National Aeronautics and Space Administration



Metal Additive Manufacturing for Rocket Engines: Successes and Failures

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21 March 2023

Additive Manufacturing Typical Process Flow



Qualification

Proper AM process selection requires an integrated evaluation of all process lifecycle steps

Metal Additive Manufacturing Development for Rocket Engines





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Laser Powder Bed Fusion (L-PBF) Copper Alloys combined with other AM processes to provide bimetallic



Directed Energy Deposition







L-PBF of complex components, new alloy developments for harsh environment







EXPL DREMC

Successful hot-fire testing of full-scale additive manufacturing (AM) Part to be flown on SLS RS-25 RS-25 Pogo Z-Baffle – Used existing design with AM to reduce complexity from <u>127 welds to 4 welds</u>



Large Scale LP-DED Nozzle Development





60" (1.52 m) diameter and 70" (1.78 m) height with integral channels 90 day deposition





95" (2.41 m) dia and 111" (2.82 m) height Near Net Shape Forging Replacement

Reference: P.R. Gradl, T.W. Teasley, C.S. Protz, C. Katsarelis, P. Chen, Process Development and Hot-fire Testing of Additively Manufactured NASA HR-1 for Liquid Rocket Engine Applications, in: AIAA Propuls. Energy 2021, 2021: pp. 1–23. https://doi.org/10.2514/6.2021-3236.



Demonstrator Aerospike LP-DED Nozzle





LP-DED Aluminum 6061-RAM2



Criteria and Comparison Various Metal AM Processes

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Multi-metallic and multi-process development

Credit: RPMI



L-PBF Liner / LP-DED Jacket



L-PBF GRCop-42 to Inco 625

L-PBF Liner / Coldspray Jacket L-PBF Liner / EBW-DED Jacket



NASA's Effort on New Alloy Development



Max. Use Temp. (°C)	Alloy Family	Purpose	Novel AM Alloys	Propulsion Use
200	Aluminum	Light weighting	-	Various
750	Copper	High conductivity; strength at temperature	GRCop-42 GRCop-84	Combustion Chambers
800	Iron-Nickel	High strength and hydrogen resistance	NASA HR-1	Nozzles, Powerheads
900	Nickel	High strength to weight	-	Injectors, Turbines
1100	ODS Nickel	High strength at elevated temp; reduced creep	GRX-810 Alloy 718-ODS	Injectors, Turbines
1850	Refractory	Extreme temperature	C-103, C-103- CDS, Mo, W	Uncooled Chambers









New alloy development using various additive manufacturing processes (PBF and DED) can yield performance improvements over traditional alloys



AM Enabling New Alloy Development



GRCop-42, High conductivity and strength for high heat flux applications





NASA HR-1, high strength superalloy for hydrogen environments







GRX-810, high strength, low creep rupture and oxidation at extreme temperatures







Ref: Tim Smith, Christopher Kantzos / NASA GRC 12



GRX-810 Oxide Dispersion Strengthened (ODS) Alloy











- Long Life Additive Manufacturing Assembly (LLAMA) project goals
 - Fabrication and testing of additively manufactured high duty cycle (>50 starts) lander engine hardware (highly instrumented).
 - Determine performance of GRCop-42 alloy used for the thrust chamber.
 - Disseminate data to industry partners.
- GRCop-42
 - Copper-based (Cu-Cr-Nb) alloy w/ Cr2Nb precipitates for dispersion strengthening.
 - Developed for high heat flux environments such as combustion chambers.
- LLAMA hardware and test program
 - Laser Powder Bed Fusion (L-PBF) GRCop-42 chambers.
 - Carbon-Carbon (C-C) and Direct Energy Deposition (DED) NASA HR-1 nozzles.
 - Additively manufactured L-PBF injectors (Alloy 625 and 718).
 - Two separate test phases completed in February 2021.





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LLAMA Hardware Overview











7k-lb_f GRCop-42 chamber and Composite Nozzle





L-PBF GRCop-42 chamber from same build plate and identical processing successfully tested 51 starts and 1,000 seconds



Anomaly Background

NASA

- Tested in early 2021
- Location: MSFC Test Stand 115
- L-PBF GRCop-42 chamber
 - 8 starts and 83.3 seconds total before separation.
 - No issues observed in prior chamber test data.
- Carbon-Composite experimental nozzle
 - Untested and possessed a noticeable crack.
 - Deemed an acceptable risk for test.





Multiple L-PBF Chambers Built and Tested

• EOS M400 L-PBF printer

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- Certified GRCop-42 powder lot
- 4 chambers on the build
 - 3 w/ identical designs for LLAMA
- Computed Tomography Scans
 - No observations from data prior to HIP
 - Did not specifically look for witness lines – focused on powder removal verification
- Post-processing
 - C1 HIP, EB weld manifold, exterior polishing
 - C2 HIP, EB weld manifold
 - C3 HIP, EB weld manifold, chemically milled

C1 - Untested





C2- Separated



Anomaly Timeline







Separated Chamber







<u>**Goal:**</u> Determine the root cause and impacts of build interruptions that occur following stop and start during the chamber build process. Areas of concern were at witness lines (chamber with 51 cycles) and separation (back-up chamber) during testing.

- 1. Created a plan to evaluate this process issue.
- 2. Evaluated fabrication process and build records of the chambers.
- 3. Completed NDE and evaluations of chambers.
 - Digital images, microscopic images, and measurements for witness line reference locations.
 - Fracture surface of chamber C2 examined for any noticeable defects or fracture points.
 - MSFC NDE team completed CT scan of all three chambers.
- 4. Sectioned chambers for metallography.
- 5. Completed metallography to understand region of concern and "good" regions.
 - Fabricated tensile and fatigue specimens with **representative witness lines.** (full length specimens from original build not available)

6. Completed tensile and fatigue testing of material from chambers (micro) and witness specimens.

Completed fractography and microstructural characterization of (sectioned) chambers and samples.
Reporting of lessons learned and recommendations.



Witness Lines Matched to Build Timeline





Color adjusted in photos to highlight witness lines 24



C3 & C2 Separation Comparison



PK129 Test 18



C2 - separated chamber (9 starts)

PK129 Test 13



C3 - 51 start chamber (full life cycle)

Optical Images of Chambers Post-Test



• Unpolished external surfaces.

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- Top (power outage) and bottom (powder overflow) witness lines did not appear to have many detrimental defects.
- Middle witness line on chamber 1, there were some large lack of fusion defects that appeared to line up with the restart line.
- Chamber 3 no defects visible at the surface beyond the crack that had already developed after test.

Middle witness line: C1 (untested) Surface was polished using CMP





Middle witness line: C3 (51 starts)





Test Specimens: Chamber Sectioning, Test Bars





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Bar	Chamber Restart Replicated	Witness Line Replicated	Restart
A	None	Control	None
11	Ttone	Section	TYONE
В	Empty Overflow	Middle and	Chamber
Ъ	D Empty Overnow	Lower	Open
С	Power Outage	Upper	Chamber
			Closed

Optical Images of Chamber Sections

Α



Label	Section	Porosity
А	Upper Witness Line	0.748%
В	Middle Witness Line	1.906%
С	Control Section	0.511%
D	Lower Witness Line	1.743%
E	Tensile Bar	0.006%

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- Samples taken from un-tested chamber (C1).
- Tensile bar built separately as part of investigation.
- Proper HIP of chambers was confirmed.
- Porosity is evident throughout samples.
- Clear congregation of porosity around witness lines.
- Porosity reduces load bearing capacity (reduced area) and can act as stress concentrators/crack initiators.



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Optical Images of Section





C1 (Leaked), Middle Witness Line





C1 (Leaked), Chamber Control





Build Direction



Combined Microtensile & Tensile Results

 Room temp tensile testing conducted on ASTM E8 specimens (0.25" dia gage) from witness bars with various restarts

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- Testing at 1200F for ASTM E8 round bars showed similar trends
- Fracture surfaces appeared similar
- Microtensile testing conducted at room temp on section from chamber (C1 and C3) at witness line and nonwitness





Fractography of Samples after Mechanical Testing



Microtensile

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- Granular fracture surface, indicating a brittle fracture
- Irregular shape porosity, indicating lack of fusion



- Granular fracture surface, indicating a brittle fracture
- Laser scan pattern clearly visible

1T3B (Chamber Control)



- Overloaded fracture surface and necking, indicating a more ductile fracture
- Less porosity compared to witness lines







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Low Cycle Fatigue of Post-build Witness Specimens



- LCF conducted at room temperature, total strain of 0.7% and 2%
- R = -1, triangular



Strain, %	Cycles, Nf
2	924
2	840
2	529
2	846
0.7	9,760
0.7	12,597
0.7	13,531
0.7	12,796
2	668
2	593
2	986
2	878
0.7	5,579
0.7	10,805
0.7	13,426
0.7	12,899
2	369
2	437
2	559
2	882
0.7	14,038
0.7	12,514
0.7	14,499
0.7	13,624

*4 samples per test case





- Chamber experienced tensile overload during hot-fire at the witness line that had a higher degree of voids.
- The L-PBF GRCop-42 chambers built under LLAMA had higher porosity (1-2%) that congregated more at witness lines causing lack of fusion.
- Granular surfaces, unmelted particles, and irregular pores were observed in microtensile specimens (sectioned) from chambers.
- Areas affected by build interruptions must be properly evaluated and dispositioned. AM machine restarts represent a risk, and appropriate restart procedures should be developed and followed to maintain material quality.
 - Witness specimens using different types of restarts showed similar tensile strengths and LCF results.
- Build log indicated no issues with parameters, but *an issue* (parameters, lens, etc) caused the porosity and HIP did not fully close these voids.
- Demonstrates the process sensitive nature of AM parts and build interruptions need to be properly documented, fully evaluated, and properly dispositioned.



- Build interruptions in GRCop-42 components do not inherently possess weakened material properties if a restart procedure is properly executed.
- Full height specimens should be built with all components to characterize the material.
- While not subject to NASA-STD-6030, this chamber provides a good case study on why it is important that AM materials used in critical applications adhere to NASA-STD-6030 standards and the need for robust process development, in-depth material evaluation, and process controls.



- The proper planning...
- High quality ingredients...
- Controlled environments and parameters...
- Properly planned post-processing...
- Repeatable and controlled process...
- Verified process...
 - = Successful Component (Happy Customers!)







Metal Additive Manufacturing for Propulsion Applications

Edited by Paul R. Gradl, Omar R. Mireles, Christopher S. Protz, and Chance P. Garcia



PROGRESS IN ASTRONAUTICS AND AERONAUTIC

Timothy C. Lieuwen, Editor-in-Chief Volume 263



https://arc.aiaa.org/doi/book/10.2514/4.106279

Online version and hardcopy available

P. R. Gradl, O. Mireles, C.S. Protz, C. Garcia. (2022). *Metal Additive Manufacturing for Propulsion Applications*. AIAA Progress in Astronautics and Aeronautics Book Series. <u>https://arc.aiaa.org/doi/book/10.2514/4.106279</u>

Additive manufacturing (AM) processes are proving to be a disruptive technology and are grabbing the attention of the propulsion industry. AM-related advancements in new industries, supply chains, design opportunities, and novel materials are increasing at a rapid pace. The goal of this text is to provide an overview of the practical concept-toutilization lifecycle in AM for propulsion applications.

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NSV

Acknowledgements



- John Fikes
- Rapid Analysis and Manufacturing Propulsion Technology (RAMPT) Project
- Optimized and Repeatable Components using Additive (ORCA)
- Long Life Additive Manufacturing Assembly (LLAMA) Project
- Space Launch System (SLS) Program
- Nima Shamsaei
- Drew Hope
- Martin Annett
- Lynn Machamer
- RPM Innovations (RPMI)
- Tyler Blumenthal
- DM3D
- GE Research
- Bhaskar Dutta
- REM Surface Engineering
- Powder Alloy Corp
- AP&C
- Formalloy
- Auburn University (NCAME)

- Ben Williams
- Marissa Garcia
- Tim Smith / GRC Christopher Kantzos / GRC
- Tal Wammen
- Tom Teasley
- Scott Chartier
- Test Stand 115 crew
- Kevin Baker
- Matt Medders
- Adam Willis
- Nunley Strong
- Zach Taylor
- Matt Marsh
- Darren Tinker
- Dwight Goodman
- Will Brandsmeier
- Jonathan Nelson
- Bob Witbrodt
- Shawn Skinner
- Will Evans
- John Ivester

- Will Tilson
- Jim Lydon
- Brian West
- Gabe Demeneghi
- David Ellis / GRC
- Judy Schneider / UAH
- David Myers / MSFC EM21
- Scott Ragasa / MSFC EM21
- Sturbridge Metallurgical Services
- Product Evaluation Systems
- IMR Test Labs
- Robert Amaro / AMTT
- Ron Beshears
- James Walker
- Steve Wofford
- Johnny Heflin
- Mike Shadoan Keegan Jackson
- Many others in Industry, commercial space and academia

The Challenges with AM Processes

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There are a lot of inputs and steps in the AM lifecycle that must go right to meet the expected geometry

Methodical AM Process Selection



- What is the **alloy** required for the application?
- What is the **overall part size**?
- What is the feature resolution and internal complexities?
- Is it a **single alloy or multiple**?
- What are programmatic requirements such as cost, schedule, risk tolerance?
- What are the end-use environments and properties required?
- What is the **qualification/certification** path for the application/process?

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Various criteria for selecting AM techniques

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- NASA has baselined HIP for data generation (per NASA 6030).
- HIP does add process time and cost but can improve performance of alloys for tensile and fatigue based on comparison studies.





S. Ghiaasiaan, N. Ahmad, P. Gradl, S. Shao, N. Shamsaei Additively Manufactured Haynes 282 via L-PBF: Effect of Post-processing on Mechanical Properties at Different Temperatures

AM Alloys and Processes In-work



Material 🔽	Process
Haynes 282	L-PBF
Haynes 282	LP-DED
Hastelloy X	L-PBF
Hastelloy X	LP-DED
Inconel 625	L-PBF
Inconel 625	LP-DED
Inconel 625	LW-DED
Inconel 625	AW-DED
Inconel 718	L-PBF
Inconel 718	LP-DED
Inconel 718	AW-DED
Inconel 939	L-PBF
Haynes 230	L-PBF
Haynes 230	LP-DED
Haynes 214	L-PBF
Haynes 233	L-PBF
Haynes 233	LP-DED

55+ Alloys in characterization

Material 🔽	Process	
NASA HR-1	L-PBF	
NASA HR-1	LP-DED	
JBK-75	L-PBF	
JBK-75	LP-DED	
CoCr	L-PBF	
CoCr	LP-DED	
Invar 36	LP-DED	
Stellite 21	LP-DED	
316L	LP-DED	
15-5	LP-DED	
17-4	L-PBF	
17-4	LP-DED	
Scalmalloy	L-PBF	
6061-RAM2	L-PBF	
6061-RAM2	LP-DED	
F357	L-PBF	
F357	LP-DED	
1000-RAM10	L-PBF	
AlSi10Mg	L-PBF	
AlSi10Mg	LP-DED	
7A77	L-PBF	

Material 🔽	Process 🔽
Monel K500	LP-DED
Monel K500	L-PBF
GRCop-42	L-PBF
GRCop-42	LP-DED
GRCop-84	L-PBF
C-18150	L-PBF
Ti6Al-4V	L-PBF
Ti6Al-4V	LP-DED
Ti6Al-4V	LW-DED
Ti6Al-4V	EBW-DED
Ti6242	L-PBF
Ti6242	LP-DED
GRX-810	L-PBF
GRX-810	LP-DED
Haynes 214-ODS	L-PBF
C-103	LP-DED



Data example of Haynes 230 LP-DED



Power (W) Layer (μ 1070 33	height Trave m) (mn 31 1	el speed n/min) 016	Powder feed rate (g/min) 19.10
Procedure (Designation)	Temperature (°C)	e Time (hrs)	Cooling
Stress Relief (SR)	1066	1.5	Furnace cool
HIP [2]	1163/103 MPa	3	Furnace cool
Solution Annealing (SOL)	1177	3	Argon quench

[2] HIP per ASTM F3301

Data from Gradl, Mireles, Protz, Garcia. "Metal Additive Manufacturing for Propulsion Applications", AIAA Progress Series. (2022). Appendix A.





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ICP & IGF Chemical Analysis



Element

Copper

Chromium

Niobium

Oxygen

Iron

Aluminum

Silicon

Nickel

Cobalt

Phosphorus

Silver

Documentation	Cr/NB Ratio
Chamber Average	1.31
Specification Target	1.13 - 1.18
PC Results	1.15

0.01

0.01

0.01

- 7 pieces from entire length of chamber C1
- Observations:

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- Composition did not vary throughout length of chamber
- AI, Si higher than expected crucible fluxing potentially
- Ni, Co, Fe within detection limits
- O notably high can reduce conductivity and produce AI-Nb-O particles that create fatigue cracking
- Cr/Nb ratio is strong indicator of GRCop's effectiveness (precipitates for dispersion strengthening)
 - High ratio in chamber pieces
 - Results in excess chromium precipitates
 - Reduces high temp strength and creep resistance
- Important for AM GRCop parts to have consistent compositions

Fractography – Tensile Fracture

- 3 witness test bars from tensile simulating build stoppages
- Observations:

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- No major differences between fractures
- Typical cup-cone fracture surfaces common for ductile metals
- Fracture surfaces had elongated grains
- Conclusions:
 - Similar fractures track with similar properties observed in bars previously





Fractography – LCF Fracture



3 test bars from LCF

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- Fracture surfaces smeared b/c LCF had fully reversible cycles
- Closed chamber restart bar never fully fractured
- Observations:
 - Open restart: flat surface before overload failure transition, secondary crack below primary crack
 - No restart: three separate cracks jogged together

