

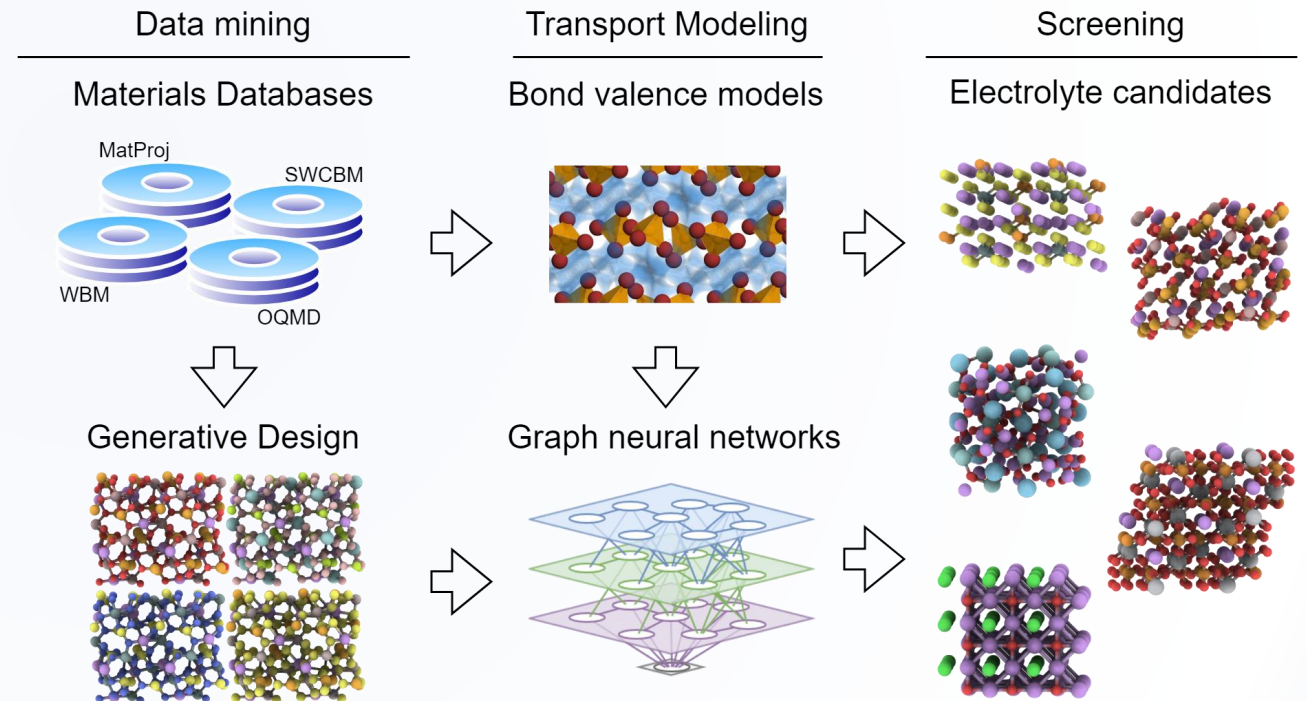
Screening for Li-based solid electrolytes using bond-valence methods and graph neural networks

Stephen R. Xie¹, Shreyas J. Honrao¹, John W. Lawson²

¹ KBR Inc at NASA Ames Research Center

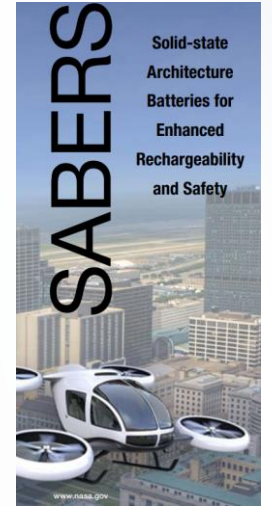
² NASA Ames Research Center, Moffett Field, CA, USA

- Pipeline for high-throughput ionic conductivity predictions
- Machine learning with conduction graphs from bond valence methods
- Medium-throughput screening with machine-learned interatomic potential



Background

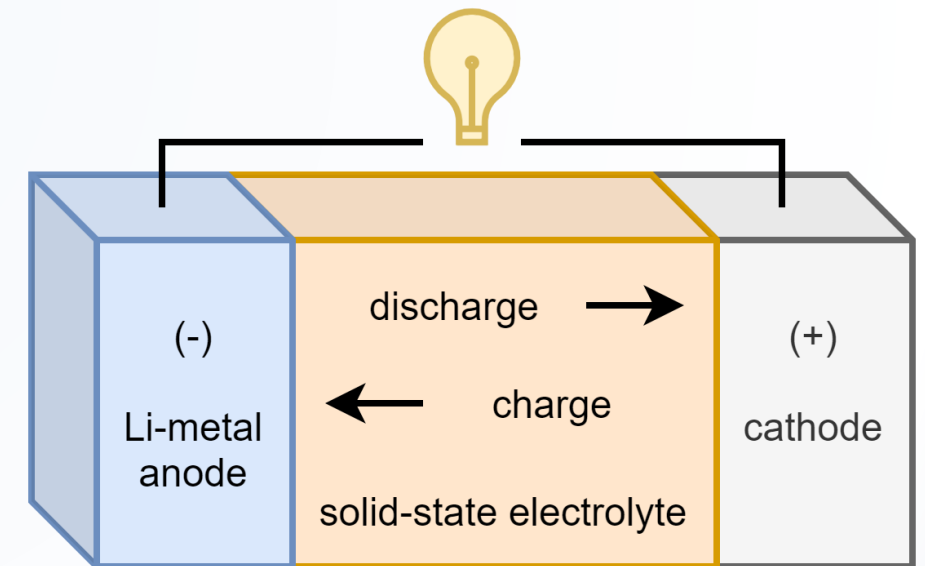
- Safe, all-electric aircraft
- All-solid-state batteries
- Solid electrolyte
 - Low electrical conductivity
 - Good electrochemical stability
 - High ionic conductivity
 - Low-cost, manufacturable, ...

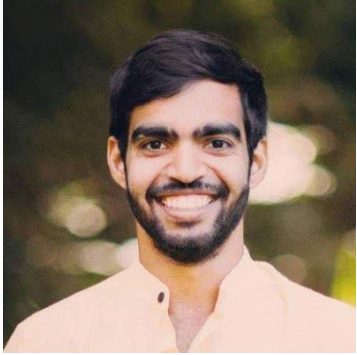


nasa.gov

Honrao, S.J., Yang, X., Radhakrishnan, B. *et al.* Discovery of novel Li SSE and anode coatings using interpretable machine learning and high-throughput multi-property screening. *Sci Rep* **11**, 16484 (2021).

<https://doi.org/10.1038/s41598-021-94275-5>

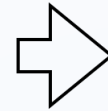




Data mining

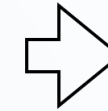
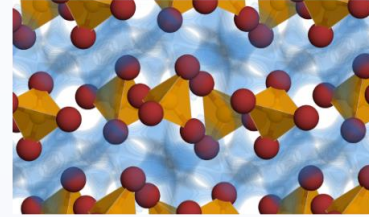
Materials Databases

MatProj



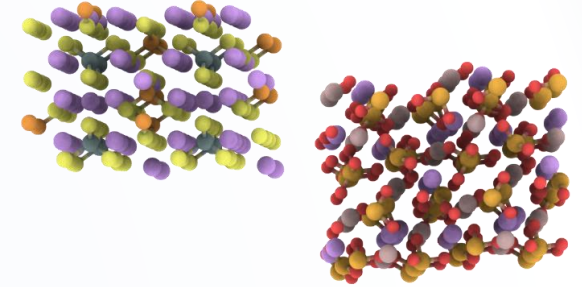
Transport Modeling

Bond valence models



Screening

Electrolyte candidates



- Database of 15,000 Li-containing materials
- Migration barriers
- Thermodynamic stability, electrochemical stability, and band gap
- Interpretable Gradient Boosting Models (GBMs)

Can we directly, efficiently compute ionic conductivity for screening?

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Predicting transport through atomistic simulations

- Mean squared displacement of Li^+

$$\text{MSD} = \frac{1}{N} \sum_{i=1}^N \langle \|\Delta r_i\|^2 \rangle$$

- Diffusion coefficient

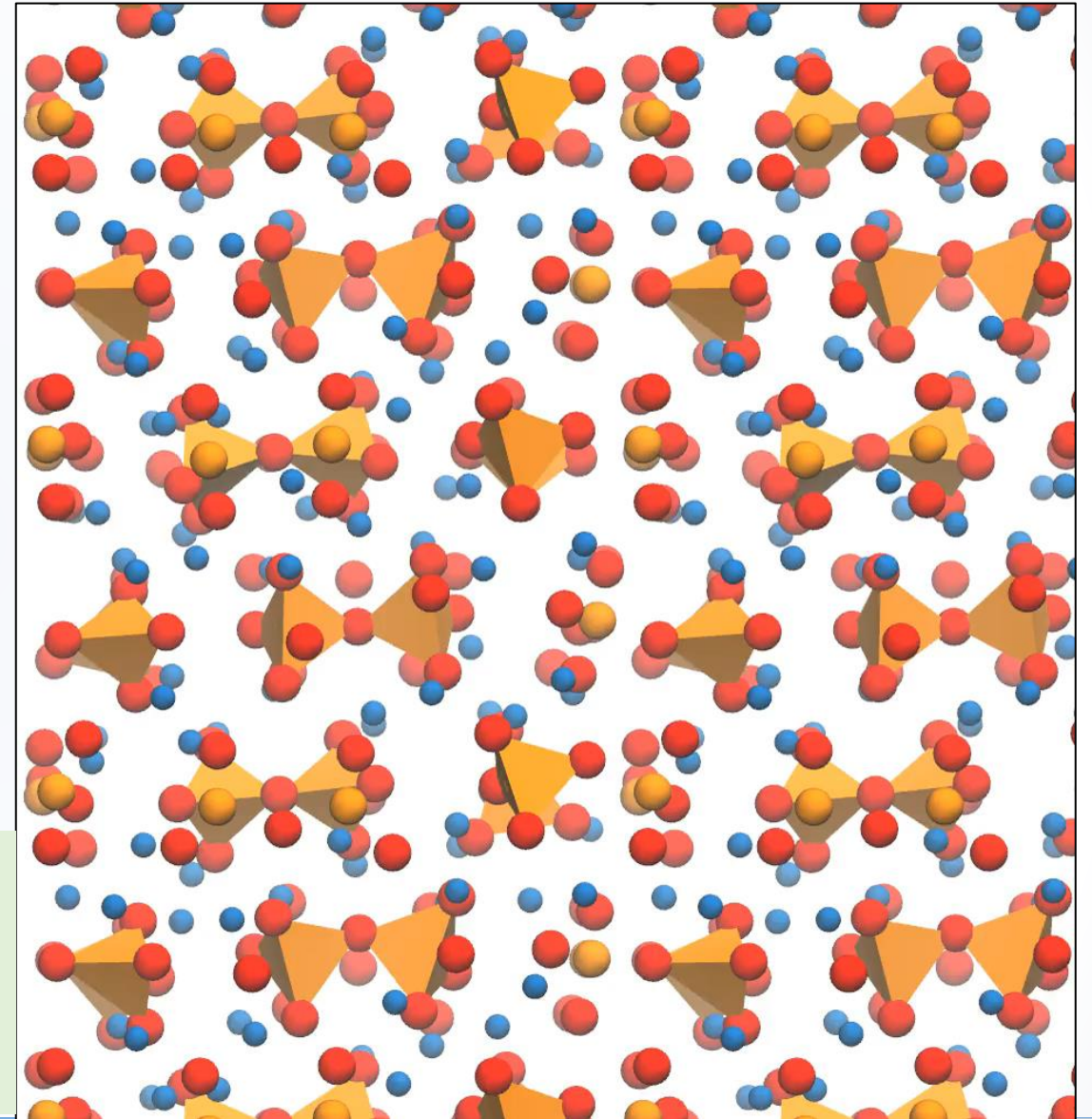
$$D_{\text{tracer}} = \lim_{t \rightarrow \infty} \left(\frac{\text{MSD}}{6t} \right), \quad D(T) = D(\infty) \exp \left(\frac{-E_a}{k_B T} \right)$$

- Ionic Conductivity

$$\sigma(T) = \frac{\rho^2 z^2}{k_B T} D(T)$$

- 100-1000 ps of simulation time for convergence
- *ab initio* molecular dynamics (AIMD)
 - 3,000-40,000 CPU-hours (\$25 - \$414)

Simulating diffusion from first principles is expensive



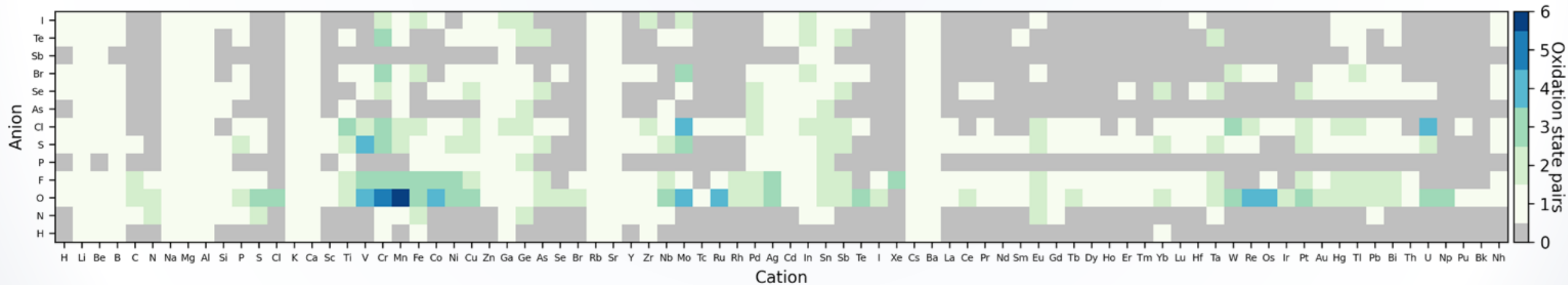
Bond valence methods with softBV

- Fast empirical force field for battery materials
- Morse-like term for cation-anion interactions
- Coulomb term for same-charge interactions
- Parametrized for hundreds of cation-anion combinations

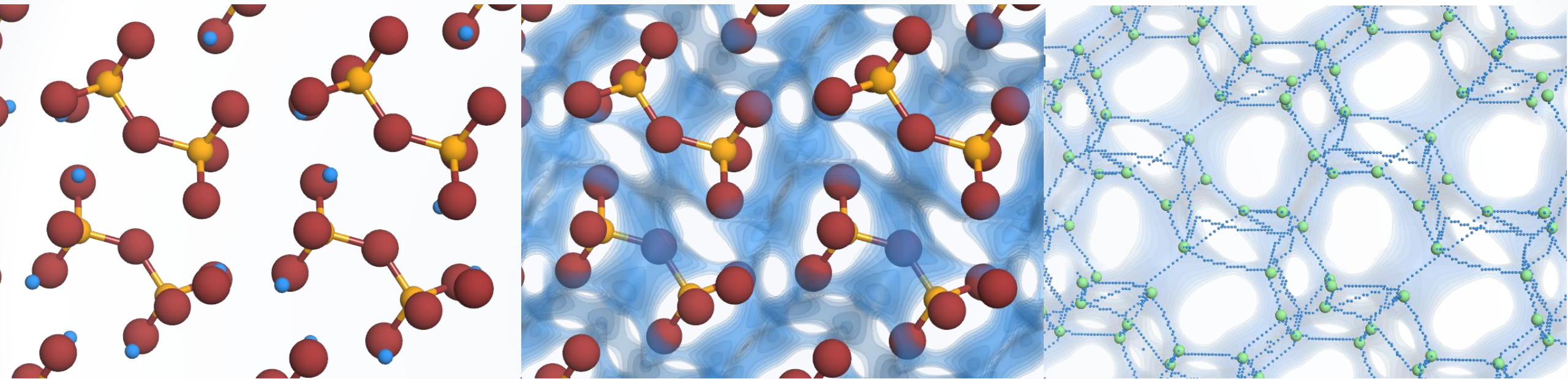
$$E_{\text{BV,Li-X}}(r_{\text{Li-X}}) = \epsilon \left((\exp [a (r_{\text{min,Li-X}} - r_{\text{Li-X}})] - 1)^2 - 1 \right) \\ = \epsilon \left[\left(\frac{s - s_{\text{min}}}{s_{\text{min}}} \right)^2 - 1 \right]$$

$$E_{\text{Coulomb,Li-B}}(r_{\text{Li-B}}) = \frac{q_{\text{Li}}q_{\text{B}}}{R} \text{erfc} \left(\frac{r_{\text{Li-B}}}{f \cdot (r_{\text{Li}} + r_{\text{B}})} \right)$$

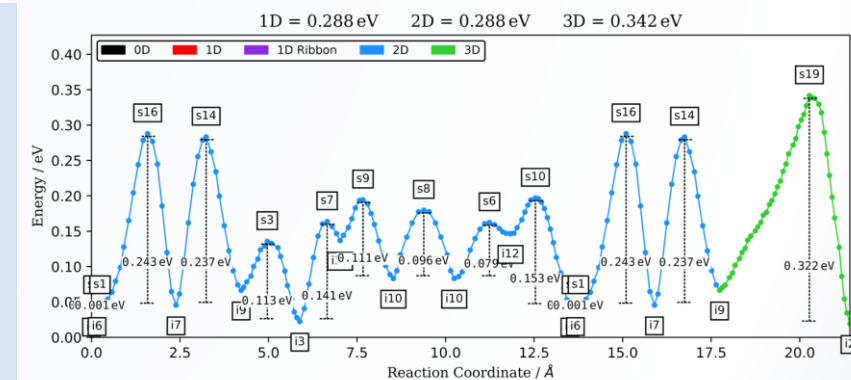
$$s_{\text{Li-X}} = \exp \left(\frac{r_{0,\text{Li-X}} - r_{\text{Li-X}}}{b_{\text{Li-X}}} \right), \quad \sum_{\text{X}} s_{\text{Li-X}} \approx V_{0,\text{Li}}$$



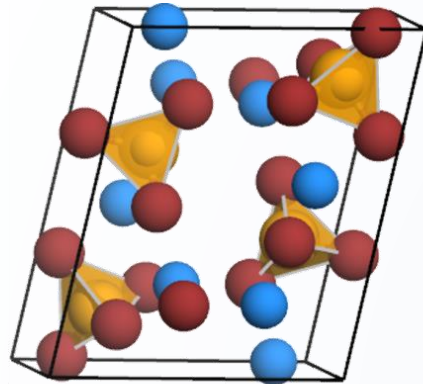
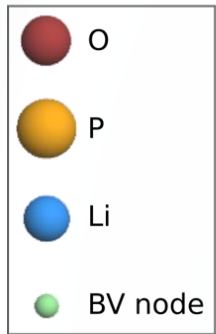
Bond valence methods with softBV



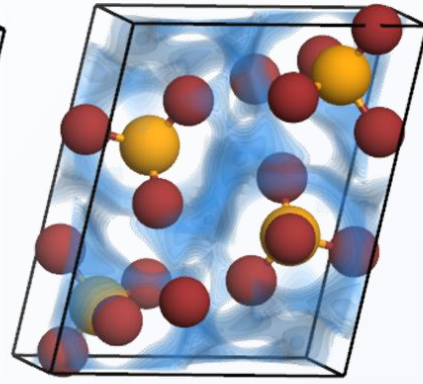
- Calculate energy landscape of Li^+ ion using the bond valence potential energy function
- Reveals low-energy pathways for Li-ion diffusion
- **No Li^+ - Li^+ interaction, no movement for non- Li^+ ions**



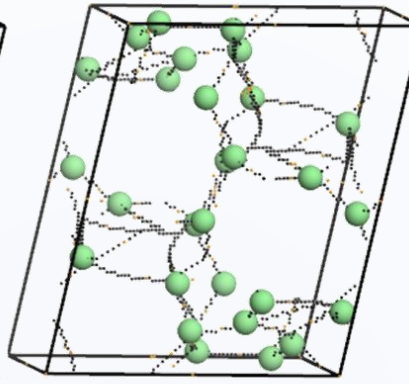
Kinetic Monte Carlo with softBV



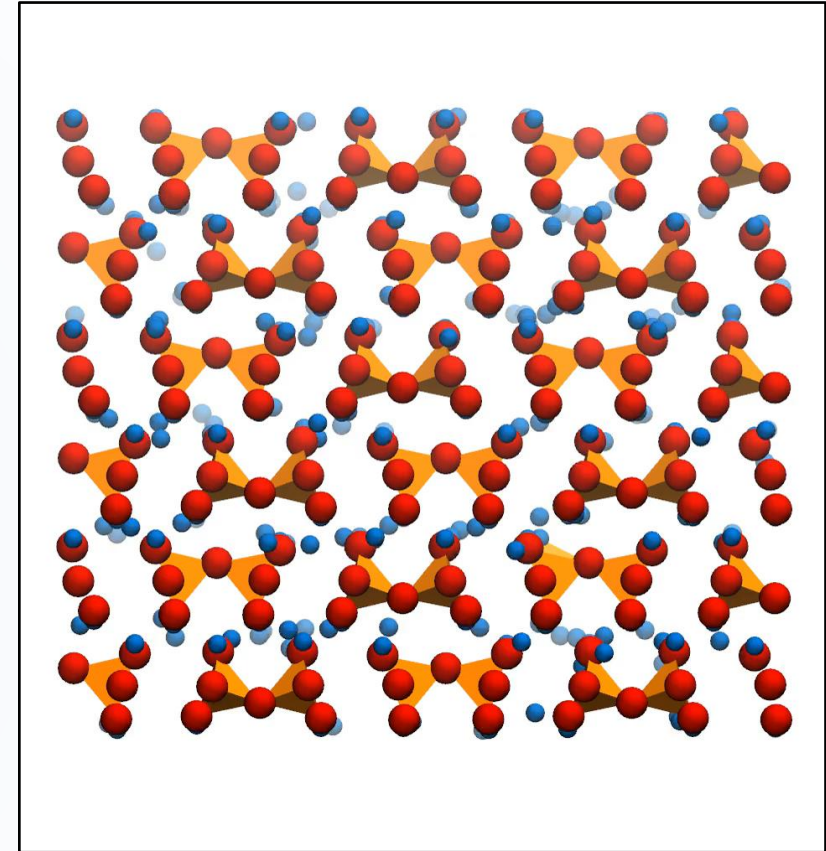
Structure



Energy
landscape



Diffusion
pathways



- Mean squared displacement of Li^+

$$\text{MSD} = \frac{1}{N} \sum_{i=1}^N \langle \|\Delta r_i\|^2 \rangle$$

- Ionic Conductivity

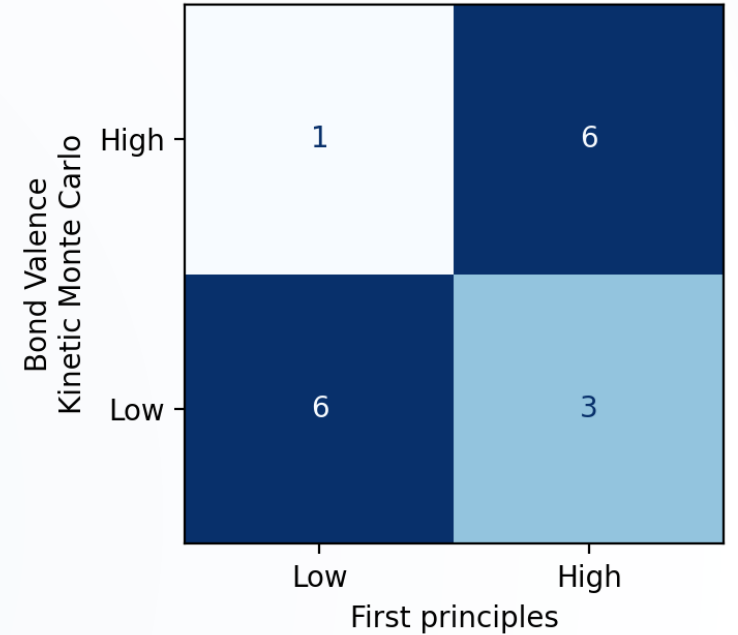
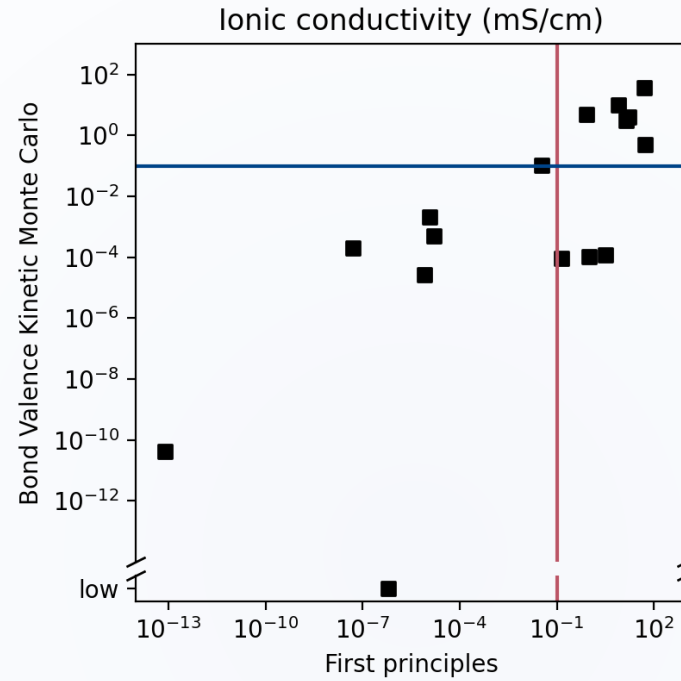
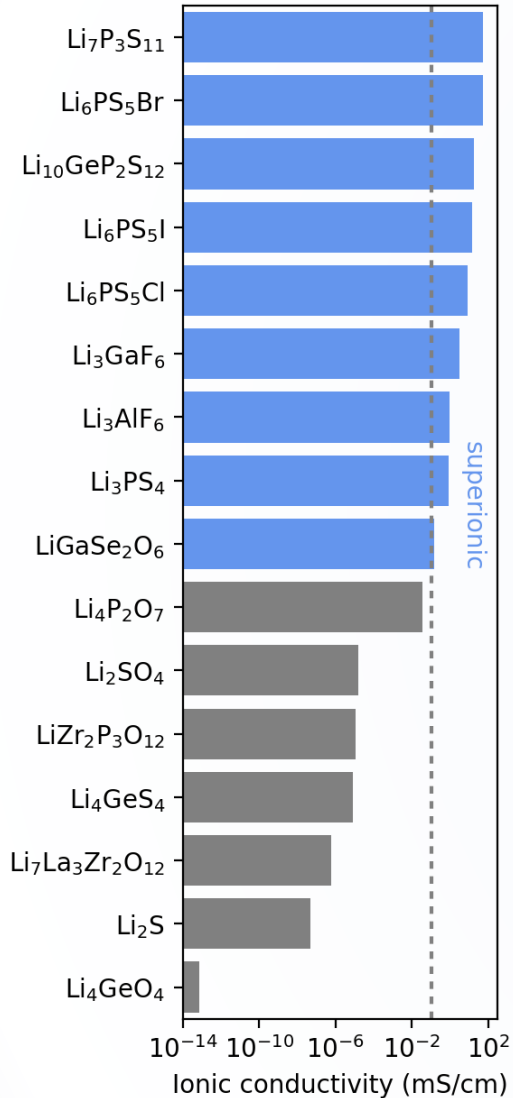
$$\sigma(T) = \frac{\rho^2 z^2}{k_B T} D(T)$$

- Site-to-site transition rate

$$p = \omega \exp\left(\frac{E_m + \Delta E}{k_B T}\right)$$

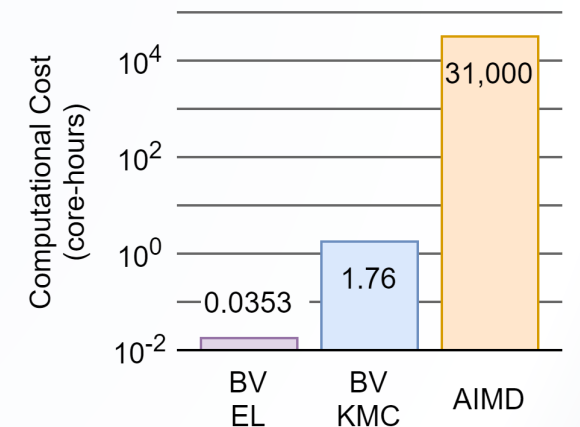
- $\text{Li}^+ - \text{Li}^+$ interactions included in ΔE
- **No movement for non- Li^+ ions**

Benchmarking Kinetic Monte Carlo

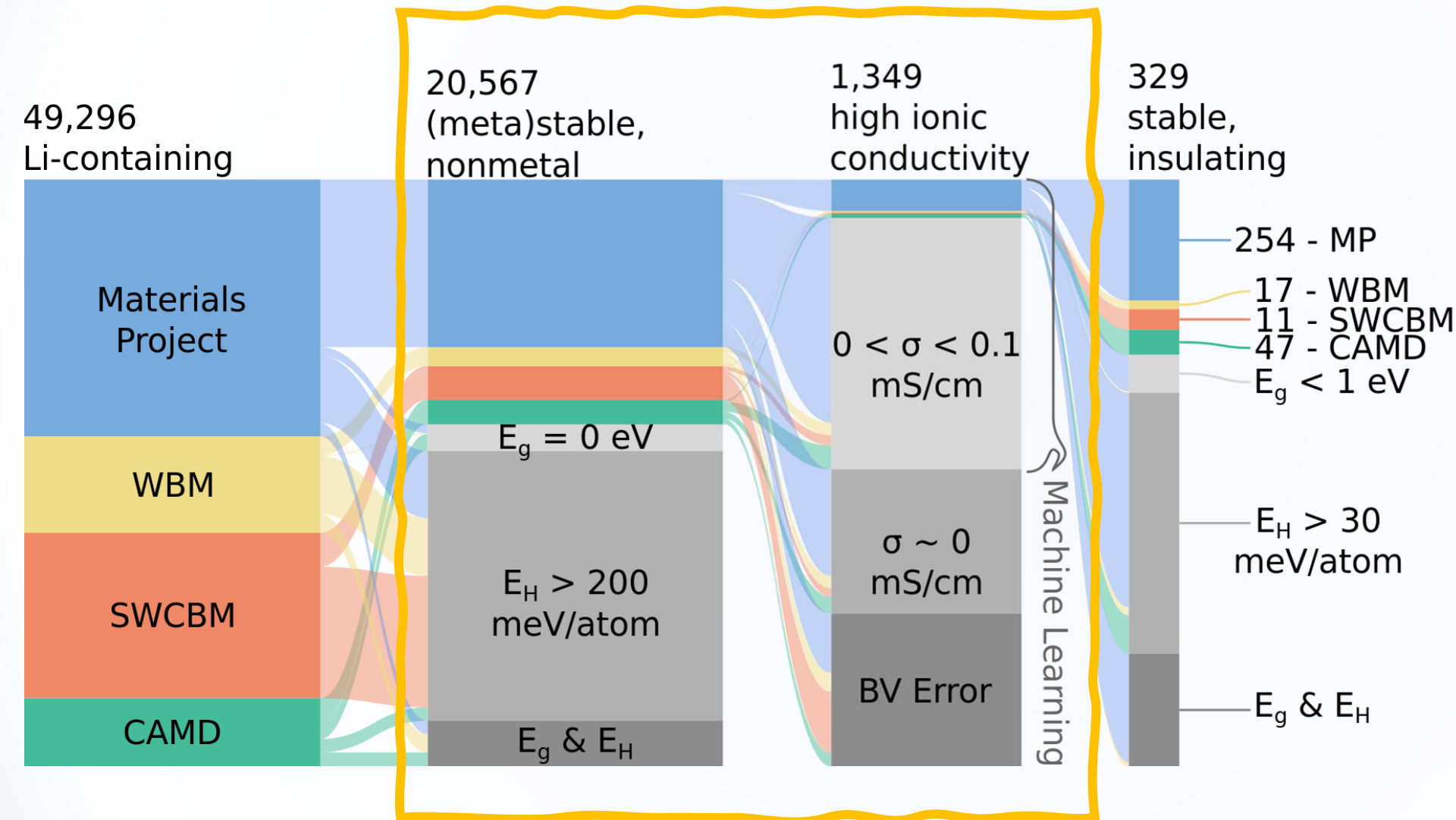


- 1 false positive
- 3 false negatives
- Precision = 0.86
- Recall = 0.67
- $F1 = 0.75$

KMC distinguishes between high and low ionic conductivity with suitable precision for screening for 1/10,000th cost of AIMD

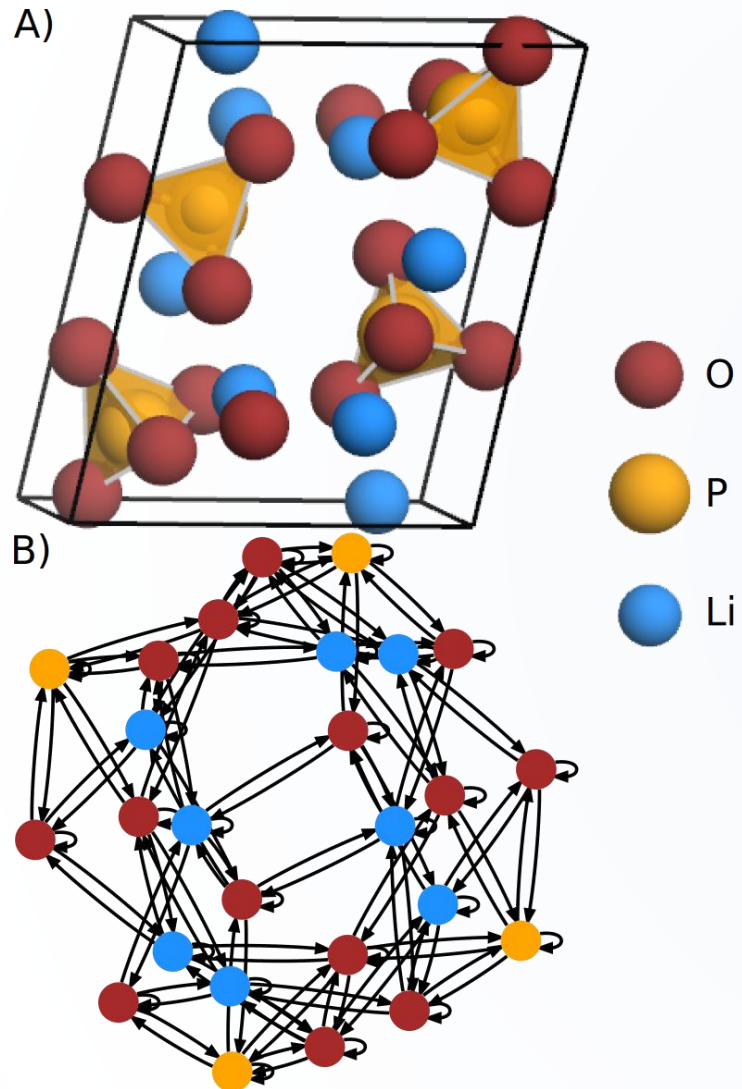


High-throughput screening with softBV



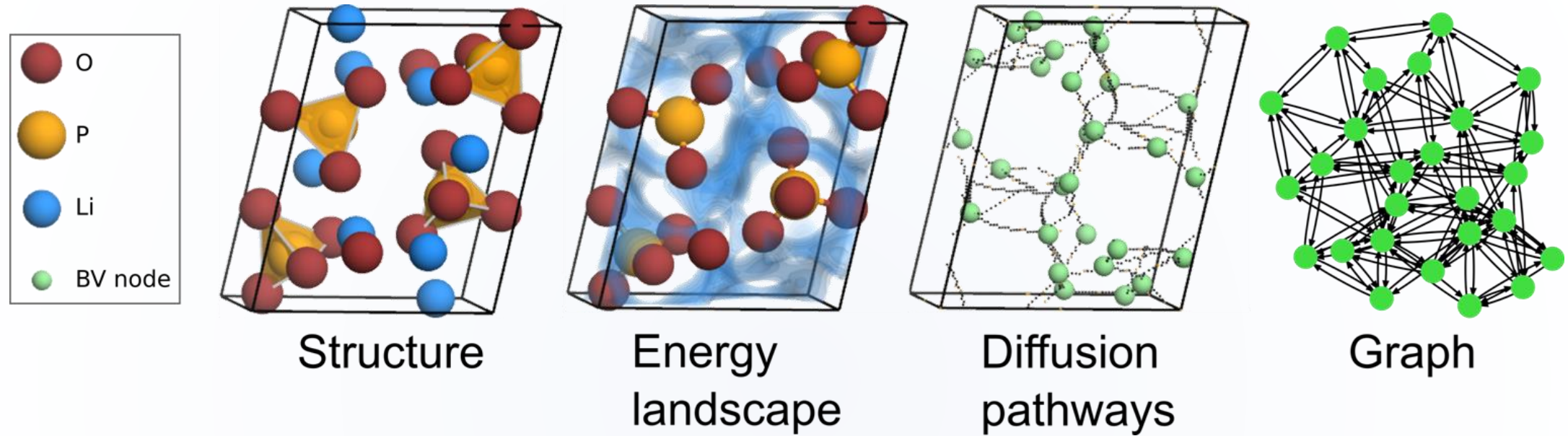
- **49,296** materials from existing databases
- **329** promising candidates with high KMC ionic conductivity
- **10,148** KMC simulations
↓
training data for machine learning

Machine learning ionic conductivity from crystal structure



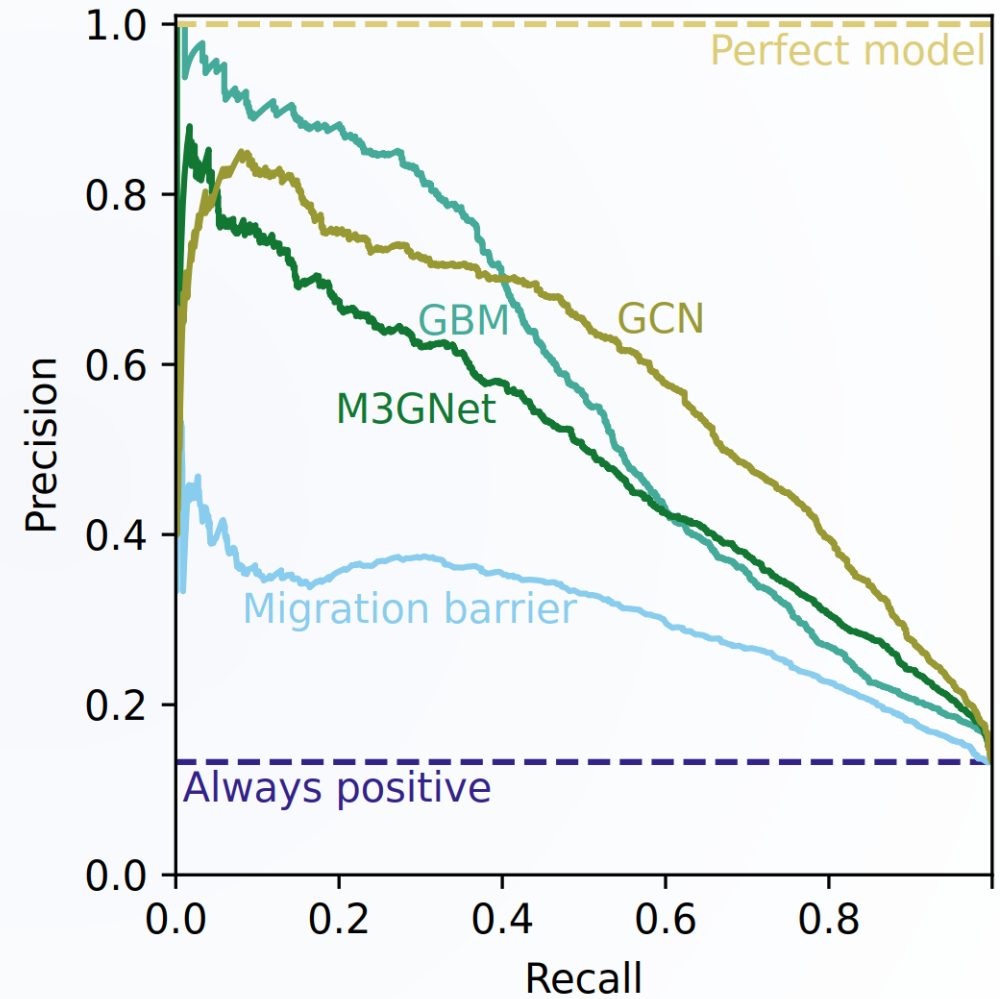
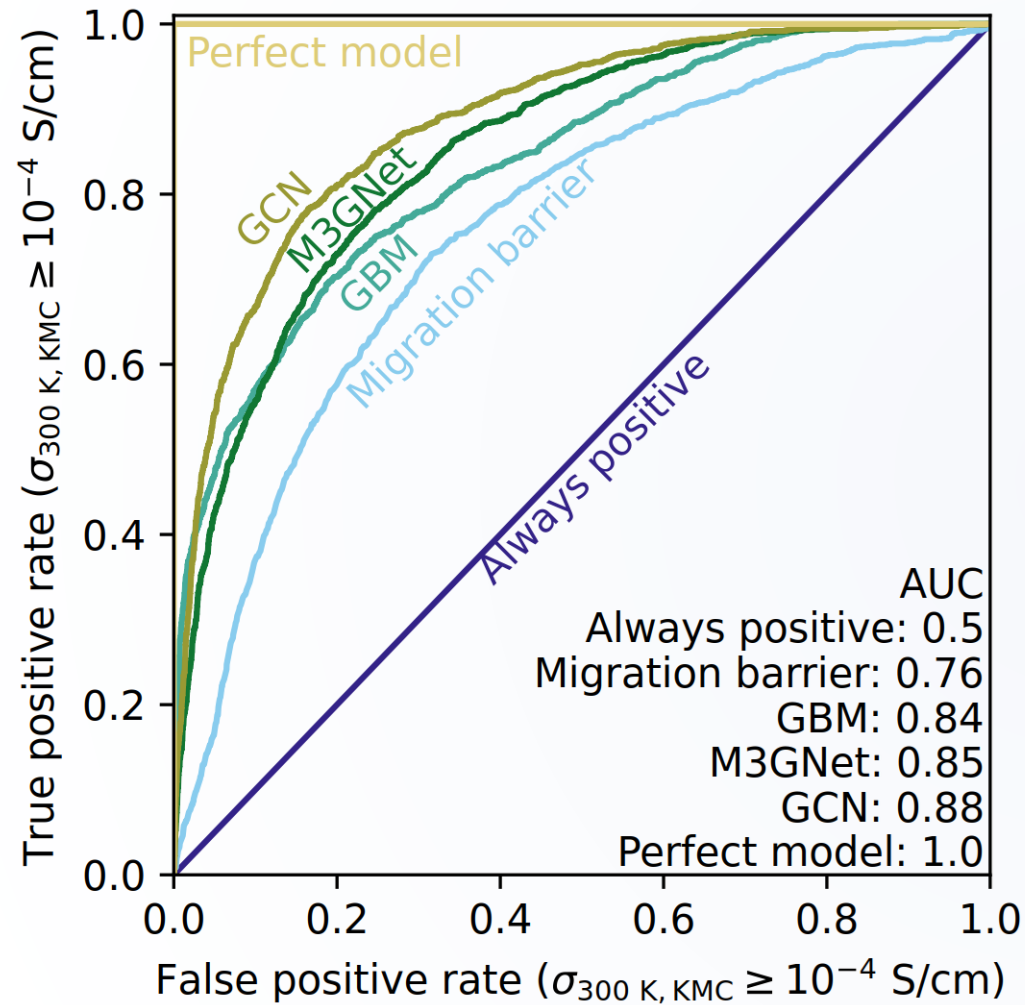
- Gradient boosting models (GBM)
 - physical quantities e.g. sublattice packing factor
 - compositional quantities e.g. Li atomic fraction
- M3GNet graph neural network
 - node vectors encoding atomic information
 - edge vectors encoding interatomic distances

Machine learning ionic conductivity from BV pathways

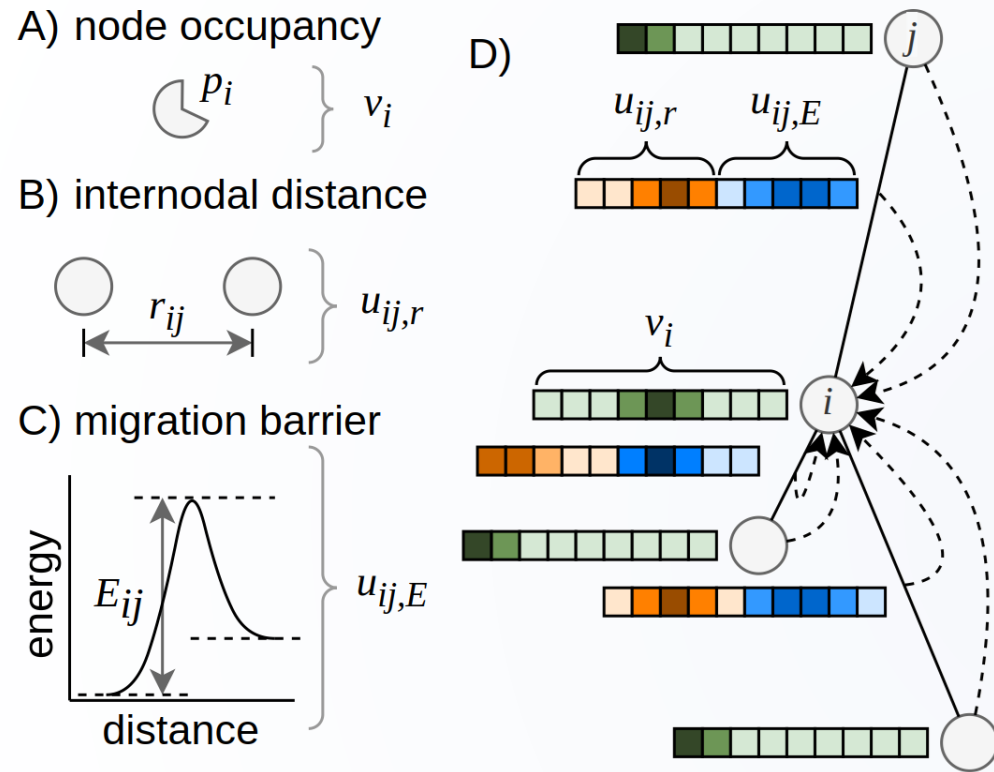


- Bond valence conduction graph
 - Nodes: minima in energy landscape
 - Edges: pathway segments
- Same graph topology used for KMC
- Can a graph convolutional network (GCN) replace KMC?

Machine learning ionic conductivity



Graph Convolutional Network

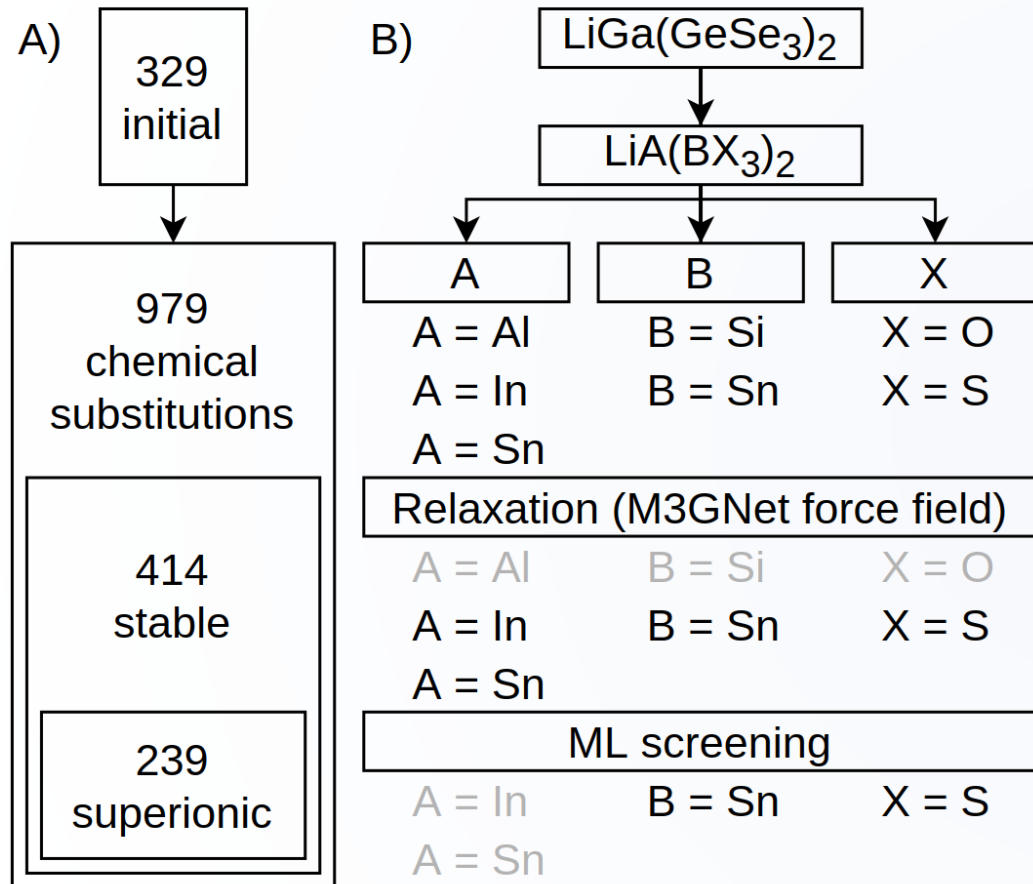


| Model performance ($\sigma_{300\text{ K}}$, KMC) | | | | |
|--|-------|-------|-------------------------|----------------------|
| | R^2 | MAE | Precision Recall=0.8 | Average Precision |
| Baseline | 0.0 | 3.638 | 0.133 | 0.133 |
| E_m linear fit | 0.039 | 3.506 | 0.227 | 0.304 |
| M3GNet | 0.560 | 1.937 | 0.306 | 0.499 |
| GBM | 0.608 | 2.411 | 0.268 | 0.569 |
| GCN | 0.691 | 1.558 | 0.395 | 0.587 |
| GCN, no E_{ij} | 0.453 | 2.074 | 0.290 | 0.455 |
| GCN, no p_i | 0.602 | 1.794 | 0.290 | 0.498 |
| GCN, no r_{ij} | 0.587 | 1.777 | 0.362 | 0.557 |

• Ablation study

- Site-to-site barriers: affect transition rates
- Occupancies: fully occupied or completely unoccupied paths do not contribute to ionic conductivity
- Distances: length of hops contributing to MSD

Generative design and Machine learning

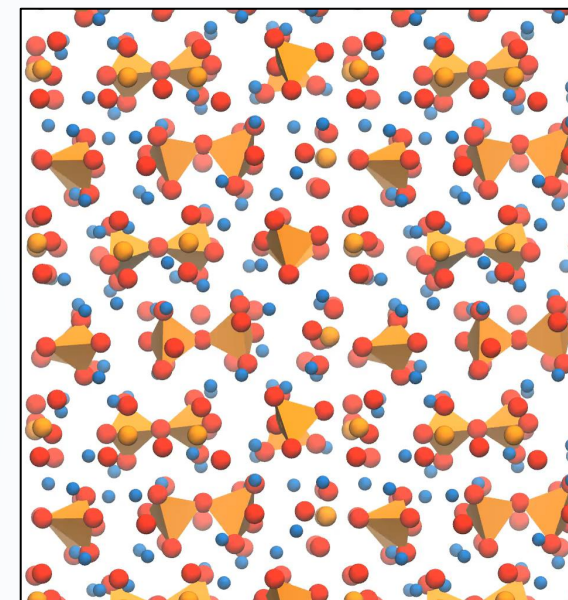
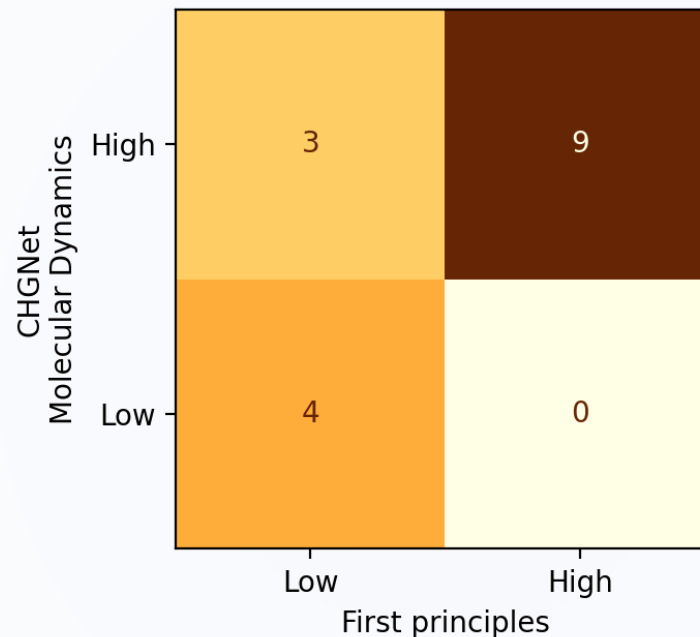
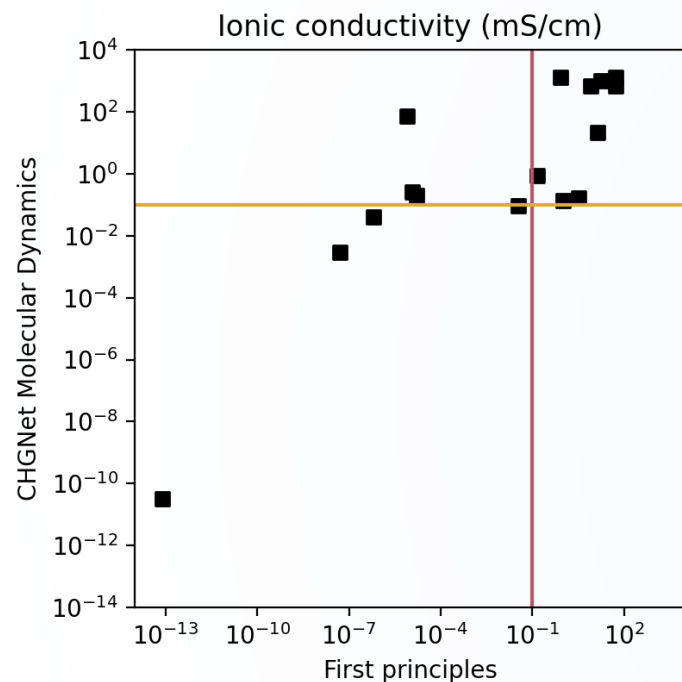


- Isovalent substitution
 - Parent structures from high-throughput screening candidates
 - Substitutions from chemical similarity rules by Wang *et al.*
- **979** materials generated from 329 candidates from high-throughput
- **239** predicted to have high ionic conductivity by GCN

Benchmarking CHGNet potential for molecular dynamics

CHGNet is a pretrained universal neural network potential.

Molecular dynamics with CHGNet accounts for more physics than bond-valence kinetic Monte Carlo.



- 3 false positives
- 0 false negatives

- Precision = 0.75
- Recall = 1.00
- $F1 = 0.86$

- CHGNet outperforms KMC
- 100x faster than *ab initio* molecular dynamics
- 100x slower than KMC

High-throughput screening

| | Source | Ionic Conductivity | | Haven Ratio | | Correlated Hopping | Anisotropic Diffusion | Hull Energy (eV/atom) | Reduction Potential (V) | ECS Window (V) |
|--|------------|--------------------------|--------|-------------|--------|--------------------|-----------------------|-----------------------|-------------------------|----------------|
| | | $\log_{10}(\text{S/cm})$ | | KMC | CHGNet | | | | | |
| | | KMC | CHGNet | | | | | | | |
| Li_2SbF_5 | WBM | 0.80 | -3.42 | 0.004 | 0.639 | | ✓ | 0.000 | 2.738 | 0.672 |
| Li_2AsF_5 | WBM | -0.12 | -3.39 | 0.004 | | | ✓ | 0.024 | 3.030 | 0.639 |
| LiZnPSe_4 | WBM | -1.11 | -0.35 | 0.544 | 0.376 | ✓ | | 0.001 | 2.119 | 0.062 |
| LiHgPSe_4 | WBM | -1.39 | -0.25 | 0.626 | | ✓ | | 0.003 | | |
| $\text{Li}_7\text{Cl}_3\text{O}_2$ | WBM | -3.41 | | 0.428 | | ✓ | | 0.028 | 0.000 | 2.767 |
| LiBSe_2 | SWCBM | -3.58 | -1.05 | 0.358 | | ✓ | | 0.000 | 1.802 | 0.018 |
| LiCuBr_2 | SWCBM | -3.98 | -3.04 | 0.570 | | | | 0.004 | 2.637 | 0.054 |
| Li_6PBrO_5 | mp-1103940 | -3.16 | -3.62 | 0.235 | 0.564 | ✓ | | 0.017 | 0.686 | 2.087 |
| $\text{Li}_2\text{Ta}_2\text{O}_3\text{F}_6$ | mp-561011 | -3.69 | -17.91 | 0.013 | 0.789 | ✓ | | 0.000 | 1.701 | 2.920 |
| $\text{Li}_{10}\text{Mg}_7\text{Cl}_{24}$ | mp-530738 | -3.89 | -1.62 | 0.009 | 0.205 | ✓ | | 0.009 | 0.886 | 2.902 |

Machine learning

| | Source | Ionic Conductivity | | Haven Ratio | | Correlated Hopping | Anisotropic Diffusion | Hull Energy (eV/atom) | Reduction Potential (V) | ECS Window (V) |
|-------------------------------------|---------|--------------------------|--------|-------------|--------|--------------------|-----------------------|-----------------------|-------------------------|----------------|
| | | $\log_{10}(\text{S/cm})$ | | | | | | | | |
| | | KMC | CHGNet | KMC | CHGNet | | | | | |
| Li_2SbBr_5 | WBM | 1.03 | -0.01 | 0.009 | 0.556 | | ✓ | 0.028 | 2.327 | 0.961 |
| $\text{Li}_5\text{I}_2\text{N}$ | WBM | -0.99 | -2.14 | 0.318 | 0.803 | ✓ | ✓ | 0.000 | 0.000 | 0.477 |
| LiGaF_4 | WBM | -2.86 | -5.80 | 0.291 | | ✓ | ✓ | 0.009 | 2.578 | 3.614 |
| LiCuI_2 | SWCBM | -2.96 | -0.61 | 0.550 | | | | 0.007 | 2.159 | 0.414 |
| $\text{LiGaSi}_4\text{O}_{10}$ | CAMD | -3.23 | | 0.065 | 0.639 | ✓ | ✓ | 0.023 | 1.814 | 1.950 |
| $\text{Li}_3\text{AlB}_2\text{O}_6$ | CAMD | -3.72 | | 0.097 | 0.210 | | | 0.027 | 1.072 | 2.226 |
| LiBI_4 | MatProj | -0.09 | | 0.014 | 0.873 | | ✓ | 0.000 | 2.366 | 0.260 |
| LiMn_2I_9 | MatProj | -0.62 | | 0.046 | | ✓ | ✓ | 0.000 | 2.410 | 2.427 |
| LiVBr_5 | MatProj | -0.96 | 0.02 | 0.286 | 0.385 | ✓ | ✓ | 0.000 | 2.646 | 2.219 |
| Li_2TiF_6 | MatProj | -3.32 | -1.27 | 0.073 | 0.246 | ✓ | ✓ | 0.026 | 2.130 | 4.170 |

Li₂BF₅ analogs

High-throughput

Machine Learning

Li₂SbF₅

Li₂SbBr₅

Li₂AsF₅

Li₅I₂N

LiZnPSe₄

LiGaF₄

LiHgPSe₄

LiCuI₂

Li₇Cl₃O₂

LiGaSi₄O₁₀

LiBSe₂

Li₃AlB₂O₆

LiCuBr₂

LiBI₄

Li₆PBrO₅

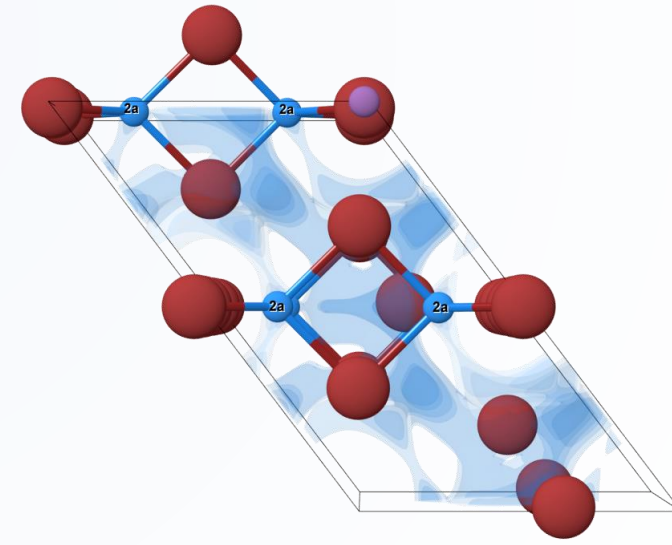
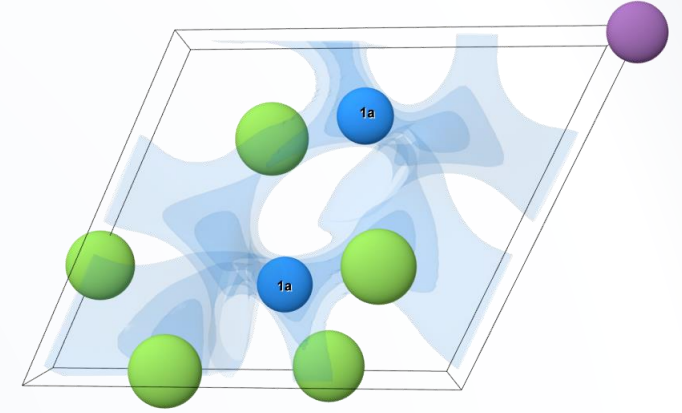
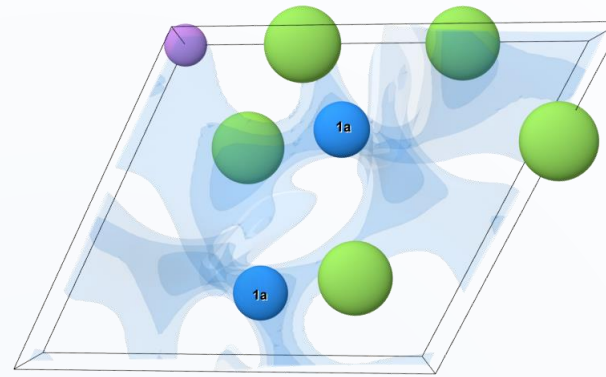
LiMn₂I₉

Li₂Ta₂O₃F₆

LiVBr₅

Li₁₀Mg₇Cl₂₄

Li₂TiF₆



LiZnPS₄ analogs

High-throughput

Machine Learning

Li₂SbF₅

Li₂AsF₅

LiZnPSe₄

LiHgPSe₄

Li₇Cl₃O₂

LiBSe₂

LiCuBr₂

Li₆PBrO₅

Li₂Ta₂O₃F₆

Li₁₀Mg₇Cl₂₄

Li₂SbBr₅

Li₅I₂N

LiGaF₄

LiCuI₂

LiGaSi₄O₁₀

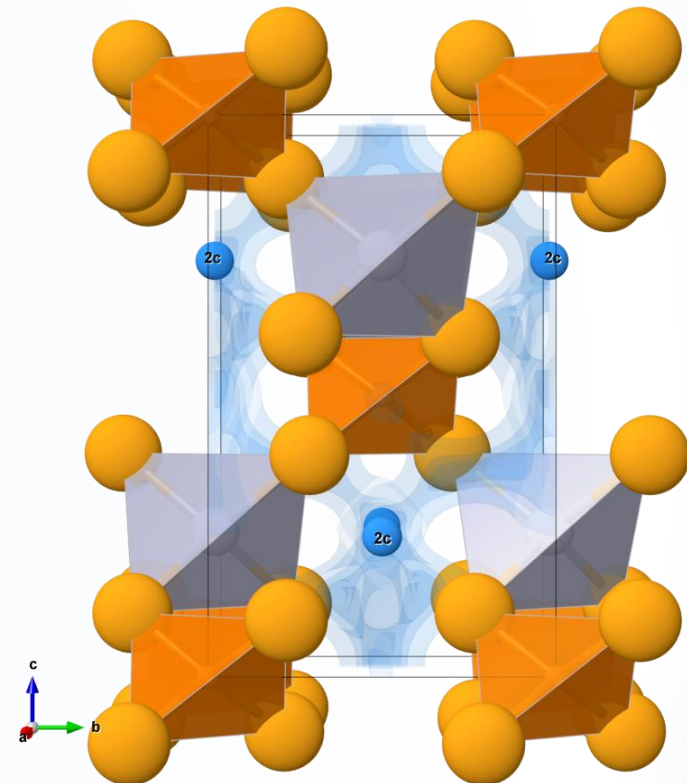
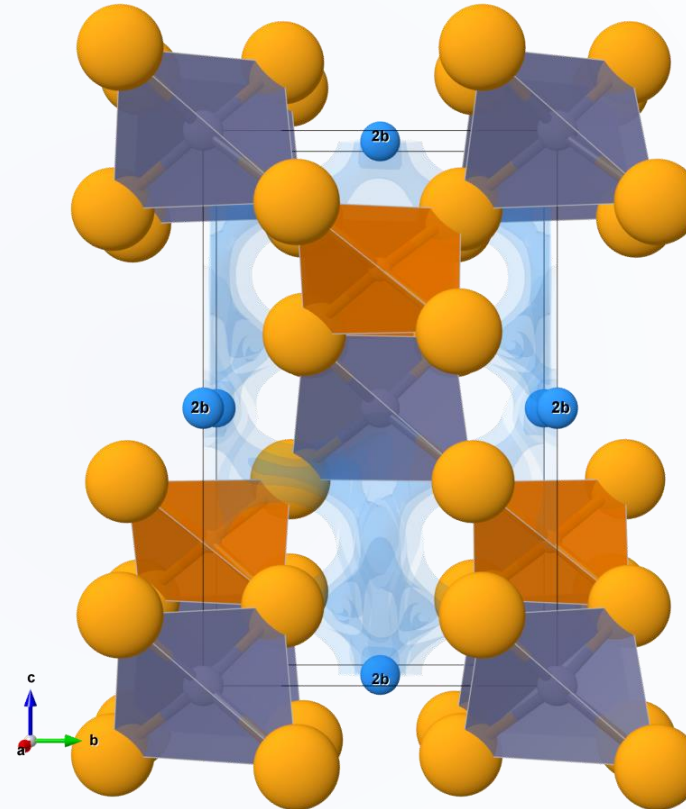
Li₃AlB₂O₆

LiBI₄

LiMn₂I₉

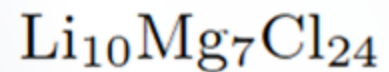
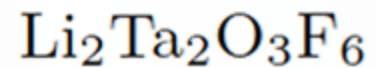
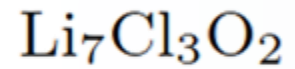
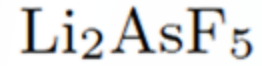
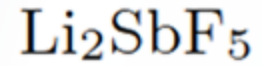
LiVBr₅

Li₂TiF₆

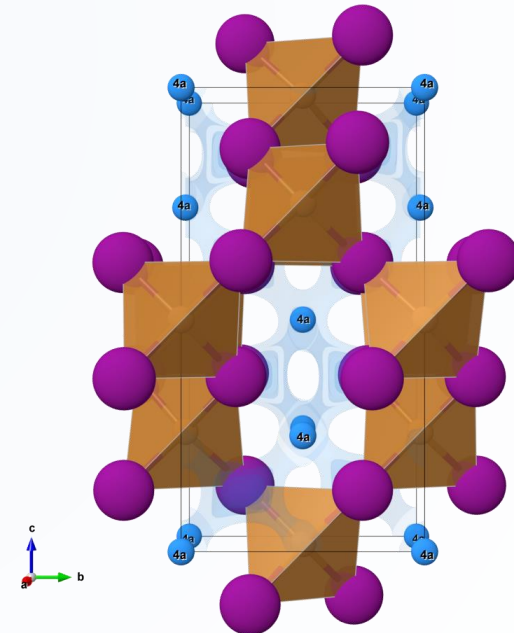
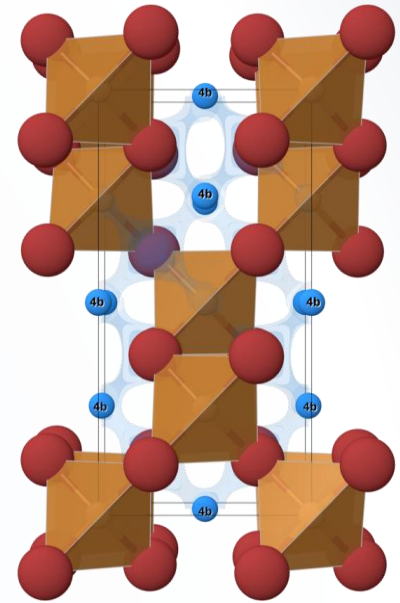
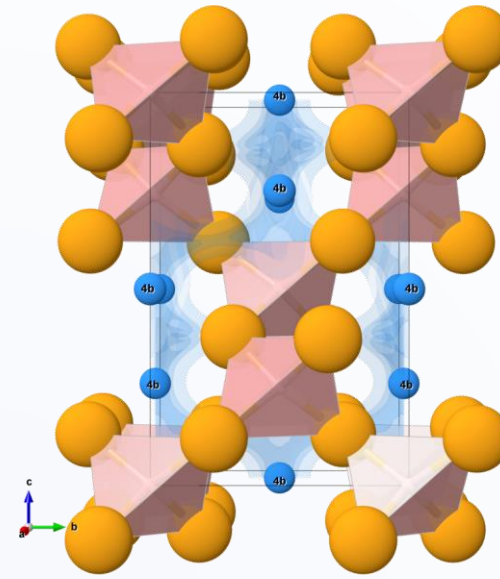
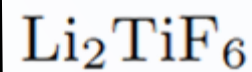
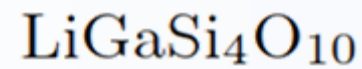
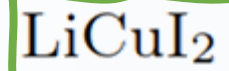
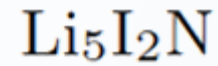
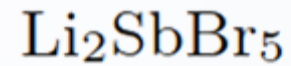


Wurtzites

High-throughput



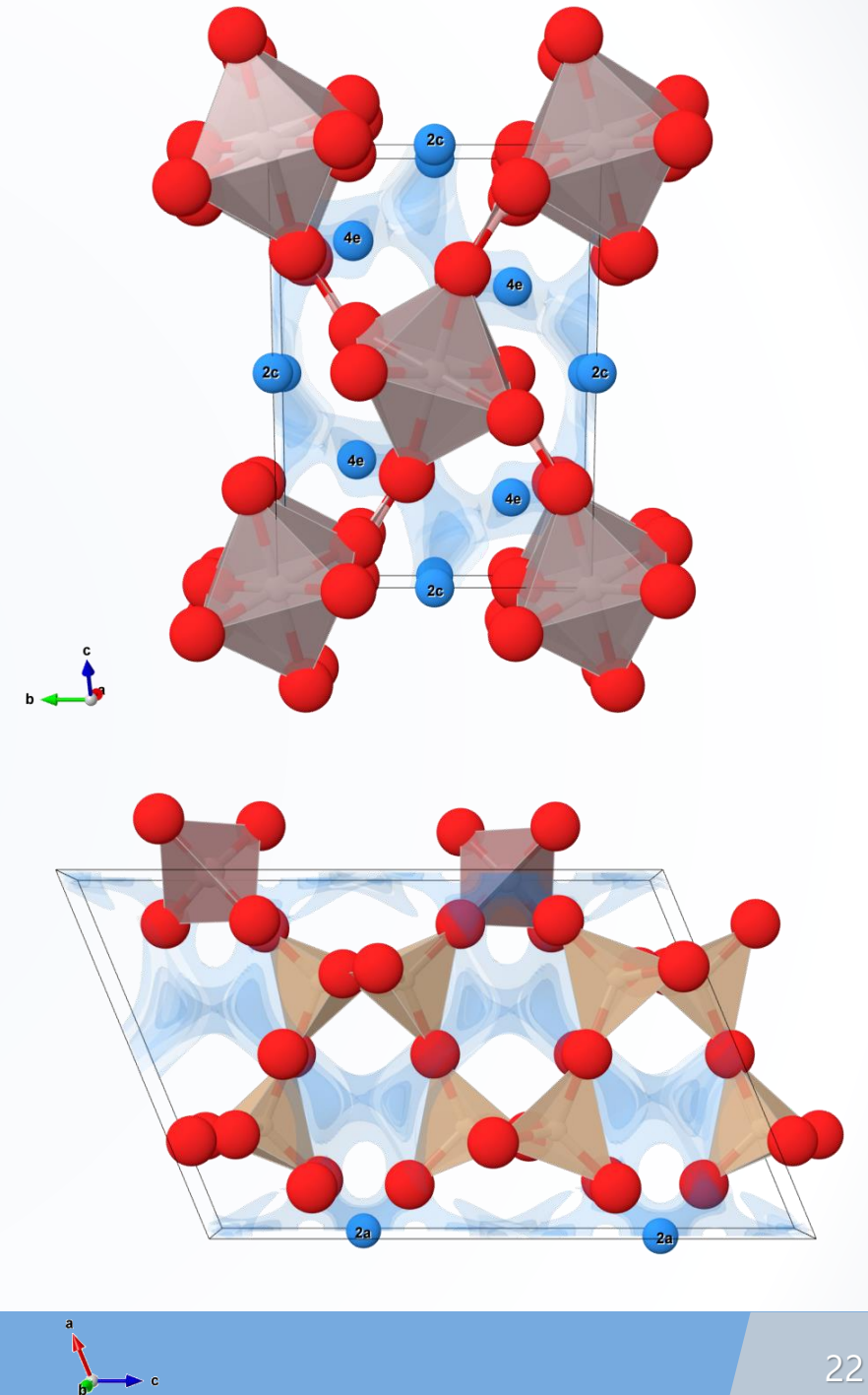
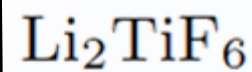
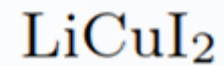
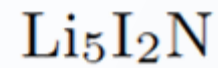
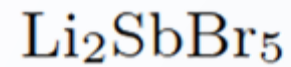
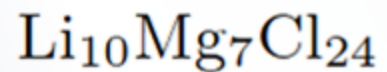
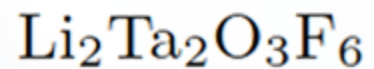
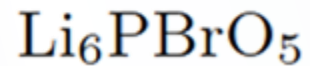
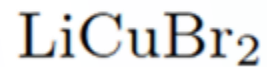
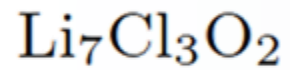
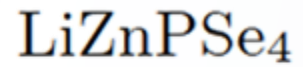
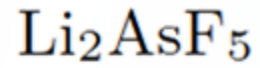
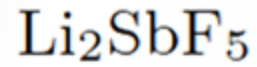
Machine Learning



Oxides

High-throughput

Machine Learning



Fluorides, Chlorides

High-throughput

Machine Learning

Li_2SbF_5

Li_2AsF_5

LiZnPSe_4

LiHgPSe_4

$\text{Li}_7\text{Cl}_3\text{O}_2$

LiBSe_2

LiCuBr_2

Li_6PBrO_5

$\text{Li}_2\text{Ta}_2\text{O}_3\text{F}_6$

$\text{Li}_{10}\text{Mg}_7\text{Cl}_{24}$

Li_2SbBr_5

$\text{Li}_5\text{I}_2\text{N}$

LiGaF_4

LiCuI_2

$\text{LiGaSi}_4\text{O}_{10}$

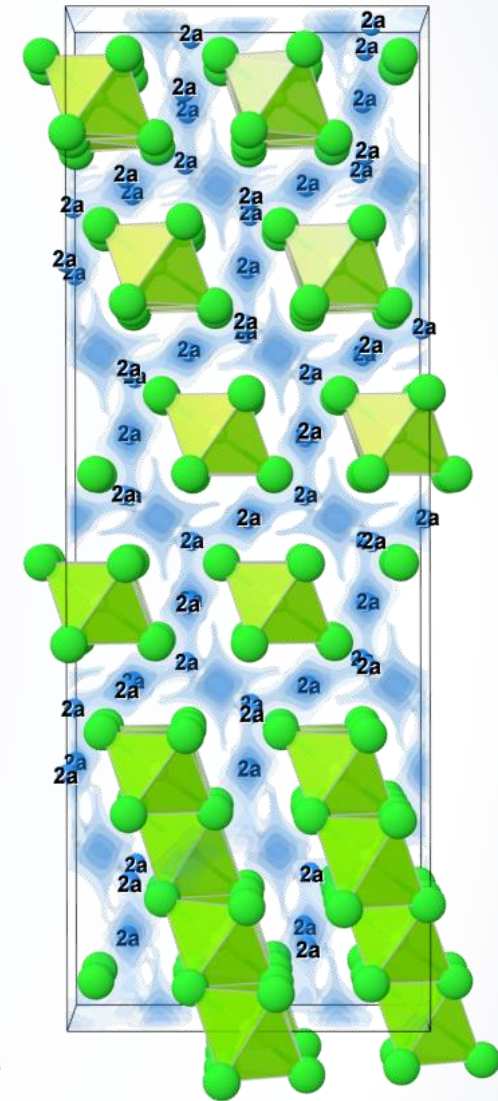
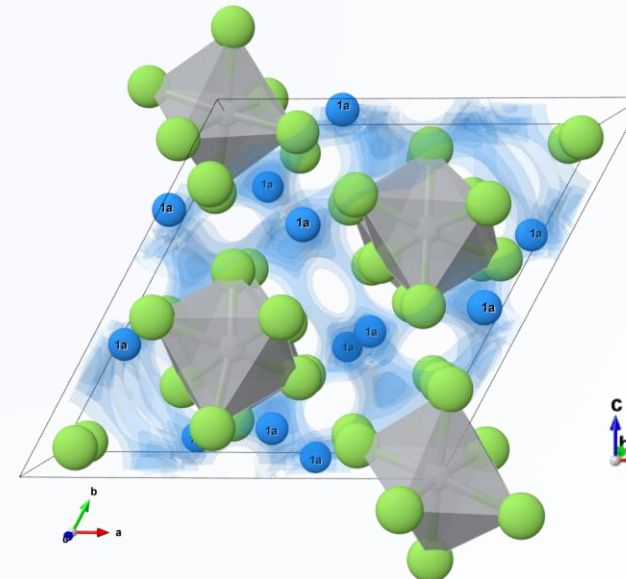
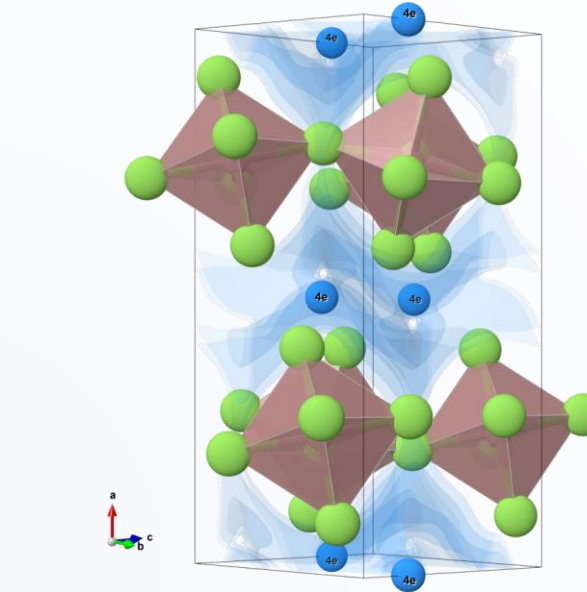
$\text{Li}_3\text{AlB}_2\text{O}_6$

LiBI_4

LiMn_2I_9

LiVBr_5

Li_2TiF_6



Bromides, Iodides

| High-throughput | Machine Learning |
|-----------------|------------------|
|-----------------|------------------|

Li_2SbF_5

Li_2AsF_5

LiZnPSe_4

LiHgPSe_4

$\text{Li}_7\text{Cl}_3\text{O}_2$

LiBSe_2

LiCuBr_2

Li_6PBrO_5

$\text{Li}_2\text{Ta}_2\text{O}_3\text{F}_6$

$\text{Li}_{10}\text{Mg}_7\text{Cl}_{24}$

Li_2SbBr_5

$\text{Li}_5\text{I}_2\text{N}$

LiGaF_4

LiCuI_2

$\text{LiGaSi}_4\text{O}_{10}$

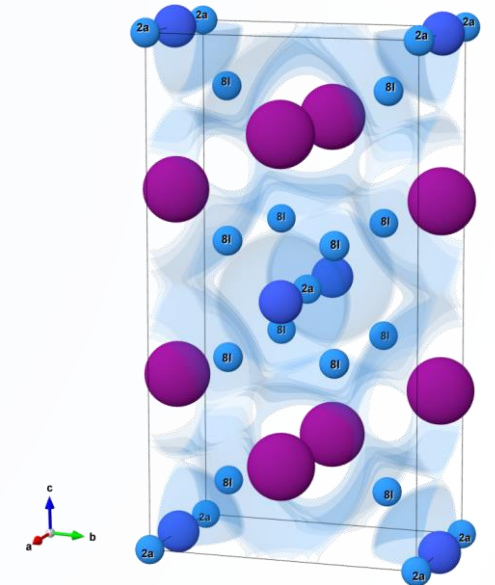
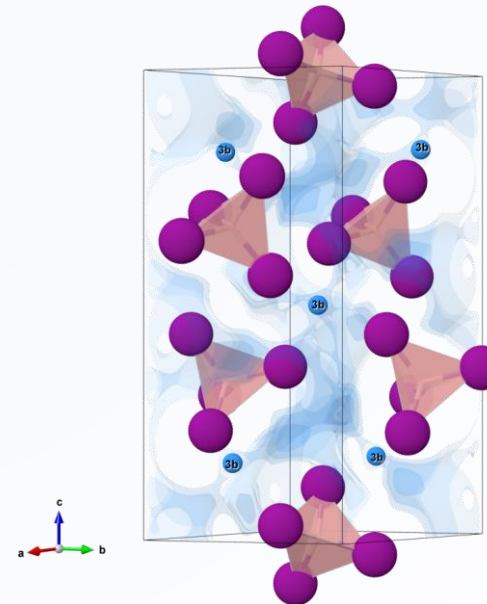
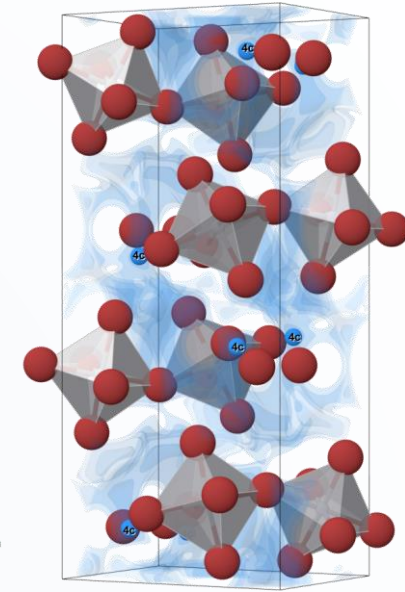
$\text{Li}_3\text{AlB}_2\text{O}_6$

LiBI_4

LiMn_2I_9

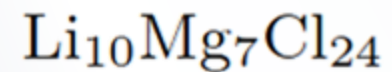
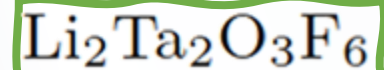
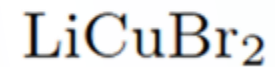
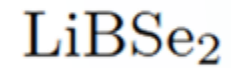
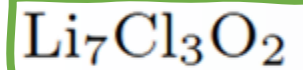
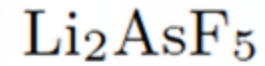
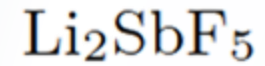
LiVBr_5

Li_2TiF_6

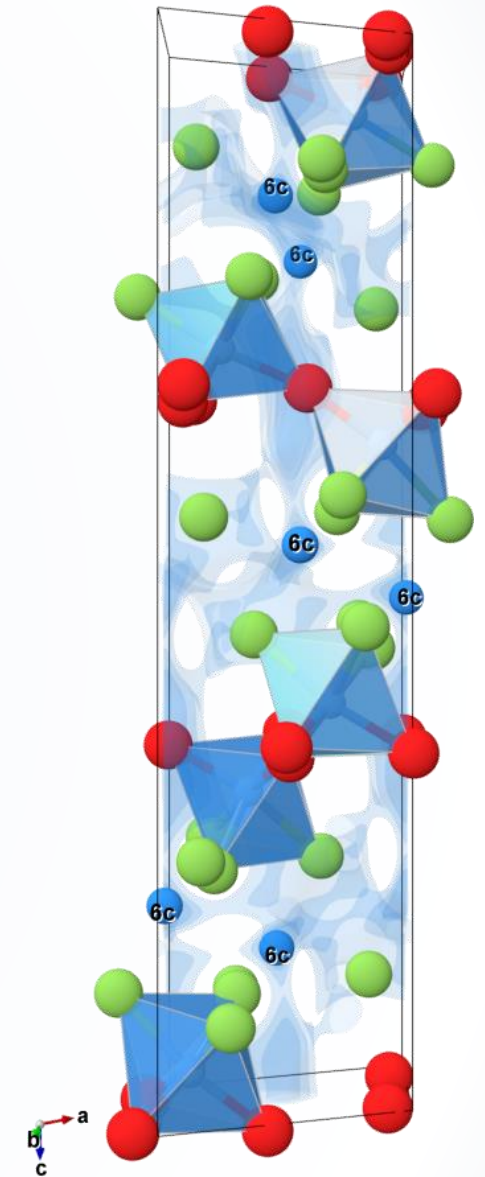
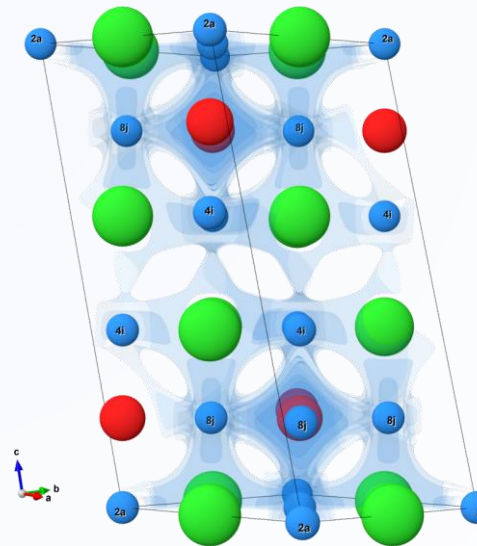
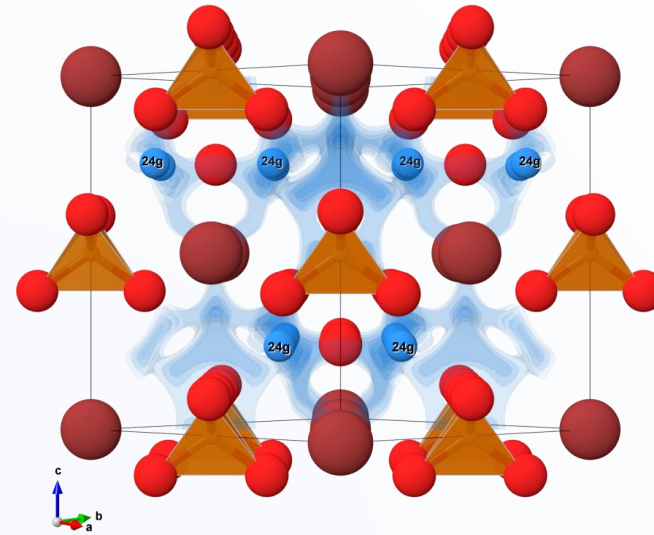
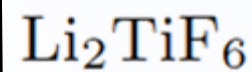
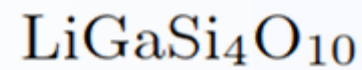
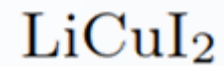
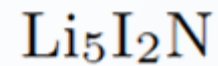
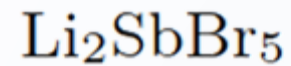


Oxohalides

High-throughput

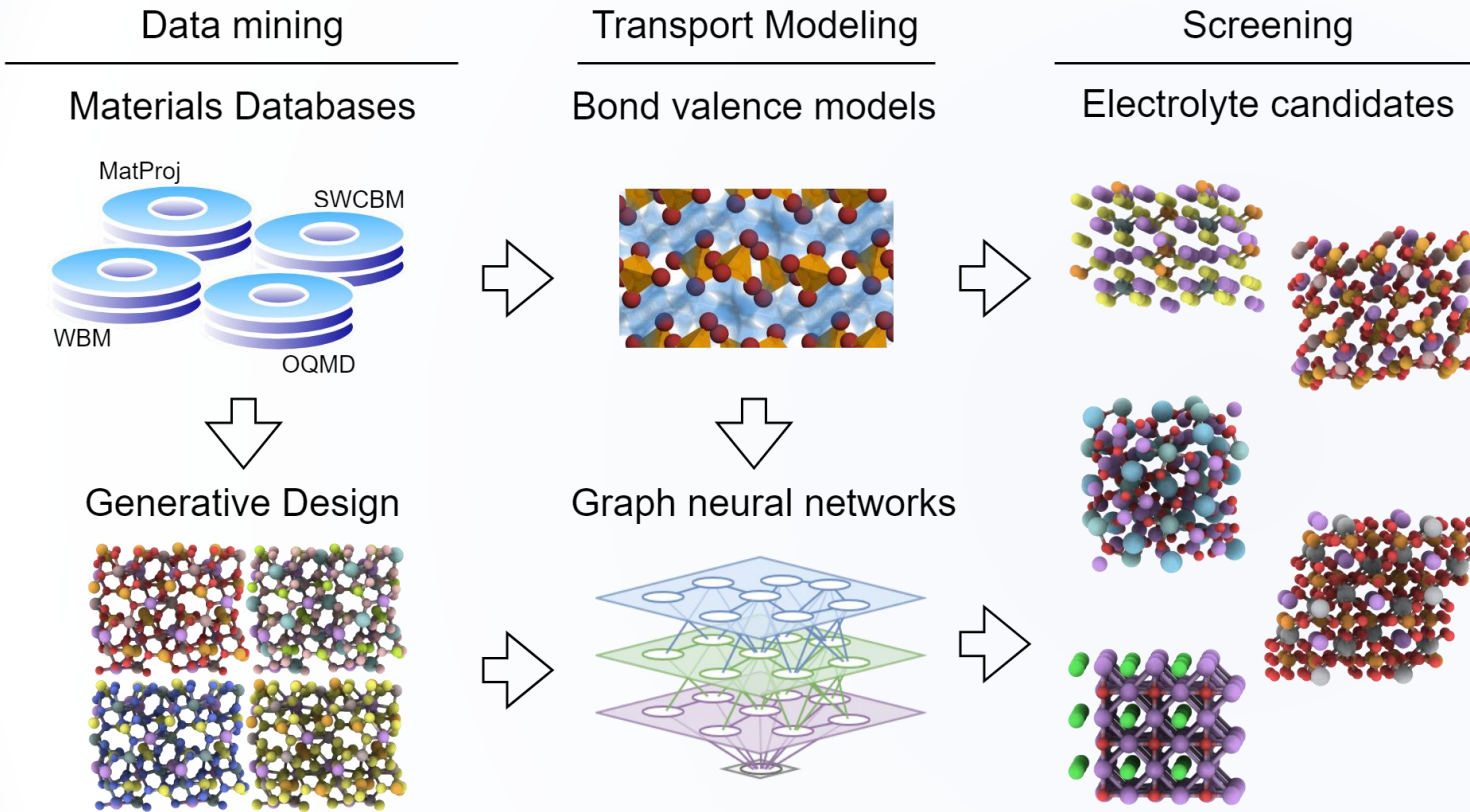


Machine Learning



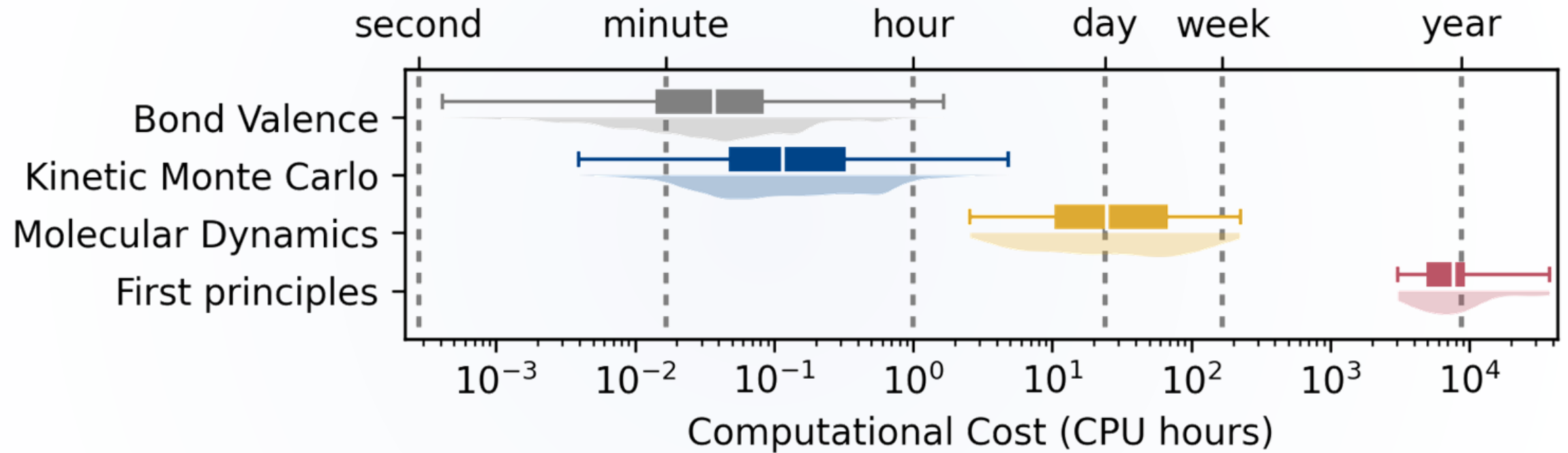


- Transformational Tools & Technology (TTT)
- Solid-state Architecture Batteries for Enhanced Rechargeability and Safety (SABERS)
- Ames Research Center
 - Junsoo Park
 - Joakim Stenlid
 - Zhigang Wu
 - Mohit Mehta
- West Virginia University
 - David Mebane



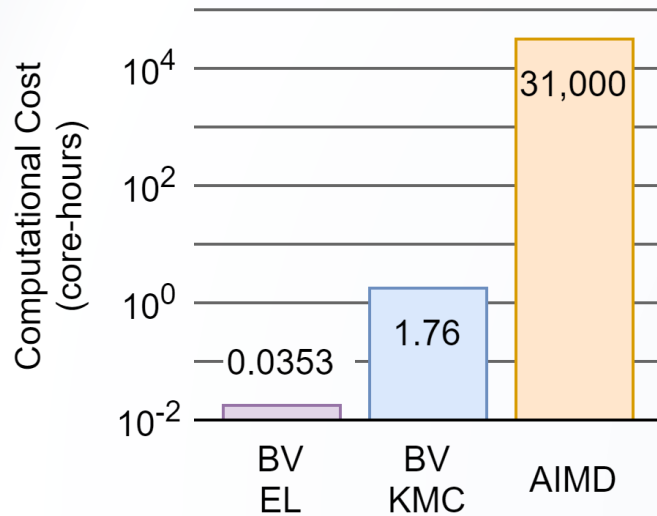


Computational cost



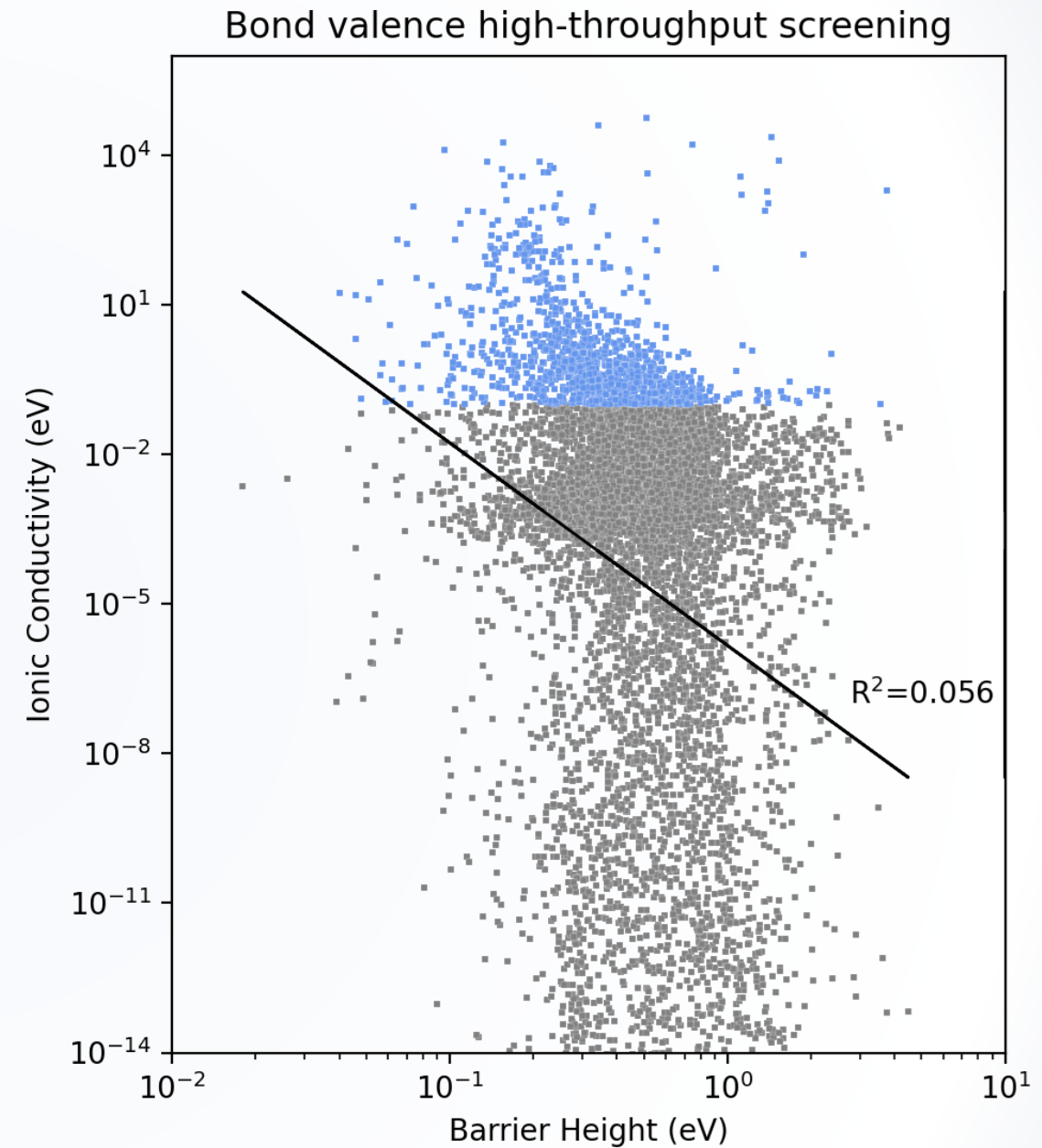
| | Energy landscape | Kinetic Monte Carlo | Molecular dynamics | First principles |
|--|------------------|---------------------|--------------------|------------------|
| Single-lithium picture | X | | | |
| Li ⁺ - Li ⁺ interactions | | X | X | X |
| Non-lithium motion | | | X | X |
| Ionic and electronic DoF | | | | X |

Kinetic Monte Carlo with softBV



$$p = \omega \exp\left(\frac{E_m + \Delta E}{k_B T}\right)$$

- Energy landscape is a single-Li⁺ picture
 - ignores Li⁺-Li⁺ interactions
 - ionic conductivity depends on mixture of occupied and unoccupied sites in pathways
- Transition rate also depends on vibration frequencies (pre-exponential factor)



Validation test of GCN trained on Kinetic Monte Carlo

