



National Aeronautics and Space Administration

Elevated Temperature Mechanical Performance of Historical Niobium Alloys

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TMS 2025. Las Vegas, NV. March 24

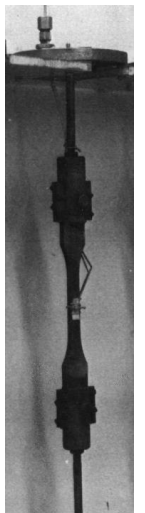
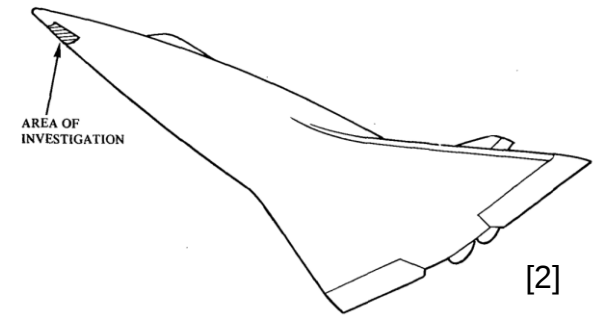


Introduction

- Niobium's unique combination of high melting point, low density, and excellent strength make it a prime candidate for re-entry vehicles and other high-temperature applications. The development of niobium (Nb) alloys was closely tied to the aerospace and defense advancements of the mid-20th century.

- **Historical Context**

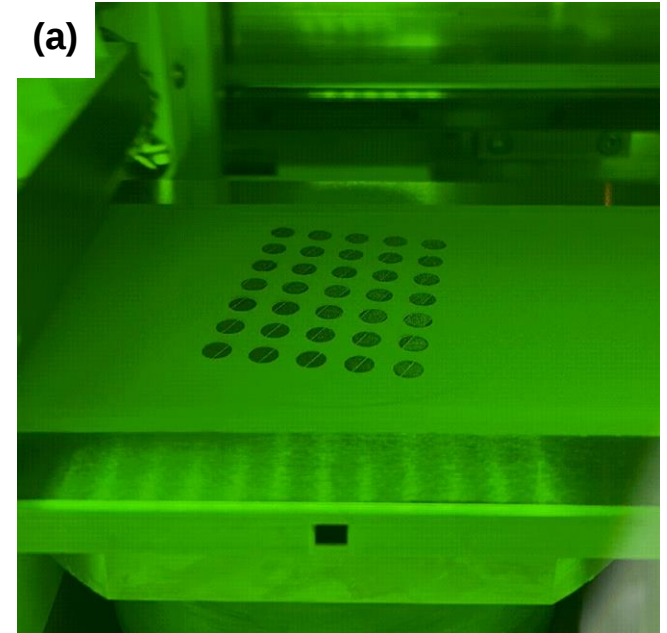
- In the 1960s, *Boeing* and *Wah Chang Corp.* initiated a cooperative effort to explore Nb-based alloys for re-entry shielding leading to the development of **C103** (Nb-10Hf-1Ti , wt%). [1]
- The early 1970s saw intense Nb alloy research driven by the space race and missile defense needs. During this time, **FS85** (Nb-28Ta-10W-1Zr) was developed by *Fansteel*. [1]
- Meanwhile, **Cb752** (Nb-10W-2.5Zr) was explored for elevated temperature structural aerospace components, yet no significant applications materialized. [1]
- Later, in the mid-to-late 1970s, *Wah Chang* introduced **WC3009** (Nb-30Hf-9W-1Zr) as an oxidation-resistant alloy, though it ultimately failed to gain widespread adoption. [1]



Motivation

- Refractory alloys, particularly Nb-based alloys, are experiencing a resurgence due to advancements in powder-based additive manufacturing.
- Researchers at NASA Glenn Research Center and Marshall Space Flight Center are actively evaluating a wide range of historical Nb alloys to identify candidates for modern aerospace applications.
- This study focuses on the performance of **C103**, **FS85**, **Cb752** and **WC3009**, assessing their mechanical properties at elevated temperatures when fabricated using Laser Powder Bed Fusion (L-PBF) additive manufacturing.
- The goal is to determine whether any of these legacy alloys retain potential for future aerospace use, especially in high-temperature environments requiring exceptional strength and creep resistance. Additionally, this property screening serves as a foundation for the development of next-generation Nb alloys.

Nb alloy (a) L-PBF processing and (b) completed parameter development build





Outline

1. Materials and Processing

- L-PBF Consolidation
- Chemical Analysis

2. Mechanical Testing

- Testing Equipment
- Elevated Temperature Tension
- Elevated Temperature Constant Load (Pseudo-Creep)

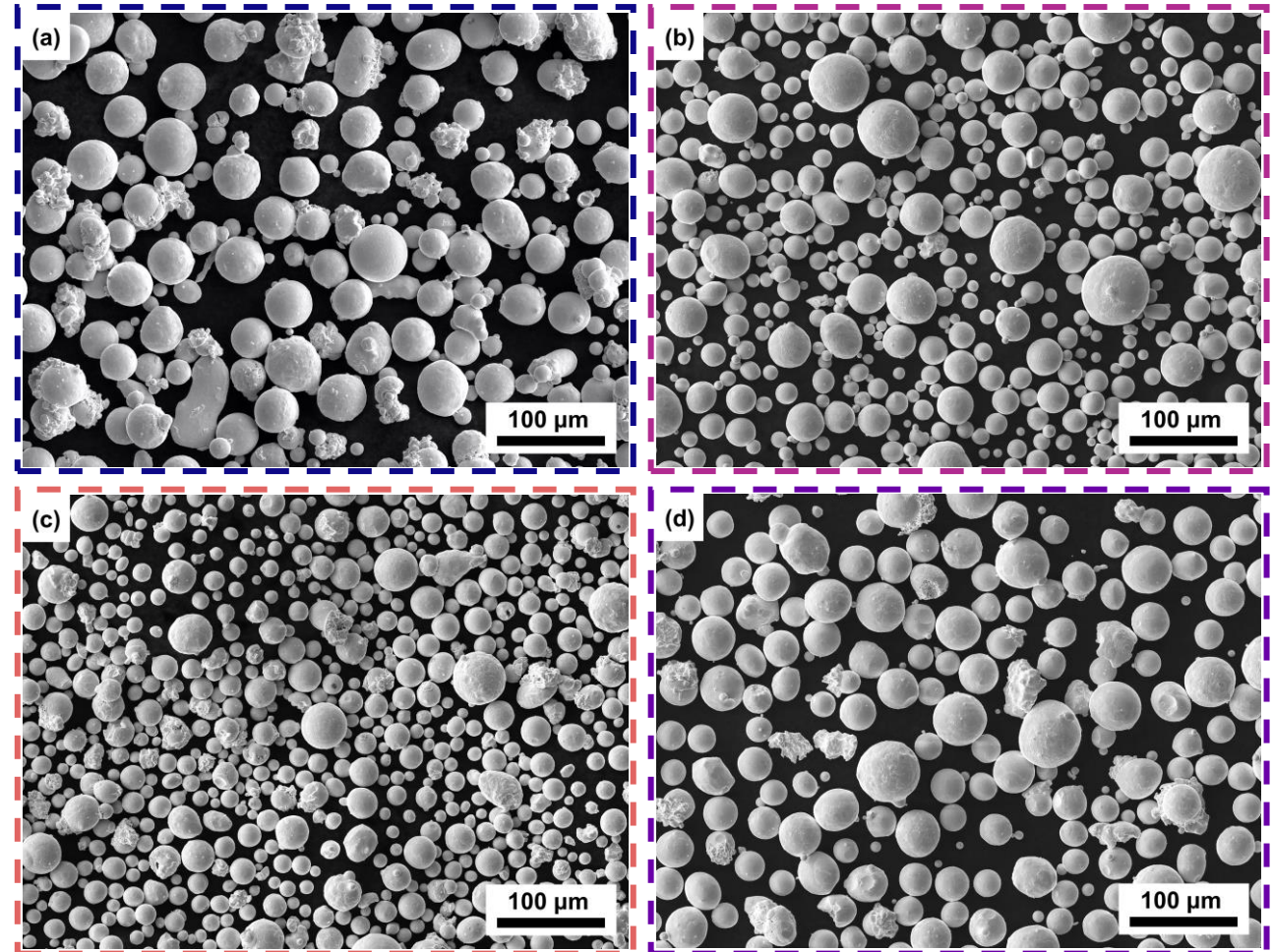
3. Comparative Evaluation

- Microstructural Thermal Stability

4. Conclusions

Powder Materials

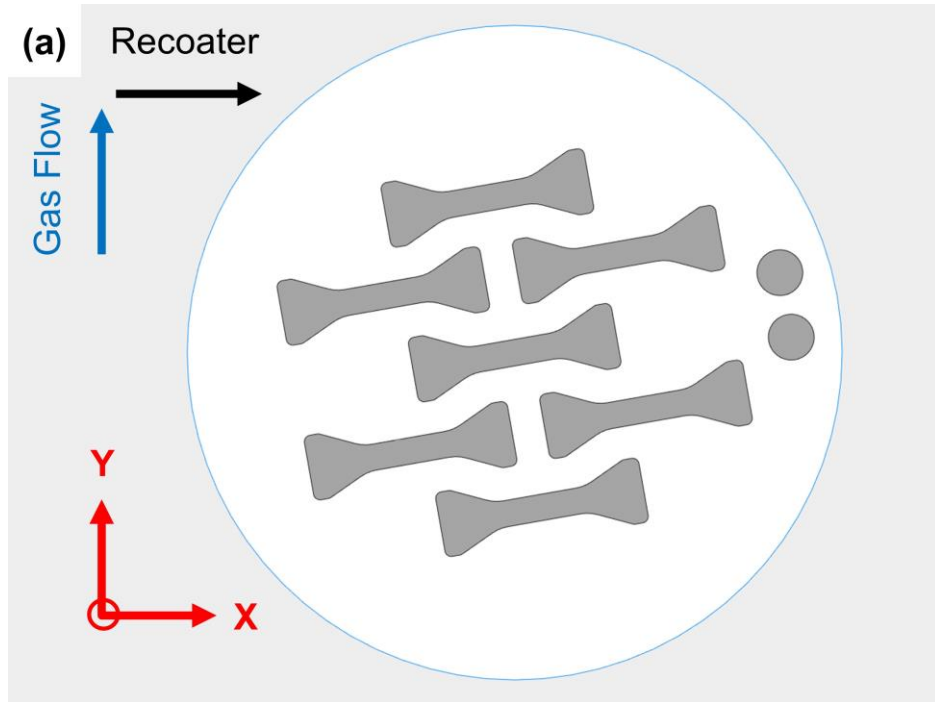
- **C103** (*Nb-10Hf-1Ti*)
 - $D_{10} = 16.3 \mu\text{m}$
 - $D_{90} = 51.5 \mu\text{m}$
- **Cb752** (*Nb-10W-2.5Zr*)
 - $D_{10} = 13.2 \mu\text{m}$
 - $D_{90} = 66.2 \mu\text{m}$ ↑
- **FS85** (*Nb-28Ta-10W-1Zr*)
 - $D_{10} = 16.4 \mu\text{m}$
 - $D_{90} = 46.8 \mu\text{m}$
- **WC3009** (*Nb-30Hf-9W-1Zr*)
 - $D_{10} = 30.7 \mu\text{m}$ ↑
 - $D_{90} = 70.7 \mu\text{m}$ ↑



Secondary electron SEM micrographs of the (a) **C103**, (b) **Cb752**, (c) **FS85**, and (d) **WC3009** Nb alloy spherical powders

Laser Powder Bed Fusion

- *EOS M100*, Nb build plate, Argon atmosphere (<10 ppm O₂), 31.5 x 10 x 1.5 mm bowties



(a) Schematic L-PBF build layout and (b) image of the completed **C103** build

L-PBF volumetric energy density (VED) and material density measurements

Material	C103	Cb752	FS85	WC-3009
VED [J/mm ³]	115	140	120	120
Arch. Density [g/cc]	8.80	9.00	10.60	10.30
Optical Density [%]	99.92	99.89	99.96	99.89

Chemical Composition

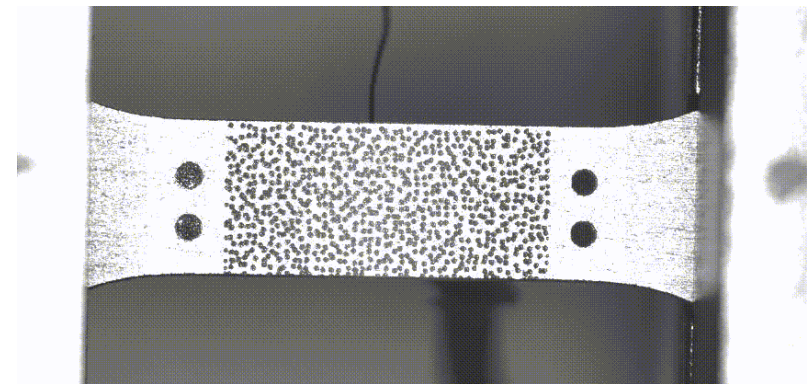
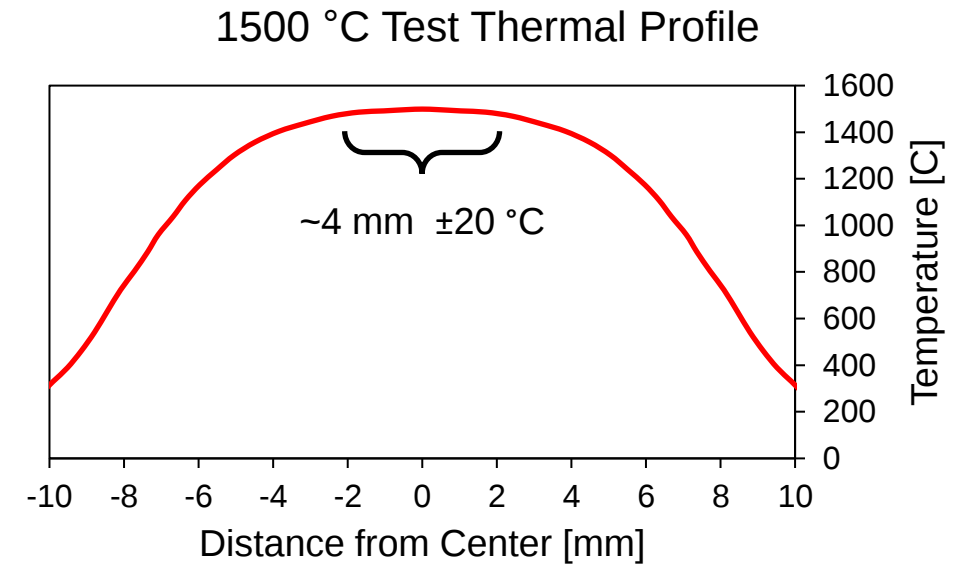
- Determined via SEM-EDS and LECO Combustion analysis

Material	Nb [wt%]	W [wt%]	Hf [wt%]	Ta [wt%]	Ti [wt%]	Zr [wt%]	O [ppm]	N [ppm]	C [ppm]
L-PBF C103 (Nb-10Hf-1Ti)	Bal.	0.27	9.85	0.36	0.92	-	425 ↑	58	19
L-PBF Cb752 (Nb-10W-2.5Zr)	Bal.	9.93	-	-	-	2.82 ↑	376 ↑	102	15
L-PBF FS85 (Nb-28Ta-10W-1Zr)	Bal.	10.05	-	27.71	-	0.16 ↓	524 ↑	115	7
L-PBF WC3009 (Nb-30Hf-9W-1Zr)	Bal.	9.39	29.53	-	-	0.45 ↓	242	58	32

A dash symbolizes a result of “not measured/detected”

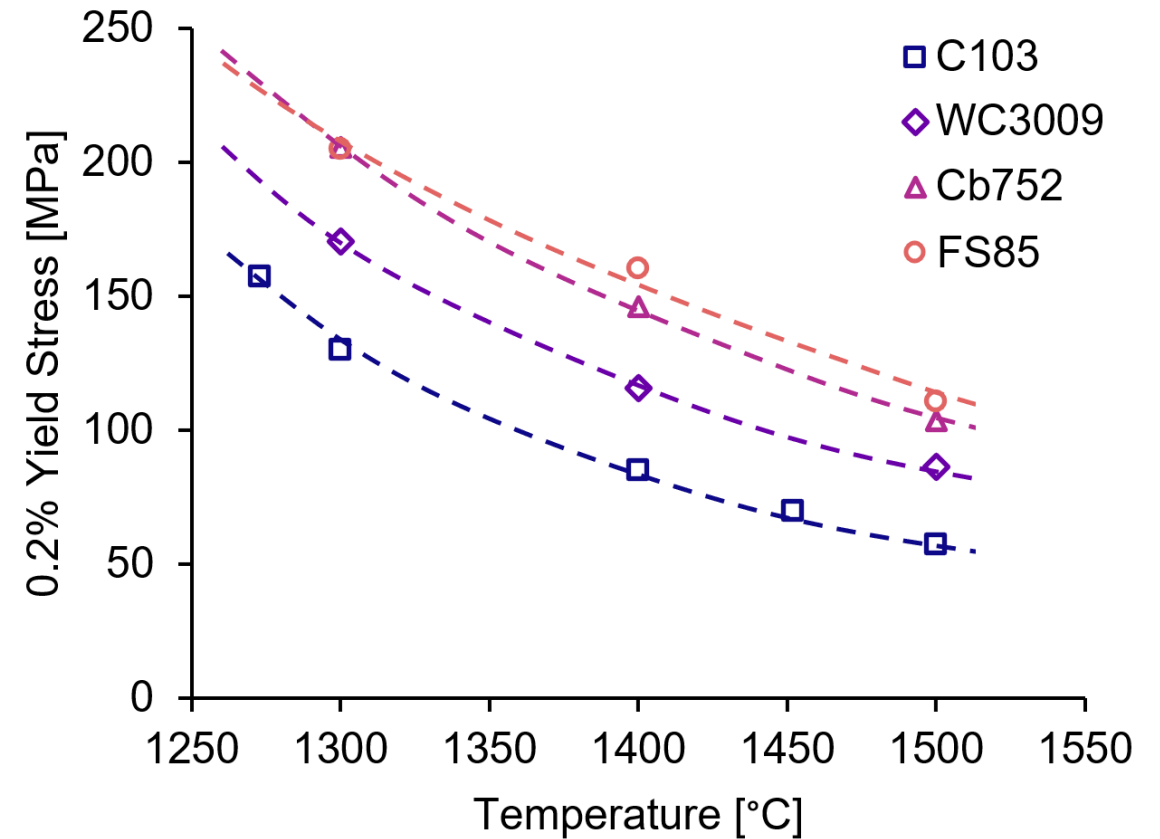
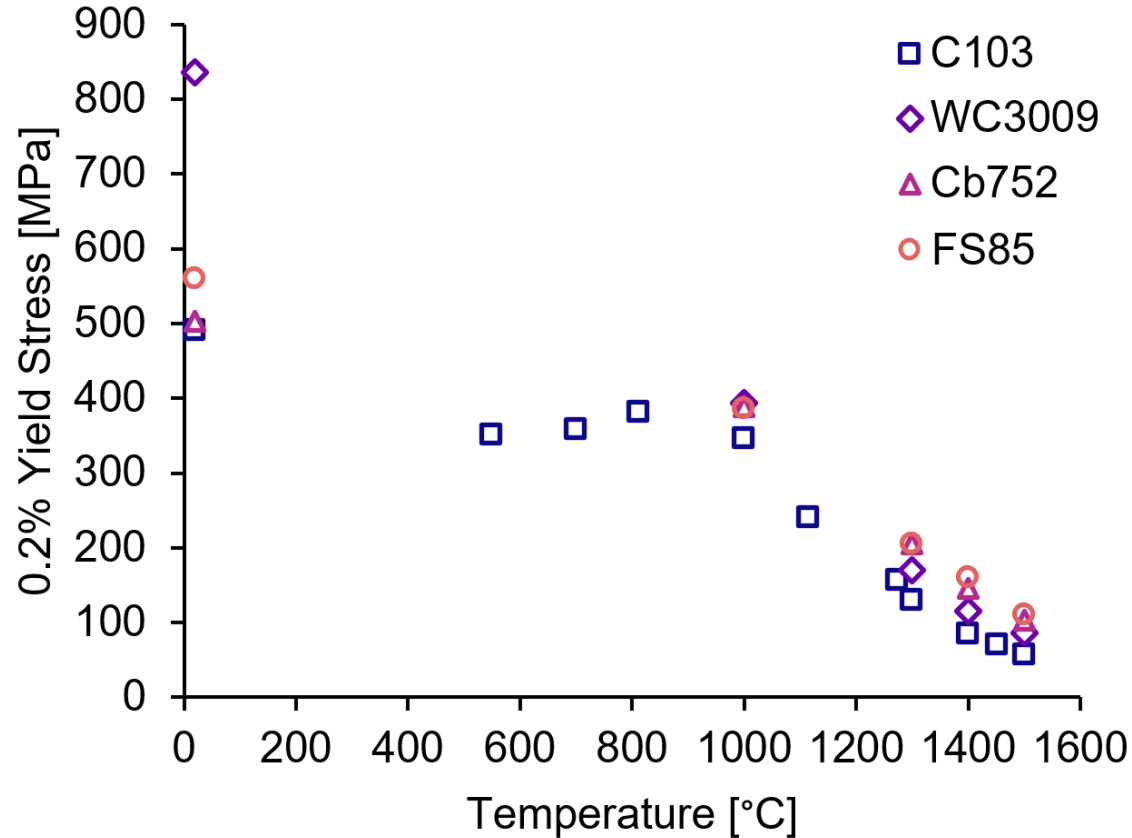
Elevated Temperature Mechanical Testing Equipment

- *Instron* Electro-Thermal Mechanical Testing (ETMT)
- Direct resistance heating (non-uniform thermal gradient)
- High-vacuum: $< 10^{-5}$ Pa ($< 10^{-7}$ Torr)
- Supports uniaxial tension & constant load (pseudo-creep)



Elevated Temperature Uniaxial Tensile Testing

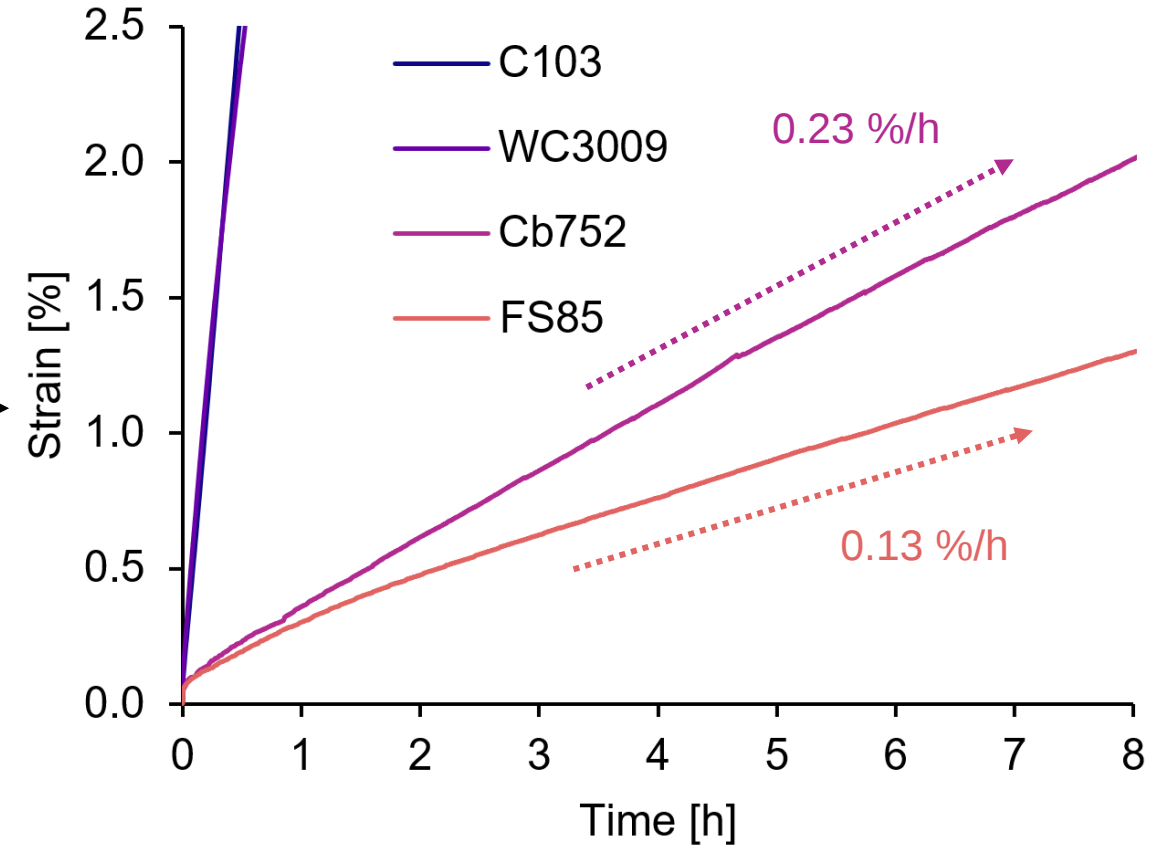
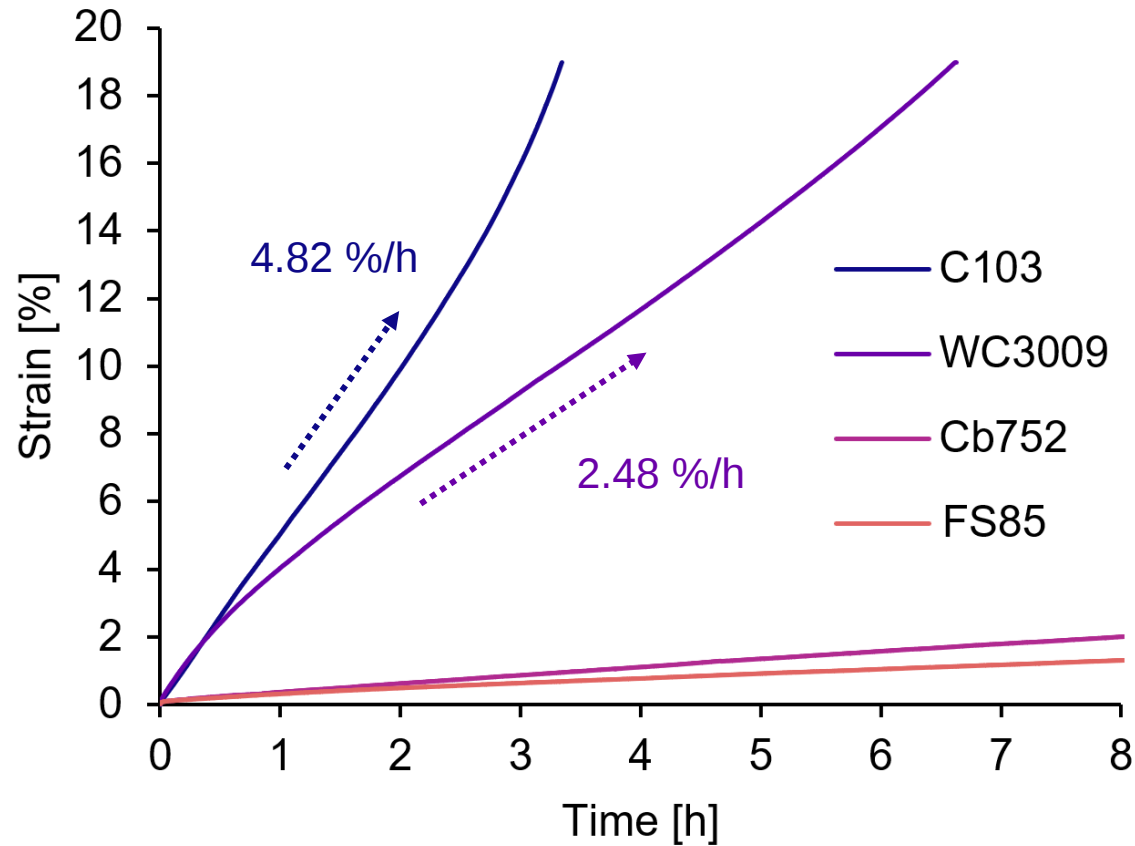
- At temperatures $> 1300^{\circ}\text{C}$: **FS85** $\approx$ **Cb752** $>$ **WC3009** $>$ **C103**



Note: 25°C data for WC3009, Cb752, and FS85 were stress relieved at 1200°C for 1 hour. All other data generated from samples stress relieved at 900°C for 1 hour.

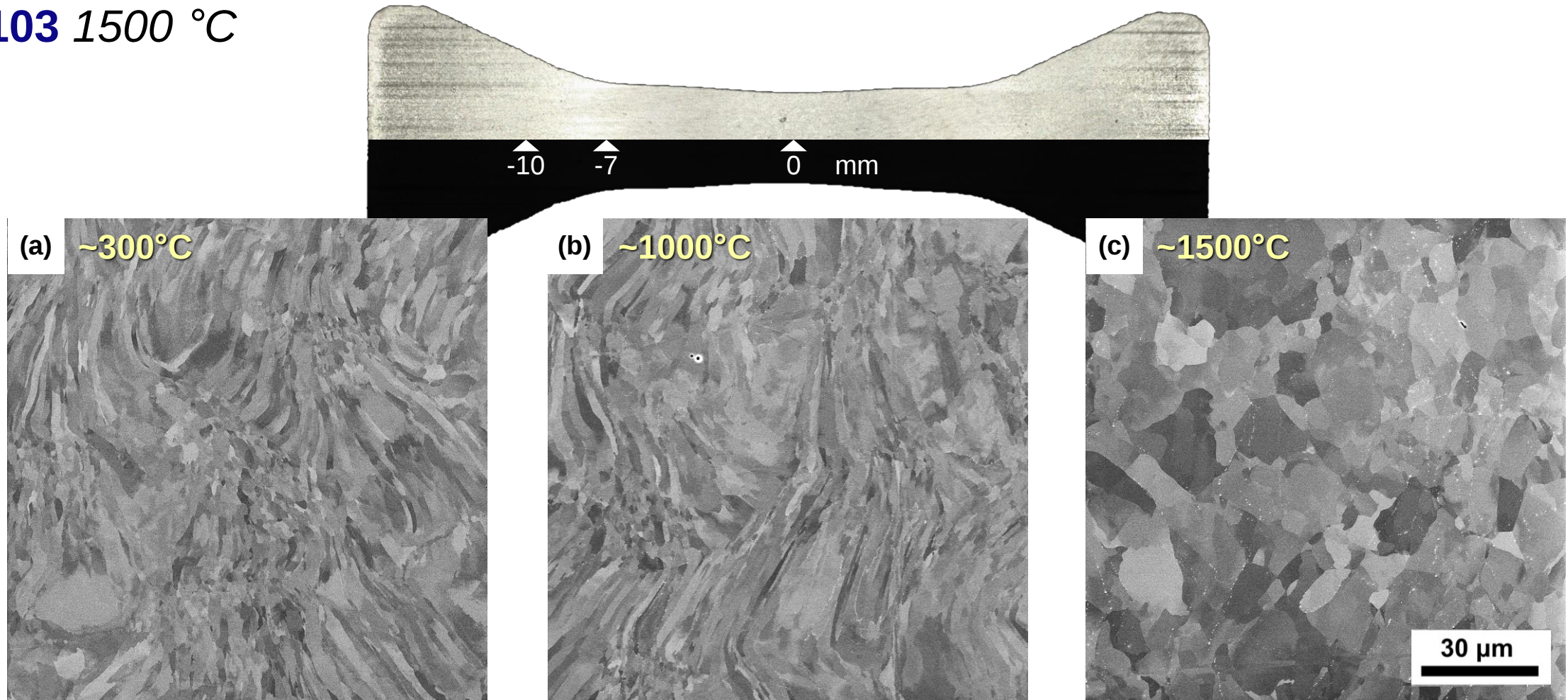
Elevated Temperature Constant Load (Pseudo-creep)

- At 1300 °C and a constant load of 50 MPa: **FS85** > **Cb752** > **WC3009** > **C103**



Microstructural Characterization: *L-PBF* > *SR* > *Tension*

- **C103** 1500 °C



Low magnification secondary electron micrographs of the 1500 °C elevated temperature L-PBF **C103** at (a) -10, (b) -7, and (c) 0 mm from central axis

Microstructural Characterization: *L-PBF* > *SR* > *Tested*

Low Mag.

C103

Cb752

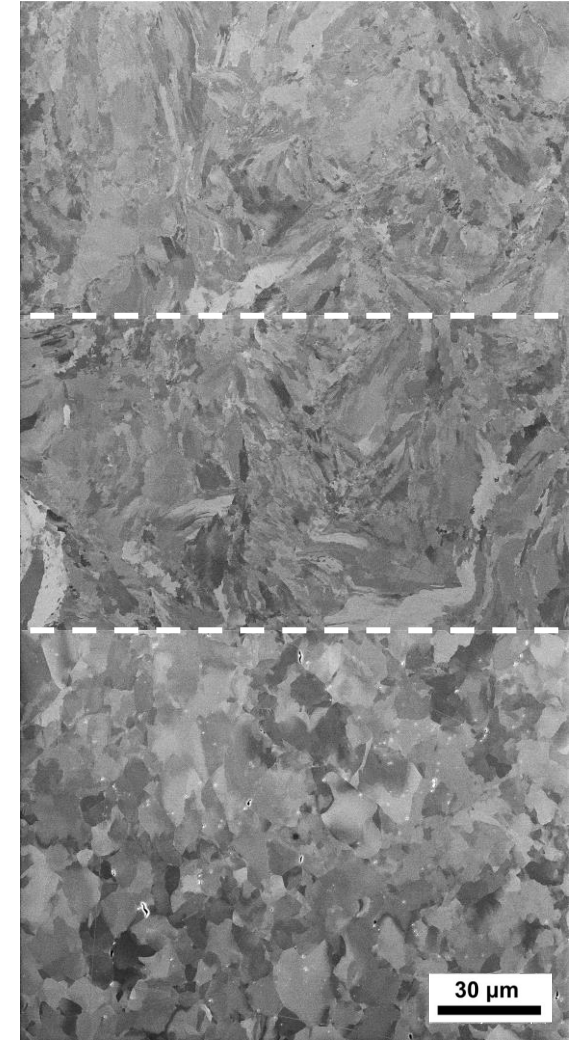
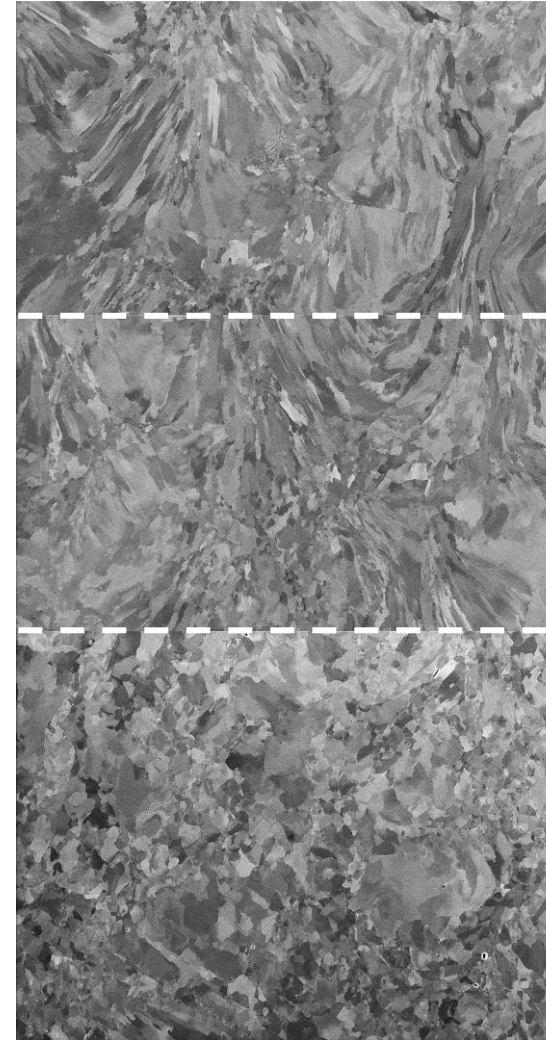
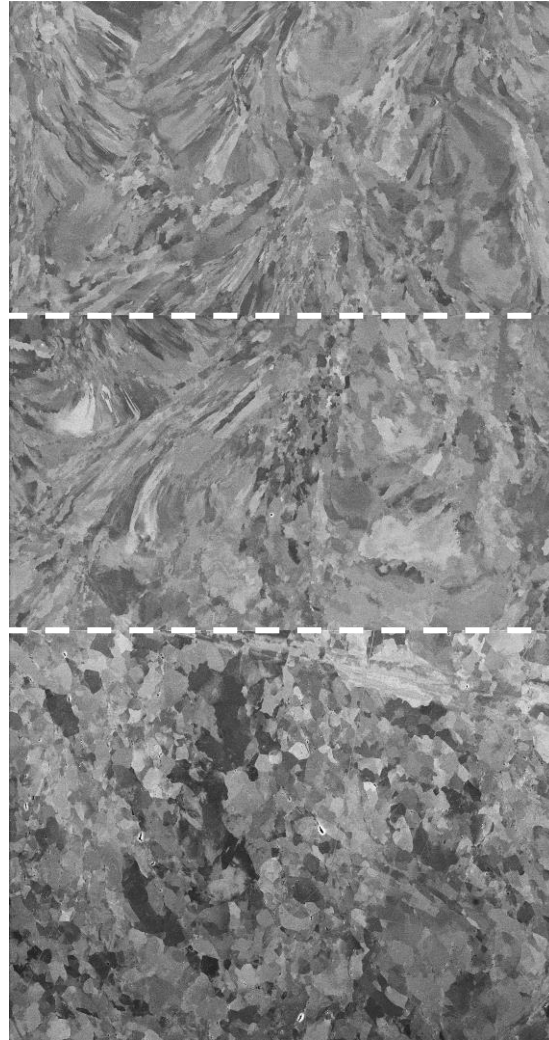
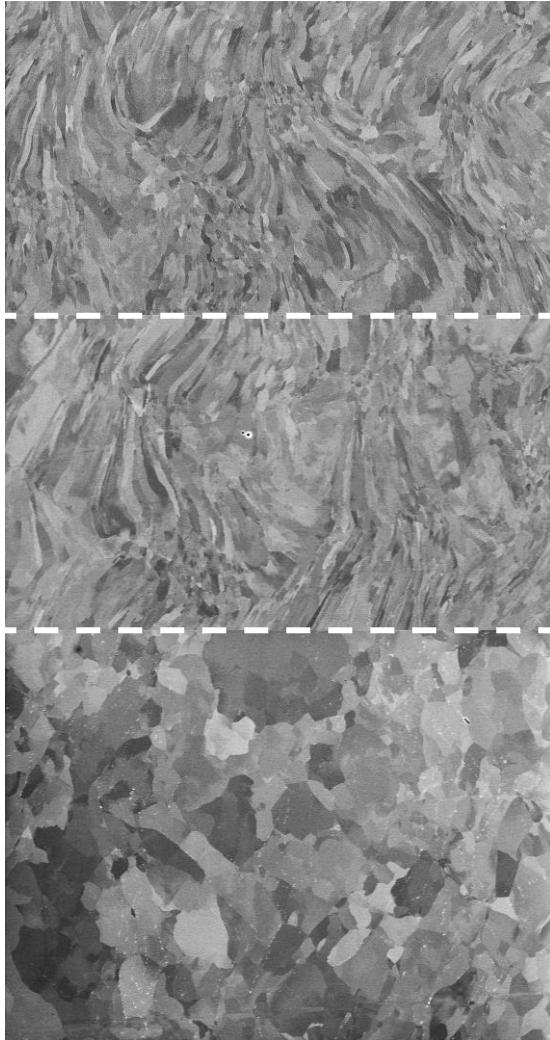
FS85

WC3009

~300 °C

~1000 °C

~1500 °C



Microstructural Characterization: *L-PBF* > *SR* > *Tested*

High Mag.

C103

Cb752

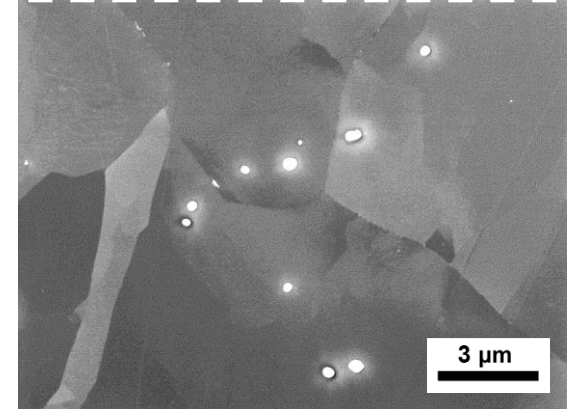
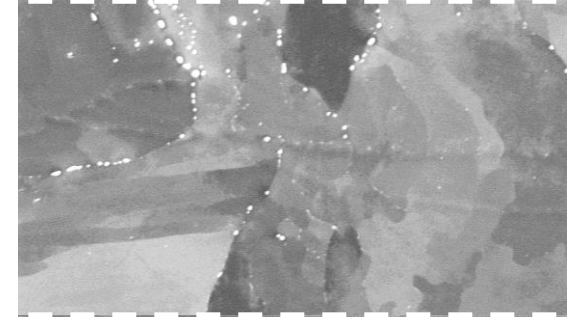
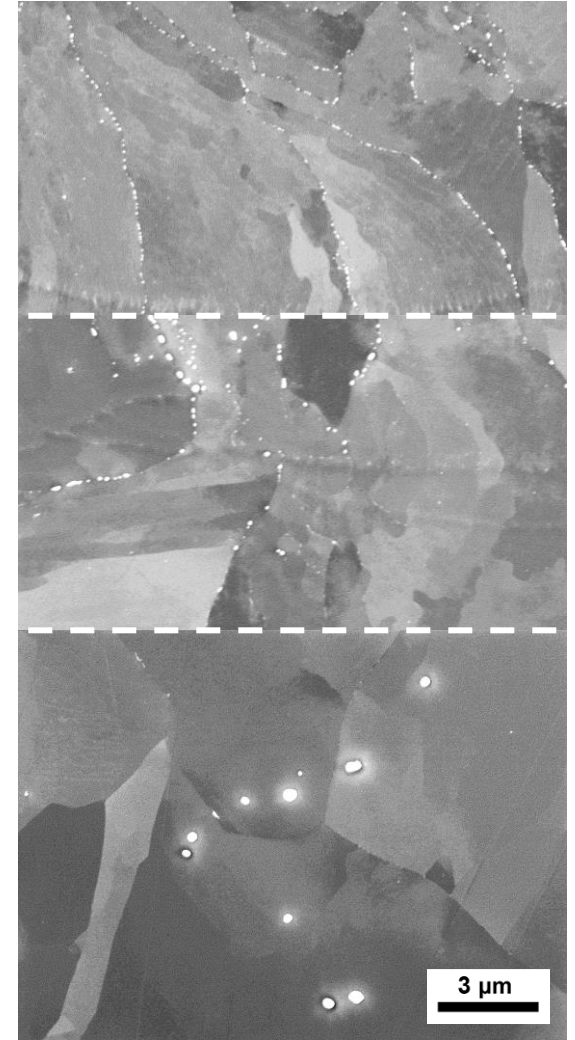
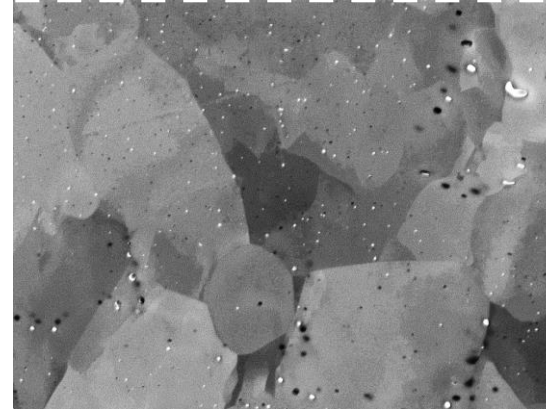
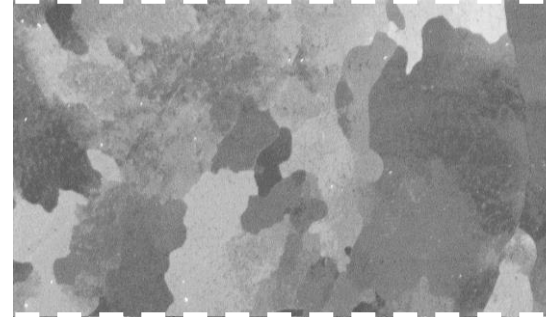
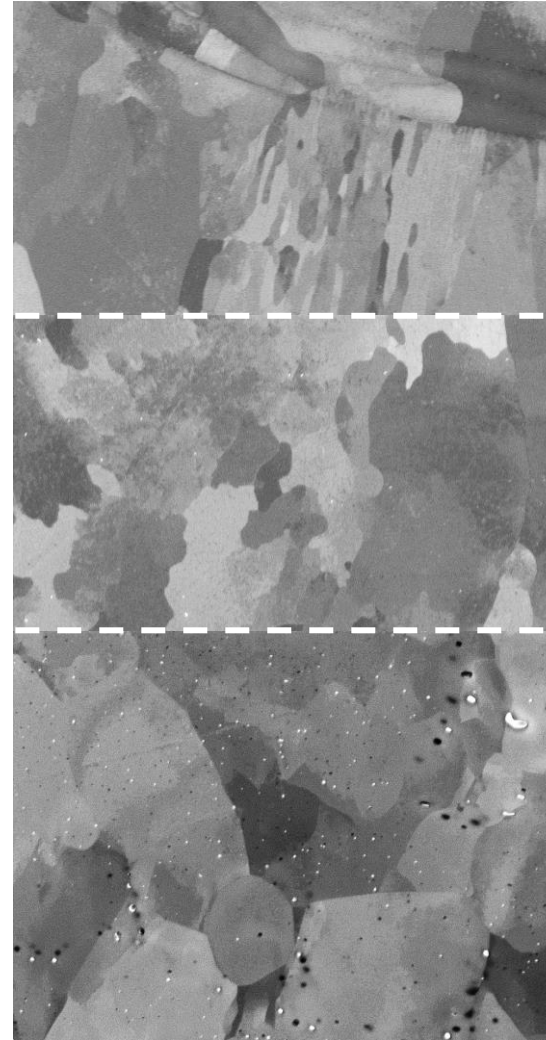
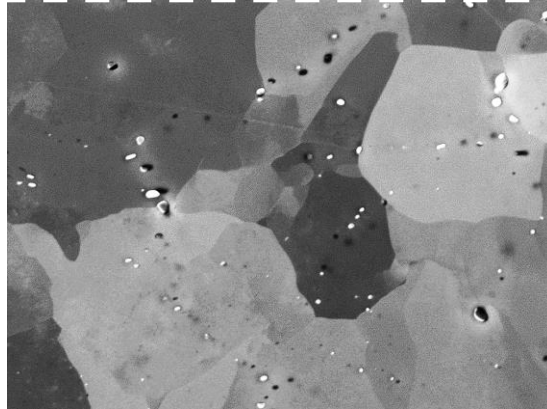
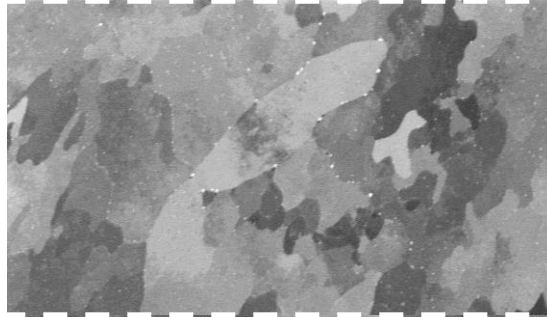
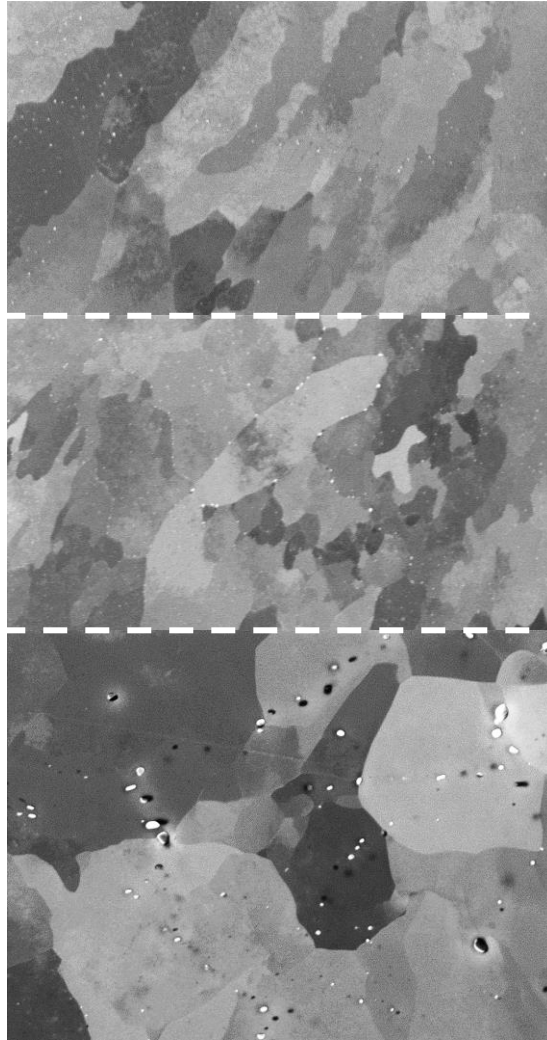
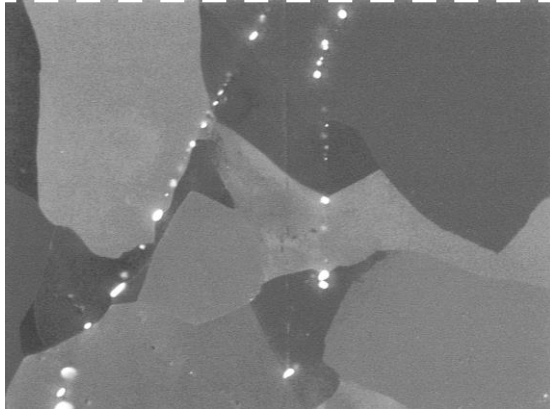
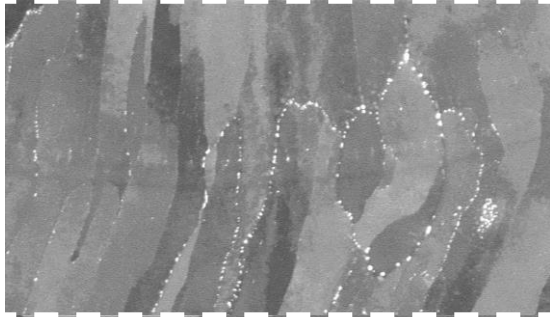
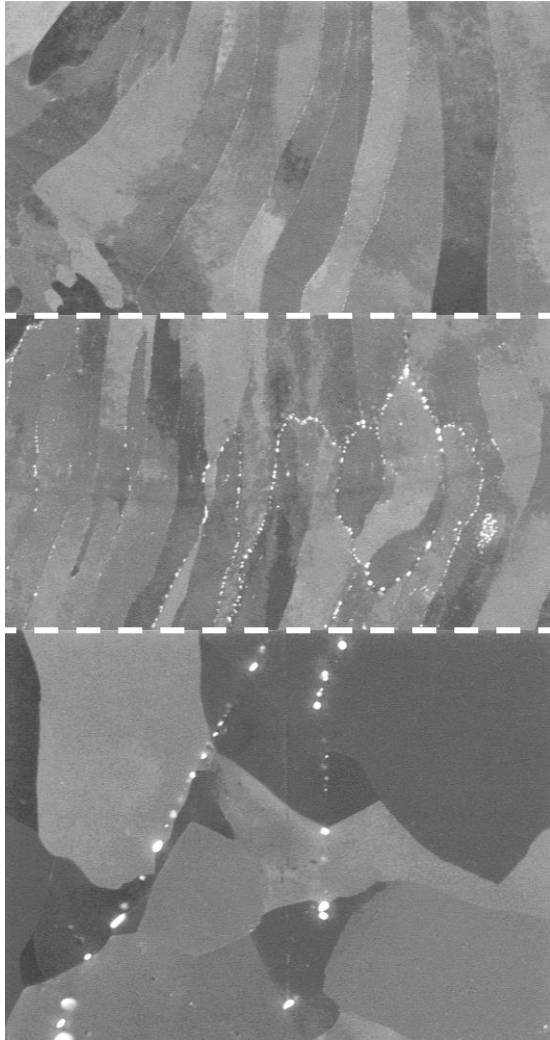
FS85

WC3009

~300 °C

~1000 °C

~1500 °C



Microstructural Characterization: *L-PBF* > *SR* > *Tested*

High Mag.

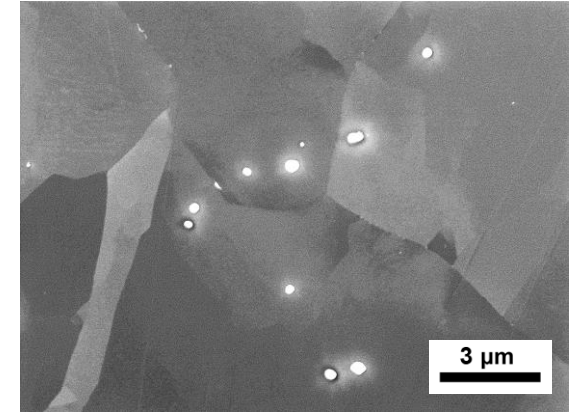
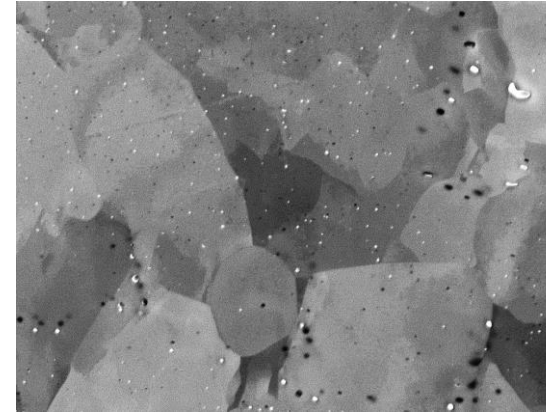
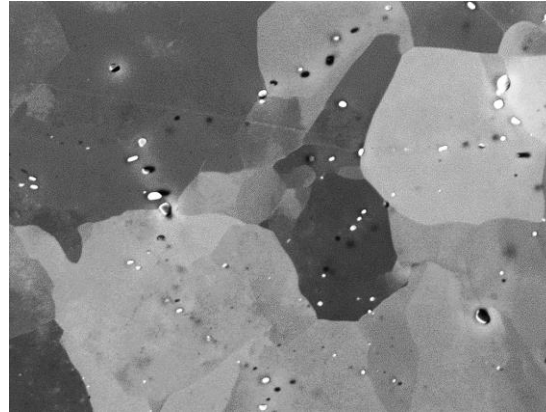
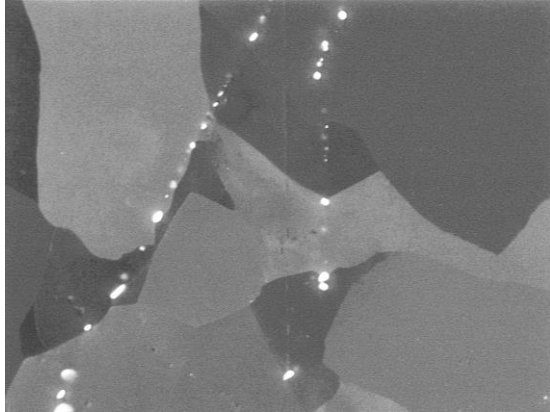
C103

Cb752

FS85

WC3009

~1500 °C



- Larger Grains
- Coarse HfO₂ not along GBs

- Smaller Grains
- Coarse ZrO₂ occasionally along GBs

- Smaller Grains
- Coarse ZrO₂ along GBs, fine ZrO₂ intragranular

- Larger Grains
- Coarse HfO₂ not along GBs

5.34 at% **Hf**
 0.24 at% O
 ~0.12 mol% HfO₂
 5.22 at% excess Hf

3.01 at% **Zr**
 0.24 at% O
 ~0.12 mol% ZrO₂
 2.89 at% excess Zr

0.20 at% **Zr**
 0.35 at% O
 ~0.17 mol% ZrO₂
 0.03 at% excess Zr

18.91 at% **Hf**
 0.14 at% O
 ~0.07 mol% HfO₂
 18.84 at% excess Hf

*Near 1:2 Zr:O ratio
 (ideal stoichiometry)

Conclusions

1. The L-PBF process successfully consolidated all examined historical niobium alloys with all relative densities $\geq 99.89\%$ (determined by optical microscopy and image analysis).
2. Elevated temperature uniaxial tensile testing determined the ranking of each L-PBF consolidated historical niobium alloy in terms of yield and ultimate tensile strength at temperatures greater than 1300 °C: **FS85** $\approx$ **Cb752** > **WC3009** > **C103**.
3. 1300 °C pseudo-creep testing performed at a constant load of 50 MPa determined the ranking of each L-PBF consolidated historical niobium alloy in terms of steady-state creep rate: **FS85** > **Cb752** > **WC3009** > **C103**.
4. Examination of the 1500 °C tensile specimen microstructure of each historical niobium alloy showed:
 - The high Hf-containing alloys (**C103** and **WC3009**) exhibited significant HfO_2 precipitate coarsening, and the precipitates were ineffective at pinning grain boundaries.
 - The Zr-containing alloys (**Cb752** and **FS85**) exhibited ZrO_2 precipitate coarsening; however, the precipitates formed a finer dispersion (particularly when the Zr:O atomic ratio was near stoichiometric (1:2)).



References

1. J. Hebda, "Niobium Alloys and High Temperature Applications," Wah Chang - An Allegheny Technologies Company, Albany, 2001.
2. W.E. Black et al., "Evaluation of Coated Columbium Alloy Heat Shields for Space Shuttle Thermal Protection System Application," *NASA CR-112119*, 1972.

