

Crystal Growth in the SUBSA furnace at the International Space Station (ISS)

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1. Introduction: Why Single Crystals ? Growth techniques.
2. *Melt flow, heat and mass transfer*
3. Using the baffle on Earth and at the IS.
4. What have we learned ? Future experiments at ISS



Peggy Whitson, July 2002



Why Single Crystals ?

Features:

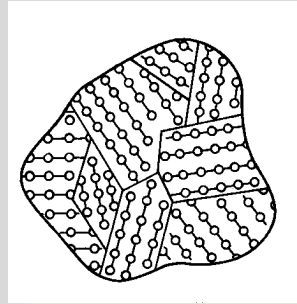
- Uniform on micro-scale
- Beauty, rarity

Research :

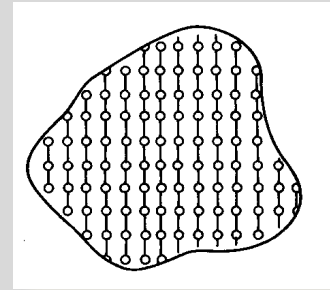
- metals,
- semiconductors,
- superconductors,
- protein crystals..

Applications:

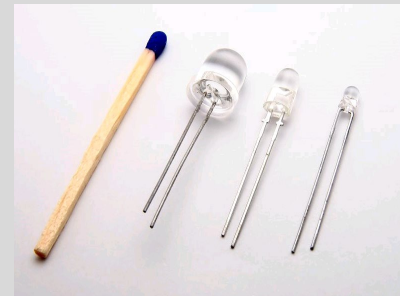
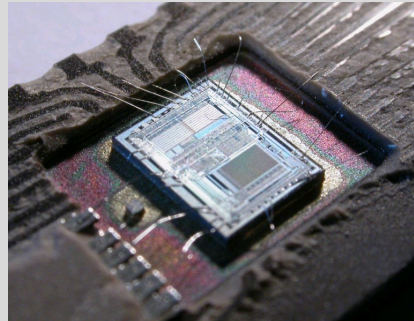
electronic/opto-
electronic devices
VLSI, LED, lasers, ..
electromechanical
transducers...
detectors



Poly



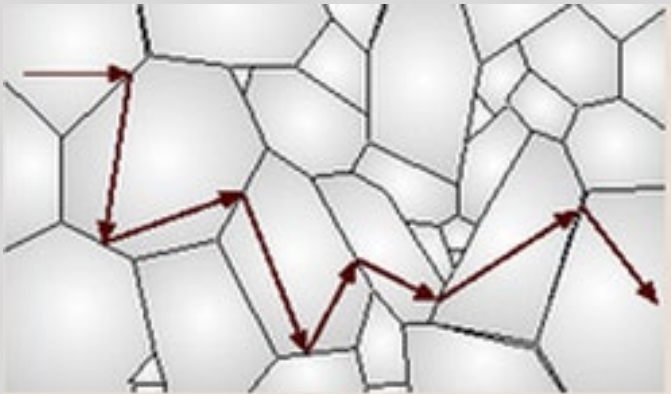
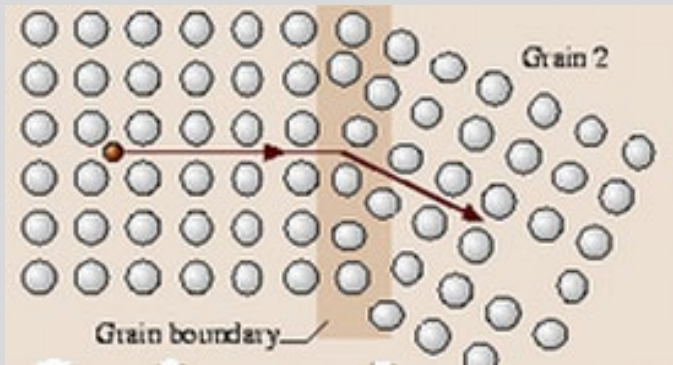
Single



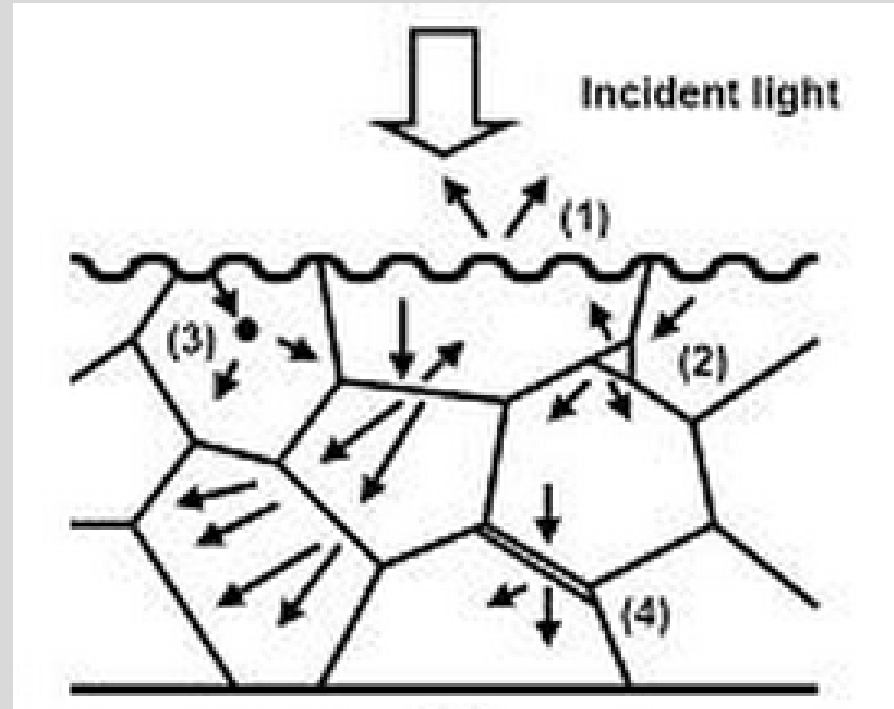
Rolls-Royce,
single-crystal,
nickel-superalloy

HPGe

Scattering of electrons and light



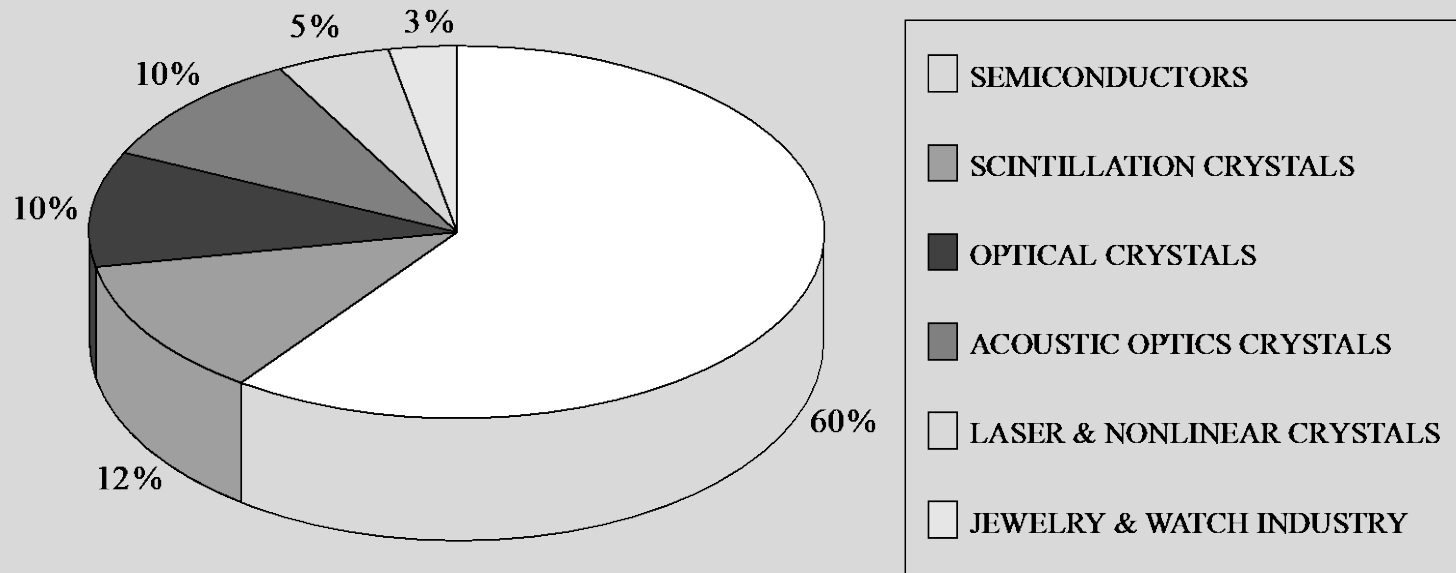
- grain boundaries



Defects/impurities:

1. Diffuse scattering
2. Pores,
3. Precipitates, etc.

Single Crystals-World Production: 20,000 tons/year



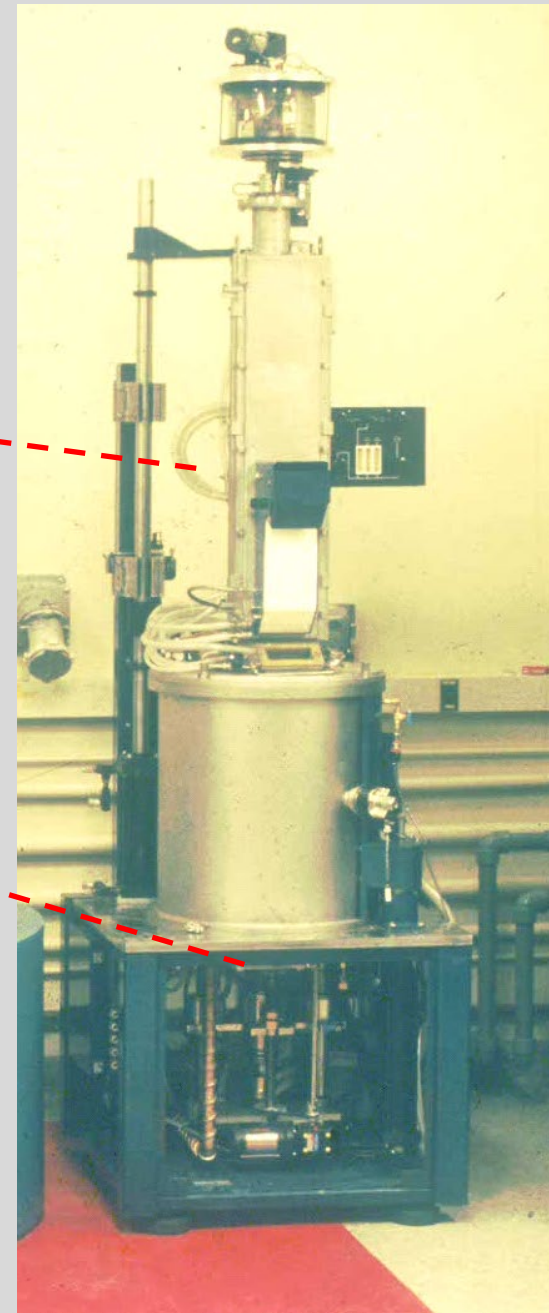
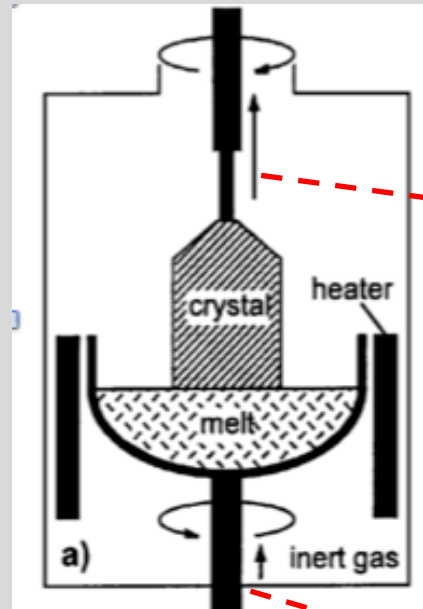
- ~60% are semiconductors silicon, GaAs, InP used for:
- **Electronics**
- **Optoelectronic** devices: emit, detect and control [light](#) (includes [gamma rays](#), [X-rays](#), [ultraviolet](#) and [infrared](#))

Czochralski Method (CZ):

1. Charge is melted

2. Seed is dipped into the melt

3. Crystal grows as atoms from the melt adhere themselves to the seed.



Governing Equations for fluid flow, heat and mass transfer

FM $\left\{ \begin{array}{l} \nabla \cdot \vec{V} = 0 \\ \frac{\partial \vec{V}}{\partial t} + (\vec{V} \cdot \nabla) \vec{V} = -\nabla p + \nabla^2 \vec{V} + Gr(T_0 - T) \vec{i}_g \end{array} \right.$

Heat $\frac{\partial T}{\partial t} + (\vec{V} \cdot \nabla) T = \frac{1}{Pr} \nabla^2 T$

Mass $\frac{\partial C}{\partial t} + (\vec{V} \cdot \nabla) C = \frac{1}{Sc} \nabla^2 C$

$Re = \frac{V L}{\nu}$

$Gr = \frac{g \beta \Delta T L^3}{\nu^2}$

$$\frac{Gr}{Re^2} \sim \frac{\text{strength of natural convective flow}}{\text{strength of forced convective flow}}$$

- Natural convection (NC) is:
1. *dominating*, for $\sqrt{Gr} \gg Re$ ← Bridgman
 2. *significant*, for $\sqrt{Gr} \sim Re$ ← CZ
 3. *negligible*, for $\sqrt{Gr} \ll Re$ ← Ideal (?)

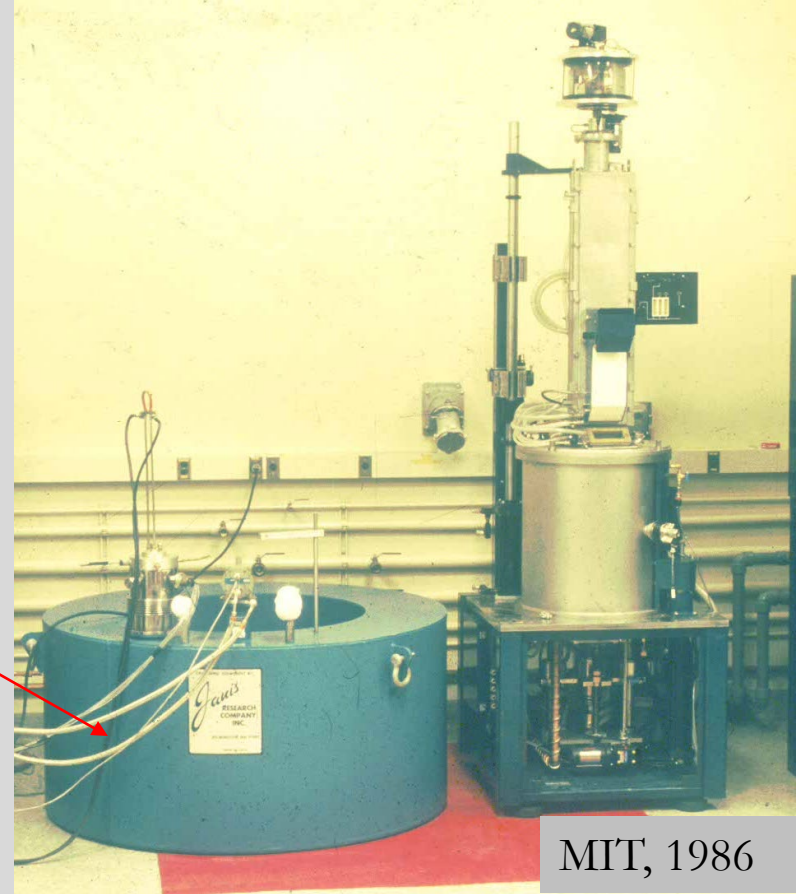
Options to control melt flow:

a) Reduce natural convection:

- Magnetic fields
- Reduced gravity (μg)
- Use a **baffle** in the melt

b) Impose forced convection

- Crystal/crucible rotation (CZ)
- Accelerated Crucible Rotation
- Coupled Vibrational Stirring
- Traveling Magnetic Fields

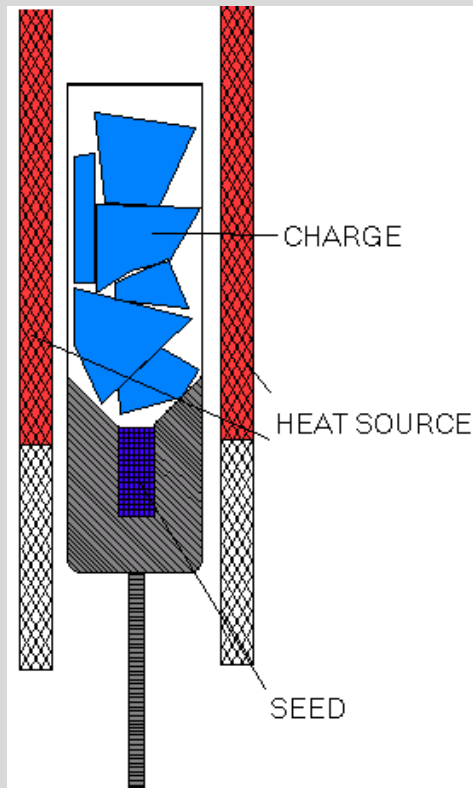


MIT, 1986

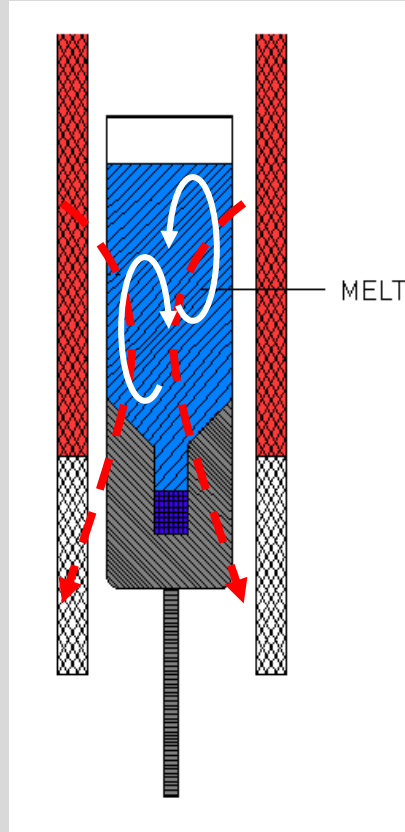
Directional solidification (Bridgman)

HOT

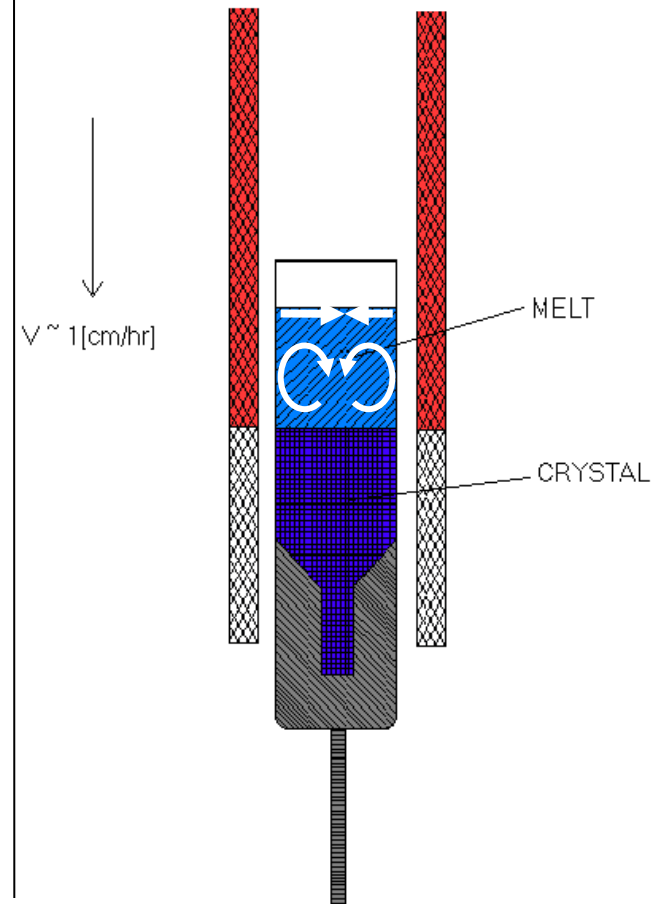
COLD



1. Melting,
2. Seeding



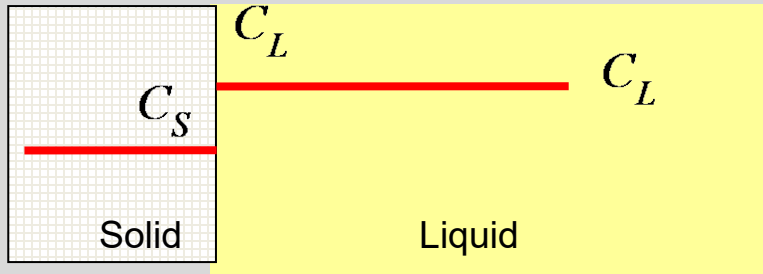
4. Growth: Interface is advanced by moving the container or the gradient



2 Distribution of impurities in the crystal

Solute partitioning at the S/L Interface

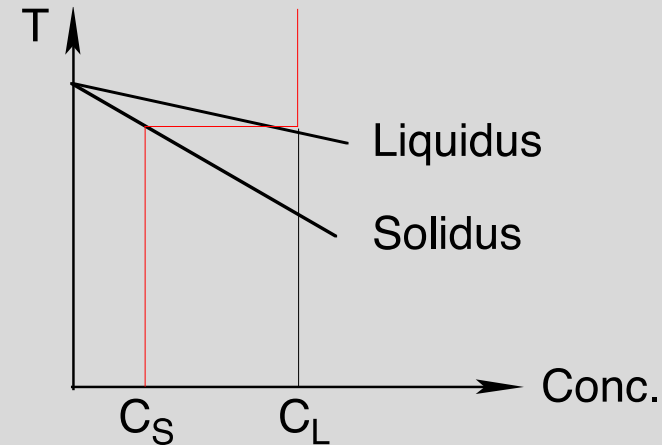
Equilibrium



Solid

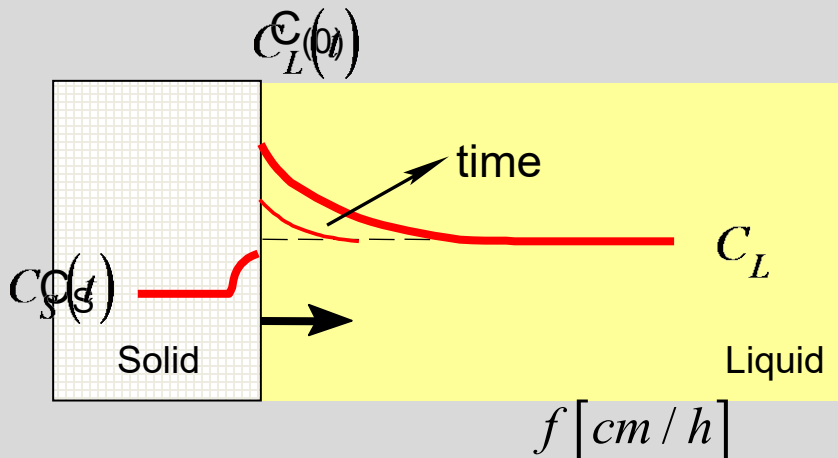
Liquid

$$\frac{C_S}{C_L} = k_0$$

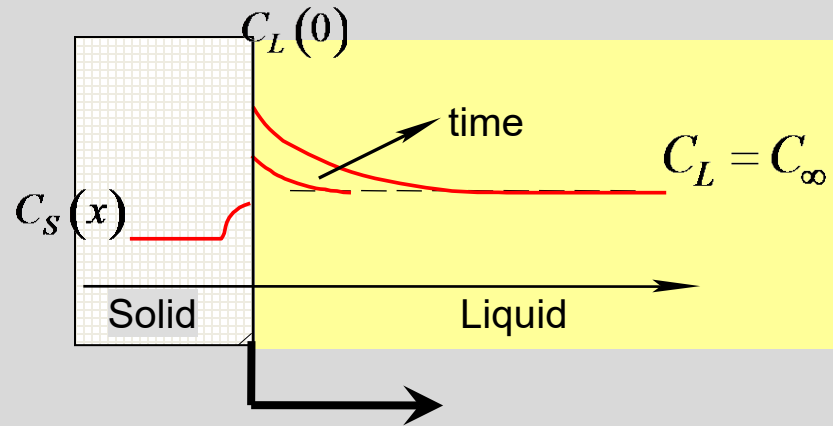


k_0 is a fundamental property; it appears in equations involving redistribution of the solute.

Beginning of motion



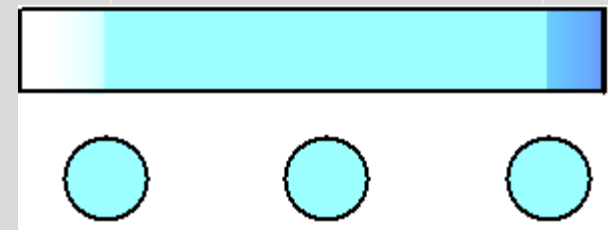
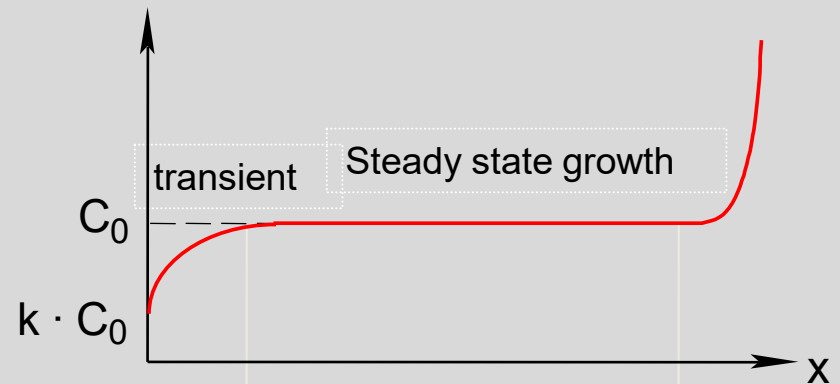
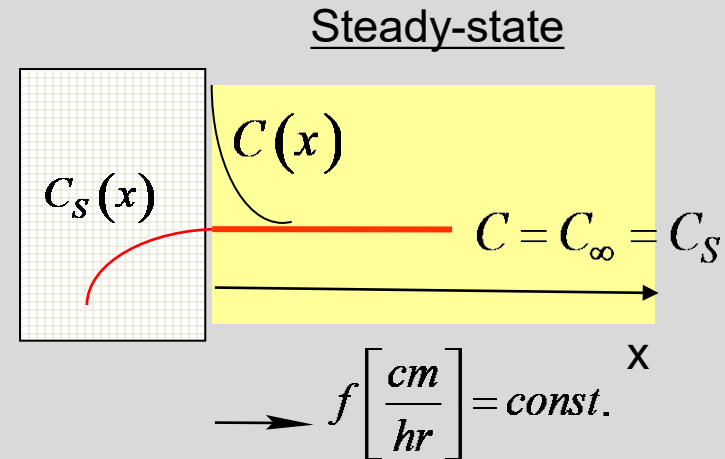
OPTION I: “Ideal Diffusion-Controlled Solidification” ($g=0$)



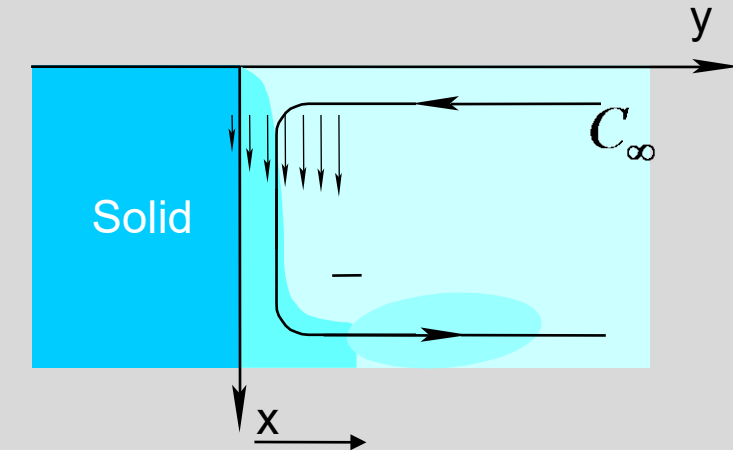
$$f \left[\frac{cm}{hr} \right] = const. = 5 mm / h$$

$$Gr = \frac{g \beta \Delta T L^3}{\nu^2} = \left(\frac{\beta}{\nu^2} \right) (g \Delta T L^3)$$

$g \rightarrow 0$

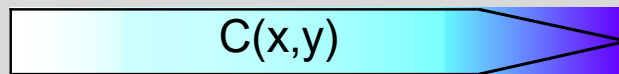
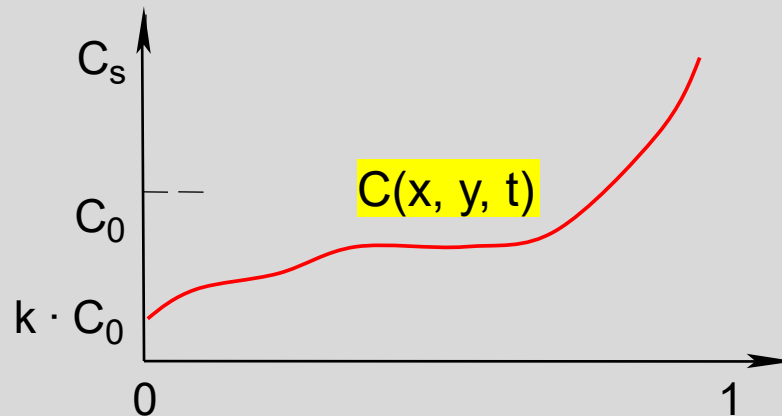


OPTION II: On Earth flow-controlled T and C in the melt



$V(t), T(t), C(t)$

no steady state



$C(x, y)$ axial and radial inhomogeneity

$$\nabla \cdot \vec{V} = 0$$

Continuity

$$\frac{\partial \vec{V}}{\partial t} + (\vec{V} \cdot \nabla) \vec{V} = -\frac{1}{\rho} \nabla p + \nu \nabla^2 \vec{V} + \frac{\vec{F}}{\rho} \quad \text{N-S}$$

$$\frac{\partial T}{\partial t} + (\vec{V} \cdot \nabla) T = \alpha \nabla^2 T \quad \text{Energy}$$

$$\frac{\partial C}{\partial t} + (\vec{V} \cdot \nabla) C = D \nabla^2 C \quad \text{Species}$$

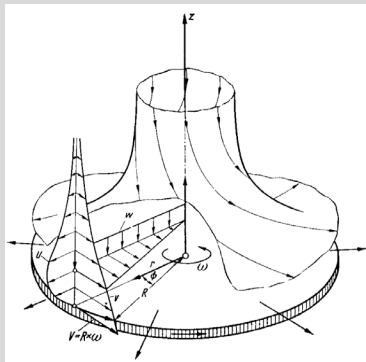
Melt flow is unavoidable

OPTION III: Perfect mixing (i.e. homogeneous melt)

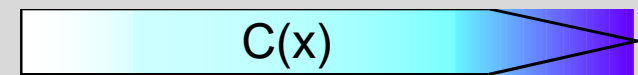
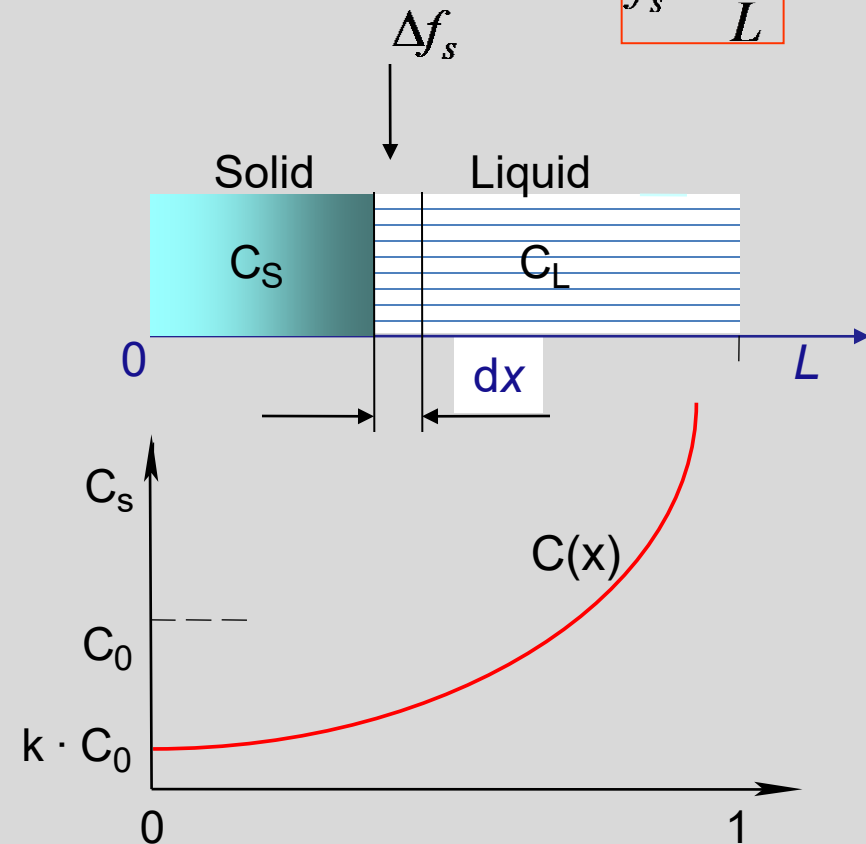
G. Gulliver (1921) and E. Scheil (1942) assumed:

- homogeneous melt
- No diffusion in solid
- Local equilibrium at S/L interface

$$C_S(x) = C_0 \cdot k \left(1 - \frac{x}{L} \right)^{(k-1)}$$



$$f_s \equiv \frac{x}{L}$$



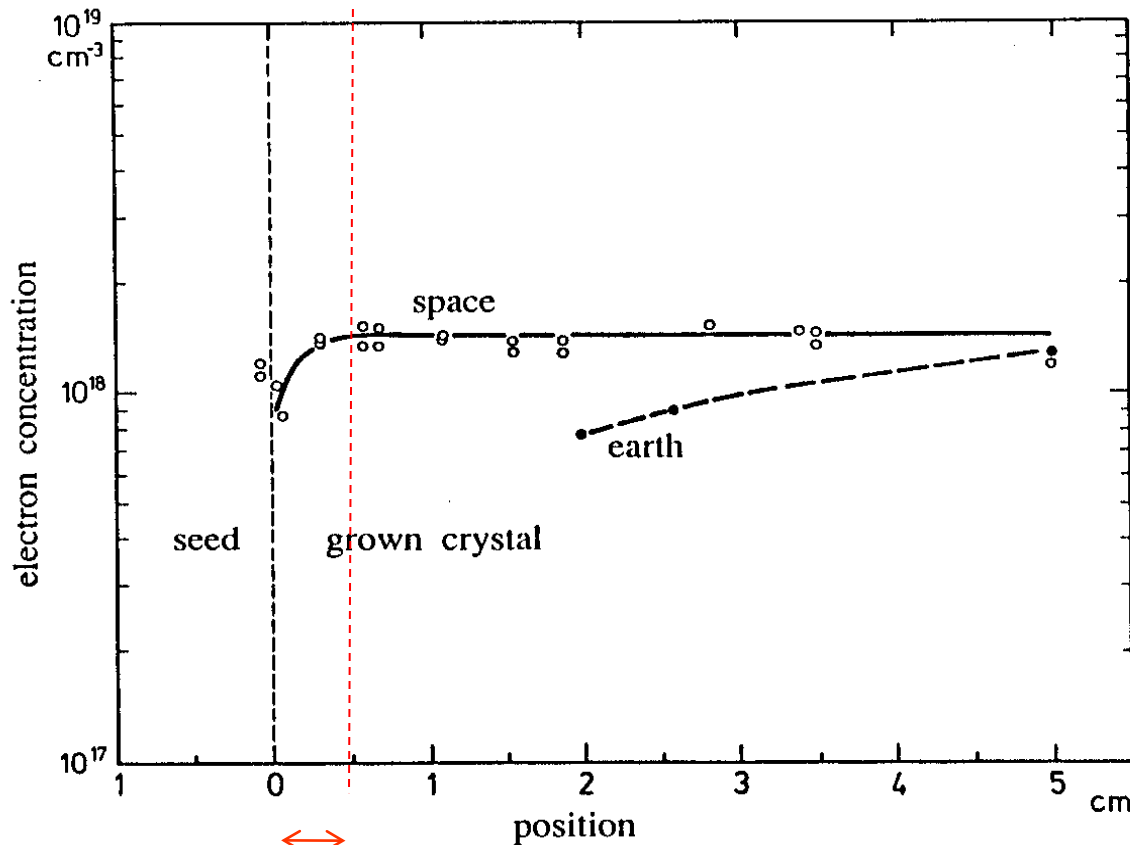
Axial inhomogeneity, $C(x)$

Witt et al. 1975, Skylab III Mission in near-earth orbit

“ideal, diffusion controlled steady state-segregation, never accomplished on earth, was achieved ...”

A.F. Witt et. Al., J. Electrochem. Soc.122 (1975) 276.

Te



Te-doped InSb ($k_0 = 0.5$)

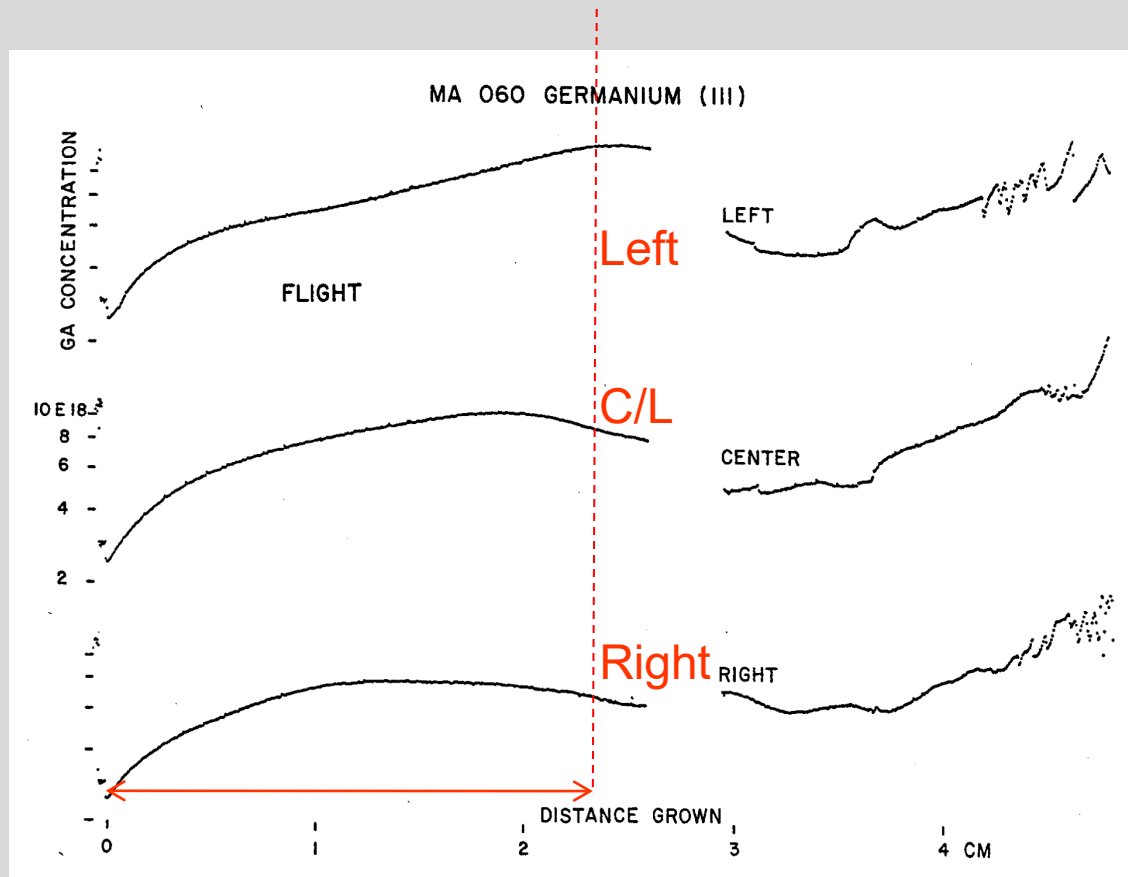
- Initial transient 0.5 cm
- Steady state

ALI

Sn doped InSb ($k_0 \sim 0.01$)

L transient

Apollo-Soyuz 1978, Witt et al.: Ga-doped Ge ($k=0.087$)



“Existing gap between theory and experimentis largely attributed to gravity-induced convection effects..”

- $k_0 \sim 0.087$;
- 3 cm/h
- Initial transient ~ 3 cm
- no Steady state

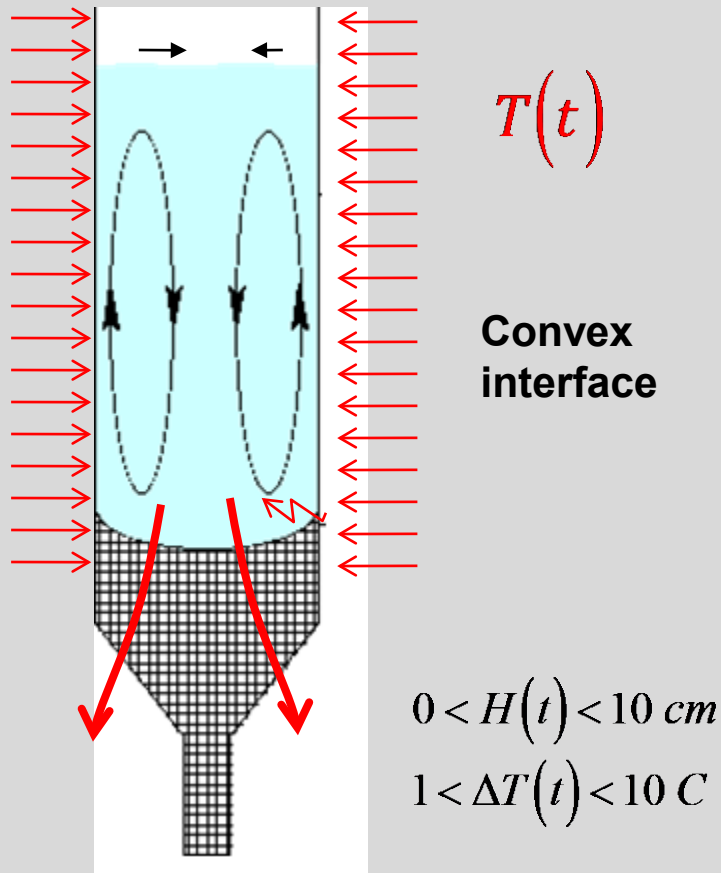
$$L_{\text{transient}} \sim 3 \text{ cm}$$

$$Gr = \frac{g\beta\Delta T L^3}{\nu^2}; \quad g \sim 10^{-4} g_0$$

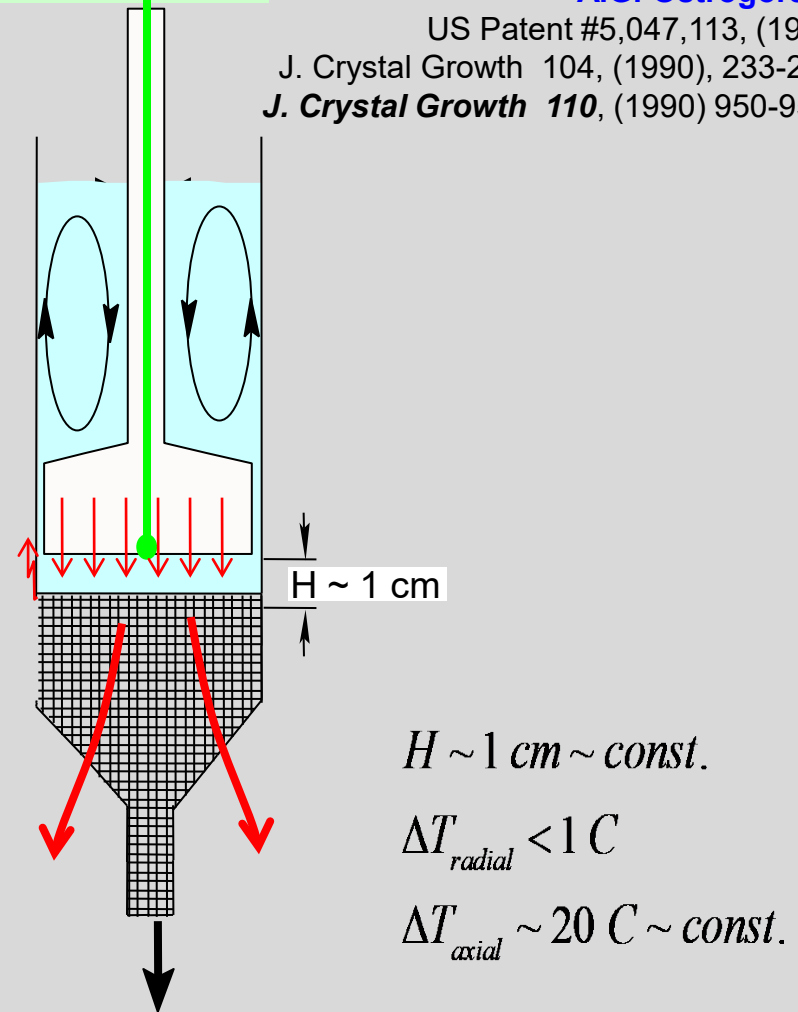
3. Using the baffle on Earth and ISS (SUBSA investigations).

Crystal Growth using a Baffle in Bridgman furnace (1990-)

A.G. Ostrogorsky,
US Patent #5,047,113, (1991)
J. Crystal Growth 104, (1990), 233-238.
J. Crystal Growth 110, (1990) 950-954



$$Gr = \frac{g \cdot \beta \cdot \Delta T \cdot H^3}{\nu^2} \sim g \cdot \Delta T \cdot H^3$$



$$\Delta T_{\text{radial}} \sim 10 \times \text{reduced}$$

$$H^3 \sim 1000 \times \text{reduced}$$

Ga-doped Ge, $k=0.087$

Bridgman

Apollo-Soyuz 1978

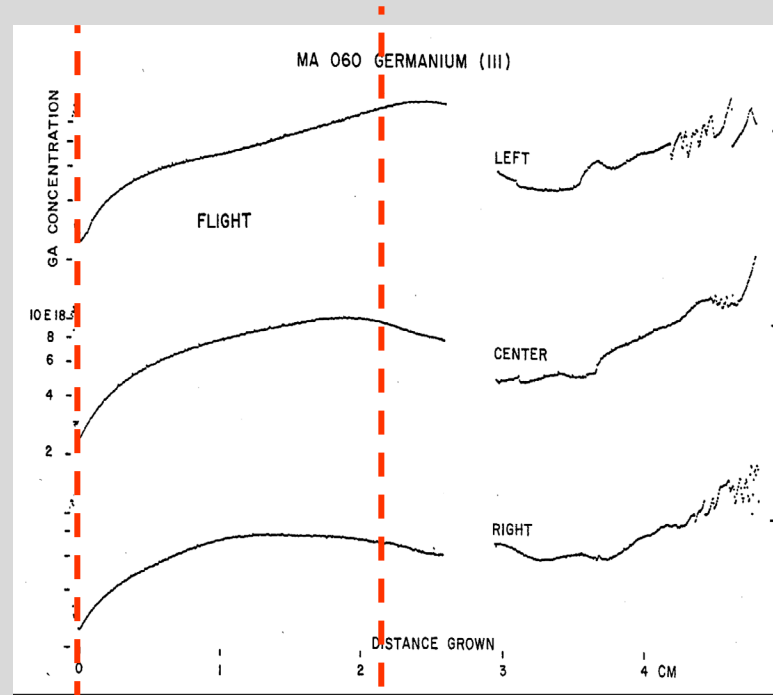
Witt et al.

$$10^{-5} g_0$$

Ge:Ga;

$V_R=30\text{mm/h}$

$D=1.7 \times 10^{-4} \text{ cm}^2/\text{s}$



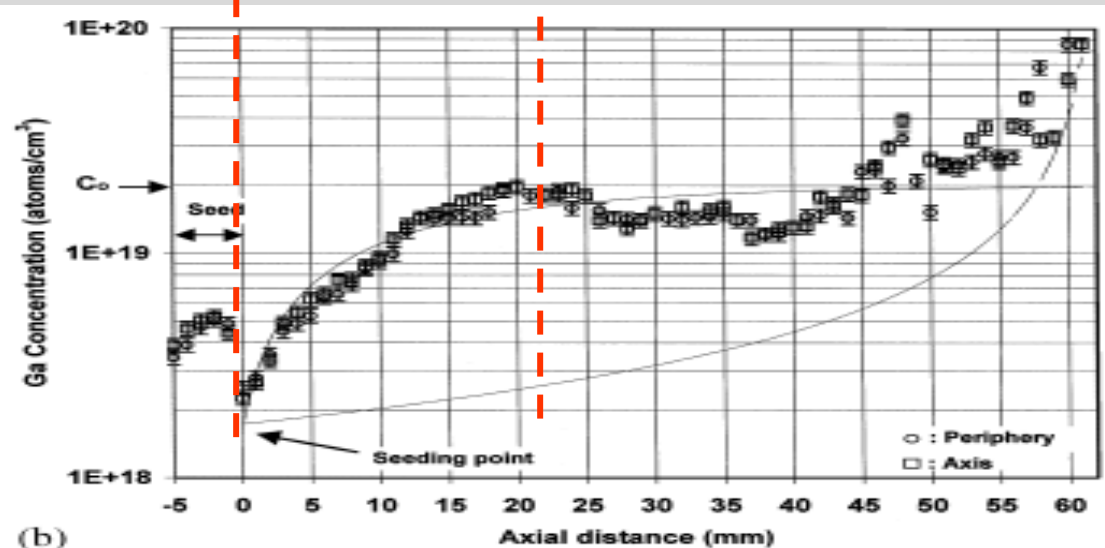
Bridgman (baffle)

Dutta and Ostrogorsky,
J. Crystal Growth **217**
(2000) 360-365.

$V_R=30\text{mm/h}$

$$g_0$$

Grown with the baffle at $1 g_0$



1. **SUBSA 1**: Solidification Using a **Baffle** in Sealed Ampoules (**InSb**)

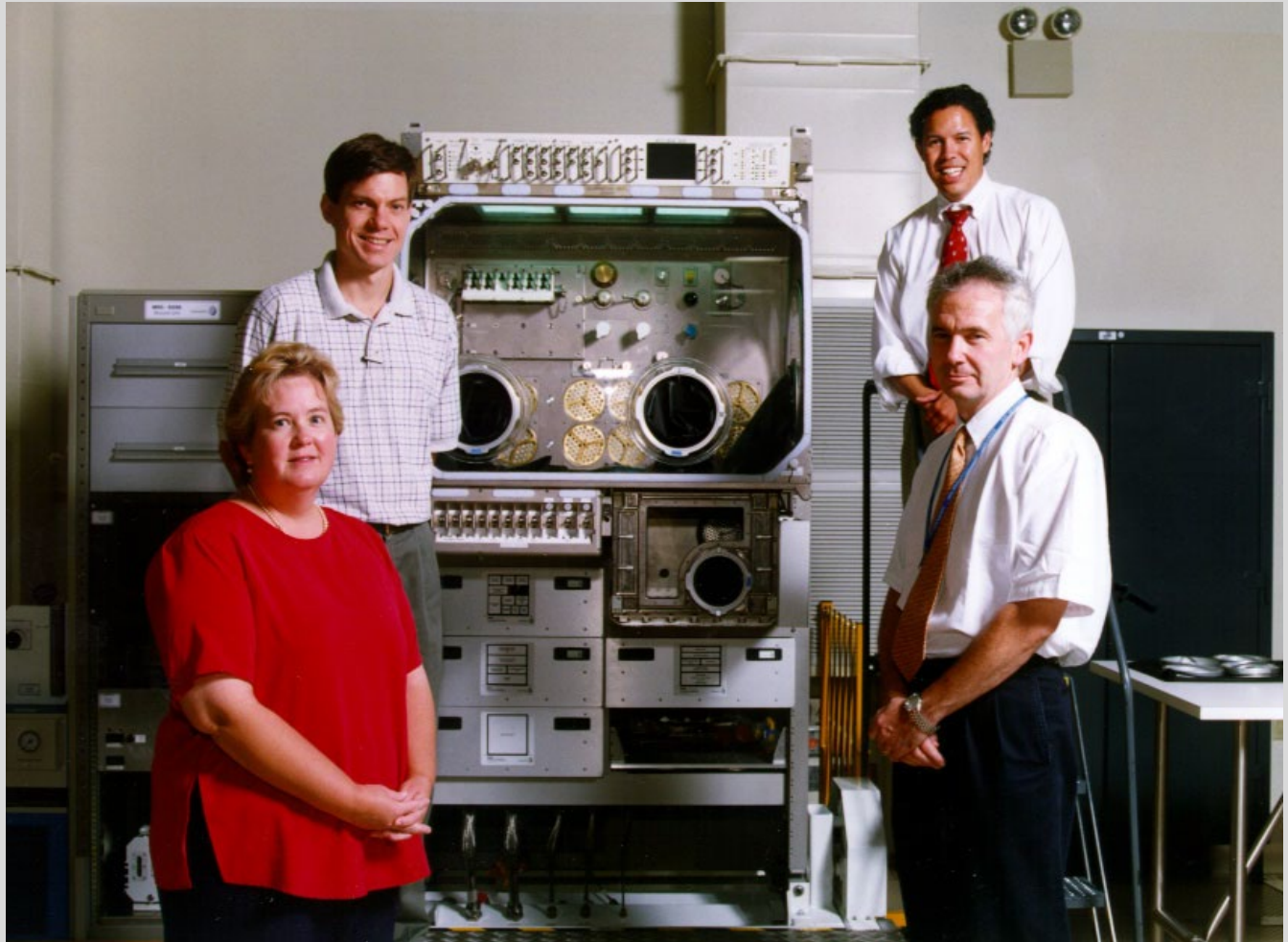
1993-2004

2. Detached Melt and Vapor Growth of InI in **SUBSA** Hardware (**InI**)

2016-present

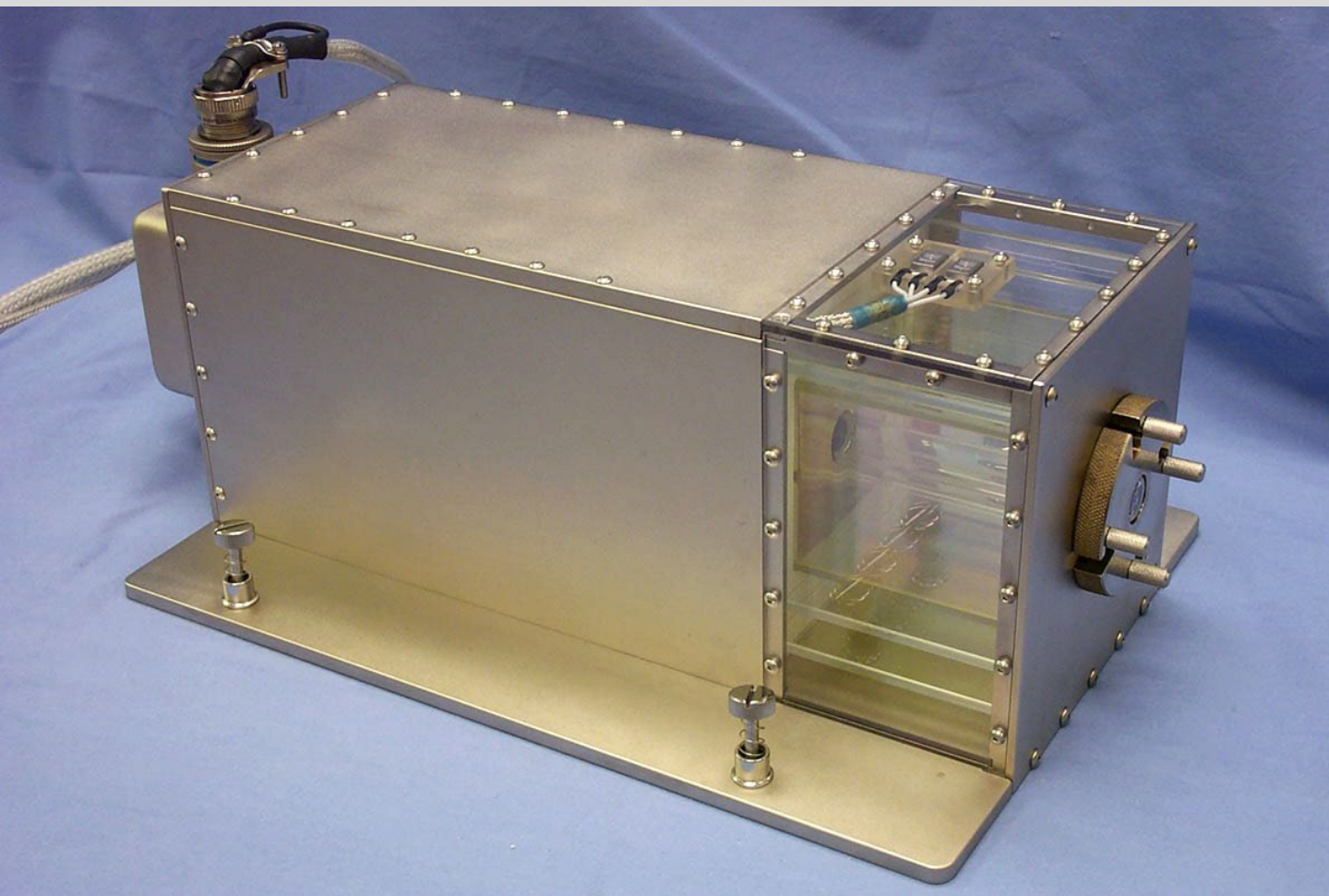
3. Diffusion Coefficients of Dopants in molten **Ge** and **Si**

2020-2025



• A.G. Ostrogorsky, PI, Rensselaer Polytechnic Institute (RPI)

SUBSA Hardware



1 DaqPad



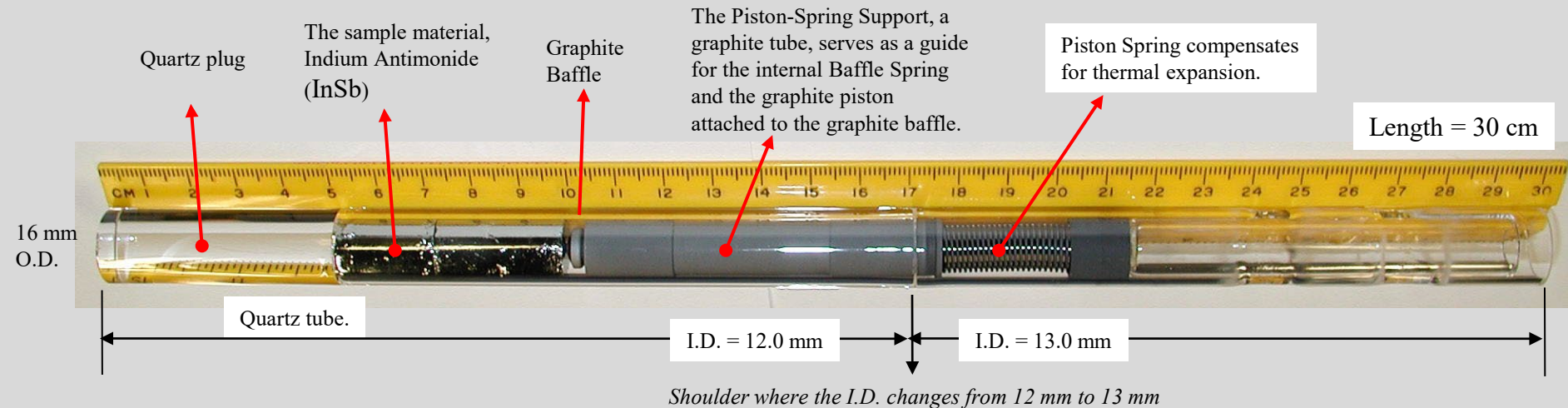
1 Process Control Module

It's a Boy !

Born May 8, 2001 at 10:35 p.m.

Weight: 14 lbs, 9 oz; Length: 15 inches

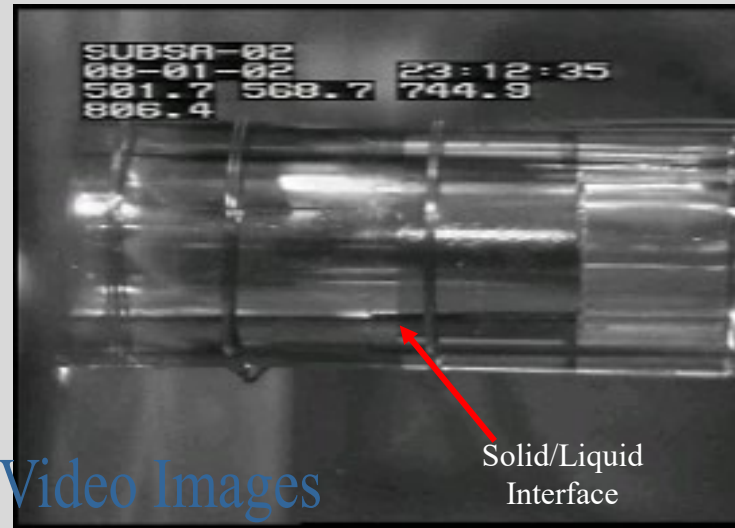
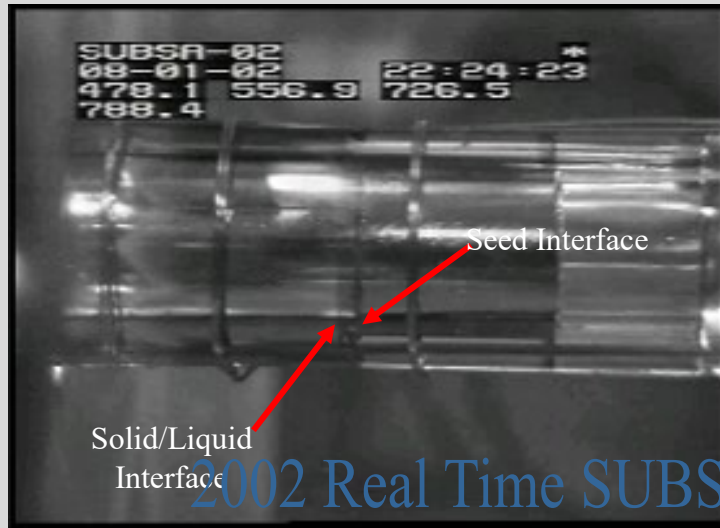
SUBSA Sample Ampoule (Baffle Assembly)



- Prepared on the ground before flight by loading the processing materials (up to 50g InSb doped with up to 50mg of Cd, Te, Ga, Zn, Ca, or As) into the ampoules. Some of the ampoules may also have up to 2g of a LiCl-KCl liquid encapsulant.
- The ampoules will be sealed under a $10E-3$ torr or better vacuum. A cartridge head and four (4) external thermocouples will then be added to finish the Sample Ampoule Assembly (SAA).

Solidification Using a Baffle in Sealed Ampoules

2002 Space Shuttle Endeavour MSG, ISS

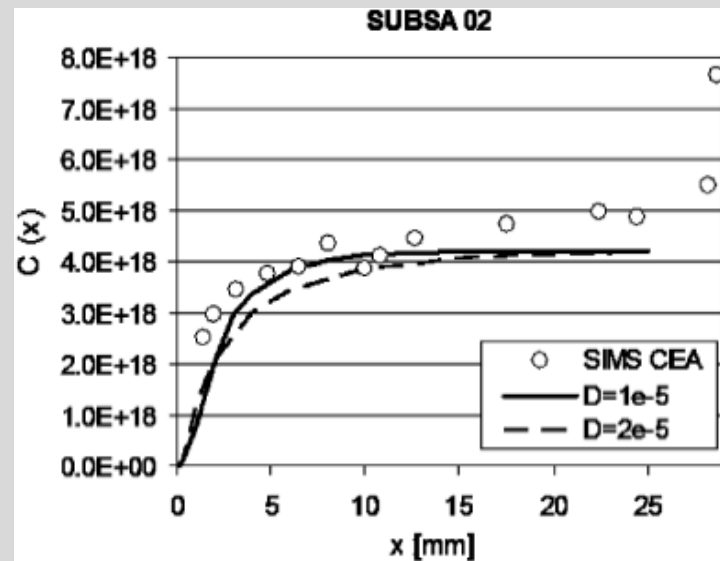
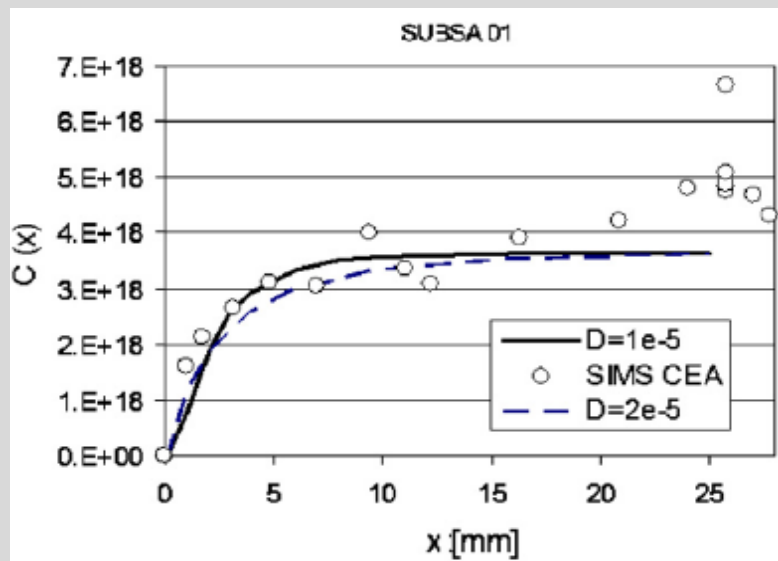
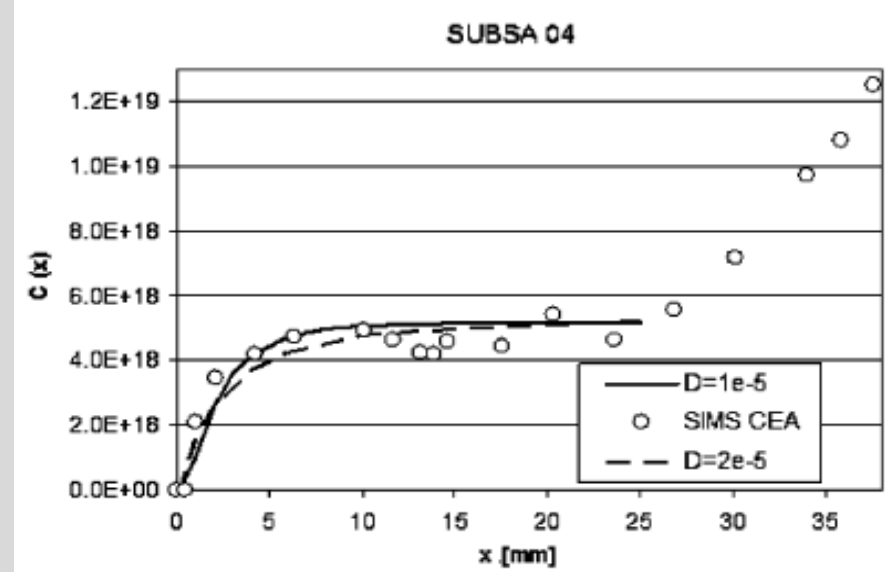
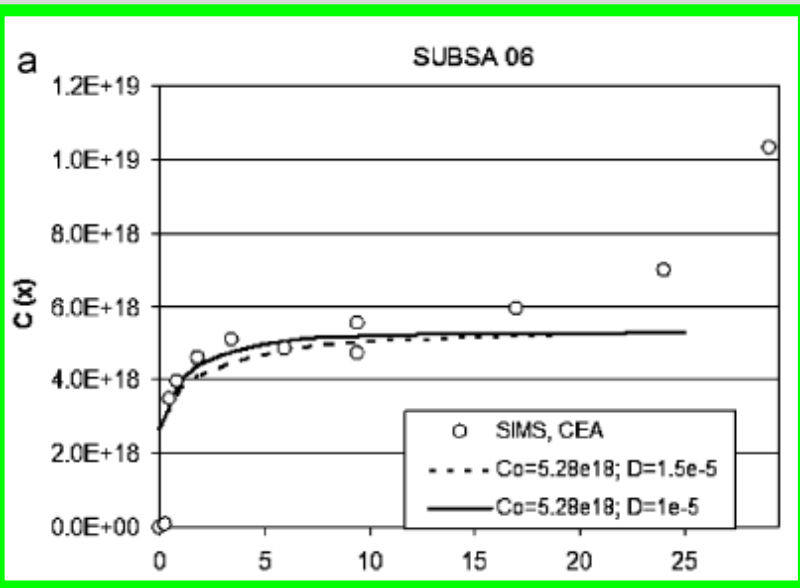


ISS, July 6, 2002, Peggy Whitson
1st experiment in MSG

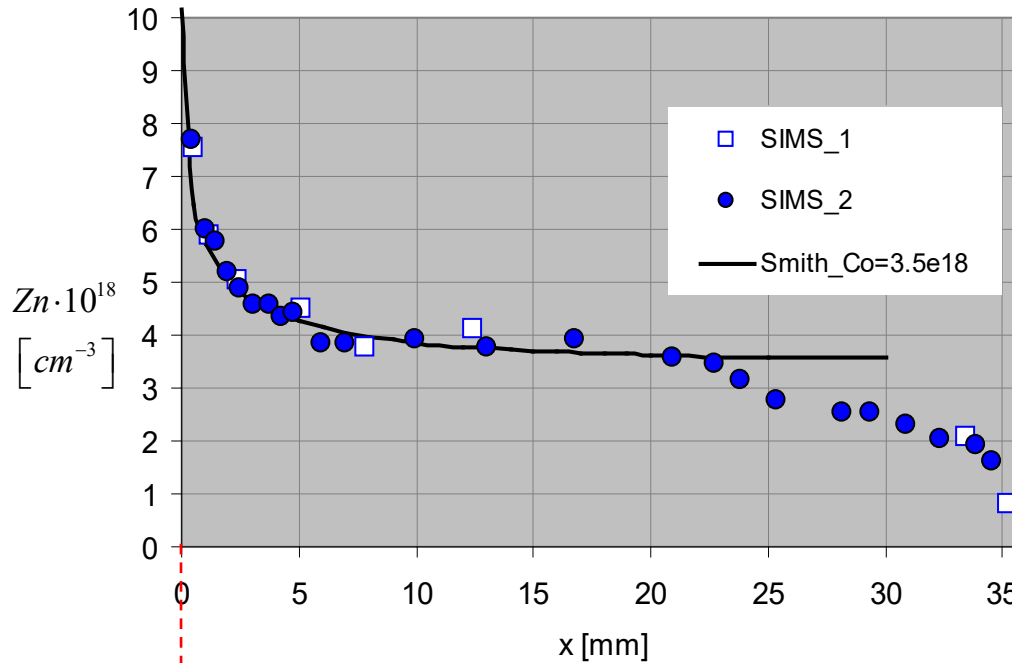


Monitoring room at RPI

SUBSA Te-doped InSb

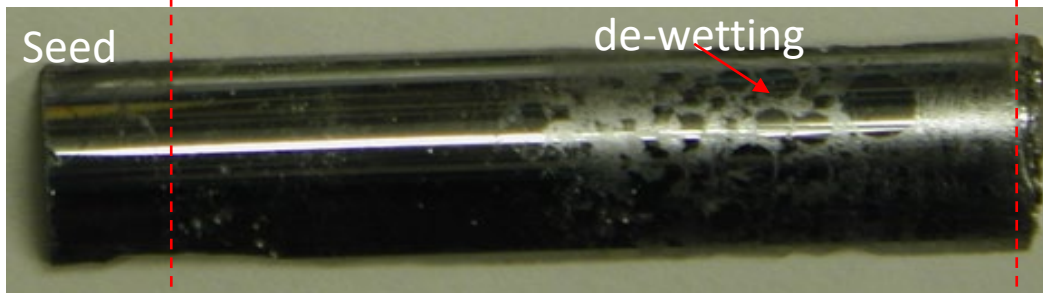


$$D = 1 \cdot 10^{-5} \left[\frac{cm^2}{s} \right]$$



Zn-doped => $k=2.9$
Te-doped => $k=0.5$

$k_0 > 1$ is proffered for growth in microgravity.



$$D = 1.2 \times 10^{-5} \text{ cm}^2/\text{s}$$

$k > 1$ is shortening the initial transient in microgravity

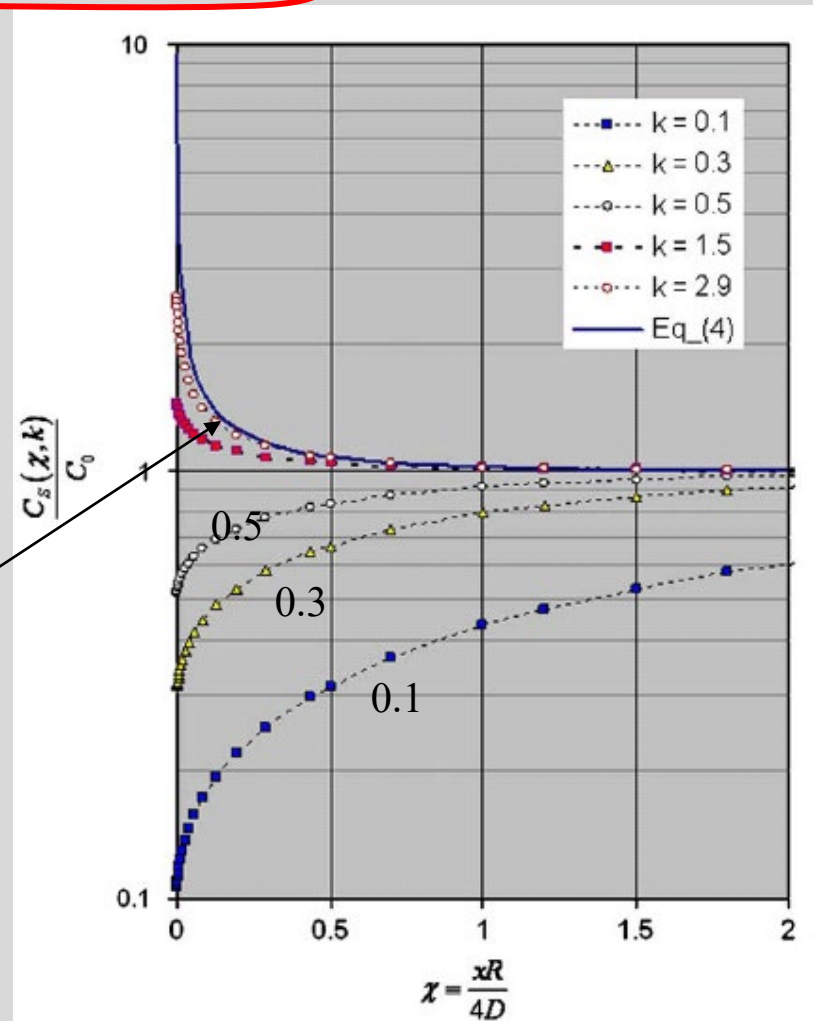
A.G. Ostrogorsky, C. Marin, M. Volz, T. Duffar, *J. Crystal Growth* (2009)

$$\frac{C_s(\chi, k)}{C_0} = \frac{1}{2} \left\{ 1 + \text{erf} \sqrt{\chi} + (2k - 1) e^{-4k(1-k)\chi} \text{erfc} \left[(2k - 1) \sqrt{\chi} \right] \right\}$$

$$\chi = \frac{xf}{4D}$$

$$\frac{C_s(\chi)}{C_0} = \phi(\chi) + F(k, \chi)$$

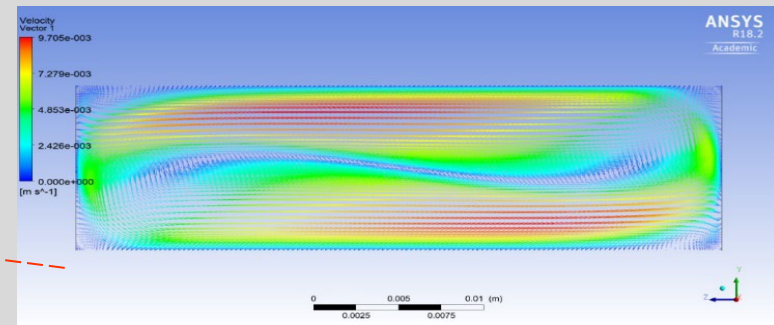
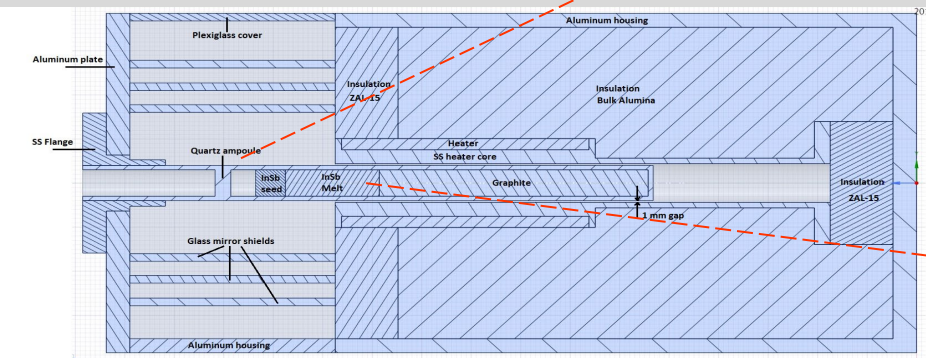
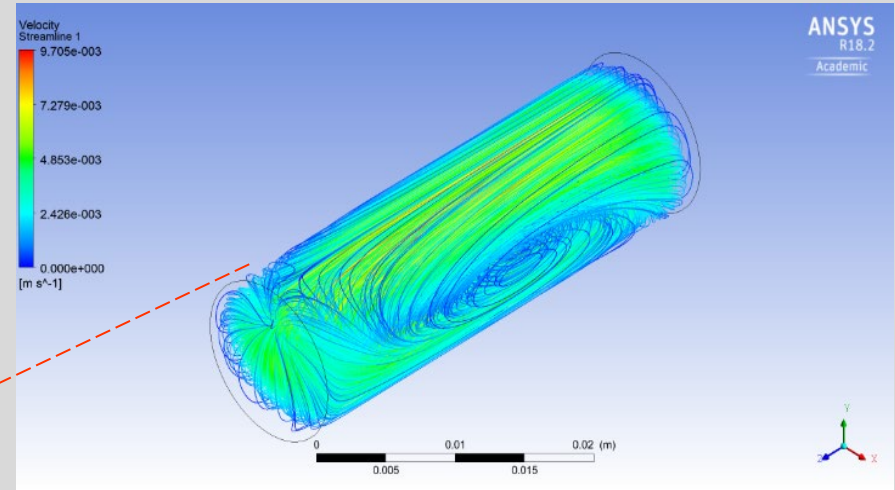
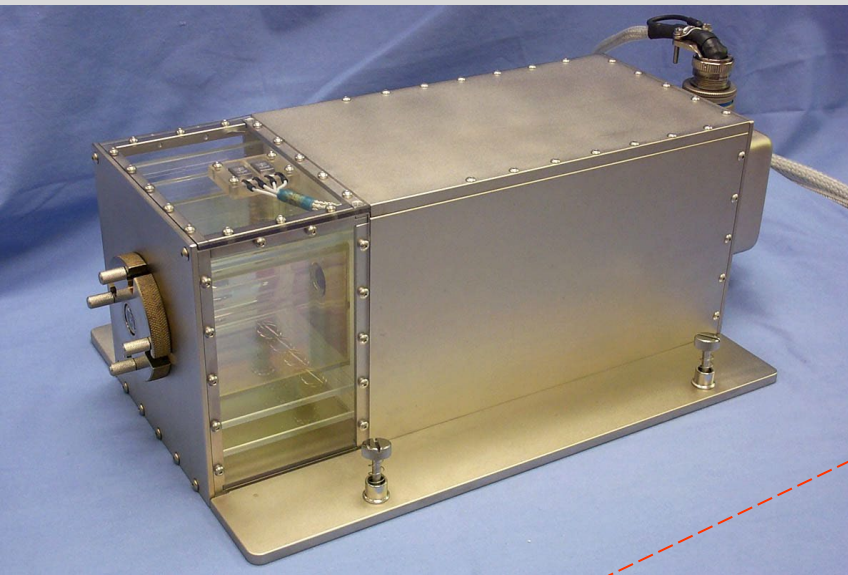
- $0 < k < 0.5$, F is negative, prolonging the transient.
- $k > 0.5$, F is positive, shortening the transients.



$$\frac{C_s(\chi)}{C_0} = \frac{1}{2} \left\{ 1 + \text{erf} \sqrt{\chi} + \frac{e^{-\chi}}{\sqrt{\pi\chi}} \right\} \quad (4)$$

A.G. Ostrogorsky, et al., *J. Crystal Growth* (2009)

Present work: detailed ANSYS model of the SUBSA furnace



μg



Reproduce the SUBSA experiments

- InSb ($k \sim 10$ W/m-K)
- InI ($k \sim 0.5$ W/m-K)
- On Earth and at ISS
- Useful for future investigations in SUBSA facility

II-SUBSA 2015-present

The refurbished SUBSA flight unit was launched in the Fall of 2017



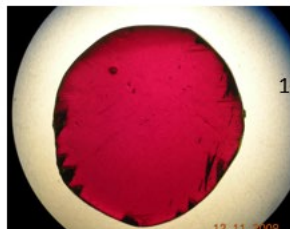
Detached Melt and Vapor Growth of **InI** in SUBSA Hardware (IIT, Ostrogorsky)

Crystal growth of **CLYC (Cs₂LiYCl₆:Ce)** in Microgravity (RMD, A. Churilov, J. P. Tower)



Why indium iodide?

- Promising semiconductor RT detector material + not toxic; MP= 360 C (perfect for SUBSA furnace)
- Developed procedures for synthesis, ZR, melt growth, vapor growth
- RPI (2006-2009); IIT (2009-present), RMD (2015).
- DoE, NNSA



15 mm diameter

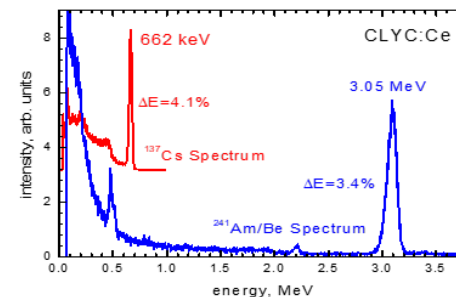
16



CLYC (Cs₂LiYCl₆:Ce)

CLYC is the first commercial dual-mode scintillator:
gamma-ray spectroscopy AND neutron detection

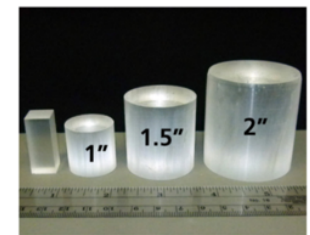
Fast, accurate isotope identification



Gamma and Neutron Spectra



CLYC crystals



CLYC instruments



RadEye

RIIDEye

Properties not known

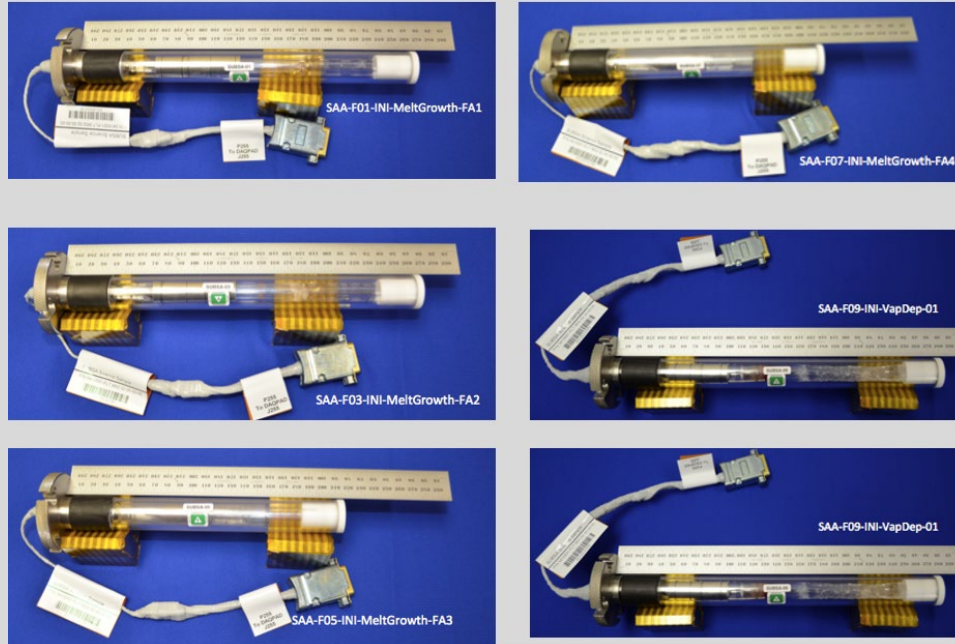
- 18 April, 2017 Launched Orbital ATK OA-7 Cygnus mission to ISS
- Four crystal growth experiments conducted in March-April 2019
- 4th melt growth experiment was conducted on January 4, 2021
- 2 vapor growth experiments were completed



David St Jacques March 2019

Detached Melt and Vapor Growth of InI in SUBSA Hardware

SUBSA 2 InI Sample Ampoule Assembly



Note:
Major difference in properties
between InSb and InI

III Diffusion Coefficients of Dopants in molten Ge and Si 2020- present

Numerical simulations require precise values of D .

$$\text{FM} \begin{cases} \nabla \cdot \vec{V} = 0 \\ \frac{\partial \vec{V}}{\partial t} + (\vec{V} \cdot \nabla) \vec{V} = -\nabla p + \nabla^2 \vec{V} + \text{Gr}(T_0 - T) \vec{i}_g \end{cases}$$

$$\text{Heat} \quad \frac{\partial T}{\partial t} + (\vec{V} \cdot \nabla) T = \frac{1}{\text{Pr}} \nabla^2 T$$

$$\text{Mass} \quad \frac{\partial C}{\partial t} + (\vec{V} \cdot \nabla) C = \frac{1}{\text{Sc}} \nabla^2 C$$

Boundary Condition (BC) at S/L interface:

$$-\frac{\partial C}{\partial x^*} \Big|_{x=0} = \text{Pe}(1 - k_0)C_0 = \text{Nu}(C_0 - C_L)$$

FC	NC	HT	MT
$\text{Re} = \frac{V \cdot L}{\nu}$	$\text{Gr} = \frac{g \beta \Delta T L^3}{\nu^2}$	$\text{Pr} = \frac{\nu}{\alpha}$	$\text{Sc} = \frac{\nu}{D}$

$$k_0 = \frac{C_s}{C_0}$$

$$\text{Pe} \equiv \frac{f \cdot L}{D}$$

$$\text{Nu} \equiv \frac{h \cdot L}{D}$$



Solidification and Quenching Low Gradient Furnace (LGF)

ACKNOWLEDGEMENTS

NASA MSFC

Dr. Martin P. Volz, Project Scientist, co-investigator

Dr. Fernando Reyes Tirado, Paul Luz, Linda Jetter

Prof. Dr. Arne Cröll, U. Freiburg, UAH.

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