

# Interpretability and Generalizability of Constitutive Models using Symbolic Regression

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## 1. Interpretable, Data-driven Machine Learning Background

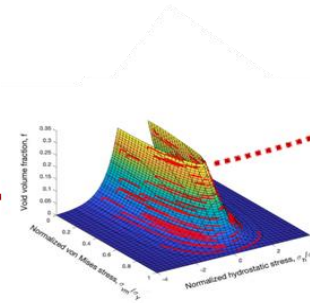
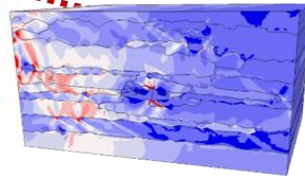
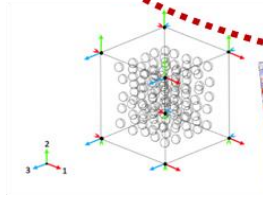
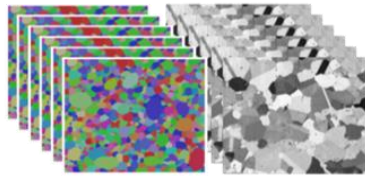
- What do we mean by ‘interpretability’ and ‘generalizability’
- Genetic Programming with Symbolic Regression (GPSR)
- Data-driven interpretable ML for constitutive models

## 2. Laying a Scientific Machine Learning Foundation

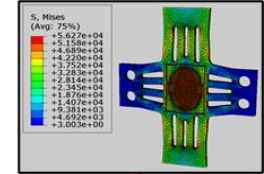
- Prescribing the model domain with Continuum Thermodynamics
- Verification studies for ML material models
- Breaking down microstructure complexity into bite-sized steps

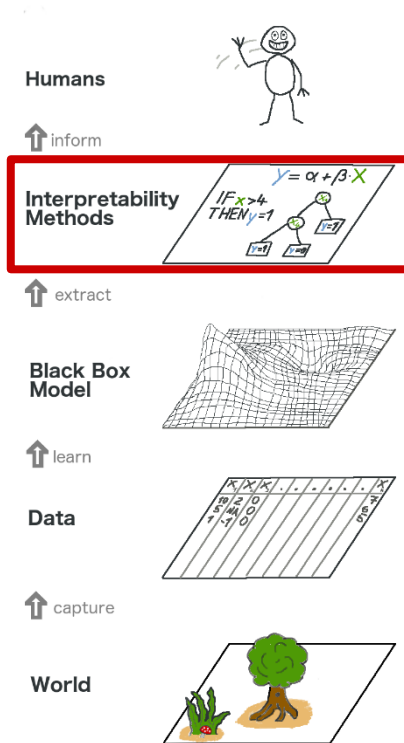
## 3. Finite Element Method Auto-implementation

- Tensor transform method for surface mapping



Gurson yield surface in  $(\sigma_h, \sigma_{vm}, f)$  space





“Interpretability is the degree to which a human can understand the cause of a decision”<sup>1</sup>

<sup>1</sup>Miller, Tim. (2017) "Explanation in artificial intelligence: Insights from the social sciences." arXiv:1706.07269.

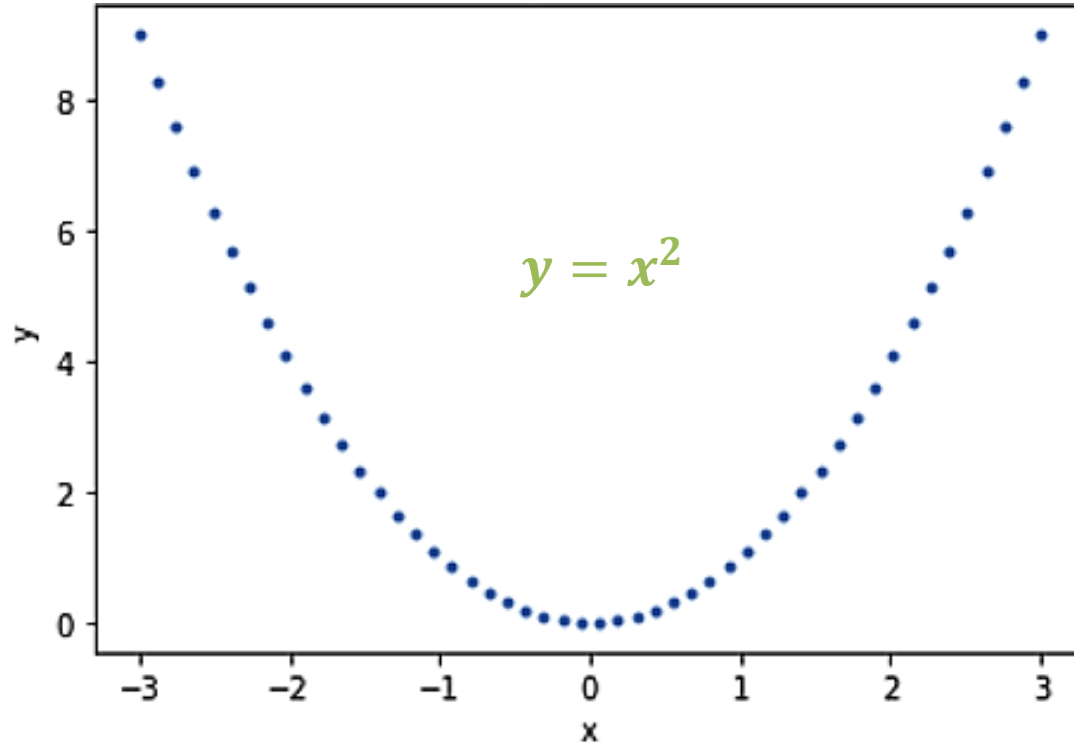
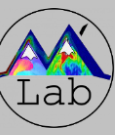
Engineers and scientists seek **interpretability** for:

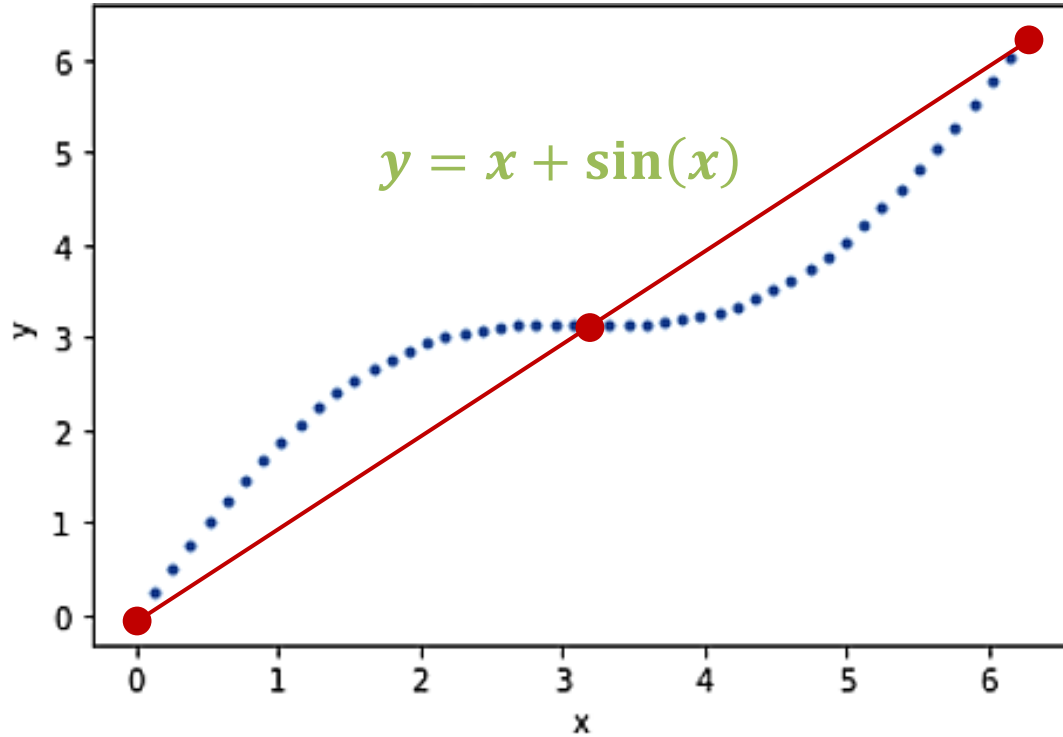
- Building trust
- Directing future data collection
- Informing feature engineering
- Informing human decision-making

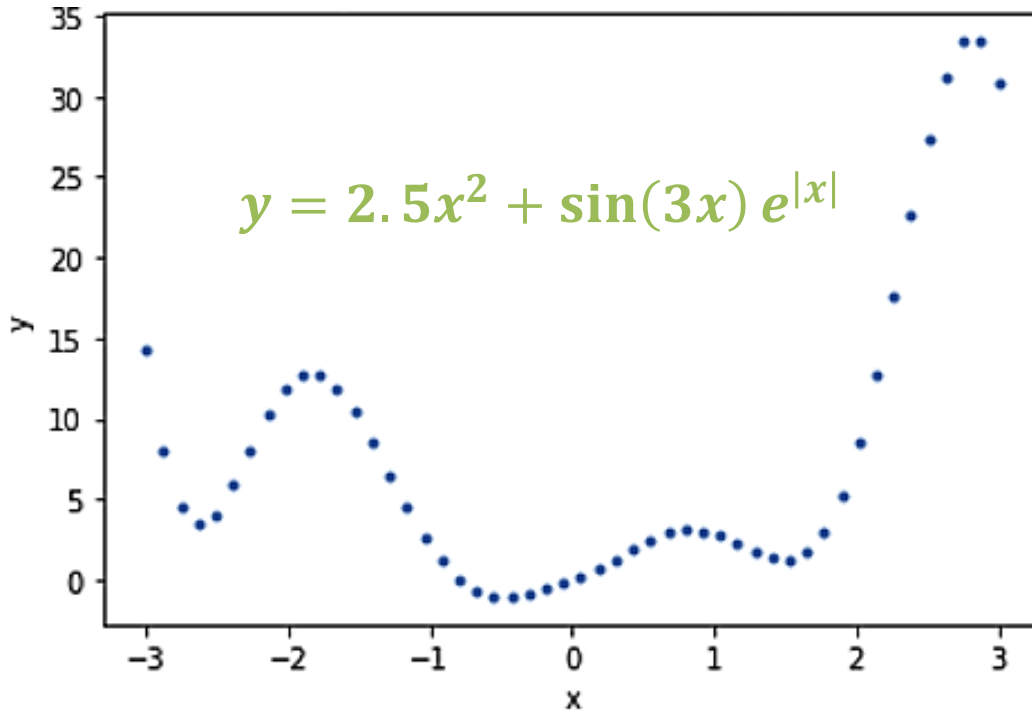
Analytic models are inherently **interpretable** and common insights, like feature sensitivities, are readily obtained.

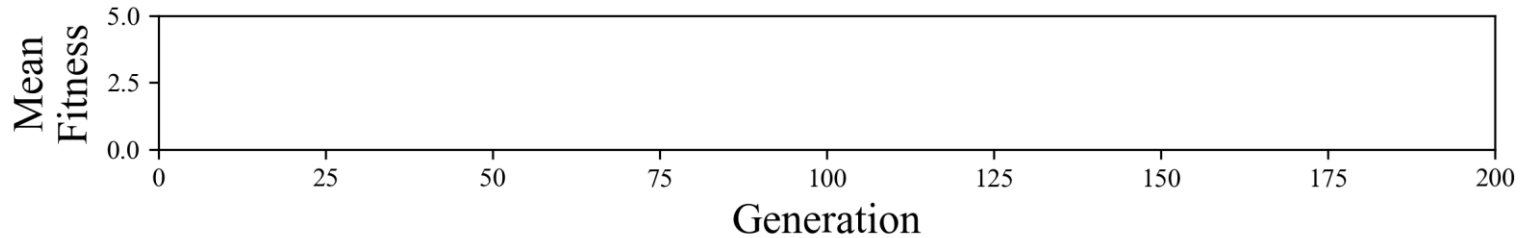
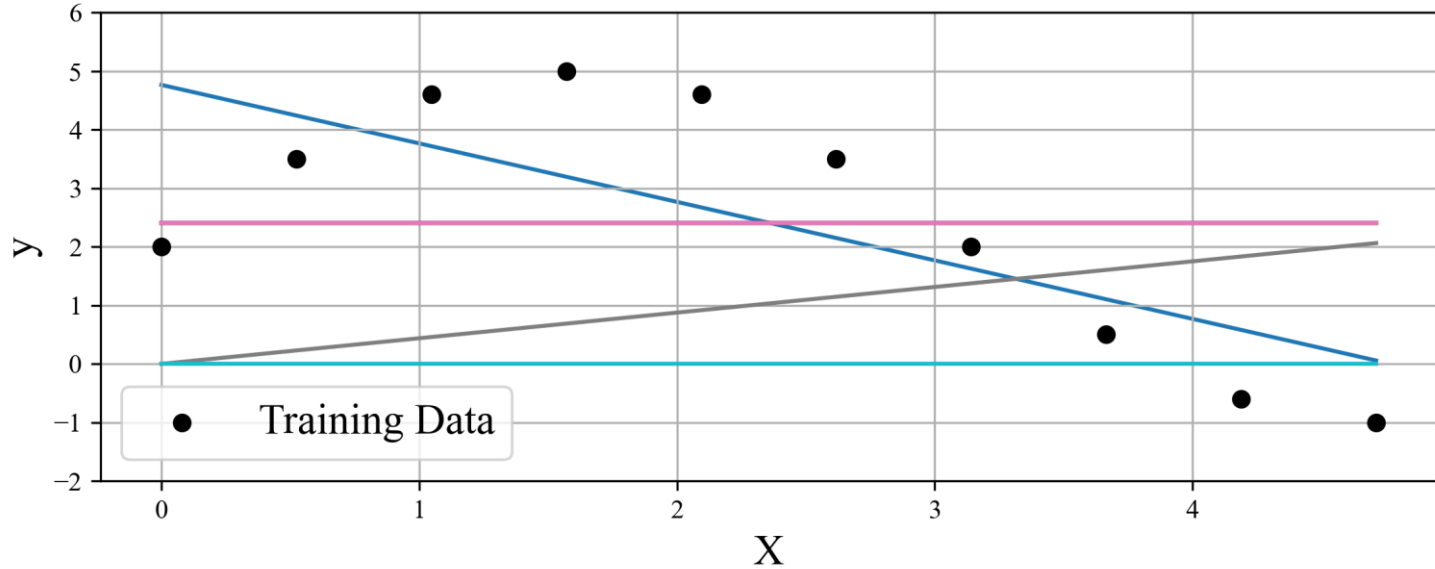
**Explainable** models go beyond the **interpretable** to enable justification of why a model is **generalizable**.

# Name That Model









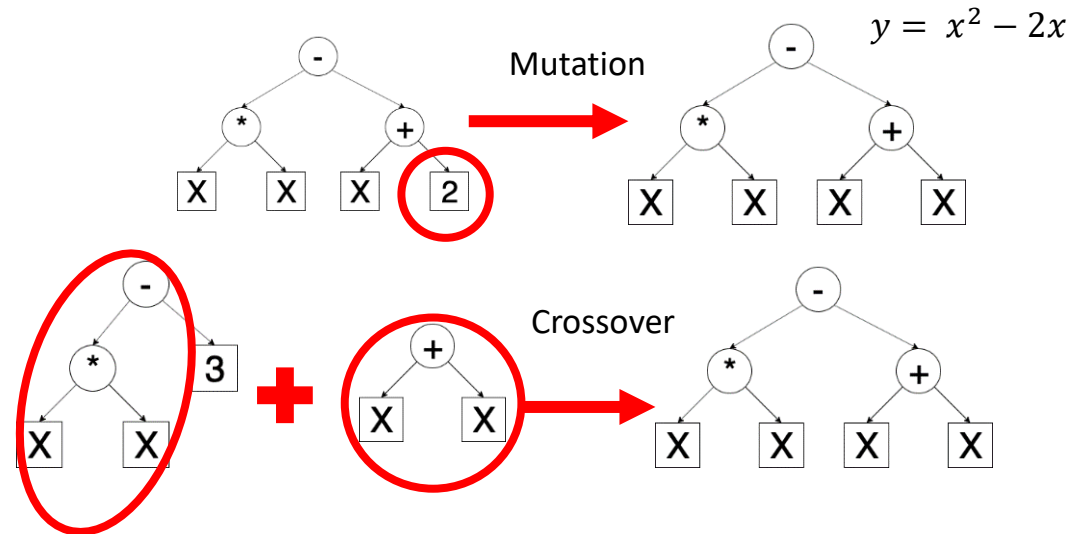
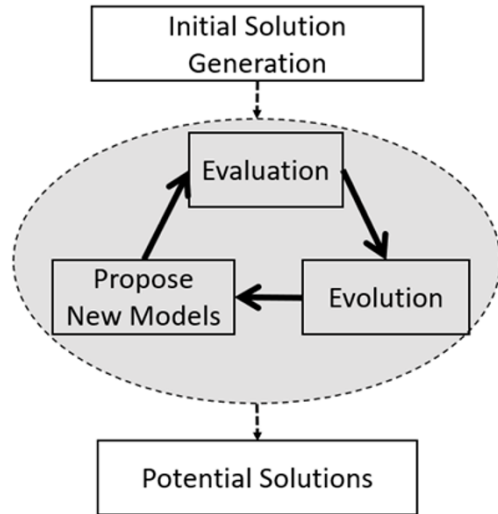
## GPSR Algorithm Concept

**Genetic Programming (GP):** Evolution of computer programs

**Symbolic Regression (SR):** Searching space of mathematical functions

**Fitness Function:** Definition of how model matches data

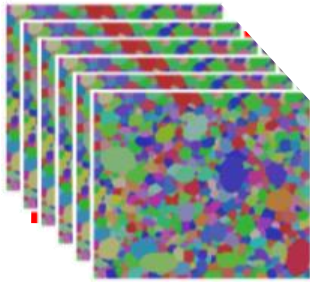
(1) generate population of equations, (2) create offspring, (3) evaluate fitness, (4) select equations





DREAM.3D

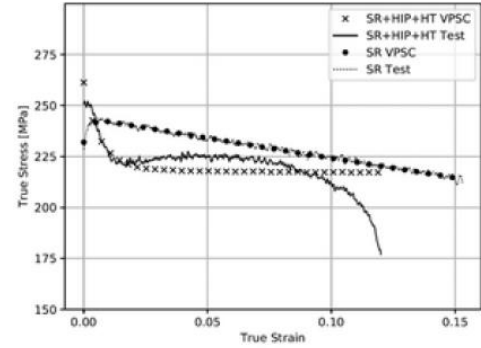
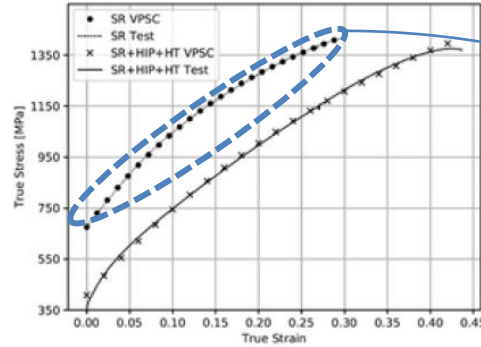
EBSD .ctf data ("Middle")



<http://dream3d.bluequartz.net/>

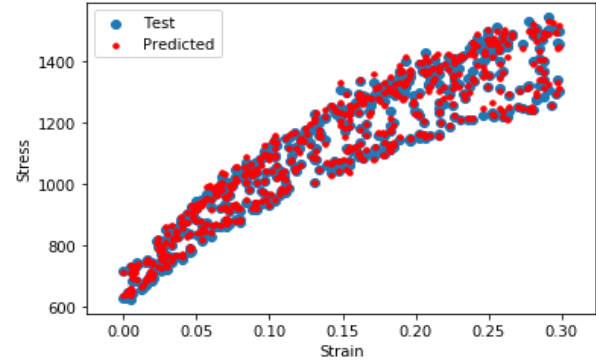
\*VPSC + scipy

Calibrate VPSC to experiment

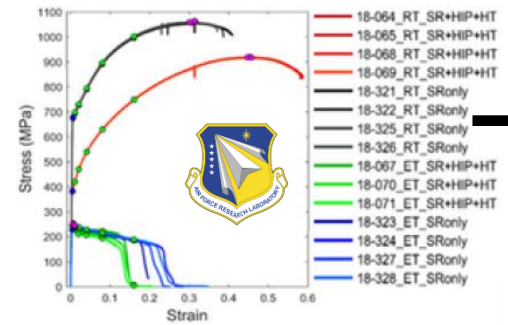


Generation 1000 simulated 'Tests'

- Subsample grain orientations
- Test GPSR model against VPSC test simulations



*\*Example for SR at room temperature*



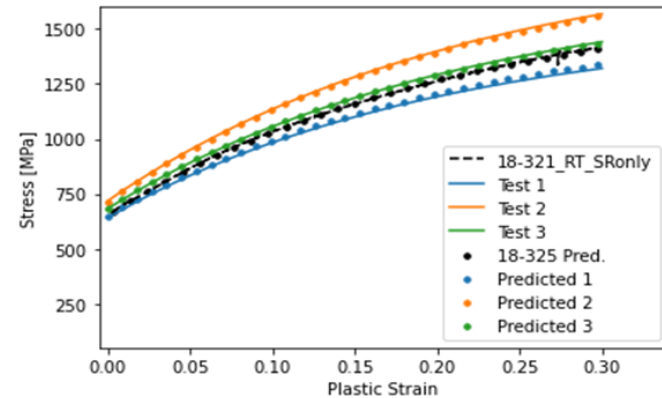
\*Viscoplastic self-consistent (VPSC)

$$\sigma = (-0.16)(-2.5E4 + 1.3E4\alpha) + \left( \frac{\beta}{\cos(\beta)} - \beta \right) (-2.5E4 + 1.3E4\alpha) + (3.6E - 3) \left( \sin(-3.7E3 - (\epsilon)) \right) (-2.5E4 + 1.3E4\alpha)$$

$$\sigma = c_1 F(\alpha) + G(\beta) F(\alpha) + H(\epsilon) F(\alpha)$$

$\epsilon$  = Plastic Strain (%)    $\alpha$  = Texture Parameter 1    $\beta$  = Texture Parameter 2

3 Randomly sampled specimen stress-strain testing data vs. GPSR model result



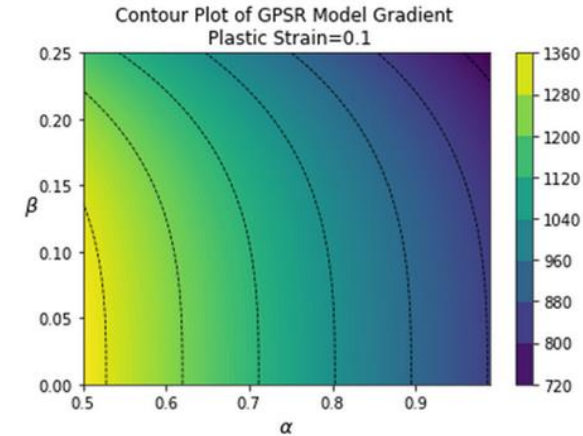
$$\frac{d\sigma_{pred}^2}{d\alpha} = c_1 * 1.26e4 + G(\beta) * 1.26e4 + H(\epsilon) * 1.26e4$$

$$\frac{d\sigma_{pred}^2}{d\beta} = \left( \frac{\beta \sin(\beta)}{\cos^2(\beta)} + \frac{1}{\cos(\beta) - 1} \right) F(\alpha)$$

CV Error Metric	Results
Training MAPE	0.64
Testing MAPE	0.63

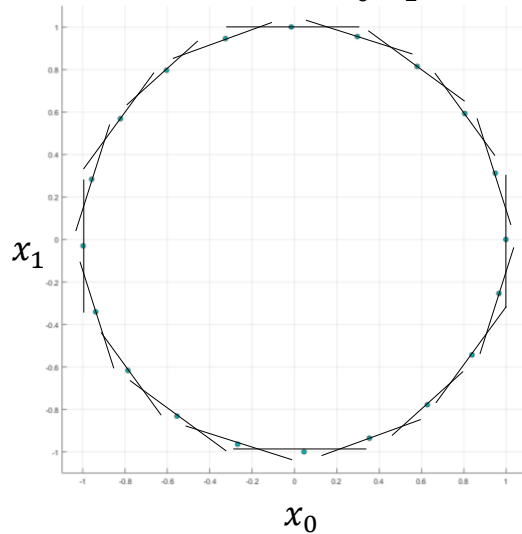
MAPE = Mean Absolute Percent Error (%)

Partial Derivatives w.r.t. texture parameters



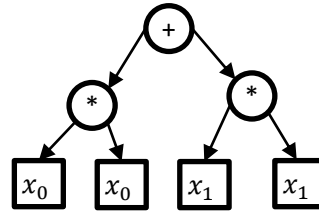
## Define fitness for plasticity (implicit) equations

Given data:  $X(x_0, x_1)$



Calculate  $\frac{\Delta x_0}{\Delta t}$  and  $\frac{\Delta x_1}{\Delta t}$

Propose model:  $f(x_0, x_1)$



Calculate  $\frac{\partial f}{\partial x_0}$  and  $\frac{\partial f}{\partial x_1}$

Calculate  $\frac{\partial f}{\partial t}$  via chain rule  $\frac{\partial f}{\partial x_0} \frac{\Delta x_0}{\Delta t}$  and  $\frac{\partial f}{\partial x_1} \frac{\Delta x_1}{\Delta t}$

average normalized deviation of  $f$ :

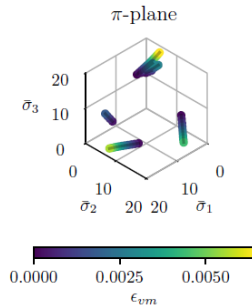
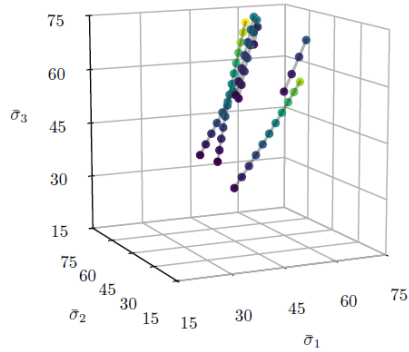
$$E(f, X) = \frac{1}{N} \sum_{i=1}^N \left| \sum_{j=1}^P \frac{\frac{\partial f}{\partial x_j} \frac{\Delta x_j}{\Delta t}}{\left| \frac{\partial f}{\partial x_j} \frac{\Delta x_j}{\Delta t} \right|} \right|^{(i)}$$

Prager Consistency Condition:

$df = 0$  at yield because  $f(\sigma, \epsilon_p) = 0$

$$df = \frac{\partial f}{\partial \sigma} : d\sigma + \frac{\partial f}{\partial \epsilon_p} : d\epsilon_p = 0$$

## Linear hardening



**Stats:**  
 70 simulated tests  
 33 points for each test  
 40 cores  
 ~1 minute for convergence

Exact solution form recovered  
 Coefficients produced result in 0.0001% relative error

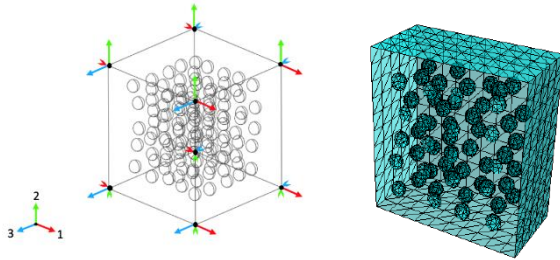
$$\begin{aligned}
 &\epsilon_{vm} - (\bar{\sigma}_2 - \bar{\sigma}_3 - (\epsilon_{vm} + \bar{\sigma}_2 - \bar{\sigma}_1)) \\
 &- ((15.2)((-0.142)(-0.142)) + \epsilon_{vm} + (\epsilon_{vm} + \bar{\sigma}_2 \\
 &- \bar{\sigma}_3 - (\epsilon_{vm} + \bar{\sigma}_2 - \bar{\sigma}_1))(\bar{\sigma}_2 - \bar{\sigma}_1) + \epsilon_{vm} - (-33.9) \\
 &+ ((\epsilon_{vm} + (-0.142)(-0.142))(\epsilon_{vm} + \epsilon_{vm}))(490100) \\
 &+ \epsilon_{vm} + \epsilon_{vm} - (\bar{\sigma}_2 - \bar{\sigma}_3 - (\epsilon_{vm} + \bar{\sigma}_2 - \bar{\sigma}_1) + \epsilon_{vm} \\
 &+ (-0.142)(-0.142) + (\bar{\sigma}_2 - \bar{\sigma}_3)(\bar{\sigma}_2 - \bar{\sigma}_3) - \epsilon_{vm})) \\
 &= \text{constant}
 \end{aligned}$$

↓  
 simplify

$$c_0 + \frac{1}{2}((\bar{\sigma}_1 - \bar{\sigma}_2)^2 + (\bar{\sigma}_2 - \bar{\sigma}_3)^2 + (\bar{\sigma}_3 - \bar{\sigma}_1)^2) - c_1\epsilon_{vm} - c_2\epsilon_{vm}^2 = \text{constant}$$

## Simulated training data:

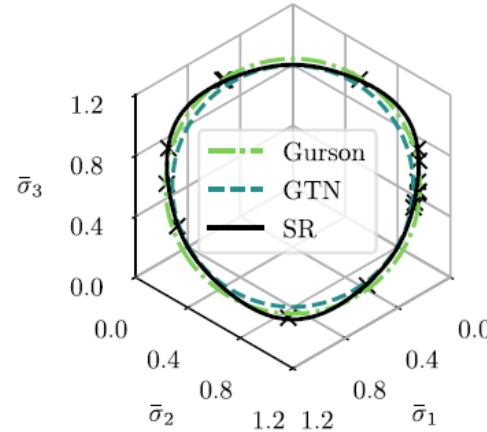
- Spherical, random void microstructure
- Proportional loading
- Matrix von Mises (perfect) plasticity
- Random perturbations of voids



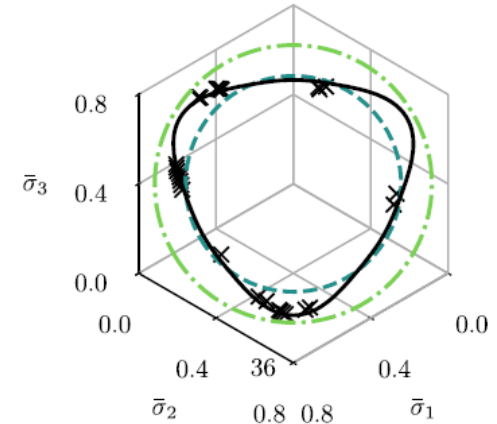
### Stats:

169 simulated tests  
145 points for each test  
160 cores  
~72 hours for convergence

$$\sigma_h = 1, \bar{\nu} = 0.0635$$



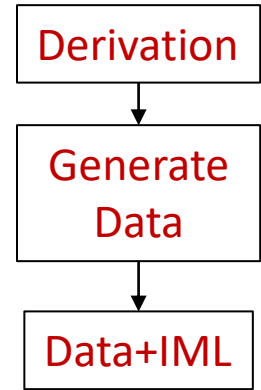
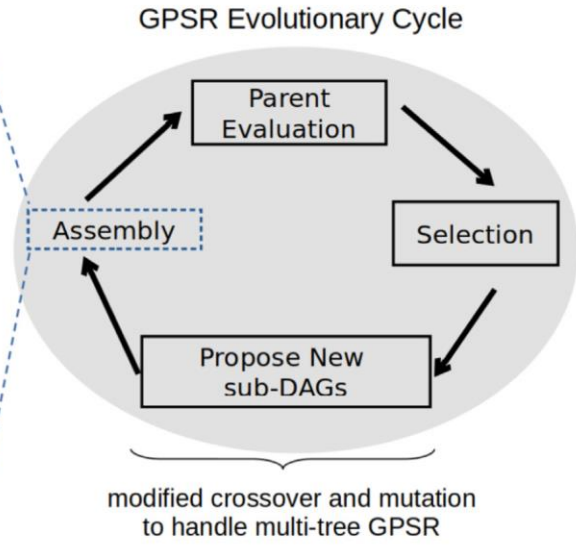
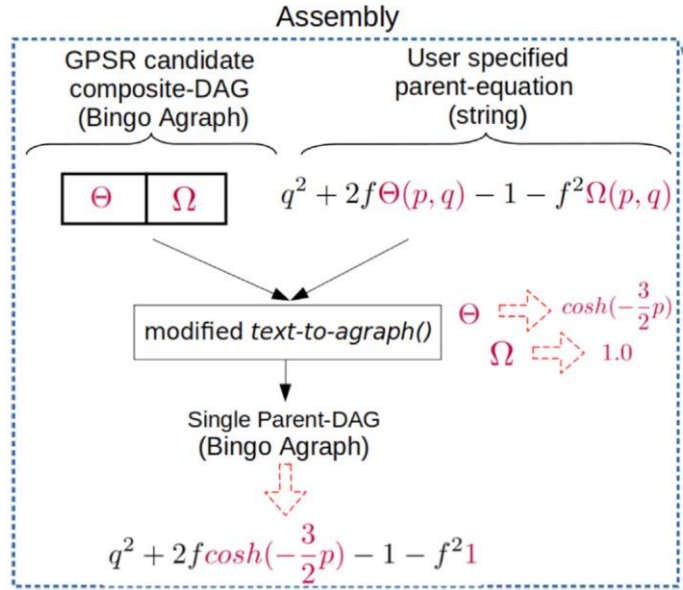
$$\sigma_h = 1.5, \bar{\nu} = 0.0645$$



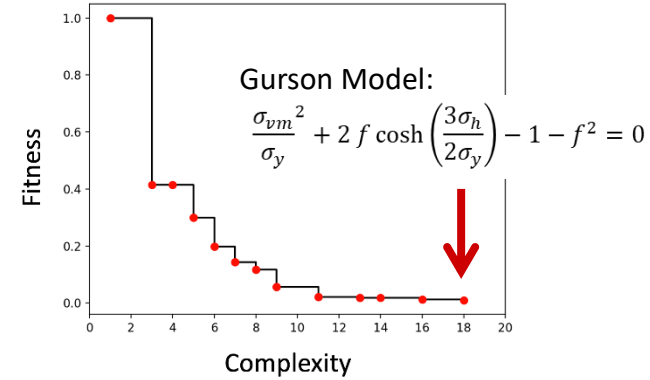
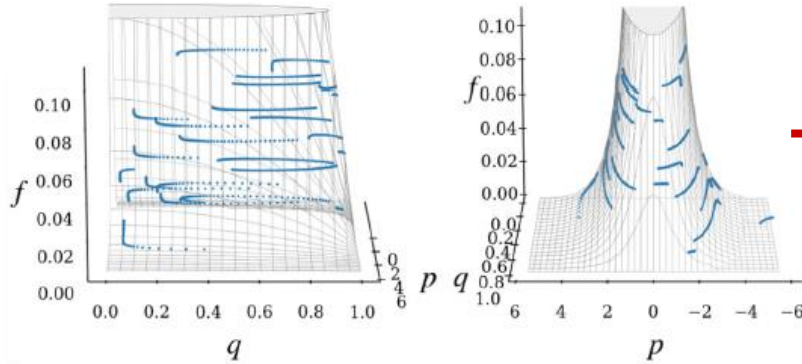
$$\begin{aligned} \Phi_{SR} = & [\bar{\sigma}_{vm}(4\bar{\sigma}_h^2 + (2\bar{\nu} - \bar{L} + c_1)\bar{\sigma}_h + c_2) \\ & - \bar{\sigma}_h(c_3\bar{\nu}^2 + c_4)](c_5\bar{\nu}^2 + 2\bar{\nu} + c_6) \\ & + c_7\bar{\nu}^2 - \bar{\sigma}_{vm}\bar{L}^2(c_8\bar{\nu}^2 - c_9) + c_{10} - c_{11} = 0 \end{aligned}$$

$$\text{Gurson yield surface: } \left(\frac{\sigma_{vm}}{\sigma_y}\right)^2 + 2f \cosh\left(\frac{3\sigma_h}{2\sigma_y}\right) - 1 - f^2 = 0$$

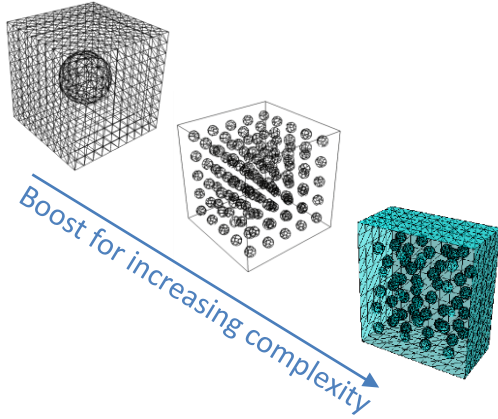
**Objective: balance foundation from analytical methods with accuracy from ML**



- 25 load cases for training
- Gurson material model generated data

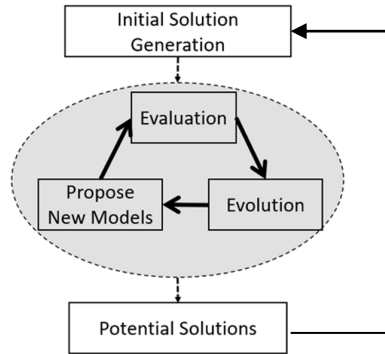


Algorithm	Generations	Median MAE	$[min, max]$ MAE	Time (h)
GPSR	10000	$1.3 \times 10^{-3}$	$[4.9 \times 10^{-6}, inf]$	7
P-GPSR	250	$1.2 \times 10^{-9}$	$[<1 \times 10^{-20}, 3 \times 10^{-6}]$	1



$$q^2 + 2f\Theta(p, q) - 1 - f^2\Omega(p, q) \quad (\text{without microstructure assumptions})$$

Assumption	Model
Void interaction	Single void, periodic BCs, void remains spherical
Void growth self similarity	Single void, periodic BCs, no void constraint



Seed  $\Theta_i$  and  $\Omega_i$   
in initial pop. for  
 $\Theta_{i+1}$  and  $\Omega_{i+1}$

$$q^2 + 2f \cosh\left(-\frac{3}{2}p\right) - 1 - f^2 \mathbf{1}$$

$\Theta_1$                        $\Omega_1$   
 $\downarrow$                                $\downarrow$

**Training data and stages:**

(with all Gurson assumptions)

$$q^2 + 2f\Theta_2 - 1 - f^2\Omega_2$$

(void interaction training data)

$$q^2 + 2f\Theta_3 - 1 - f^2\Omega_3$$

(growth self-similarity training data)





Mapping relies on the **tensor-transform method (TTM)**, which represents yield surfaces as:

$$\vec{\sigma}^T \mathbf{P} \vec{\sigma} = \sigma_i P_{ij} \sigma_j = 1$$

$$\vec{\sigma}^T = [\sigma_1, \sigma_2, \sigma_3]$$

e.g., von Mises yield criterion:

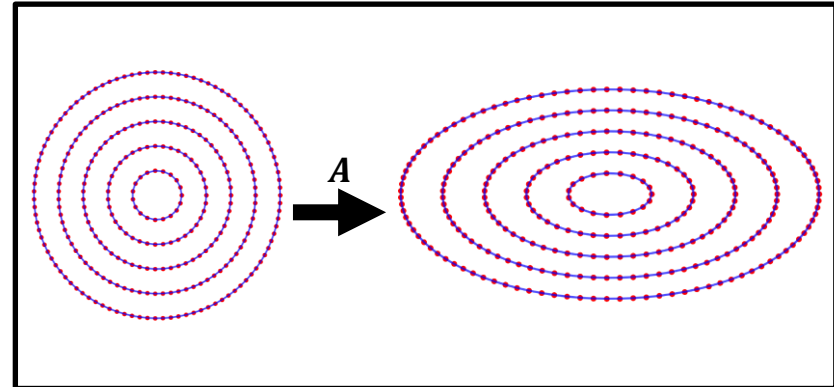
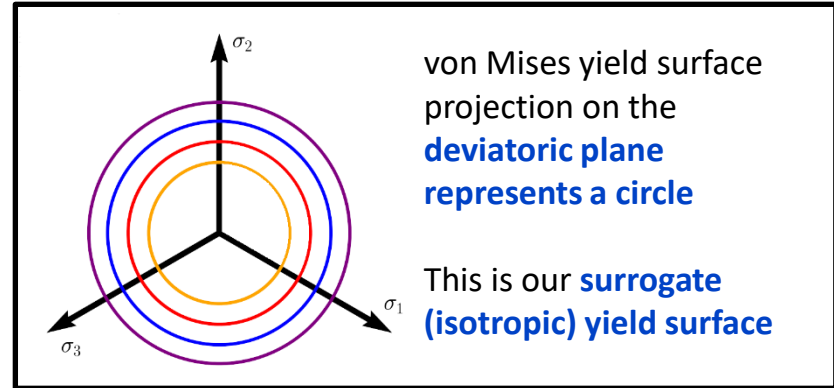
$$(\sigma_1 - \sigma_2)^2 + (\sigma_1 - \sigma_3)^2 + (\sigma_2 - \sigma_3)^2 = 2\sigma_y^2$$

$$\mathbf{P} = \frac{1}{\sigma_y^2} \begin{bmatrix} 1 & -1/2 & -1/2 \\ -1/2 & 1 & -1/2 \\ -1/2 & -1/2 & 1 \end{bmatrix}$$

TTM relies on **transforming a surrogate  $\mathbf{P}$  tensor** via:

$$\mathbf{P}_{real} = \mathbf{A}^T \mathbf{P}_{surrogate} \mathbf{A}$$

$$\vec{\sigma}^T \mathbf{A}^T \mathbf{P}_{surrogate} \mathbf{A} \vec{\sigma} = 1$$



- Each grain is a visco-plastic anisotropic ellipsoidal inclusion that has a deviatoric plastic response within a visco-plastic anisotropic Homogeneous Effective Medium (HEM).
- Stress and strain are uniform within a grain (inclusion) but can differ from the HEM values, in contrast to a Taylor model (where strain rates are equivalently imposed).

Grain strain rate is a sum slip rate on each system,  $s$ , obtained by the Schmid tensor,  $m$ , and stresses  $\sigma$ ,  $\tau^s$

$$\longrightarrow \dot{\varepsilon}_{ij}' = \dot{\gamma}_0 \sum_s m_{ij}^s \left( \frac{m^s : \sigma}{\tau^s} \right)^n = M_{ijkl} \sigma'_{kl}$$

The HEM is subject to an equivalent strain rate where an interaction tensor,  $M$ , describes the 'stiffness'

$$\longrightarrow \dot{\bar{\varepsilon}}_{ij}' = \bar{M}_{ijkl} \bar{\sigma}'_{kl}$$

Equilibrium is enforced by solving stress divergence equation (conservation of momentum):

$$\longrightarrow \sigma_{ij,j} = (\sigma'_{ij} + p\delta_{ij})_{,j} = 0$$

Yield surfaces **representing highly textured stainless steel** were generated with **VPSC** (i.e., simulated data)

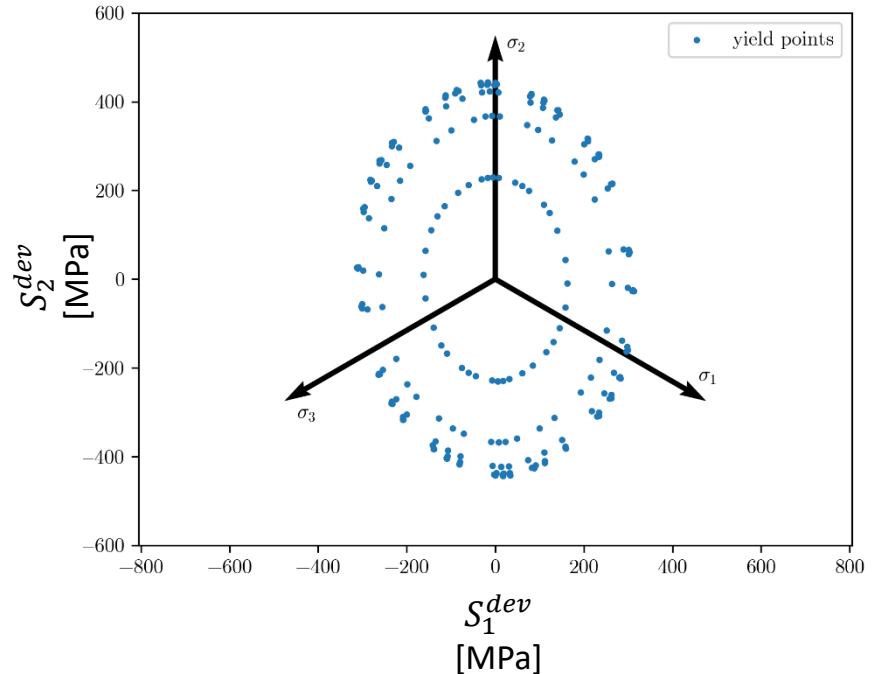
Objective: demonstrate the ability to map arbitrary yield surfaces to surrogates

Training data:

Stainless steel loaded with  $\epsilon_{vm} \in [0, 0.125]$

Yield surfaces extracted at 6 points during loading history

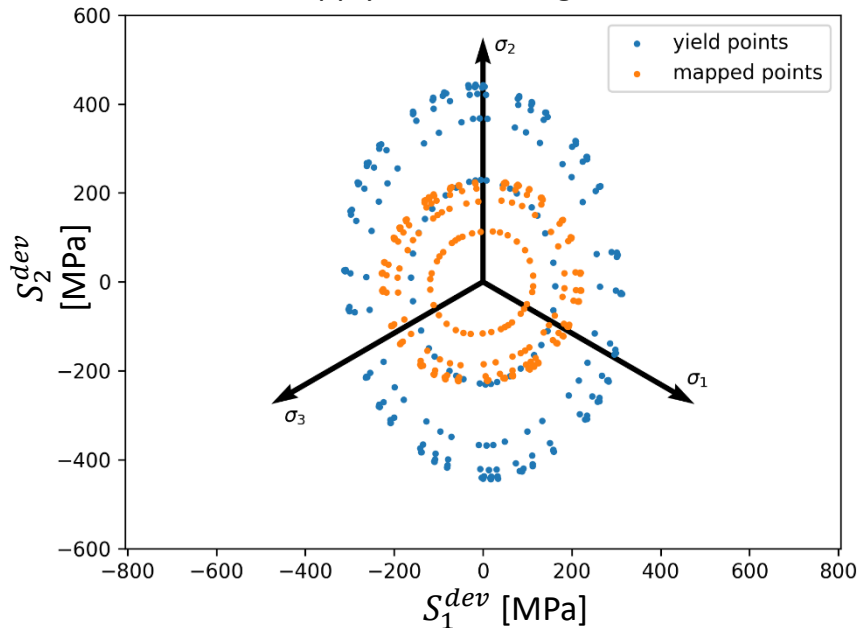
37 points per  $\epsilon_{vm}$  value



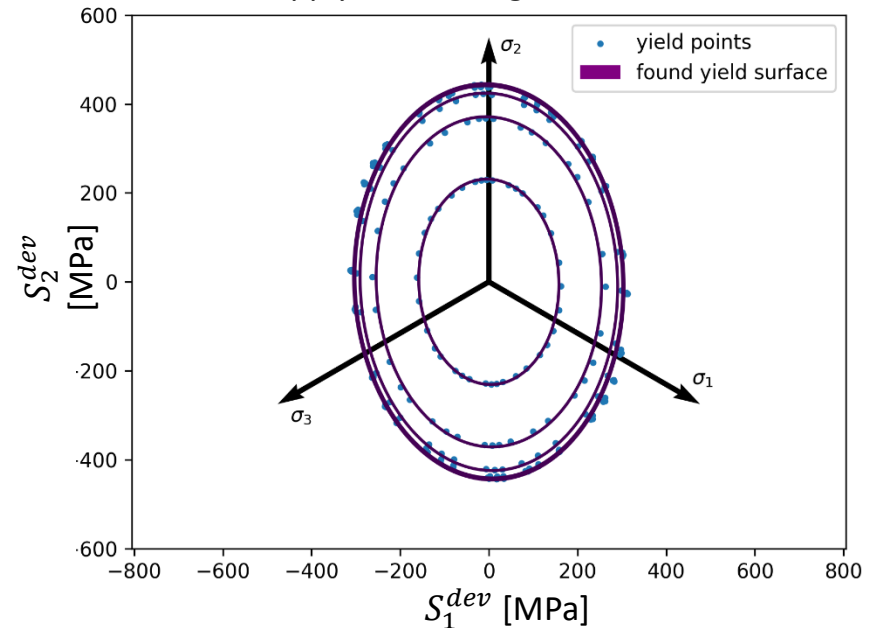
$$\mathbf{B} = \mathbf{A}^{-1} = (0.000228\epsilon_{vm} + 5.298) \begin{bmatrix} -0.225 & 0.39 & 0.23 \\ -0.093 & 0.41 & 0.075 \\ -0.258 & 0.512 & 0.139 \end{bmatrix}$$

MSE = 0.0108

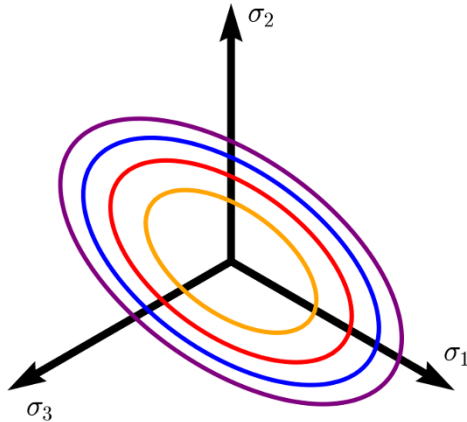
Apply  $\mathbf{B}$  to training data



Apply  $\mathbf{A}$  to surrogate surface

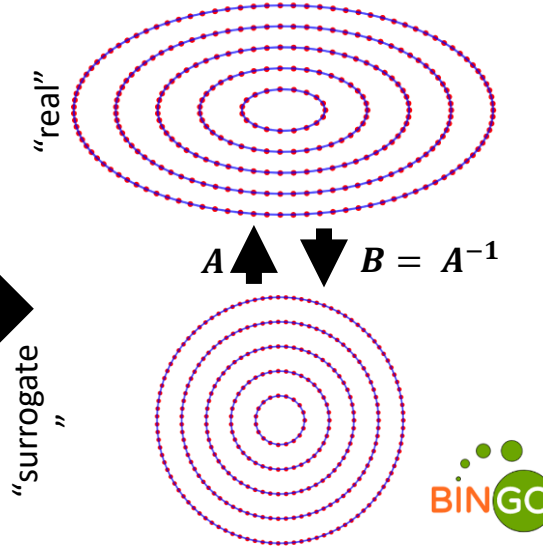


## Data collection



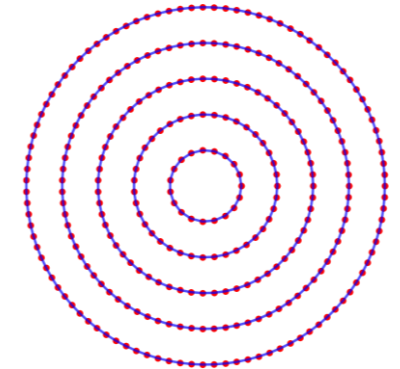
Collect yield-surface data,  
typically from experiment  
("real")

## Map to surrogate yield surface



Use IML to find mapping tensor  
 $A$  from "real" (anisotropic)  
surface to a surrogate (isotropic)  
surface similar to \*Oller et al.

## Solve FE model with surrogate model



Solve finite-element model with  
a single surrogate implemented  
as a user constitutive model  
(e.g., UMAT in ABAQUS)

## 1. Interpretable, Data-driven Machine Learning Background

- We seek interpretable and generalizable material constitutive models
- GPSR provides one means for inherent interpretability through the evolution of analytical expressions
- A purely data-driven approach results in accurate but unclear models

## 2. Laying a Scientific Machine Learning Foundation

- Partial derivation before the ML process:
  - forms a guiding parent equation
  - improves model accuracy and training performance
  - define features of importance in training data sets, and
  - promotes generalizable models
- Verification studies (i.e., checking for known analytical models) with GPSR are a natural step
- Microstructure complexity can be added iteratively to produce an accurate, complex, but still interpretable model

## 3. Finite Element Method Auto-implementation

- With the tensor transform method, only one surrogate constitutive model need be implemented in FEA and GPSR provides the transform to the real constitutive model