FATIGUE CRACK TIP DEFORMATION
AND FATIGUE CRACK PROPAGATION

by
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for
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CONTRACT REPORT

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FOREWORD

The work described herein was done at the George Sachs Fracture Research Laboratory, Syracuse University, under NASA Grant NGR-33-022-105 with Mr. Marvin Hirschberg, Materials and Structure Division, NASA-Lewis Research Center, as technical officer. The sustained support and interest of Mr. S. S. Manson and Mr. Marvin Hirschberg are appreciated.
The effects of stress ratio, prestress cycling and plate thickness on the fatigue crack propagation rate are studied on 2024-T351 aluminum alloy. Fatigue crack propagation rate increases with the plate thickness and the stress ratio. Prestress cycling below the static yield strength has no noticeable effect on the fatigue crack propagation rate. However, prestress cycling above the static yield strength causes the material to strain harden and increases the fatigue crack propagation rate.

Crack tip deformation is used to study the fatigue crack propagation. The crack tip strains and the crack opening displacements were measured from moire fringe patterns. The moire fringe patterns were obtained by a double exposure technique, using a very high density master grille (13,400 lines per inch).

The cyclic strain range, $\Delta \varepsilon$, was found to be proportional to $r^{-1/2}$. At intermediate $\Delta K$ values, i.e., region II, the measured $\Delta \varepsilon$ values are very close to those obtained by theoretical elastic calculations. At the high $\Delta K$ values, region III, due to the crack tip necking, the measured $\Delta \varepsilon$ is larger than that obtained by elastic calculation: the higher the $\Delta K$ value, the larger the difference between the measured and the calculated values.

Plastic zone size, $r_p$, was found to be proportional to $\Delta K^2$ in region II and proportional to $\Delta K^4$ in region III. If $da/dN$ is proportional to $\Delta K^2$ in region II and proportional to $\Delta K^4$ in region III, then $da/dN$ is proportional to $r_p$, i.e., $da/dN = r_p/N$, where $N$, a constant, is the number of cycles to propagate a crack through $r_p$. 
The measured crack opening displacement range, $\Delta \delta$, was found to be proportional to $r^{1/2}$. In the range of measurements, $\Delta \delta$ is proportional to $\Delta K$ in region II, and proportional to $\Delta K^2$ in region III. The data of $\Delta \delta$ were compared with those predicted by the Dugdale model and the elastic calculations. The differences in $\Delta \delta$, obtained by these two calculations, are small in the range of the measurements. However, the data seem to agree better with the elastic calculations because there was no tendency for the plot of $\Delta \delta$ versus $r$ to level off as $r$ approaches the crack tip, as predicted by the Dugdale model.
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I. INTRODUCTION

A. Fatigue Crack Propagation

Fatigue crack propagation has been studied quite extensively both from the crack length measurements on specimen surfaces and from striation spacings on fracture surfaces. The first fatigue crack propagation law which drew wide attention was the one derived by Head in 1953\(^1\). After substantial calculations and deductions, the law was given\(^2\) as

\[
\frac{da}{dN} = \frac{C_1 (\Delta \sigma \sqrt{a})^3}{(\sigma_y - \Delta \sigma) r_p^{1/2}}
\]

where \( C_1 \) is a constant, which depends on the strain-hardening modulus, Young's modulus, the yield stress \( \sigma_y \) and fracture stress of a material; \( \Delta \sigma \) is the applied stress range; \( a \) is the half crack length and \( r_p \) is the plastic zone size ahead of the crack tip, which was assumed to be constant during crack propagation. However, in 1958, Frost and Dugdale\(^3\) noticed that the plastic zone size increases with crack length.

Based on dimensional analysis, both Frost and Dugdale\(^3\) and Liu\(^4\) concluded that crack propagation rate should be directly proportional to the crack length, if crack length was the only pertinent length parameter. Their equation was given as

\[
\frac{da}{dN} = f(\Delta \sigma) a
\]
Frost and Dugdale \((3)\) observed empirically that \(f(\Delta \sigma) = C_2(\Delta \sigma)^3\), therefore their equation becomes

\[
\frac{da}{dN} = C_2(\Delta \sigma)^3 a \tag{3}
\]

On the other hand, Liu \((5)\) used the concept of total hysteresis energy absorption to failure and emphasized the damage caused by cumulative cyclic strain at a crack tip \((6)\). He found that \(f(\Delta \sigma) = C_3(\Delta \sigma)^2\), which led to

\[
\frac{da}{dN} = C_3(\Delta \sigma)^2 a \tag{4}
\]

The quantity \((\Delta \sigma)^2 a\) is the square of the stress intensity factor range, \(\Delta K^2\), for a centrally cracked infinite plate.

Paris \((7)\) noted that the stress intensity factor at a crack tip should be the factor controlling the rate of crack extension. Crack tip stress and strain will be reviewed later in this section. Later on, Paris, Gomez and Anderson \((8)\) proposed that crack propagation rate should be determined by the maximum stress intensity factor, \(K_{\text{max}}\). Subsequently, Paris and Erdogan \((2)\) used the stress intensity factor range, rather than maximum stress intensity factor, to correlate data on fatigue crack propagation rate.
Since then, the concept of crack tip stress intensity factor has been used to analyze the experimental data and a large amount of data have been collected on the correlation between fatigue crack propagation rate and stress intensity factor range. Based on experimental data, Paris and Erdogan\(^{(2)}\) proposed that fatigue crack propagation rate should be proportional to the fourth power of $\Delta K$. Their equation was therefore given as

$$\frac{da}{dN} = C_h \Delta K^4$$  \hspace{1cm} (5)

Ever since, the fourth power relation has been observed by a number of investigators. However, other values of the exponents, from 2 to 7, have also been observed.

McClintock\(^{(9)}\) used Manson-Coffin's low cycle fatigue equation and derived the fourth power relation by a model including a material structural size $\rho$. His equation is

$$\frac{da}{dN} = \frac{7.5}{\rho} \left( \frac{\varepsilon_{eq}}{\varepsilon_P} \right)^2 r_p^2$$ \hspace{1cm} (6a)

By replacing $r_p = (1/2\pi)(\Delta K/\sigma_y)^2$ and $\varepsilon_{eq} = \sigma_y/E$ one obtains

$$\frac{da}{dN} = C_5 \frac{\Delta K^4}{\varepsilon_P^2 (E\sigma_y)^2 \rho}$$ \hspace{1cm} (6b)
where \( \varepsilon_F \) is the true fracture strain for monotonic fracture; \( \varepsilon_Y \) is the yield strain and \( E \) is Young's modulus.

Rice\(^{10,11}\) also derived the fourth power relation from a rigid plastic strip model, which assumed that plastic deformation is limited to a strip of material ahead of a crack tip. His equation is

\[
\frac{da}{dN} = \frac{5\pi(1-\nu^2)\varepsilon_Y \sigma_Y}{96U^{*}} \left( \frac{\Delta K}{\sigma_Y} \right)^4
\]

(7)

where \( \nu \) is the Poisson's ratio and \( U^{*} \) is the postulated critical energy to create a new surface. A realistic physical explanation of Rice's rigid plastic strip model was the model of localized necking at a crack tip as suggested by Liu\(^6\).

Krafft\(^{12}\) assumed that fatigue crack propagation rate has a fourth power dependence on \( \Delta K \). On the basis of an empirical correlation between the plane strain fracture toughness and the plastic flow property of a material, he proposed an equation for fatigue crack propagation rate

\[
\frac{da}{dN} = \frac{16 \times 10^6 f(\gamma)}{7E^3 \frac{K_{IC}}{n}} K_{max}^4 = C K_{max}^4
\]

(8a)

where \( f(\gamma) = (1+\gamma)^4 \left[ 1-(1-\gamma)^2 \right] \), \( \gamma = 1-(K_{min}/K_{max}) \) and \( n = \) strain hardening exponent. Later on the equation was modified\(^{13}\)

\[
\frac{da}{dN} = \left[ \frac{10^8 n}{E} \right] \gamma^2 \left( \frac{K_{max}}{K_{IC}} \right)^4 d_T
\]

(8b)
where \( d_T = (KIC/En)^2/2\pi \) which is the process zone size.

Frost and Dixon\(^{(14)}\) assumed that crack propagation rate is equal to the increase in crack length of the ellipse perimeter between the loading and the unloading conditions. They gave the equation as

\[
\frac{da}{dN} = \frac{(\Delta \sigma)^2 a}{E^2} (\ln \frac{\alpha}{\Delta \sigma} - 1) \tag{9}
\]

Bibly, Cottrell and Swinden\(^{(15)}\) used a continuous dislocation model to calculate the crack tip plastic displacement, \( \delta_0 \). In their model, flow occurs only in a narrow strip ahead of a crack tip. The crack tip opening displacement was given as

\[
\delta_0 = \frac{\mu a}{\pi \alpha} \ln \left( \frac{\xi}{a} \right) \tag{10}
\]

where \( \mu \) is the shear modulus and \( \xi \) is the sum of the crack length and the length of the plastic zone. This dislocation model is equivalent to the Dugdale's strip necking model. The equations for crack opening displacement, \( \delta \) were given by Goodier and Field\(^{(17)}\), using the Dugdale's strip necking model:

\[
\delta = 2V(x,a) = \frac{2\mu \xi}{\pi E} \left[ \cos \theta \log \frac{\sin^2(\theta_2-\theta)}{\sin^2(\theta_2+\theta)} + \right.
\]

\[
\left. \cos \theta_2 \log \frac{(\sin \theta_2+\sin \theta)^2}{(\sin \theta_2-\sin \theta)^2} \right] \tag{11a}
\]
where \( \cos \theta_2 = \frac{a}{\ell} \)

\[
\cos \theta = \frac{x}{\ell} \text{ for } |x| < \ell, \quad -\pi < \theta < \pi
\]

\[
\theta_2 = \frac{\pi \Delta \sigma / 2\sigma_Y}{1 + \left( \frac{1}{r_p} \right)^{1/2}}
\]

and by Rice (11) as

\[
\delta = 2V(x,0) = \frac{(\kappa+1)\sigma_Y r_p}{\pi \mu} \left\{ \left(1 - \frac{x}{r_p}\right)^{1/2} - \frac{x}{2 r_p} \log \left[ \frac{1 + \left( \frac{x}{r_p} \right)^{1/2}}{1 - \left( \frac{x}{r_p} \right)^{1/2}} \right] \right\}
\]

where \( 0 < x < r_p \) and \( r_p = (\pi/\theta)(K/I\sigma_Y)^2 \) (11b)

the crack tip opening displacement, \( \delta_0 \) is given as

\[
\delta_0 = 2V(0,0) = \frac{(\kappa+1)\sigma_Y r_p}{\pi \mu} = \frac{1}{E\sigma_Y} K^2_I \quad \text{for plane stress (11c)}
\]

Weertman (18) applied the Bibly, Cottrell and Swinden's crack tip plastic displacement to the problem of fatigue crack propagation. He adopted a fracture criterion which stated that a crack will grow whenever the sum of all the cyclic plastic displacements at the crack tip exceed the critical value \( \delta^* \), and a growing crack will stop at a point where the total plastic displacement at the crack tip is less than \( \delta^* \).

The equation was given as

\[
\frac{da}{dN} = \frac{0.4 (\Delta \sigma \sqrt{a})^4}{(\sigma_Y \delta^*) \mu \sigma_Y^2}
\] (12)
McEvily and Johnston\(^{(19)}\) also proposed the following empirical equation, which is similar to Equation (12),

\[
\frac{da}{dN} = \frac{\Delta K^4}{\sigma_Y^2 \sigma_u (\sigma_Y^2 + \sigma_u)}
\]

\[(13)\]

where \(\sigma_Y\) is the 0.2 percent offset yield strength, \(\sigma_u\) is the ultimate tensile strength and \(\varepsilon_m\) is the engineering strain at maximum load.

Lardner\(^{(20)}\) also applied the theory of Bibby, Cottrell and Swinden's to derive a fatigue crack propagation law. He assumed that a crack can propagate during the unloading and the equation was given as

\[
\frac{da}{dN} = \left(1-\nu\right) \frac{\Delta K^2}{4\mu \sigma_Y^2}
\]

\[(14)\]

McClintock\(^{(21)}\) suggested that if the fatigue crack propagation rate is related to the crack opening displacement by a proportionality constant \(C\), one can write

\[
\frac{da}{dN} = C \Delta \delta = \frac{4C^2 (\Delta \sigma) a}{E \sigma_Y^2}
\]

\[(15)\]

Tomkins\(^{(22)}\) assumed that during the tensile half of the cycle, plastic flow occurred in two narrow shear bands of length \(D\), radiating at \(\pm 45^\circ\) from the crack plane at the crack tip. Because of the shear
strain concentrated in these two narrow bands, the Dugdale model of plastic cohesive forces can be applied to the fatigue problem. He further assumed that the rate of fatigue crack propagation is equal to the crack tip opening displacement, i.e.

\[
\frac{da}{dN} = \delta_o = D\Delta \varepsilon_p
\]

(16a)

where \( \Delta \varepsilon_p \) is the applied plastic strain range, which is related to the cyclic stress by a power relationship. Using the plastic zone size equation for \( D^{(23)} \), Tomkins derived a fatigue crack propagation equation for low stress region,

\[
\frac{da}{dN} = C_8(\Delta \sigma)^3 \sigma_m a
\]

(16b)

where \( \sigma_m \) is the applied mean stress.

Analyzing the effect of stress ratio, \( R \), to the fatigue crack propagation rate, Broek and Schijve\(^{(24)}\) proposed an exponential equation for the case of a centrally cracked aluminum plate. Their equation has the form of

\[
\frac{da}{dN} = C_9\left(K'_{\text{max}}\right)^3 \exp\left(-C_{10}R\right)
\]

(17a)

where \( K'_{\text{max}} = \sigma_{\text{max}} \sqrt{a\left[1+40\left(\frac{a}{W}\right)^2\right]^{1/3}} \), and \( W \) is the width of the specimen.
Equation (17a) can also be written in terms of the stress intensity factor range as

\[
\frac{da}{dN} = C_9 \left( \frac{\Delta K'}{1-R} \right)^3 \exp (-C_{10} R) \quad (17b).
\]

Forman, Kearney and Engle\(^{(25)}\), considering the instability of the crack growth when the stress intensity factor approaches the fracture toughness, \(K_c\), proposed an equation for fatigue crack propagation rate as

\[
\frac{da}{dN} = \frac{C_{11} (\Delta K)^n}{(1-R)K_c - \Delta K} \quad (18).
\]

Hudson and Scardina\(^{(26)}\) also studied the effect of stress ratio on fatigue crack propagation in 7075-T6 aluminum alloy sheet. They found that all the data from tests at negative \(R\) values fell into a relatively narrow scatter band with the data from the \(R = 0\) tests, which indicates that the compression portion of the loading cycle did not significantly affect the crack propagation rate. On the other hand, for a positive \(R\), they found the higher the stress ratio at a given value of \(\Delta K\), the higher the fatigue-crack-propagation rate. This spread in rates was small at the lower \(\Delta K\) value, but became progressively larger as the \(\Delta K\) value was increased. They also concluded that the equation of Forman et al.\(^{(25)}\) gave the best fit to their data.

Both striation spacings and specimen surface crack propagation rates have been studied on the same specimens by Pelloux\(^{(27)}\), Hertzberg
and Paris\textsuperscript{(28)} and Kershaw and Liu\textsuperscript{(29)}. Pelloux\textsuperscript{(27)} and Kershaw and Liu\textsuperscript{(29)} found that the fatigue crack propagation rates observed on the specimen surface was faster than the fatigue crack propagation rates measured from the striation spacings. Pelloux has attributed this difference to the extensive cracking of brittle second phase particles ahead of the main crack. The effect of these brittle second phase particles has been analyzed by McClintock\textsuperscript{(9,20)}. On the other hand, Hertzberg and Paris\textsuperscript{(28)} have not found a significant difference between propagation rates based upon striation spacing and those based upon surface measurements.

Applying notch fracture analysis to a sharp crack and considering that cyclic process of a fatigue crack as a continuing re-initiation of fracture, Weiss\textsuperscript{(30)} derived a crack propagation equation for strain controlled fatigue as

$$\frac{da}{dN} = a\left(\frac{\varepsilon_N}{\varepsilon_{FF}}\right)^{n+1} - \frac{n}{2}$$

(19)

where $\varepsilon_N$ is nominal net section strain; $\varepsilon_{FF}$ is equal to $\sigma_{FF}/E$. $\sigma_{FF}$ is equal to or at least related to the endurance limit; $n$ is Neuber's equivalent particle size for sharp notches and $n$ is strain hardening exponent.

This equation was obtained, based on the assumption that $\frac{da}{dN}$ was equal to the distance beyond the crack tip, over which the stress exceeded $\sigma_{FF}$. 
Typical fatigue crack propagation data which correlate $\frac{da}{dN}$ with $\Delta K$ can be divided into three regions as shown by Liu and Iino\(^{(31)}\). In region I, the low $\Delta K$ region, the slope of the curve is around $\frac{1}{4}$ and the crack propagation rate is usually less than a few micrometers per cycle. In region III, the high $\Delta K$ region, the slope of the curve is greater than $\frac{1}{4}$. The fracture mode in this region has been changed from normal flat mode to shear mode. Therefore, the fatigue crack propagation rate is strongly affected by specimen thickness and necking process. Region III is a region close to the final failure of the specimen. As a consequence, it is less important than other regions.

In region II, the intermediate $\Delta K$ region, the slope of the curve is close to $2$. Crack propagation in this region is under the normal fracture mode and is not complicated by a shear or necking process. Most of the investigation in this report concerns this region. Only part of the investigations are extended to region III or region I.

B. Stresses and Strains Near a Crack Tip

In the paper of "Analysis of Stresses and Strains Near the End of a Crack Traversing a Plate," Irwin\(^{(32)}\) derived a group of equations which were identical to those of Williams\(^{(33)}\), except that he emphasized only the first term of an infinite series. The equations were given as
\[ \sigma_y = \left( \frac{E \mu^2}{2\pi r} \right)^{1/2} \cos \frac{\theta}{2} \left( 1 + \sin \frac{\theta}{2} \sin \frac{3\theta}{2} \right) \]

\[ \sigma_x = \left( \frac{E \mu^2}{2\pi r} \right)^{1/2} \cos \frac{\theta}{2} \left( 1 - \sin \frac{\theta}{2} \sin \frac{3\theta}{2} \right) \tag{20a} \]

\[ \tau_{xy} = \left( \frac{E \mu^2}{2\pi r} \right)^{1/2} \cos \frac{\theta}{2} \sin \frac{\theta}{2} \cos \frac{3\theta}{2} \]

\[ \delta = \frac{8}{E} \left( \frac{E \mu^2}{2\pi r} \right)^{1/2} \]

for plane stress, \( \tag{20b} \)

where \( r \) is the distance from crack tip, \( \theta \) is the angle from x axis, \( \delta \) is the crack opening displacement, and \( \mu \) is the strain energy release rate. The coordinates of the crack tip are shown in Figure (1a). Equation (20a) shows that the stresses ahead of a crack tip are inversely proportional to the square root of the distance from the crack tip, while the crack opening displacement is proportional to the square root of \( r \).

By defining \( (E \mu^2)^{1/2} = K \) as the stress intensity factor, Equation (20) becomes

\[ \sigma_y = \frac{K}{\sqrt{2\pi r}} \cos \frac{\theta}{2} \left( 1 + \sin \frac{\theta}{2} \sin \frac{3\theta}{2} \right) \]

\[ \sigma_x = \frac{K}{\sqrt{2\pi r}} \cos \frac{\theta}{2} \left( 1 - \sin \frac{\theta}{2} \sin \frac{3\theta}{2} \right) \tag{21a} \]

\[ \tau_{xy} = \frac{K}{\sqrt{2\pi r}} \cos \frac{\theta}{2} \sin \frac{\theta}{2} \cos \frac{3\theta}{2} \]
Thus the stress distribution at the vicinity of a crack tip are completely described by the new term $K_i$, the stress intensity factor. The stress intensity factor, which describes the stress fields near the crack tip, is a function of applied stress, crack length and overall specimen geometry. The general formula for the stress intensity factor range can be written as

$$
\Delta K = \Delta \sigma (\pi a)^{1/2} f(a)
$$

where $\Delta \sigma$ is the applied stress range, $a$ is the half crack length and $f(a)$ is the dimensionless function of geometry, which is a stress field correction factor for the finite geometry. In the literature, one can easily find various formulas for $K$ for various specimen geometry and loading conditions.

A number of analytical solutions to the problems of crack tip plasticity have been given in the past years. Some of these results will be compared and discussed with the moire strain measurements in a subsequent section of this investigation.

Hult and McClintock \(^{(35)}\) introduced the mode III plasticity analysis. They calculated the strain distribution around a crack tip for the anti-plane shear loading. For non-strain hardening materials, they found that plastic strain was inversely proportional to the distance, $r$, from a crack tip. The relation was given as

$$
\delta = \frac{8K_i}{E} \left( \frac{r}{2\pi} \right)^{1/2}
$$

(21b)
where \( \tau_y \) is the yield stress in shear, \( \mu \) is the shear modulus of elasticity and \( R_a \) is the size of plastic zone.

Equation (23) showed that strain singularity within plastic region was inversely proportional to the distance from crack tip. Throughout the whole plastic region, \( \tau_{yz} \) was taken to be equal to the shear yield stress.

The material of the above analysis is perfectly plastic. The strain distribution ahead of a crack tip was proportional to \( r^{-1} \) instead of \( r^{-1/2} \) which was derived from the linear elastic model.

Most of the metallic materials behaved neither perfectly plastic nor perfectly elastic. They exhibit a certain degree of strain hardening. If the stress-strain relationship in the plastic region follows the well known power law as

\[
\sigma = C_{12} \varepsilon^n
\]

and applying Neuber's notch stress analysis to analyze the plastic flow, Weiss (36) obtained stress and strain distributions in front of a sharp notch

\[
\frac{\varepsilon_y}{\varepsilon_N} = C_{13} (x)^{-1/n+1}
\]

\[
\frac{\sigma_y}{\sigma_N} = C_{14} (x)^{-n/n+1}
\]

where \( x \) is the distance from notch root, \( \sigma_y \) and \( \varepsilon_y \) are the stress and strain at \( x \) respectively, and \( \sigma_N \) and \( \varepsilon_N \) are the net section stress and net section strain respectively.
Hutchinson (37) studied the tensile crack in a hardening material under both plane strain and plane stress conditions. He found that the magnitude of the tensile stress ahead of the crack in plane strain is larger than the magnitude of that in plane stress. Using a stress-strain relation as \( \epsilon = \sigma + C_{15} \sigma^2 \), suggested by Ramberg and Osgood to describe the plastic flow around a crack tip, Hutchinson also obtained two equations for stress and strain similar to those of Weiss.

Almost at the same time, Rice and Rosengren (38) investigated the plane strain deformation near a crack tip in a power law hardening material. They also found the same crack tip strain singularity as that of Weiss and Hutchinson. The product of stress and strain exhibits a singularity of \( 1/r \) for all materials.

Swedlow (39) and Swedlow, Williams and Yang (40) used the finite element method to analyze the stress-strain distributions around a crack tip in a plate under plane stress condition. They observed that, in the early stage of loading, both stress and strain are proportional to the same singularity of \( r^{-1/2} \) as predicted by the linear elastic analysis. However, the stress and strain singularities changed their character and deviated from the value of \( r^{-1/2} \) as loading progressed. At a later stage of loading, the strain singularity gradually levels off toward some stable value of \( r^{-1/n+1} \) as predicted by Weiss (36), Hutchinson (37) and Rice and Rosengren (38) while the stress singularity becomes less severe and levels off toward some stable value which is slightly above the value of \( r^{-n/n+1} \).
On examining the stress and strain singularity results presented above, one finds that within a plastic zone, strain is a more meaningful and easily measurable quantity; and the strain singularity at a crack tip depends strongly on the stress-strain relation. An elastic model gives the strain singularity of $r^{-1/2}$, a perfectly plastic model gives the strain singularity of $r^{-1}$, while a strain hardening model gives the strain singularity of $r^{-1/(n+1)}$. To verify these analytical models, strain measurements ahead of but close to a crack tip are needed. Moire strain measurements with very high density grille seem capable to provide such information.

In this investigation, the effects of stress ratio, plate thickness and prestress on the fatigue crack propagation rate in 2024-T351 aluminum alloys were studied. Here the effect of stress ratio on the fatigue crack propagation rate, as well as the low cycle fatigue data and the cyclic stress-strain curve for 7075-T6 aluminum alloy are also reported. The data on 7075-T6 aluminum alloy were analyzed in terms of the cyclic fracture ductility. On the other hand, based on the results of prestress effect on the fatigue crack propagation rate for 2024-T351 aluminum alloy, a simple equation for fatigue crack propagation rate expressed in terms of plastic zone size is formulated. The validity of this equation was substantiated by the plastic zone size obtained from the moire strain measurements at the tip of a fatigue crack.

Moire strain measurements were made on the plates of 2024-T351 aluminum alloy. Moire fringes were obtained by double exposure
technique. Strain distribution ahead of a crack tip as well as crack opening displacements behind a crack tip were measured from these fringe patterns. Plastic zone size, $r_p$, and crack opening displacements, $\delta$, were determined for each stress intensity factor $K$. The data of $r_p$ and $\delta$ were correlated with fatigue crack propagation rate.
II. FATIGUE CRACK PROPAGATION

Fatigue crack propagation has been studied on 2024-T351 and 7075-T6 aluminum alloys. Centrally cracked plate specimens either 4 inches or 6 inches wide were tested. A jeweler's saw cut of 0.25 to 0.50 inches long served as a crack starter. The specimen geometry is shown in Figure (1b). For 2024-T351 aluminum alloy, the effects of stress ratio, plate thickness and prestress-cycling on fatigue crack propagation rate were investigated. The results will be compared and correlated with the crack tip plastic zone size, $r_p$, which was measured from the moire fringe patterns. For 7075-T6 aluminum alloy, the effect of stress ratio on the rate of fatigue crack propagation, as well as the low cycle fatigue data and cyclic stress-strain curve are reported. The data will be analyzed in terms of the cyclic fracture ductility.

It has been shown\(^{(31)}\) that fatigue crack propagation rate data can be correlated by three line segments. In the low $\Delta K$ region, i.e. region I, the slope of the line is approximately $4$ to $5$; in the intermediate $\Delta K$ region, i.e. region II, the slope is close to $2$; and in the high $\Delta K$ region, i.e. region III, the slope is equal to or larger than $4$. Figure (2a) shows the plot of $\frac{da}{dN}$ versus $\Delta K$ for 7075-T6 aluminum alloy in region I and region II. The data were collected from five specimens. The applied stress ranges and specimen dimensions are indicated in the figure. All the specimens were tested under an applied stress ratio, $R = 1/10$. The slopes are $5$ and $2.1$ in region I and region II, respectively. The transition from region I to region II takes place at $\Delta K = 6$ksi $\sqrt{in}$, where $\frac{da}{dN}$ is equal to $4$ micro-inches per cycle.

A. The Effect of Stress Ratio on Fatigue Crack Propagation

A study of stress ratio effect on fatigue crack propagation was conducted on both 2024-T351 and 7075-T6 aluminum alloys. Two stress ratios,
R = 1/10 and 1/3, were tested on 2024-T351 aluminum alloy; and three stress ratios, 1/10, 1/3 and 1/2, were tested on 7075-T6 aluminum alloy. Two specimen thicknesses, 0.050 inches and 0.25 inches of 2024-T351 aluminum alloy were used, while only one specimen thickness, 0.125 inches, was used for 7075-T6 aluminum alloy. In Figures (2b), (2c) and (2d), fatigue crack propagation rate, da/dN versus stress intensity factor range, ΔK, were plotted. The applied stress ratios and applied stress ranges were all indicated in the figures. Figures (2b) and (2c) are the plots of 2024-T351 aluminum alloy for 0.050 inches thick and 0.25 inches thick specimens respectively. Figure (2d) is the plot of 7075-T6 aluminum alloy. The data shown in the figure fall in region II and region III. They indicate the following:

(a) The slopes of the lines, which correlated da/dN with ΔK, increase with stress ratios. In region II, the values of the slopes are 2.0 and 2.7 for R = 1/10 and 1/3 respectively for both 2024-T351 and 7075-T6 aluminum alloys; and 3.4 for R = 1/2 for 7075-T6 aluminum alloy. In region III, the slopes of the 2024-T351 aluminum alloy data are 4 and 5.4 for R = 1/10 and 1/3 respectively; they are twice the values of the slopes in region II.

(b) The higher the stress ratio at a given value of ΔK, the higher the rate of fatigue crack propagation. This spread in rates is smaller at lower ΔK values but it becomes progressively larger as the ΔK value increases. For instance, at ΔK = 7.3 ksi√in in Figure (2c), da/dN is 5.4 micro-inches per cycle for all three values of R. However, at ΔK = 15 ksi√in, the da/dN values are 25, 40 and 72 micro-inches per cycle for R = 1/10, 1/3 and 1/2, respectively. This phenomenon was also observed by Hudson and
Scardina\textsuperscript{(26)}. The increase in $da/dN$ as $R$ increases is caused by the increase in $K_{\text{max}}$. At the same value of $\Delta K$, $K_{\text{max}}$ at $R = 1/3$ is 1.35 times the value at $R = 1/10$, and $K_{\text{max}}$ at $R = 1/2$ is 1.8 times the value at $R = 1/10$.

(c) All the lines for region II converge at $\Delta K = 7.3$ ksi $\sqrt{\text{in}}$. Therefore, one can write an empirical equation that correlates the data as

$$\frac{da}{dN} = C_{16} \left(\frac{\Delta K}{7.300}\right)^n$$

where $n$ are the values of slopes, which depend on the stress ratio $R$, i.e. 2.0, 2.7 and 3.4 for $R = 1/10$, 1/3 and 1/2 respectively. $C_{16}$ is the value of $da/dN$ at $\Delta K = 7.3$ ksi $\sqrt{\text{in}}$. The values of $C_{16}$ are 2.8 and 4.6 micro-inches per cycle for 0.050 inches and 0.25 inches thick 2024-T351 aluminum alloy, respectively; and 5.4 micro-inches per cycle for the 0.125" thick 7075-T6 aluminum alloy.

(d) Comparing the data of 2024-T351 aluminum alloy of Figures (2b) and (2c) for the same stress ratio, one found that the thicker the specimen is, the faster a crack will propagate. This was also observed by Broek and Schijve\textsuperscript{(24)} and Barsom\textsuperscript{(11)}. At the same $\Delta K$ value and the same $R$-value, the $da/dN$ value of the thicker 2024-T351 aluminum specimen is 1.61 times that of the thinner specimen. The value of 1.61 is the ratio of the $C_{16}$ values for thick and thin specimens in the Equation (26). This is explained by the fact that the stresses and strains near a crack tip in the thick specimen are closer to the plane strain condition than in the thin one. As pointed out by Hutchinson\textsuperscript{(37)}, the magnitude of the tensile stress at a crack tip is higher in the plane strain case than in the plane stress case. This high tensile
stress in a thick specimen enhances crack propagation.

(e) The transition point from region II to region III takes place at \(\Delta K = 21 \text{ ksi} \sqrt{\text{in}}\) for 2024-T351 aluminum alloy. The thickness does not seem to have any significant effect on the transition point. Clark and Wessel found that the transition point for their WOL specimen of 5456-H321 aluminum alloy was \(\Delta K = 25 \text{ ksi} \sqrt{\text{in}}\).

B. Effect of Prestress-Cycling on Fatigue Crack Propagation

The effects of prestress-cycling on fatigue crack propagation have been studied on 2024-T351 aluminum alloy. Each specimen was cyclically prestressed, at certain stress range for a given number of cycles. After the prestress cycling, a slot was introduced into the specimen and the effect on subsequent fatigue crack propagation rate was investigated. The specimen geometry is shown in Figure (3a). Due to the load limit of our fatigue machine (20,000 lbs), a reduced test section was necessary in order to apply a meaningful prestress range, \(\Delta \sigma\). Two thicknesses, 0.10 inches and 0.08 inches, were tested. Four prestress ranges were used in this investigation. They were 22,000±18,000 psi, 27,500±22,500 psi, 33,000±27,000 psi and 35,750±29,250 psi. The corresponding fatigue lives at these four prestress ranges were found to be 220,000, 46,000, 12,000 and 5,500 cycles respectively. The first two prestress ranges were below the static yield strength of the materials, while the last two exceeded it. The maximum stress levels and the stress ranges have to be reset frequently, during the process of setting-up the prestress levels, in order to reach the desired levels. This was especially true for the prestress tests above the static yield strength. After the initial setting, the prestress levels were checked and adjusted if necessary. The number of prestress cycles at the prestress level were also shown for each test in the figure. At the two lower prestress levels, both below the static yield strength of the material, very slight plastic
deformation occurred and the material was not hardened. On the other hand, at the two high prestress levels, the static yield strength was exceeded, plastic deformations were accumulated and the material was cyclically strain hardened. Figure (3b) shows the recording of the load-strain curve for prestress at 35,750 ± 29,250 psi. The specimen started to yield and strain hardened at the first cycle, when the stress reached the static yield strength. After the first cycle, the specimen showed no noticeable cyclic deformation up to ten cycles. The specimen was then put into the fatigue machine, and prestressed up to 3,300 cycles. After prestress-cycling, a central slot of 2a = 0.20 inches was machined into each specimen, and all the specimens were set at the applied stress range of 11,000 ± 9,000 psi. The stress ratios of the prestress cycling and for the fatigue crack propagation tests were all kept at 1/10. Specimens 1, 2 and 3 were 0.100 inches thick and specimens 4, 5, 6 and 7 were 0.080 inches thick.

Fatigue crack propagation in these specimens were measured. Since the specimen thickness affects fatigue crack propagation, the data of these two groups of specimens were plotted in separate figures. In Figures (4a) and (4b), the half crack length, a, was plotted versus the difference of the number of cycles of load and the number of cycles of load at a = 0.23 inches, N. In this way, the change of crack propagation rates caused by cyclic prestressing can be easily observed.

In Figure (4a) specimen No. 1 was not prestressed. It served as a reference to detect the effect of prestress on fatigue crack propagation rate. Specimens No. 2 and 3 are both prestressed at 22,000 ± 18,000 psi
at 160,000 and 200,000 cycles respectively. This prestress is below the static yield strength. The data of specimens No. 1 and 3 coincide with each other, while the crack in specimen No. 2 propagated more slowly than in specimen No. 1. The irregularity could be caused by the scatter of the material property. The average fatigue crack propagation rate of specimen No. 2 was 10 percent lower than that of specimen No. 1. The deviation is well within the scatter band of fatigue crack propagation data. The effect of a low prestress level, if any, must be small.

Figure (4c) shows the data of da/dN versus ΔK of these three specimens. All the data points fall pretty well within a band.

Figure (4b) shows the data of specimens 4, 5, 6 and 7. Specimen 4 was not prestressed. Specimen 5, 6 and 7 were prestressed at 27,500 ± 22,500 psi, 33,000 ± 27,000 psi and 35,750 ± 29,250 psi, respectively. The number of prestress cycles are listed in the figure. The data indicate that the prestress at 27,500 ± 22,500 psi has a negligible effect on fatigue crack propagation rate. The prestress at 33,000 ± 27,000 psi increases the average fatigue crack propagation rate by one third while prestress at 35,750 ± 29,250 psi increases the average fatigue crack propagation rate by one half. In Figure (4d) the data of da/dN versus ΔK were plotted. In region II the data of specimen 4 and 5 shows a slope of 2, but that of specimen 6 has a slope of 2.6. Specimen 7 shows a much steeper slope. The data clearly show that the two lower prestress levels, which are below the static yield strength have negligible effects on the fatigue crack propagation rate. However, prestress above static
yield strength increases fatigue crack propagation rate considerably. These observations lead one to conclude that cyclic deformation at very low levels makes a negligible contribution to fatigue damage of a material element. The high strain cycles, when a material element moves close to a crack tip, cause much severer damage. Therefore, the damage caused strain cycles, when a material element was outside of $r_p$, can be neglected.

Based on the blunting model for fatigue crack propagation and with the suggestion that the crack opening displacement, $\delta$, given by the Dugdale model, is a measure of bluntness, the crack propagation rate was written as

$$\frac{da}{dN} = D_\gamma \delta = \frac{4C_\gamma \Delta K^2}{E\sigma_Y}$$ (15)

Accordingly, if prestress-cycles increase $\sigma_Y$, as in the case of 2024-T351 aluminum alloy, the $da/dN$ should be decreased. But the experimental data shows the contrary. Both $da/dN$ and $\sigma_Y$ were increased by prestress cycling. This contradiction suggests that blunting mechanism alone cannot be attributed as the only cause of fatigue crack propagation.

C. Low Cycle Fatigue Data for 7075-T6 Aluminum Alloy

Low cycle fatigue data for 7430 fully annealed steel and 2024-T351 aluminum alloy were collected by Liu and Iino. Based on their data, the rate of fatigue crack propagation was given in terms of ductility, $M$, cyclic yield stress and cyclic yield strain,
where $C_{17}$ is a proportional constant equal to 1.15.

In this report, the work was extended to 7075-T6 aluminum alloy. Smooth specimens, with a diameter of 0.247 inches, were used. The gauge length was 0.3 inches. Figure (5a) shows the plot of total strain range, $\Delta \varepsilon_T$, versus the number of cycles to fracture, $N_T$. The slope was found to be -0.45. The slopes for the curves of 2024-T351 aluminum alloy and 1340 fully annealed steel are -0.49. By the extrapolation of the line in Figure (5a), the cyclic fracture ductility for 7075-T6 aluminum alloy was found to be 0.25. Figure (5b) shows the curve of total cyclic stress range, $\Delta \sigma_T$, versus total longitudinal cyclic strain range, $\Delta \varepsilon_T$ for both 7075-T6 and 2024-T351 aluminum alloys. The cyclic yield strength range, $\Delta \sigma_Y(c)$, and the cyclic yield strain range, $\Delta \varepsilon_Y(c)$, for 7075-T6 aluminum alloy were found to be 170,000 psi and 0.015 respectively.

By putting all the values of $C_{17}$, $\Delta \sigma_Y(c)$, $\Delta \varepsilon_Y(c)$, and $M$ into Equation (27), an empirical equation of fatigue crack propagation rate for 7075-T6 aluminum alloy was obtained.

$$\frac{da}{dN} = C_{17} \left( \frac{\Delta K}{\Delta \sigma_Y(c)} \right)^2 \left( \frac{\Delta \varepsilon_Y(c)}{M} \right)^2 \tag{27}$$

By comparing the values of $da/dN$, obtained from the substitution of $\Delta K$ values into Equation (28), with the experimental values as shown in Figure (2a), the correlation is surprisingly good in the intermediate
ΔK region, region II, where the slope of the line is 2. For instance, at \(\Delta K = 10 \text{ ksi} \sqrt{\text{in}}\), Equation (28) predicts that \(\frac{da}{dN}\) is equal to 14.3 micro-inches per cycle, while the average experimental value is 10.5 micro-inches per cycle. The predicted values fall into the upper band of the measured values.
III. CRACK TIP DEFORMATION

A. Experimental Procedures

Cyclic deformations, both strains ahead of a crack tip and crack opening displacements behind a crack tip were measured. The material used in this investigation was 2024-T351 aluminum alloy. Two stress ratios, $R = 1/10$ and $1/3$, were studied. Four inch wide centrally cracked plate specimens, 0.25 inches or 0.050 inches thick, were tested. The as received plate was 0.25 inches thick. The 0.05 inches thick specimens were thinned down on one side. All the specimens were made from the same plate. The geometry of specimens is shown in Figure (6).

The specimens were carefully polished, cleaned and dried. A coat of photo-resist was sprayed on these cleaned and dried surfaces. A high density master grille of 13,400 lines per inch was used. Moire patterns were obtained by the double exposure technique. Both total maximum strain, $\varepsilon_{\text{max}}$, and strain amplitude, $\Delta \varepsilon$, were measured.

As the crack propagated toward a point, the strain at the point increased because of the decrease in $r$. If plastic deformation took place, a cyclically induced creep strain, $\varepsilon_{cs}$ occurred. The total maximum strain $\varepsilon_{\text{max}}$ is the sum of the cyclically induced creep strain, $\varepsilon_{cs}$ and the strain range, $\Delta \varepsilon$. To measure the total maximum strain, the photo-resist coating was first exposed to a grille master at zero load. In order to avoid the residual strains caused by the earlier cycles, the crack was allowed to grow more than five times the plastic zone size estimated by the Irwin's formula, $r_p = (1/2\pi) (\Delta K/\sigma_y)^2$. 
Then the second exposure was made at the maximum load. The moire pattern
gave the sum of the accumulated strains, as the crack propagated into
the region where the measurements were made, as well as the strain caused
by the last loading cycle. After the measurements of the total maximum
strains were made, the moire pattern was erased. The specimen surface
was cleaned, dried and coated with photo-resist again. At the same crack
length and at same values of $\Delta K$ and $K_{\text{max}}$, the strain ranges were measured.
The first exposure was made at $\sigma_{\text{min}}$ and the second one at $\sigma_{\text{max}}$. Therefore
at each set of values of $K_{\text{max}}$ and $\Delta K$, the total maximum strains, $\varepsilon_{\text{max}}$
and the strain ranges, $\Delta \varepsilon$ were measured. All the measurements were made
along the crack line. Figure (7a) shows a set of fringe pictures to
measure $\varepsilon_{\text{max}}$ and $\Delta \varepsilon$.

Figure (7b) shows the definition of the strain quantities. $\varepsilon_{\text{min}}$ is
the difference between $\varepsilon_{\text{max}}$ and $\Delta \varepsilon$. It corresponds to the strains at
$K_{\text{min}}$ including the accumulated strains caused by the earlier cycles.
When the specimen was completely unloaded, i.e. $K = 0$, the residual
strain was the accumulated strain $\varepsilon_{\text{acc}}$, caused by all the previous stress
cycles. Therefore $\varepsilon_{\text{min}}$ is the sum of $\varepsilon_{\text{acc}}$ and the strain caused by $K_{\text{min}}$
of the last cycle alone. $\varepsilon_{\text{acc}}$ can also be viewed as the cyclically in-
duced creep strain, $\varepsilon_{\text{cs}}$.

The basic principle of moire strain analysis can be found else-
where (45,46). Knowing the pitch, $p$, of the master grille and the moire
fringe spacing, $d_y$, in the $y$-direction (see Figure (1) for reference
system), the average tensile strain, $\varepsilon_y$, over the fringe spacing is cal-
culated from
For small strain and high density grille, i.e. \( d_y \gg p \), Equation (29a) can be simplified as

\[
\varepsilon_y = \frac{p}{(d_y - p)}
\]

(29b)

Note that moire strain measurement is basically a displacement measurement method. It measures the average strain over a finite gauge length equal to the inter-fringe spacing. The resolution of the method is directly dependent on the grille density. The high grille density used can easily resolve a strain of 0.1 percent. Because of the small fringe spacing, the gauge lengths are much shorter. Therefore the measurements in an area of high strain gradient are much more accurate.

B. Crack Tip Strain

The cyclic strain ranges, \( \Delta \varepsilon \), and the maximum strains \( \varepsilon_{\text{max}} \), ahead of a crack tip were measured and plotted against the distance from the crack tip on a log-log scale. Figures (8a, 8b) and (9a, 9b) show the strain measurements at stress ratios \( R = 1/10 \) and \( 1/3 \) respectively. The specimen thickness is 0.05 inches. Figures (10a, 10b) and (11a, 11b) are the ones for a specimen thickness, \( t = 0.25 \) inches, and at stress ratios \( R = 1/10 \) and \( 1/3 \) respectively. Close to a crack tip, the data at a given \( \Delta K \) or \( K_{\text{max}} \) value can be correlated by a straight line. All the lines for \( \Delta \varepsilon \) have the slope of \(-0.5\), regardless of the stress ratio and
specimen thickness. The cyclic strain range at a point is the change of
the strain at the point, as the applied stress changes from minimum to
maximum values. The measurements in Figure (8a) were made on three speci-
mens. The applied maximum stresses are in the neighborhood of 15,000 psi.
The stress ratio is 1/10. Since the applied stresses are considerably
below the yield strength of the materials, the plastic zone sizes must
be quite small in comparison to their corresponding crack lengths. In
this case, the deformations are close to the values given by the elastic
solution. One can write the strain in y-direction, $\varepsilon_y$ along the crack
line, for the plane stress model as

$$
\varepsilon_y = \frac{(\sigma_y - \nu \sigma_x)}{E} = \frac{\sigma_y (1 - \nu)}{E}
$$

$$
= \frac{K (1 - \nu)}{E \sqrt{2\pi r}}
$$

where $K$ is the stress intensity factor, $E$ is the elastic modulus, $\nu$
is Poisson's ratio and $r$ is the distance from the crack tip. The ex-
pression of $\sigma_x$ and $\sigma_y$ are given by Equation (21a) with $\theta = 0$. In
order to calculate $\Delta \varepsilon$, $\Delta K$ instead of $K$, should be used in Equation (30).

In the lower $\Delta K$ range, i.e. below 23 ksi $\sqrt{\text{in}}$, the measured strains
agree closely with the strains calculated from the elastic solution of
Equation (30), with $E = 10.1 \times 10^6$ psi and $\nu = 0.3$. On the average,
the measured values are approximately 10 percent higher than the cal-
culated ones. As $\Delta K$ increases beyond this value, the deviation of
the measured and the calculated values increases rapidly. At $\Delta K = 34.2 \text{ ksi} \sqrt{\text{in}}$, the measured values are 1.33 times the calculated elastic strains as shown in Figure (12). The solid lines are the measured strains replotted from Figure (8a). The dashed lines are the calculated ones. These high measured strains are caused by crack tip necking.

Figure (8b) shows the maximum strain, $e_{\max}$, measured at five $K_{\max}$ values. The slopes of the lines decrease as the values of $K_{\max}$ increase. They are all less than -0.5 and vary from -0.66 to -0.78, as listed in the figure. The average value is -0.71. This variation of the slopes of the lines agrees qualitatively with the analysis given by Swedlow, Williams and Yang. However it should be pointed out that the calculation by Swedlow et al. was made for a monotonic loading and the measured strains were the accumulated strains under a cyclic load with a positive mean load.

Figure (13) shows two sets of the measured accumulated maximum strains, $e_{\max}$, which were replotted from Figure (8b) as the solid lines. The dashed lines are the elastic strains calculated from Equation (30) for the same $K_{\max}$ values of the measured lines. For $K_{\max} = 13.4 \text{ ksi} \sqrt{\text{in}}$ at $r = 0.01$ inches, the measured $e_{\max}$ is 1.3 times the calculated value. The deviation decreases as $r$ increases. At $r = 0.03$ inches, the measured $e_{\max}$ is less than 1.1 times the calculated value. At the higher values of $K_{\max}$, the deviation is much larger. For $K_{\max} = 25.4 \text{ ksi} \sqrt{\text{in}}$, at $r = 0.01$ inches, the $e_{\max}$ is 2.9 times the calculated value.

Figure (14) shows the plot of strain versus the distance from a
crack tip. Two pairs of $\varepsilon_{\text{max}}$ and $\Delta \varepsilon$ curves were plotted. The data were obtained from Figures (8a) and (8b). For each pair, the values of $\Delta K$ and $K_{\text{max}}$ are nearly the same. They are approximately $13.4$ ksi $\sqrt{\text{in}}$ and $20.7$ ksi $\sqrt{\text{in}}$. The difference is less than 2 percent. The values of $\Delta K$ and $K_{\text{max}}$ are listed in the figure. The differences between these two curves of each pair are the accumulated residual strains, $\varepsilon_{\text{acc}}$, as a specimen is unloaded. It is expected that $\varepsilon_{\text{acc}}$ will increase as $R$ increases. As $r$ increases and strain decreases, the two curves of each pair converge. Both of these two pairs of curves, converge at $\varepsilon \approx 0.0021$. At this level of strain, there is no accumulation of plastic deformation, even if the mean load is positive.

If the stress-strain characteristics of the material are symmetrical in tension and compression, one can conclude that for a completely reversed stress cycle, if the strain amplitude is less than 0.0021, there is no noticeable cyclic plastic deformation. If fatigue damage nucleation is caused by repeated cyclic plastic deformation, this strain amplitude must correspond to the endurance limit of the material, i.e. 21,000 psi. The measured endurance limit of 2024-T3 aluminum alloy for a completely reversed stress cycle is 20,000 psi, as given in the Alcoa Aluminum Handbook of 1959. If this correspondence is valid, a convenient and quick means to measure endurance limit can be developed. However, additional data are needed to substantiate this conclusion, especially the investigation of effects of the residual stresses near a crack tip on the strain distribution.
For a K value equal to 20.7 ksi $\sqrt{\text{in}}$, these two curves converge at $r = 0.14$ inches. At any point further away, the deformation is elastic. At a point, where $r$ is slightly less than 0.14 inches, plastic deformation takes place, even though the local stress at the point is less than one half of the static yield strength of the material. The deviation of the actual strain from the linear elastic relation is so minute that it is very difficult to detect whether or not the load is monotonic. At $r=0.01$ inches, $\varepsilon_{\text{max}}$ is 2.3 times $\Delta\varepsilon$. For the case $K = 13.4$ ksi $\sqrt{\text{in}}$, the difference between $\varepsilon_{\text{max}}$ and $\Delta\varepsilon$ is much smaller. These two curves converge at a distance $r = 0.04$ inches. The larger difference in $\varepsilon_{\text{max}}$ and $\Delta\varepsilon$, at a high K value is caused by crack tip necking.

Figure (15) shows the plot of $\varepsilon_{\text{max}}$ and $\Delta\varepsilon$ versus $r$ in linear scale. Both were measured at the same crack length. The value of $K_{\text{max}}$ is 25.5 ksi $\sqrt{\text{in}}$ while $\Delta K$ is 22.8 ksi $\sqrt{\text{in}}$. The upper solid curve is the measured $\varepsilon_{\text{max}}$ curve. It is the maximum strain of the cyclic deformation. The dashed curve is $\Delta\varepsilon$. The lower solid curve is the difference between $\varepsilon_{\text{max}}$ and $\Delta\varepsilon$, therefore it is the minimum strain, $\varepsilon_{\text{min}}$, curve. If the crack length is very long in comparison with the distance to a point ahead of a crack tip, the K value does not change much as the crack propagates toward the point. Therefore the decrease in $r$ is linearly proportional to the number of cycles of loading. Consequently, the abscissa axis can be considered as a time scale. For the point at $r = 0.1$ inches, one can view $r = 0.1$ inches...
as time zero. The time scale moves toward the left of the figure. The figure indicates clearly that as time goes on $\varepsilon_{\text{max}}$, $\Delta \varepsilon$, and $\varepsilon_{\text{min}}$ increase rapidly.

Near a crack tip, both $\varepsilon_{\text{max}}$ and $\Delta \varepsilon$ curves can be plotted as straight lines on a log-log scale, with different slopes. The data can be fitted by the empirical equations as,

\[
\frac{\varepsilon_{\text{max}}}{\varepsilon_Y(s)} = \left(\frac{r}{r_{\text{max}}(s)}\right)^{n_1}
\]

\[
\frac{\Delta \varepsilon}{\Delta \varepsilon_Y(c)} = \left(\frac{r}{r_p(c)}\right)^{n_2}
\]

The strain at any time can be written as,

\[
\varepsilon = \varepsilon_{\text{mean}} + \frac{\Delta \varepsilon}{2} \sin (2\pi N)
\]

and

\[
\varepsilon_{\text{mean}} = \varepsilon_{\text{max}} - \frac{\Delta \varepsilon}{2}
\]

After substitution, one obtains

\[
\varepsilon = \varepsilon_Y(s) \left(\frac{r}{r_{\text{max}}(s)}\right)^{n_1} - \frac{1}{2} \Delta \varepsilon_Y(c) \left(\frac{r}{r_p(c)}\right)^{n_2} \left[1 - \sin(2\pi N)\right]
\]

Equation (33) is the cyclic strain history ahead of a crack tip in Figure (15).
Figures (9a) and (9b) show the plots of $\Delta \varepsilon$ and $\varepsilon_{\text{max}}$ for $R = 1/3$ and $t = 0.05$ inches. The slopes of the lines in Figure (9a) for $\Delta \varepsilon$ are all equal to $-0.5$. The measured strains in this case are about 20 percent higher than the calculated elastic strains. This increase in the measured strains are much higher than those of $R = 1/10$, i.e. 10 percent. This increased deviation is caused by crack tip necking enhanced by the high stress ratio. At higher values of $R$, the $\varepsilon_{\text{max}}$ is much higher than $\Delta \varepsilon$. In other words, at the same value of $\Delta K$, the $\varepsilon_{\text{max}}$ for $R = 1/3$ is much higher than the $\varepsilon_{\text{max}}$ for $R = 1/10$. For example, at $\Delta K = 22.8 \text{ ksi } \sqrt{\text{in}}$ for $R = 1/10$, at $t = 0.05$ inches, $\varepsilon_{\text{max}}$ is twice $\Delta \varepsilon$, i.e. 0.006 and 0.003 respectively. For the case of $R = 1/3$, at $\Delta K = 22.5 \text{ ksi } \sqrt{\text{in}}$, $\varepsilon_{\text{max}} = 0.0135$ and $\Delta \varepsilon = 0.0032$, that is $\varepsilon_{\text{max}} = 4.2 \Delta \varepsilon$. This higher value of $\varepsilon_{\text{max}}$ enhances necking for the higher value of $R$. The slopes of the lines in Figure (9b) for $\varepsilon_{\text{max}}$ vary from $-0.65$ to $-0.8$. The average slope is $-0.77$.

Figure (10a) and (10b) are the plotted data for specimens tested at $R = 1/10$ and $t = 0.25$ inches. On the average, the measured values of $\Delta \varepsilon$ are approximately 25 percent higher than the calculated elastic strains. The measured values of $\Delta \varepsilon$ for 0.25 inches thick specimens are 15 percent higher than the 0.050 inches thick specimens at the same corresponding $\Delta K$ values. This agrees with the finding that fatigue crack propagation rate is faster in thick specimens than in thin ones.

Figure (16) shows the comparison of $\varepsilon_{\text{max}}$ for $t = 0.25$ inches.
and 0.050 inches specimens at three $K_{\text{max}}$ values, and at $R = 1/10$. The $K_{\text{max}}$ values of each pair of lines are nearly equal to each other. The data are taken from Figures (8b) and (10b). The solid lines represent the thick specimens and the dashed lines represent the thin ones. The values of $K_{\text{max}}$ are shown in the figure. At lower $K$ values, the measured strains, $\varepsilon_{\text{max}}$, are close to each other. As $K_{\text{max}}$ is increased to 20 ksi $\sqrt{\text{in}}$ the $\varepsilon_{\text{max}}$ of thin specimens is nearly 1.2 times that of thick one. The $\varepsilon_{\text{max}}$ values at $K_{\text{max}} = 25.5$ ksi $\sqrt{\text{in}}$ for thin specimens is nearly 1.5 times that of the thick ones at $K_{\text{max}} = 23.7$ ksi $\sqrt{\text{in}}$, even though the difference in $K_{\text{max}}$ is only 7.6 percent. The data indicate that the $\varepsilon_{\text{max}}$ values increase much more rapidly for thin specimens as $K_{\text{max}}$ values increase. This is another indication of crack tip necking. At high $K_{\text{max}}$ value, the thicker specimen, i.e. $t = 0.25$ inches, has more constraint and less tendency to neck than the thinner one. Therefore it has lower values of $\varepsilon_{\text{max}}$.

Figures (11a) and (11b) show $\Delta \varepsilon$ and $\varepsilon_{\text{max}}$ for specimens tested at $R = 1/3$ and $t = 0.25$ inches. In Figure (11a), the slopes of the lines are - 0.5. On the average, the measured values of $\Delta \varepsilon$ are 33 percent higher than the calculated elastic strains. In Figure (11b), the slopes of the lines vary from - 0.5 for the low values of $K_{\text{max}}$ to - 0.8 at higher values of $K_{\text{max}}$. The deviation of the measured strains from the calculated elastic strains increases with $K_{\text{max}}$.

C. Plastic Zone Size

As a crack propagates toward a point, the strain at the point
increases. If a point is far away from a crack tip, the deformation is elastic. As a point moves to \( r_p(e) \), permanent plastic deformation begins to accumulate. Plastic deformation occurs only during the loading half cycle, if the maximum cyclic effective stress exceeds the static yield strength, \( \sigma_Y(s) \). Plastic deformation will take place during the unloading half cycle, if the cyclic effective stress range, \( \Delta \sigma_{\text{eff}}(c) \), exceeds the cyclic yield strength range, \( \Delta \sigma_Y(c) \). The cyclic stress-strain curve of 2024-T351 aluminum alloy is shown in Figure (5b). The cyclic yield strength range, \( \Delta \sigma_Y(c) \), is 134,000 psi, which is more than two times higher than the static yield strength, \( \sigma_Y(s) \), i.e. 52,000 psi.

The stress \( \sigma_Y \) at a point along a crack line can be calculated from Equation (30) if \( \epsilon_y \) is known. Assuming \( \sigma_x = \sigma_y \) and \( \sigma_z = 0 \), the effective stress is

\[
\sigma_{\text{eff}} = \left\{ \frac{1}{2} \left[ (\sigma_x - \sigma_y)^2 + (\sigma_y - \sigma_z)^2 + (\sigma_z - \sigma_x)^2 \right] \right\}^{1/2}
\]

\[= \sigma_y \] (34)

With \( \sigma_Y(s) = 52,000 \) psi, \( E = 10.1 \times 10^6 \) psi and \( v = 0.3 \), the static yield strain, \( \varepsilon_Y(s) \) was calculated from Equation (30) as

\[
\varepsilon_Y(s) = \left( \frac{1-v}{E} \right) \sigma_Y(s) = \left( \frac{1-0.3}{10.1 \times 10^6} \right) \times 52,000 = 3.6 \times 10^{-3} \] (35a)
Similarly, the cyclic yield strain range, $\Delta \varepsilon_Y(c)$, was calculated with $\Delta \sigma_Y(c) = 134,000$ psi, $E = 10.1 \times 10^6$ psi and $v = 0.3$.

$$\Delta \varepsilon_Y(c) = \frac{(1-v)}{E} \Delta \sigma_Y(c) = 9.3 \times 10^{-3} \tag{35b}$$

Using $\varepsilon_Y(s) = 3.6 \times 10^{-3}$, $\Delta \varepsilon_Y(c) = 9.3 \times 10^{-3}$ and the data of strain distribution curves in Figures (8a) to (11b), both static plastic zone size, $r_p(s)$, and cyclic plastic zone size, $r_p(c)$, can be obtained experimentally for each stress intensity value. Figure (17a) shows the plot of static plastic zone size, $r_p(s)$, versus stress intensity range, $\Delta K$ for the thin specimen of 0.050 inches thick. The dashed line is the calculated value. For the case of $R = 1/10$, at $\Delta K = 11.6$ ksi $\sqrt{\text{in}}$, the measured $r_p(s)$ is equal to 0.0085 inches. At $\Delta K = 22.8$ ksi $\sqrt{\text{in}}$, measured $r_p(s)$ is equal to 0.037 inches.

Within the range of $\Delta K = 11.6$ ksi $\sqrt{\text{in}}$ to 22.8 ksi $\sqrt{\text{in}}$, the average value of $r_p(s)$ is 12 percent higher than the calculated one. As the value of $\Delta K$ becomes larger than 22.8 ksi $\sqrt{\text{in}}$, the measured values of $r_p(s)$ start to deviate from the line having a slope of 2 to the line with a slope of 4. At $\Delta K = 34.2$ ksi $\sqrt{\text{in}}$, the measured $r_p(s)$ is about 61 percent higher than the calculated one. The empirical equations for these two line segments can be written as

$$r_p(s) = C_1 8 \left(\frac{\Delta K}{\sigma_Y(s)}\right)^2 = 0.18 \left(\frac{\Delta K}{52,000}\right)^2 = 66.56 \times 10^{-6} \Delta K^2 \text{ micro-inch}$$

for region II \hspace{1cm} (36a)
and
\[ r_p(s) = C_{19} \left( \frac{\Delta K}{\sigma_{Y(s)}} \right)^4 = 0.8 \left( \frac{\Delta K}{52,000} \right)^4 = 10.9 \times 10^{-14} \Delta K^4 \text{ micro-inch} \]

for region III \ (36b).

For the case of \( R = 1/3 \), the measured \( r_p(s) \) is about 38 percent higher than the calculated value. The difference between \( R = 1/10 \) and \( 1/3 \) is 23 percent.

Figures (17b) shows the plot of cyclic plastic zone size, \( r_p(c) \) versus stress intensity range, \( \Delta K \). The data was obtained from Figures (8a) and (9a). At lower \( \Delta K \) values, the lines in Figures (8a) and (9a) have to be extrapolated to determine the cyclic plastic zone size. The characteristics of both lines for \( R = 1/10 \) and \( 1/3 \) are the same as those in Figure (17a) as expected. The slopes of the lines for \( R = 1/10 \) also change from 2 to 4 at \( \Delta K = 24 \text{ ksi } \sqrt{\text{in}} \).

On the average the values of \( r_p(c) \) are about 6.7 times lower than those of \( r_p(s) \). The empirical equations for these two line segments of \( R = 1/10 \) can also be written as

\[ r_p(c) = C_{20} \left( \frac{\Delta K}{\sigma_{Y(c)}} \right)^2 = 0.12 \left( \frac{\Delta K}{134,000} \right)^2 \]

\[ = 6.68 \times 10^{-6} \Delta K^2 \text{ micro-inch} \]

for region II \ (37a).
\[ r_p(c) = C_{21} \left( \frac{\Delta K}{\Delta \sigma_Y(c)} \right)^h = 5 \left( \frac{\Delta K}{134,000} \right)^h \]
\[ = 1.55 \times 10^{-1} \Delta K^h \text{ micro-inch} \]

for region III (37b).

Figure (18) shows the plot of static plastic zone size, \( r_p(s) \) versus stress intensity range \( \Delta K \) for 0.25 inches thick specimens. Only the data for region II were shown. The slopes for both lines of \( R = 1/10 \) and \( 1/3 \) are also 2. For the case of \( R = 1/10 \), the measured \( r_p(s) \) is 1.55 times higher than the calculated one, while the \( r_p(s) \) for \( R = 1/3 \) is also 23 percent higher than that of \( R = 1/10 \). Comparing Figure (18) with Figure (17a) for the same stress ratio \( R \), one finds that the values of \( r_p(s) \) for the thick specimens are 35 percent higher than that of thin ones. However, the differences between the values of \( r_p(s) \) for \( R = 1/10 \) and \( 1/3 \) are the same for both thick and thin specimens, i.e. 23 percent. This observation leads one to formulate an empirical equation for \( r_p \) as

\[ r_p(s) = C_{22} \frac{\Delta K^2}{1-R} \quad (38) \]

where \( C_{22} \) is a constant dependent on thickness.

Figure (19) shows the maximum static plastic zone size, \( r_{\text{max}}(s) \) versus the maximum stress intensity factor \( K_{\text{max}} \). The data were obtained from the maximum strain distribution in Figures (8b), (9b),
(10b) and (11b) with $\varepsilon_Y = 3.6 \times 10^{-3}$. The specimen cross-sectional area and the applied stress ratios are all shown in the figure. The overall slope for all of the data is 2.6. However, if one examines more closely, it seems that every individual set of data can be correlated by a line segment with a slope of 2, except the one with the cross sectional area, $A = 4'' \times 0.050''$ and $R = 1/10$, which were obtained from three different specimens. In toto, the data indicate that the effects of thickness and stress ratio to the maximum plastic zone size is not severe.

D. Crack Opening Displacement

The relative crack opening displacement between two points, one on the upper and the other on the lower crack surfaces, can be measured from the moire fringe pattern and calculated as

$$\delta = pN_o$$

where $p$ is the pitch of the master grille, which is equal to $1/13,400$ inches. $N_o$ is the total number of fringes along the crack surface between these two points.

There are two models to calculate the crack opening displacement i.e. linear elastic model and Dugdale's strip necking model. The linear elastic model gives

$$\delta = 2K(\frac{r}{2\pi})^{1/2} \frac{h}{E}$$

for plane stress. (21b)
On the other hand, Dugdale model predicts that crack tip opening displacement, \( \delta \), is proportional to the square of the stress intensity factor as shown in Equation (11c). The crack opening displacement, \( \delta \), is given by Goodier and Field as

\[
\delta = 2V(x,a) = \frac{2c_y \lambda}{\pi E} \left[ \cos \theta \log \frac{\sin^2(\theta_2 - \theta)}{\sin^2(\theta_2 + \theta)} + \frac{\cos^2(\theta_2 + \theta)}{(\sin \theta_2 + \sin \theta)^2} \cos \theta_2 \log \frac{\sin \theta_2}{\sin \theta_2 - \sin \theta} \right]
\]

Figures (20a) and (20b) show the plots of the crack opening displacement range, \( \Delta \delta \), versus the distances from a crack tip for 0.050 inches thick specimen. \( \Delta \delta \) was measured as the applied stress was increased from \( \sigma_{\text{min}} \) to \( \sigma_{\text{max}} \). Therefore the measured \( \Delta \delta \) corresponds to the stress intensity factor range, \( \Delta K \). Figure (20a) is for \( R = 1/10 \) and Figure (20b) is for \( R = 1/3 \). Each line in the figures corresponds to one \( \Delta K \) value. All the lines have a slope of 0.5. This means that crack opening displacement is proportional to the square root of the distance from a crack tip, which is predicted by the linear elastic model shown in Equation (21b).

Figure (20c) shows the plot of \( \Delta \delta \) at \( r = 0.04 \) inches versus \( \Delta K \) for both cases of \( R = 1/10 \) and \( R = 1/3 \). The data were obtained from Figures (20a) and (20b). With \( E = 10.1 \times 10^6 \text{ psi} \), the dashed line was calculated using the elastic model of Equation (21b). The slope of the dashed line is one. In region II, i.e. the intermediate \( \Delta K \)
values, the lines, correlating $\Delta \delta$ with $\Delta K$, have a slope of one for both $R = 1/10$ and $R = 1/3$. This agrees with the prediction of the linear elastic model. On the average, the measured values of $\Delta \delta$ for $R = 1/3$ are 6 percent less than the calculated ones; and those for $R = 1/10$, the measured values of $\Delta \delta$ are 2½ percent below the calculated ones. In region III, i.e. high $\Delta K$ values, the data of $R = 1/10$ can be correlated with a straight line having a slope of two.

Figures (20d) and (20e) are another two sets of data of $\Delta \delta$ versus $r$ for 0.25 inches thick specimens. Figure (20d) is for $R = 1/10$ and Figure (20e) is for $R = 1/3$. The slopes of the lines in both figures are all 0.5. This indicates that crack opening displacement is proportional to the square root of distance from the crack tip, regardless of the effects of specimen thickness and applied stress ratio. Figure (20f) shows the plot of $\Delta \delta$ at $r = 0.04$ inches versus their corresponding $\Delta K$ values for both $R = 1/10$ and $R = 1/3$. The data were obtained from Figures (20d) and (20e). The dashed line in the figure was calculated from the elastic model using Equation (21b). Similar to thin specimens, the lines, correlating $\Delta \delta$ with $\Delta K$ in region II, can be correlated with a straight line having a slope equal to one. On the average, the measured $\Delta \delta$ values for $R = 1/10$ are 10 percent below the calculated ones; and those for $R = 1/3$ are 8 percent higher than the calculated ones. In toto, the measured $\Delta \delta$ for $R = 1/3$ is 18 percent higher than those at $R = 1/10$. This difference is the same as in the case of 0.050 inches thick specimens. Comparing Figures (20c) and (20f) for the effect of thickness on the values of $\Delta \delta$ at the same stress.
ratio, one finds that $\Delta \delta$ of the thicker specimens, $t = 0.25$ inches, is $1\frac{1}{4}$ percent higher than that of thinner specimens for both $R = 1/10$ and $R = 1/3$.

Figure (21a) shows the relative maximum crack opening displacement, $\delta_{\text{max}}$, versus the distance from the crack tip for the specimens 0.050 inches thick at $R = 1/10$. Figure (21b) are those for $R = 1/3$. The values of $\delta_{\text{max}}$ were measured from the same pictures of moire fringe patterns which were used to measure the $\epsilon_{\text{max}}$. The corresponding $K_{\text{max}}$ values for each line are indicated in the figures. The slopes of these lines are close to 0.5. They vary from 0.5 to 0.6. Figures (22a) and (22b) are another two sets of data which show $\delta_{\text{max}}$ versus $r$ for 0.25 inches thick specimens. The values of $K_{\text{max}}$ and $R$ are also shown in the figures. For the case of $R = 1/10$, all the slopes of the lines are equal to or very close to 0.5. While in the case of $R = 1/3$, the slopes of the lines vary from 0.5 to 0.6.

Figure (22c) is the plot of $\delta_{\text{max}}$ at $r = 0.04$ inches versus $K_{\text{max}}$ for both thin and thick specimens. The data are obtained from Figures (21a), (21b), (22a), and (22b). The applied stress ratios and specimen thicknesses are shown in the figure. The dashed line is the calculated line using Equation (21b) of elastic model. The data for both thin and thick specimens can be correlated with a straight line having a slope equal to one regardless of the stress ratio.

Figures (22d) and (22e) show the comparison of $\Delta \delta$ predicted by elastic model and Dugdale model with the measured values. Figure
(22d) is for thin specimen and Figure (22e) is for thick one. The measured data for two ΔK values are shown in each figure. The solid and the dashed lines are calculated using Dugdale model and elastic model respectively. The cyclic yield stress range, $\Delta \sigma_Y$, is used for Dugdale's model.

The experimental points seem to agree better with the calculated displacements of the elastic crack. However the differences in displacements calculated by these two models are small in the range of the measurements. A straight line correlates well with the measured displacements. There is no tendency to level off as r approaches the crack tip.
IV. DISCUSSION

A. Crack Tip Deformation and Fatigue Crack Propagation

Based on dimensional analysis, Frost and Dugdale (3) and Liu (4) have pointed out that fatigue crack propagation is proportional to the crack length if the crack length is the only pertinent length parameter. Subsequently, Liu (5) has shown that if the plastic zone size is very small or if the ratio $\Delta\sigma/\sigma_Y$ is low, the crack tip stresses and strains can be characterized by $\Delta K$ and $R$. This conclusion is correct, if the crack tip stress and strain distributions are not affected by crack tip necking. In this case, $r_p$ is proportional to $\Delta K^2$, if $R$ remains constant. The crack tip region can be scaled by $r_p$ so that at the homologous points, i.e., at the geometrically similar points, the stresses and strains are identical. If a crack increment is proportional to $r_p$, the stresses and strains within the area of the crack increment must be the same regardless of the value of $\Delta K$ or the size of $r_p$. If crack propagation is caused by the stresses and strains, $da/dN$ must be proportional to $r_p$. That is

$$\frac{da}{dN} = C_{23} r_p \quad (39a)$$

By applying the cumulative damage rule within the plastic region ahead of a crack tip, Lehr and Liu (47) and Liu and Iino (31) derived a fatigue crack propagation equation for a cracked plate. They showed that fatigue crack propagation rate was proportional to the plastic zone size as
\[
\frac{da}{dN} = C_{24} \left( \frac{M}{\varepsilon_y(c)} \right)^{1/Z} r_p
\]  
(39b)

where \( C_{24} \) is a proportional constant, \( M \) is the strain range corresponding to the fatigue life of one cycle, \( Z \) is the slope of \( \log \Delta \varepsilon_T \) versus \( \log N_f \) and \( \varepsilon_y(c) \) is the cyclic yield strength found from the strain controlled fatigue test.

Assuming that the true mechanical conditions around a crack tip are represented by the plastic deformations, Erdogan and Roberts (48) proposed a model of fatigue crack propagation as

\[
\frac{da}{dN} = C_{25} (r_p)^\alpha
\]  
(40a)

where \( C_{25} \) and \( \alpha \) are constants for a given material, and \( r_p \) is a characteristic length of the plastic zone at the crack tip.

Later on, Roberts and Erdogan (49), studying the effect of mean stress on fatigue crack propagation, generalized the above equation to

\[
\frac{da}{dN} = C_{26} (r_{max})^{\alpha_1} (r_p)^{\alpha_2}
\]  
(40b)

where \( C_{26}, \alpha_1 \) and \( \alpha_2 \) are constants for a given material and \( r_{max} \) and \( r_p \) are the maximum and range values of the plastic zone size, respectively, as measured along the prolongation of the crack.
Now, let us assume that fatigue damage occurs only inside of the plastic region. As we have shown prestress cycling below the static yield strength has a negligible effect on the rate of fatigue crack propagation. In other words, fatigue damage occurs mainly after an element enters the plastic zone boundary. Before the elements enter the plastic zone boundary, there is only negligible fatigue damage to the elements. If \( r_p \ll a \), \( \Delta K \) is essentially constant during the time period after a material element enters \( r_p \) and before it arrives the crack tip. Therefore, one can assume \( da/dN \), within \( r_p \), is essentially constant.

If

\[
da/dN = C_{23} r_p
\]

then

\[
\int_a^{a+r_p} da = C_{23} r_p \int_0^{N_r} dN
\]

or

\[
C_{23} = 1/N_r
\]

therefore

\[
da/dN = \frac{r_p}{N_r} \tag{41}
\]

where \( N_r \), a constant, is the number of cycles to propagate a crack through the length of \( r_p \). The validity of Equation (41) will be discussed and clarified by \( r_p \) values obtained from the moire strain measurements ahead of a crack tip and \( da/dN \), measured on the specimen surfaces.

Figure (23a) shows the plot of \( da/dN \) versus \( \Delta K \), as well as \( r_p(s) \) versus \( \Delta K \) for 0.050 inches thick specimens at \( R = 1/10 \). The data were obtained from Figures (2b) and (17a). In the lower \( \Delta K \) region the data give a slope of two for both lines, while the data in the higher \( \Delta K \) region give a slope of four. The transitions take place at \( \Delta K \) equal to
$21 \text{ ksi } \sqrt{\text{in}} \text{ and } 24 \text{ ksi } \sqrt{\text{in}} \text{ for } \frac{\text{da}}{\text{dN}} \text{ and } r_p \text{ respectively. The data in Figure (23a) give the empirical constant } N_r \text{ for region II.}$

$$N_r = \frac{r_p(s)}{(\text{da}/\text{dN})} = 1.08 \times 10^3 \text{ cycles}$$

or

$$\frac{\text{da}}{\text{dN}} = \frac{r_p(s)}{N_r} = \frac{C(\frac{\Delta K}{\sigma_y})^2}{N_r} = \frac{0.18}{1.08 \times 10^3} \left( \frac{\Delta K}{52,000} \right)^2$$

$$= 61.6 \times 10^{-9} \Delta K^2 \text{ micro-inch/cycle} \quad (42) \text{.}$$

For the data in region III, the empirical constant $N_r$ is

$$N_r = \frac{r_p(s)}{(\text{da}/\text{dN})} = 0.75 \times 10^3 \text{ cycles}$$

or

$$\frac{\text{da}}{\text{dN}} = \frac{r_p(s)}{N_r} = \frac{C(\frac{\Delta K}{\sigma_y})^4}{N_r} = \frac{0.8}{0.75 \times 10^3} \left( \frac{\Delta K}{52,000} \right)^4$$

$$= 14.6 \times 10^{-17} \Delta K^4 \text{ micro-inch/cycle} \quad (43) \text{.}$$

Similar data for 0.25 inches thick specimens at $R = 1/10$ are shown in Figure (23b). The data give the empirical constant $N_r$ for region II.

$$N_r = \frac{r_p(s)}{(\text{da}/\text{dN})} = 1.0 \times 10^3 \text{ cycles}$$

or

$$\frac{\text{da}}{\text{dN}} = \frac{r_p(s)}{N_r} = \frac{C(\frac{\Delta K}{\sigma_y})^2}{N_r} = \frac{0.26}{1.0 \times 10^3} \left( \frac{\Delta K}{52,000} \right)^2$$

$$= 96.2 \times 10^{-9} \Delta K^2 \text{ micro-inch/cycle} \quad (44) \text{.}$$
The values of $N_r$ for thick and thin specimens in region II are found to be $1.0 \times 10^3$ cycles and $1.08 \times 10^3$ cycles respectively. They have only an 8 percent difference. This amount of difference is well within the experimental error. Therefore, one can conclude that the number of cycles, $N_r$, to propagate a crack through the plastic zone size, $r_p$, is a material constant.

The data for $R = 1/3$ are plotted in Figures (24a) and (24b). The slopes of the line that correlate $da/dN$ with $\Delta K$ and the line that correlates $r_p$ with $\Delta K$ do not agree. The slope of the former line is 2.6 and that of the latter is 2, for both 0.050 inches and 0.25 inches thick specimens.

In Figure (24a) the dashed line is the fatigue crack propagation rate predicted from Equation (42). The measured and predicted lines intersect at $\Delta K = 11.5 \text{ ksi } \sqrt{\text{in}}$, where the crack propagation rate is equal to 10 micro-inches per cycle. At $\Delta K = 20 \text{ ksi } \sqrt{\text{in}}$, the $da/dN$ are 30 micro-inches per cycle and 40 micro-inches per cycle for predicted and measured values respectively. In Figure (24b), the measured $da/dN$ deviates from that given by Equation (44). At $\Delta K = 7 \text{ ksi } \sqrt{\text{in}}$ the predicted value of $da/dN$ is 1.4 times the measured one. While at $\Delta K = 18 \text{ ksi } \sqrt{\text{in}}$, the measured value of $da/dN$ is 1.4 times the predicted one.

The correlation of $da/dN$ with $r_p$ will be examined in terms of the characterization of crack tip stresses and strains by $\Delta K$ and $R$. As we have pointed out earlier, the crack tip cyclic deformation is
characterized by \( \Delta K \) and \( R \), if the deformation is not affected by necking. In this case, the region ahead of a crack tip can be scaled by the characteristic length \( r_p \), so that as a crack grows, the deformation at geometrically similar points is identical.

Cyclic deformation can be specified by \( \Delta \varepsilon \) and \( \varepsilon_{\text{max}} \). In Figures (25a to 25d), both \( \Delta \varepsilon \) and \( \varepsilon_{\text{max}} \) are plotted against \( r/r_p(s) \). The data are obtained from Figures (8a to 11b). The \( r_p(s) \) is the distance for \( \varepsilon_Y(s) = 3.6 \times 10^{-3} \). For each pair of \( \Delta \varepsilon \) and \( \varepsilon_{\text{max}} \) curves, the same value of \( r_p(s) \) is used. Since all the lines for \( \Delta \varepsilon \) have the same slope, i.e. \( -0.5 \), all the data are consolidated into a single line. In Figure (25a), the data for 0.050 inches thick specimens at \( R = 1/10 \) are shown. The three lines of \( \varepsilon_{\text{max}} \) data at \( \Delta K = 16.2, 13.3 \) and 11.6 ksi \( \sqrt{\text{in}} \) fall into a narrow band. This indicates clearly that the region ahead of a crack tip can be scaled by \( r_p(s) \) so that at geometrically similar points, the cyclic strains, i.e. \( \Delta \varepsilon \) as well as \( \varepsilon_{\text{max}} \), are identical.

As discussed earlier, if \( r_p \) can be used to scale the region near a crack tip, \( da/dN \) should be proportional to \( r_p \) and \( K^2 \). As indicated by the crack growth data in Figure (2b) all these three values of \( \Delta K \) are in region II, and \( da/dN \) is indeed proportional to \( \Delta K^2 \). For the line at \( K_{\text{max}} = 25.5 \text{ ksi \( \sqrt{\text{in}} \)}, the \( \varepsilon_{\text{max}} \) curve is considerably above the other three. This \( K_{\text{max}} \) value is above the transition point of the \( da/dN \) curve in Figure (23a). Because the \( \varepsilon_{\text{max}} \) is higher than the value given by the scaling law, it is reasonable to expect that \( da/dN \) is faster than that given by Equation (44), i.e. \( da/dN \) proportional to \( \Delta K^2 \). Figure (25b) shows similar data for 0.25 inches thick specimens.
at \( R = 1/10 \). All the lines of \( \varepsilon_{\text{max}} \) data fall into a narrow band, too. This is another indication that the region ahead of a crack tip can be scaled by \( r_p(s) \), so that at geometrically similar points the cyclic strains are identical.

Figures (25c) and (25d) show the data of \( R = 1/3 \) for 0.050 inches and 0.25 inches thick specimens, respectively. Unlike the data of \( R = 1/10 \), the data of \( R = 1/3 \) do not fall into a narrow band. The data indicate that at geometrically similar points ahead of a crack tip, the maximum cyclic strains, \( \varepsilon_{\text{max}} \), are not identical, due to the effect of higher stress ratios. The higher the value of \( K_{\text{max}} \), the higher the value of \( \varepsilon_{\text{max}} \) at geometrically similar points. As a consequence, one would expect that the measured values of \( da/dN \) for \( R = 1/3 \) would increase more rapidly than those of \( R = 1/10 \), and that the slope of \( da/dN \) versus \( \Delta K \) would deviate from two.

This deviation is also illustrated in Figure (25e), which shows the plot of \( (da/dN)/r_p(s) \) versus \( r_{\text{max}}(s) \). The values of \( da/dN \) are obtained from Figures (2b) and (2c), while the values of \( r_p(s) \) and \( r_{\text{max}}(s) \) are obtained from Figures (8a) to (11b), with \( \varepsilon_{Y(s)} = 3.6 \times 10^{-3} \). The values of \( (da/dN)/r_p(s) \) at \( R = 1/10 \) are almost constant for both thick and thin specimens. The average values are \( 1.02 \times 10^{-3} \) per cycle and \( 0.85 \times 10^{-3} \) per cycle for thick and thin specimens respectively. However, at \( R = 1/3 \), the values of \( (da/dN)/r_p(s) \) increase with increasing \( r_{\text{max}}(s) \). The slopes of the lines, correlating \( (da/dN)/r_p(s) \) with \( r_{\text{max}}(s) \), are 0.41 and 0.62 for thick and thin specimens respectively.
Based on these experimental data, one can write an empirical equation as

\[
\frac{da}{dN} \frac{1}{r_p(s)} = C_{23} \frac{\alpha_1}{r_{\text{max}}(s)}
\]

or

\[
\frac{da}{dN} = C_{23} r_p(s) \frac{\alpha_1}{r_{\text{max}}(s)}
\]

where \( C_{23} \) is the proportional constant, \( \alpha_1 \) is the slope in Figure (25e), which is zero at \( R = 1/10 \) and 0.41 and 0.62 for thick and thin specimens respectively, at \( R = 1/3 \).

Equation (45) is similar to Equation (40b), which was proposed by Roberts and Erdogan(49).

B. Crack Opening Displacement and Fatigue Crack Propagation

In this investigation, it has been shown that crack opening displacement range, \( \Delta \delta \), is proportional to the square root of the distance from a crack tip, and it is linearly proportional to the stress intensity factor range. The measured values agree better with the calculated elastic displacements of the crack surfaces. Schaeffer, Liu and Ke(44) have shown that the crack opening displacements given by the Dugdale model agree well with the measured values in a thin steel specimen. In a thin specimen, necking takes place at a crack tip and the crack tip necking causes a thin narrow strip of extensive plastic
deformation. The strip necking zone agrees with the main feature of the Dugdale model. As specimen thickness increases, the size of the strip necking zone decreases. For a thick specimen, the region of extensive plastic deformation must be diffused rather than a thin narrow strip. Therefore, even if crack opening displacement exists at a crack tip, its value must be much smaller than that given by the Dugdale model.

Laird (21) proposed a crack tip blunting model for fatigue crack propagation. The crack tip opening displacement, $\delta_o$, given by the Dugdale model (11) or by the Wells (50) calculation is often used as a measure of the crack tip "blunting". In either calculation, the crack tip opening displacement is inversely proportional to the yield strength, $\sigma_Y$, of a material. According to the model, an increase in $\sigma_Y$ reduces crack tip opening displacement and $da/dN$. The study of the effects of pre-stress clearly indicates the contrary. The static yield strength of the 2024-T351 aluminum alloy is 52,000 psi, and the cyclic yield strength range, $\Delta\sigma_Y(c)$, is 134,000 psi. The material cyclically strain hardens. As the prestress level increases, $\Delta\sigma_Y(c)$ increases. Yet this increase in $\Delta\sigma_Y(c)$ causes an increase in $da/dN$. Furthermore according to the model, the crack tip opening displacement should be less in the interior of a specimen than on the specimen surface. If $da/dN$ is linearly proportional to crack tip opening displacement, the crack should advance faster on the surface than in the interior. All the observations indicate otherwise. This contradiction certainly warrants additional studies in this important area of fatigue crack propagation research.
V. SUMMARY AND CONCLUSIONS

1. Fatigue crack propagation rate is proportional to $\Delta K^2$, if the region ahead of a crack tip can be scaled by the characteristic length, $r_p$, so that as a crack grows, the deformation at geometrically similar points is identical.

2. Fatigue crack propagation rate is affected by thickness and stress ratio. The larger the plate thickness or stress ratio is, the faster the crack propagation rate will be. This spread in rate, due to the stress ratio effect, is small at lower $\Delta K$ values but becomes progressively larger as $\Delta K$ is increased. This means that the slopes of the crack propagation rate increase with increasing stress ratio. In these studies, the slopes in region II are found to be 2.0, 2.7 and 3.4 for $R = 1/10$, 1/3 and 1/2, respectively.

3. For 2024-T351 aluminum alloy, prestress cycling below the static yield strength has no noticeable effect on the fatigue crack propagation rate. However, prestress cycling above the static yield strength causes hardening and increases the fatigue crack propagation rate. This observation leads one to conclude that fatigue damage occurs mainly after the material elements ahead of a crack have entered the plastic region. Before the material elements enter the plastic boundary, there is negligible damage to the elements. Therefore, fatigue crack propagation rate can be written as

$$\frac{da}{dN} = \frac{r_p}{N_r}$$

(41)
where \( N_r \) is the number of cycles to propagate a crack through \( r_p \).

4. Under the plane stress condition, the cyclic strain distribution ahead of a crack tip, measured by moire method, agrees very well with the calculation from linear elastic theory, when the values of \( \Delta K \) are small. This indicates that the high density grille moire technique is a very useful tool for the measurements of surface strains around a crack tip.

5. The measured crack tip strains can be correlated with a straight line in a log-log plot. The slopes of these lines for \( \Delta \varepsilon \) are all equal to \(-0.5\), while those for \( \varepsilon_{max} \) are varied from \(-0.5\) to \(-0.8\). The cyclic strain history ahead of a crack tip can be written as

\[
\varepsilon = \varepsilon_{mean} + \frac{\Delta \varepsilon}{2} \sin (2\pi N) \\
= \varepsilon_Y(s) \left( \frac{r}{r_{max(s)}} \right)^{n_1} - \frac{1}{2} \Delta \varepsilon_Y(c) \left( \frac{r}{r_{p(c)}} \right)^{n_2} [1 - \sin(2\pi N)]
\]

where \( \varepsilon_{mean} \) is the cyclic mean strain, \( \varepsilon_Y(s) \) is the static yield strain, \( \Delta \varepsilon_Y(c) \) is the cyclic yield strain range, \( r_{max(s)} \) is the static maximum plastic zone size, \( r_{p(c)} \) is the cyclic plastic zone size, \( n_1 \) varies from \(-0.5\) to \(-0.8\), and \( n_2 \) is \(-0.5\).

6. The plastic zone size, \( r_p \) is found to be proportional to \( \Delta K^2 \) in region II for all the cases investigated. In region III, it is found to be proportional to \( \Delta K^4 \) for the case of 0.050 inches at \( R = 1/10 \).
If \( \frac{da}{dN} \) is proportional to \( r_p \), one obtains the following two equations

\[
\frac{da}{dN} = \frac{r_p}{N} = C_3 \Delta K^2 \quad \text{for region II} \tag{4}
\]

\[
\frac{da}{dN} = \frac{r_p}{N} = C_4 \Delta K^4 \quad \text{for region III} \tag{5}
\]

The last equation is the fourth power law of fatigue crack propagation, proposed by Paris and Erdogan \(^{(2)}\).

7. The effect of stress ratio on the fatigue crack propagation rate can also be written as

\[
\frac{da}{dN} = C_{23} \cdot r_p \frac{r_{\max}}{r_{\max}} \tag{45}
\]

where \( a_1 \) is a constant which is equal to or larger than zero. This is the equation proposed by Roberts and Erdogan \(^{(49)}\).

8. The crack opening displacement range, \( \Delta \delta \), is found to be proportional to the square root of the distance from a crack tip. At a fixed distance near a crack tip, \( \Delta \delta \) is linearly proportional to \( \Delta K \) in region II for all the cases investigated, and is proportional to \( \Delta K^2 \) in region III for the case of 0.050 inches thick at \( R = 1/10 \).

9. In the region investigated, i.e. between 0.01 inches and 0.1 inches from a crack tip, the predicted \( \Delta \delta \) by the Dugdale model and the elastic
model is very close. However, the experimental points seem to agree better with the calculated \( \Delta \delta \) by the elastic model because there is no tendency to level off as \( r \) approaches the crack tip.
REFERENCES


FIGURE 1a. CRACK AND COORDINATES
FIGURE 1b. A CENTRALLY CRACKED PLATE SPECIMEN.
FIGURE 2a. FATIGUE CRACK PROPAGATION RATE FOR 7075-T6 ALUMINUM ALLOY
FIGURE 2b. FATIGUE CRACK PROPAGATION RATE FOR 0.050 INCHES THICK 2024-T351 ALUMINUM ALLOY
FATIGUE CRACK PROPAGATION RATE FOR 0.25 INCHES THICK
2024-T351 ALUMINUM ALLOY

2024-T351 Al
A = 4" x 0.25"
R = 1/10 = 14,000 PSI
R = 1/3 = 5,000 PSI
R = 1/3 = 5,000 PSI

STRESS INTENSITY FACTOR RANGE - ΔK (KSI IN)

CRACK PROPAGATION RATE - \( \frac{dN}{dP} \) (IN/CYCLE)
FIGURE 2d. FATIGUE CRACK PROPAGATION RATE FOR 0.125 INCHES THICK 7075-T6 ALUMINUM ALLOY
FIGURE 3a. SPECIMEN GEOMETRY FOR PRESTRESS TESTS
2024-T351 Al
A = 4" x 0.080"
\( \Delta \sigma_0 = 35,750 \pm 29,250 \text{ PSI} \)

**FIGURE 3b. LOAD-STRAIN TRACING FOR PRESTRESS CYCLES**
FIGURE 4a. EFFECT OF PRESTRESS CYCLES ON FATIGUE CRACK PROPAGATION, 0.100 INCHES THICK 2024-T351 ALUMINUM ALLOY
2024-T351 Al
A = 4" x 0.080"
R = 1/10

FIGURE 4b. EFFECT OF PRESTRESS CYCLES ON FATIGUE CRACK PROPAGATION,
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(a) ACCUMULATED DEFORMATION

(b) CYCLIC DEFORMATION RANGE

FIG. (7a) MOIRE PATTERNS NEAR A CRACK TIP - 2024-T351 ALUMINUM ALLOY, $\Delta K = 28.3 \text{ ksi \sqrt{in.}}$, $K_{max} = 32.3 \text{ ksi \sqrt{in.}}$. 
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FIGURE 11a. STRAIN RANGE DISTRIBUTION AHEAD OF A CRACK TIP FOR 0.25 INCHES THICK 2024-T351 ALUMINUM ALLOY AT R=1/3
2024-T351 Al
A = 4" x 0.25"
R = 1/3

\( K_{\text{max}} \text{ (KSI \sqrt{IN}) SLOPE} \)

- \( 25.9 \quad -0.80 \)
- \( 23.2 \quad -0.76 \)
- \( 21.1 \quad -0.80 \)
- \( 18.7 \quad -0.75 \)
- \( 15.6 \quad -0.70 \)
- \( 13.4 \quad -0.60 \)
- \( 11.6 \quad -0.50 \)

**FIGURE 11b.** MAXIMUM STRAIN DISTRIBUTION AHEAD OF A CRACK TIP FOR 0.25 INCHES THICK 2024-T351 ALUMINUM ALLOY AT R=1/3
FIGURE 12. COMPARISON OF THE CALCULATED AND MEASURED STRAIN RANGE DISTRIBUTION AHEAD OF A CRACK TIP FOR 0.050 INCHES THICK 2024-T351 ALUMINUM ALLOY AT R=1/10
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Figure 17a. Static plastic zone size versus stress intensity factor range for 0.050 inches thick 2024-T351 aluminum alloy.
2024-T351 Al
A = 4" x 0.25"

- R = 1/10
- R = 1/3
--- CALCULATED

FIGURE 18. STATIC PLASTIC ZONE SIZE VERSUS STRESS INTENSITY FACTOR RANGE FOR 0.25 INCHES THICK 2024-T351 ALUMINUM ALLOY
FIGURE 19. MAXIMUM STATIC PLASTIC ZONE SIZE VERSUS THE MAXIMUM STRESS INTENSITY FACTOR FOR 2024-T351 ALUMINUM ALLOY
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2024-T351 Al
$A = 4'' \times 0.25''$
$R = 1/3$
Figure 20f. Relative crack opening displacement range at r=0.04 inches versus the stress intensity factor range for 0.25 inches thick 2024-T351 aluminum alloy.
FIGURE 21a. RELATIVE MAXIMUM CRACK OPENING DISPLACEMENT VERSUS THE DISTANCE FROM A CRACK TIP FOR 0.050 INCHES THICK 2024-T351 ALUMINUM ALLOY AT R=1/10
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FIGURE 23b. FATIGUE CRACK PROPAGATION RATE AND STATIC PLASTIC ZONE SIZE VERSUS STRESS INTENSITY FACTOR RANGE FOR 0.25 INCHES THICK 2024-T351 ALUMINUM ALLOY AT R=1/10

2024-T351 Al
A = 4" x 0.25"
R = 1/10

da

dN

F  

r_p(s)
FIGURE 24a. FATIGUE CRACK PROPAGATION RATE AND STATIC PLASTIC ZONE SIZE VERSUS STRESS INTENSITY FACTOR RANGE FOR 0.050 INCHES THICK 2024-T351 ALUMINUM ALLOY AT R=1/3
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