hypothesized to improve aerogel thermal stability.
3. Increased dopant concentration from 15 to 30 mol% M/(M+Zr) reduces densification of the pore structure (Gd, Y perform best).



# Looking Forward

1. Wider availability of material property data (surface energy, cation diffusivity, etc.) may help understand source(s) of variability in aerogel thermal stability.



2. Alternative strategies for zirconia-based aerogels synthesis need to be identified including the potential of nucleating agents and post-synthetic modifications such as surface capping strategies