EFFECT OF SPECIMEN THICKNESS
ON FATIGUE-CRACK-GROWTH BEHAVIOR
AND FRACTURE TOUGHNESS OF 7075-T6
AND 7178-T6 ALUMINUM ALLOYS

by C. Michael Hudson and J. C. Newman, Jr.

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Stress-intensity methods were used to analyze the experimental results. For a given thickness and value of \( R \), the rate of fatigue-crack growth was essentially a single-valued function of the stress-intensity range for 7075-T6 and 7178-T6. An empirical equation developed by Forman, Kearney, and Engle (in Trans. ASME, Ser. D: J. Basic Eng., vol. 89, no. 3, Sept. 1967) fit the 7075-T6 and 7178-T6 crack-growth data reasonably well.
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SUMMARY

A study was made to determine the effects of specimen thickness on fatigue-crack growth and fracture behavior of 7075-T6 and 7178-T6 aluminum-alloy sheet and plate. Specimen thicknesses ranged from 5.1 to 12.7 mm (0.20 to 0.50 in.) for 7075-T6 and from 1.3 to 6.4 mm (0.05 to 0.25 in.) for 7178-T6. The stress ratios R used in the crack-growth experiments were 0.02 and 0.50. For 7075-T6, specimen thickness had relatively little effect on fatigue-crack growth. However, the fracture toughness of the thickest gage of 7075-T6 was about two-thirds of the fracture toughness of the thinner gages of 7075-T6. For 7178-T6, fatigue cracks generally grew somewhat faster in the thicker gages than in the thinnest gage. The fracture toughness of the thickest gage of 7178-T6 was about two-thirds of the fracture toughness of the thinner gages of 7178-T6.

Stress-intensity methods were used to analyze the experimental results. For a given thickness and value of R, the rate of fatigue-crack growth was essentially a single-valued function of the stress-intensity range for 7075-T6 and 7178-T6. An empirical equation developed by Forman, Kearney, and Engle (in Trans. ASME, Ser. D: J. Basic Eng., vol. 89, no. 3, Sept. 1967) fit the 7075-T6 and 7178-T6 crack-growth data reasonably well.

INTRODUCTION

Fatigue cracks of various sizes have been discovered during the service life of many aircraft structures. As a result, the predictions of fatigue-crack-growth rates and fracture toughness of parts containing fatigue cracks have become of considerable interest to aircraft designers and operators. In order to make such predictions, the effects of a wide range of parameters must be understood. Many of these parameters, such as component configuration, stress ratio, loading sequence, and environment, have already been investigated at NASA Langley Research Center and are reported in references 1 to 7. However, relatively little research has been conducted on the effects of
material thickness on fatigue behavior. Consequently, a series of axial-load fatigue-crack-growth and fracture-toughness experiments were conducted on 7075-T6 and 7178-T6 aluminum-alloy specimens ranging in thickness from 5.1 to 12.7 mm (0.20 to 0.50 in.) and from 1.3 to 6.4 mm (0.05 to 0.25 in.), respectively. These materials were selected because of their frequent use in aircraft construction.

Stress-intensity methods were used to analyze the data because these methods have shown great promise for predicting fatigue-crack propagation and fracture in complex structures. For example, Poe (ref. 8) showed that fatigue-crack growth in stiffened panels can be predicted from stress-intensity parameters and the data from tests of simple sheet specimens.

An empirical equation developed by Forman, Kearney, and Engle (ref. 9) was fitted by least-squares techniques to the fatigue-crack-propagation data. This equation fit the fatigue-crack-growth data generated in a previous study of stress-ratio effects reasonably well (ref. 3).

**SYMBOLS**

The units used for the physical quantities defined in this paper are given in both the International System of Units (SI) and the U.S. Customary Units. The measurements and calculations were made in the U.S. Customary Units. Factors relating the two systems are given in reference 10 and those used in the present investigation are presented in appendix A.

- \(a\) half-length of a central symmetrical crack, mm (in.)
- \(a_i\) half-length of crack at start of a fracture-toughness test, mm (in.)
- \(C\) constant in fatigue-crack-growth equation
- \(da/dN\) rate of fatigue-crack growth, nm/cycle (in./cycle)
- \(E\) Young's modulus of elasticity, GN/m\(^2\) (psi)
- \(e\) elongation in 51-mm (2-in.) gage length, percent
- \(K_{cn}\) critical stress-intensity factor, MN/m\(^{3/2}\) (psi-in\(^{1/2}\))
- \(K_{max}\) maximum stress-intensity factor, MN/m\(^{3/2}\) (psi-in\(^{1/2}\))
K_{min}  \quad \text{minimum stress-intensity factor, } \text{MN/m}^{3/2} \left(\text{psi-in}^{1/2}\right)

\Delta K  \quad \text{stress-intensity-factor range, } \text{MN/m}^{3/2} \left(\text{psi-in}^{1/2}\right)

N  \quad \text{number of load cycles}

n  \quad \text{exponent in fatigue-crack-growth equation}

P_a  \quad \text{amplitude of load applied in a cycle, } N \text{ (lbf)}

P_f  \quad \text{maximum load applied to specimen during fracture-toughness test, } N \text{ (lbf)}

P_m  \quad \text{mean load applied in a cycle, } N \text{ (lbf)}

P_{max}  \quad \text{maximum load applied in a cycle, } P_m + P_a, N \text{ (lbf)}

P_{min}  \quad \text{minimum load applied in a cycle, } P_m - P_a, N \text{ (lbf)}

R  \quad \text{ratio of minimum stress to maximum stress}

S_a  \quad \text{alternating gross stress, } P_a/wt, \text{MN/m}^2 \text{ (psi or ksi)}

S_f  \quad \text{maximum gross stress applied to specimen during fracture-toughness test, } P_f/wt, \text{MN/m}^2 \text{ (psi)}

S_m  \quad \text{mean gross stress, } P_m/wt, \text{MN/m}^2 \text{ (psi or ksi)}

S_{max}  \quad \text{maximum gross stress, } P_{max}/wt, \text{MN/m}^2 \text{ (psi)}

S_{min}  \quad \text{minimum gross stress, } P_{min}/wt, \text{MN/m}^2 \text{ (psi)}

t  \quad \text{specimen thickness, mm (in.)}

w  \quad \text{specimen width, mm (in.)}

\alpha  \quad \text{secant correction factor for stress intensity in a finite width panel, } \sqrt{\sec \frac{\pi a}{w}}

\sigma_u  \quad \text{ultimate tensile strength, } \text{MN/m}^2 \text{ (ksi)}

\sigma_y  \quad \text{yield strength (0.2-percent offset), } \text{MN/m}^2 \text{ (ksi)}
SPECIMENS, TESTS, AND PROCEDURES

Specimens

Through-crack test specimens were made from three thicknesses each of 7075-T6 and 7178-T6 aluminum alloys. The thicknesses and tensile properties of these alloys are listed in table I. The tensile specimens used to obtain these properties met ASTM Standards (ref. 11). The nominal chemical compositions of the two alloys are shown in table II.

The specimen configuration used in both the crack-propagation and fracture-toughness tests is shown in figure 1. These specimens were 292 mm (11.5 in.) wide and 889 mm (35.0 in.) long. The longitudinal axes of all specimens were parallel to the rolling direction of the material. A notch 2.54 mm (0.10 in.) long by 0.25 mm (0.01 in.) wide was cut into the center of each specimen by use of an electrical discharge machining process. The heat-affected zone resulting from this process is less than 0.25 mm (0.01 in.) wide. Consequently, after crack initiation, all of the material through which the fatigue crack propagates is unaltered by the cutting process.

A reference grid (ref. 12) was photographically printed on the surface of the specimen for crack-propagation monitoring. The spacing between grid lines was 1.3 mm (0.050 in.). Metallographic examination and tensile tests conducted on 7075-T6 specimens bearing the grid indicated no detrimental effect on the material.

Testing Machines

Three axial-load fatigue-testing machines were employed in this investigation. The capabilities of these machines are listed in the following table:

<table>
<thead>
<tr>
<th>Machine type</th>
<th>Maximum load capacity</th>
<th>Operating frequency used</th>
<th>Machine described in —</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>kN</td>
<td>lbf</td>
<td>Hz</td>
</tr>
<tr>
<td>Subresonant</td>
<td>89</td>
<td>20 000</td>
<td>30</td>
</tr>
<tr>
<td>Hydraulic</td>
<td>1334</td>
<td>300 000</td>
<td>1 to 5</td>
</tr>
<tr>
<td>Combination:</td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>As subresonant unit</td>
<td>467</td>
<td>105 000</td>
<td>14</td>
</tr>
<tr>
<td>As hydraulic unit</td>
<td>587</td>
<td>132 000</td>
<td>0.7 to 1.0</td>
</tr>
</tbody>
</table>

The 1334-kN (300 000-lbf) tester described in the preceding table was also used for fracture-toughness tests requiring loads in excess of 534 kN (120 000 lbf). A hydraulic
axial-load universal testing machine was used for fracture-toughness tests requiring lower loads. This universal machine had a load capacity of 534 kN (120 000 lbf).

**Test Procedure**

Axial-load fatigue-crack-propagation experiments were conducted at stress ratios \( R \) of 0.02 and 0.50. The maximum gross stresses in these experiments ranged from 69 to 276 MN/m\(^2\) (10 to 40 ksi) for 7075-T6 and from 52 to 155 MN/m\(^2\) (7.5 to 22.5 ksi) for 7178-T6. The alternating and mean loads were kept constant throughout each test. The fatigue-crack-growth data were obtained by observing crack growth through 10 power microscopes. The number of cycles required to propagate the crack to each grid line was recorded so that crack-propagation rates could be determined.

Fracture-toughness data were obtained two ways. Most of these data came from standard toughness tests in which fatigue-cracked specimens were monotonically loaded to failure at a load rate of 2.2 kN/sec (30 000 lbf/min). The remainder of these data came from fatigue-crack-propagation tests which were continued up to specimen failure. In these tests, the maximum load in the fatigue-crack-propagation test was assumed to be the load at failure.

When a centrally cracked sheet specimen is loaded in axial tension, transverse compressive stresses are generated near the crack surface (ref. 15). These compressive stresses can buckle thin specimens out of the plane of the sheet near the crack. The increase in stress-intensity factor due to this buckling is difficult to calculate; consequently the thinner gage specimens \((t = 5.1 \text{ mm (0.20 in.) for 7075-T6 and } t = 1.3 \text{ and } 4.1 \text{ mm (0.05 and 0.16 in.) for 7178-T6)}\) were clamped between oiled guide plates (ref. 16) to restrain buckling. The thicker specimens did not buckle; therefore guide plates were not used.

**RESULTS AND DISCUSSION**

**Fatigue-Crack-Growth Experiments**

The results of the fatigue-crack-growth experiments on the 7075-T6 and 7178-T6 specimens are presented in table III. This table gives the average number of cycles required for a through-crack to propagate from a half-length of 2.54 mm (0.10 in.) to the listed half-lengths. Fatigue-crack-growth rates were determined graphically from crack-growth curves which were faired through the data of table III.

The fatigue-crack-growth curves for the 7075-T6 specimens of different thicknesses are presented in figure 2. At eight of nine stress levels, fatigue cracks propagated fastest in the 5.1-mm-thick (0.20-in.) 7075-T6 specimens. However, for a given stress level,
the ratio of the maximum to the minimum number of cycles required to reach a given crack length never exceeded 1.7, thereby indicating a relatively small thickness effect.

The fatigue-crack-growth curves for the 7178-T6 specimens are presented in figure 3. At six of seven stress levels, fatigue cracks propagated slowest in the 1.3-mm-thick (0.05-in.) 7178-T6 specimens. For a given stress level, the ratio of the maximum to the minimum number of cycles required to reach a given crack length never exceeded 2.7, thereby indicating a moderate thickness effect.

Fatigue-crack-growth curves for 7075-T6 and 7178-T6 specimens of about the same thickness (5.1 and 4.1 mm (0.20 and 0.16 in.), respectively) and tested at the same values of $S_{\text{max}}$ and $R$ are shown in figure 4. For a given stress level, the ratio of the maximum to the minimum number of cycles required to reach a given crack length never exceeded 1.7. In two instances fatigue cracks grew fastest in 7075-T6, and in the two other instances, fastest in 7178-T6. Thus, in the thickness range of 4 to 5 mm (0.16 to 0.20 in.), the two alloys appear about equally resistant to fatigue-crack propagation.

Inspection of the fracture surfaces of the specimens (fig. 5, for example) indicated that intermittent bursts of crack growth (referred to hereinafter as "pop-in" (ref. 17)) occurred in the interior of specimens having thicknesses as small as 4.1 mm (0.16 in.). The dark areas in figure 5 indicate pop-in. The light areas indicate normal, microscopic fatigue-crack growth. The reason for this pop-in is not understood at this time.

The fatigue-crack-growth data in table III were analyzed by using stress-intensity methods (see appendix C). For a given thickness and value of $R$, the rate of fatigue-crack growth was a single-valued function of the stress-intensity range for 7075-T6 and 7178-T6 (fig. 6).

An empirical fatigue-crack-growth equation developed by Forman, Kearney, and Engle (ref. 9) was fitted to the test data. This equation has the form

$$\frac{da}{dN} = \frac{C(\Delta K)^n}{(1 - R)K_{\text{cn}} - \Delta K}$$

(The symbol $K_{\text{cn}}$ is denoted by $K_c$ in ref. 9.)

The empirical constants $C$ and $n$ were determined by using least-squares techniques to fit the equation to the data. When these constants were determined in SI Units, $\Delta K$ and $K_{\text{cn}}$ were given in MN/m$^{3/2}$ and $da/dN$ was given in nm/cycle. When $C$ and $n$ were computed in U.S. Customary Units, $\Delta K$ and $K_{\text{cn}}$ were given in psi-in$^{1/2}$ and $da/dN$ was given in in./cycle. The values of $C$ and $n$ determined for the different thicknesses are listed in the following table:
Equation (1) fit the test data reasonably well.

### Fracture-Toughness Experiments

The results of the fracture-toughness experiments on the 7075-T6 and 7178-T6 specimens are listed in table IV. This table gives the half-length of the crack at the start of the fracture-toughness test $a_i$, the maximum gross stress applied to the test specimen during the fracture-toughness test $S_f$, and the critical stress-intensity factor $K_{cn}$. This factor was calculated by using the equation

$$K_{cn} = \left(\frac{P_f}{wt}\right)^{1/n}a_i$$  

(2)

where $\alpha$ is given in appendix C.

The values of $K_{cn}$ for the various thicknesses are plotted against $a_i$ in figure 7. Analysis of the data in figure 7 indicates that the fracture toughness of the 12.7-mm-thick (0.50-in.) 7075-T6 was, on the average, about two-thirds of the fracture toughness of the thinner gages of 7075-T6. The average fracture toughness of the 6.4-mm-thick (0.25-in.) 7178-T6 was about two-thirds of the fracture toughness of the thinner gages of 7178-T6. Figure 7 also indicates that $K_{cn}$ increased with increasing crack length. A similar variation of $K_{cn}$ with crack length occurred in tests on through-cracked 2014-T6 and 2219-T87 aluminum alloys (ref. 18).

Values of $K_{cn}$ for 7075-T6 and 7178-T6 specimens of about the same thickness (5.1 and 4.1 mm (0.20 and 0.16 in.)) are plotted against $a_i$ in figure 8. The fracture toughness of 7075-T6 was about 20 percent higher than the fracture toughness of 7178-T6.

### CONCLUSIONS

A study was made to determine the effects of specimen thickness on fatigue-crack growth and fracture behavior of 7075-T6 and 7178-T6 aluminum-alloy sheet and plate.
The 7075-T6 specimens had thicknesses of 5.1, 9.7, and 12.7 mm (0.20, 0.38, and 0.50 in.); the 7178-T6 specimens had thicknesses of 1.3, 4.1, and 6.4 mm (0.05, 0.16, and 0.25 in.). The stress ratios $R$ (ratio of the minimum stress to the maximum stress) used in these experiments were 0.02 and 0.50. The experimental results were analyzed by using stress-intensity methods, and an empirical equation was fitted to the data. The following conclusions can be drawn from this study:

1. For 7075-T6, material thickness had relatively little effect on fatigue-crack growth. The fracture toughness of the 12.7-mm-thick (0.50-in.) 7075-T6 was about two-thirds of the fracture toughness of the thinner gages of 7075-T6.

2. For 7178-T6, fatigue cracks generally grew somewhat faster in the thicker gages than in the thinnest gage. The fracture toughness of the 6.4-mm-thick (0.25-in.) 7178-T6 was about two-thirds of the fracture toughness of the thinner gages of 7178-T6.

3. For a nominal thickness of 5.1 mm (0.20 in.), fatigue cracks in 7075-T6 and 7178-T6 propagated to a given crack length in approximately the same number of cycles. For the same nominal thickness, the fracture toughness of 7075-T6 was about 20 percent higher than the fracture toughness of 7178-T6.

4. During the fatigue-crack-growth tests, intermittent bursts of crack growth (pop-in) occurred in the interior of the 7075-T6 and 7178-T6 specimens having thicknesses $\geq 4.1$ mm (0.16 in.). The reason for this pop-in is not understood at present.


6. For a given thickness and value of $R$, the rate of fatigue-crack growth was essentially a single-valued function of the stress-intensity range for 7075-T6 and 7178-T6.

Langley Research Center,
National Aeronautics and Space Administration,
APPENDIX A

CONVERSION OF SI UNITS TO U.S. CUSTOMARY UNITS

The International System of Units (SI) was adopted by the Eleventh General Conference on Weights and Measures held in Paris in 1960 (ref. 10). Conversion factors required for units used herein are given in the following table:

<table>
<thead>
<tr>
<th>Physical quantity</th>
<th>SI Unit (a)</th>
<th>Conversion factor (b)</th>
<th>U.S. Customary Unit</th>
</tr>
</thead>
<tbody>
<tr>
<td>Force</td>
<td>newtons (N)</td>
<td>0.2248</td>
<td>lbf</td>
</tr>
<tr>
<td>Length</td>
<td>meters (m)</td>
<td>.3937 x 10^2</td>
<td>in.</td>
</tr>
<tr>
<td>Stress</td>
<td>newtons per sq meter (N/m^2)</td>
<td>.145 x 10^-6</td>
<td>ksi = 10^3 lbf/in^2</td>
</tr>
<tr>
<td>Stress intensity</td>
<td>newtons per meter^3/2 (N/m^3/2)</td>
<td>.9099 x 10^-6</td>
<td>ksi-in^1/2</td>
</tr>
<tr>
<td>Frequency</td>
<td>hertz (Hz)</td>
<td>60</td>
<td>cpm</td>
</tr>
</tbody>
</table>

^prefixes and symbols to indicate multiples of units are as follows:

<table>
<thead>
<tr>
<th>Multiple</th>
<th>Prefix</th>
<th>Symbol</th>
</tr>
</thead>
<tbody>
<tr>
<td>10^-9</td>
<td>nano</td>
<td>n</td>
</tr>
<tr>
<td>10^-3</td>
<td>milli</td>
<td>m</td>
</tr>
<tr>
<td>10^3</td>
<td>kilo</td>
<td>k</td>
</tr>
<tr>
<td>10^6</td>
<td>mega</td>
<td>M</td>
</tr>
<tr>
<td>10^9</td>
<td>giga</td>
<td>G</td>
</tr>
</tbody>
</table>

^Multiply value given in SI Unit by conversion factor to obtain equivalent in U.S Customary Unit.
APPENDIX B

DESCRIPTION OF 1334-kN (300 000-lbf) FATIGUE TESTER

The 1334-kN (300 000-lbf) machine is an analog closed-loop servohydraulic fatigue-testing system. A schematic diagram of the loading system is shown in figure 9. To use this system, the operator first sets in the desired mean load by adjusting the mean-load potentiometer. Then the desired alternating load is set by adjusting the alternating-load potentiometer (which controls the amplitude of the function generator signal).

The voltages from the mean-load potentiometer and the function generator are combined to form a command signal which is fed into the servoloop summing point. The voltage from a transducer — either the load cell or the linearly variable displacement transducer (LVDT) — is also fed into this summing point. The command and transducer voltages are summed and suitably amplified to form a signal which drives the servovalve. This servovalve directs oil to the appropriate side of the hydraulic cylinder to obtain the commanded load. Load repeatability for this testing system is ±0.5 percent of the applied load.

Loads are monitored by comparing on an oscilloscope the output voltage from the load cell (or LVDT) with an adjustable bias voltage which corresponds to the desired load level for the test. When the sum of these voltages is zero, the desired load is on the test specimen. (This comparison is made at both the maximum and minimum loads in the cycle.) The accuracy of this monitoring system is better than ±0.1 percent of full scale.
APPENDIX C

FATIGUE-CRACK-GROWTH ANALYSIS

The fatigue-crack-growth data were correlated by the stress-intensity methods. Paris (ref. 19) hypothesized that the rate of fatigue-crack growth was a function of the stress-intensity range; that is

$$\frac{da}{dN} = f(\Delta K) \quad (C1)$$

where

$$\Delta K = K_{\text{max}} - K_{\text{min}} \quad (C2)$$

For centrally cracked specimens subjected to a uniformly distributed axial load

$$K_{\text{max}} = \alpha s_{\text{max}} \sqrt{a\pi} \quad (C3)$$

and

$$K_{\text{min}} = \alpha s_{\text{min}} \sqrt{a\pi} \quad (C4)$$

The term $\alpha$ is a factor intended to correct for the finite width of the specimen (ref. 20) and is given by

$$\alpha = \sqrt{\sec \frac{\pi a}{w}} \quad (C5)$$
REFERENCES


TABLE I.- AVERAGE TENSILE PROPERTIES OF ALUMINUM ALLOYS TESTED

<table>
<thead>
<tr>
<th>t</th>
<th>σ_u</th>
<th>σ_y</th>
<th>e, %</th>
<th>E</th>
<th>No. of tests</th>
</tr>
</thead>
<tbody>
<tr>
<td>mm</td>
<td>in.</td>
<td>MN/m²</td>
<td>ksi</td>
<td>MN/m²</td>
<td>ksi</td>
</tr>
<tr>
<td>5.1</td>
<td>0.20</td>
<td>595</td>
<td>86.3</td>
<td>542</td>
<td>78.6</td>
</tr>
<tr>
<td>9.7</td>
<td>.38</td>
<td>574</td>
<td>83.3</td>
<td>528</td>
<td>76.6</td>
</tr>
<tr>
<td>12.7</td>
<td>.50</td>
<td>598</td>
<td>86.7</td>
<td>551</td>
<td>79.9</td>
</tr>
</tbody>
</table>

7178-T6

<table>
<thead>
<tr>
<th>t</th>
<th>σ_u</th>
<th>σ_y</th>
<th>e, %</th>
<th>E</th>
<th>No. of tests</th>
</tr>
</thead>
<tbody>
<tr>
<td>mm</td>
<td>in.</td>
<td>MN/m²</td>
<td>ksi</td>
<td>MN/m²</td>
<td>ksi</td>
</tr>
<tr>
<td>1.3</td>
<td>0.05</td>
<td>608</td>
<td>88.2</td>
<td>564</td>
<td>81.8</td>
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<tr>
<td>4.1</td>
<td>.16</td>
<td>624</td>
<td>90.5</td>
<td>586</td>
<td>85.0</td>
</tr>
<tr>
<td>6.4</td>
<td>.25</td>
<td>622</td>
<td>90.2</td>
<td>593</td>
<td>86.0</td>
</tr>
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</table>

TABLE II.- NOMINAL CHEMICAL COMPOSITIONS OF ALUMINUM ALLOYS TESTED

<table>
<thead>
<tr>
<th>Aluminum alloy</th>
<th>t</th>
<th>Element, percent by weight</th>
</tr>
</thead>
<tbody>
<tr>
<td></td>
<td>mm</td>
<td>in.</td>
</tr>
<tr>
<td>7075-T6</td>
<td></td>
<td></td>
</tr>
<tr>
<td>5.1</td>
<td>0.20</td>
<td>0.11</td>
</tr>
<tr>
<td>9.7</td>
<td>.38</td>
<td>.11</td>
</tr>
<tr>
<td>12.7</td>
<td>.50</td>
<td>.11</td>
</tr>
<tr>
<td>7178-T6</td>
<td></td>
<td></td>
</tr>
<tr>
<td>1.3</td>
<td>0.05</td>
<td>0.11</td>
</tr>
<tr>
<td>4.1</td>
<td>.16</td>
<td>.08</td>
</tr>
<tr>
<td>6.4</td>
<td>.25</td>
<td>.08</td>
</tr>
</tbody>
</table>

Bal. = Balanced
### TABLE III - AVERAGE NUMBER OF CYCLES REQUIRED TO EXTEND CRACKS FROM A HALF-LENGTH OF 2.54 mm (0.10 in.) TO VARIOUS LENGTHS

(a) 7075-T6

<table>
<thead>
<tr>
<th>t (hr)</th>
<th>S_m (kN/m²)</th>
<th>S_a (kN/m²)</th>
<th>Loading frequency</th>
<th>R</th>
<th>3.81 mm (0.15 in.)</th>
<th>5.08 mm (0.20 in.)</th>
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<th>25.40 mm (1.00 in.)</th>
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| 0.38  | 9.7 | 140.7 | 20.40 | 135.1 | 19.60 | 1.0 | 60 | 0.02 | 327 | 492 |
|-------|-----|-------|-------|-------|-------|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|
|       | 105.5 | 15.30 | 101.4 | 14.70 | 1.0 | 60 | 0.02 | 705 | 1050 |
|       | 70.3 | 10.20 | 67.6 | 9.80 | 1.0 | 60 | 0.02 | 2200 | 3720 |
|       | 52.7 | 7.65 | 50.7 | 7.35 | 1.0 | 60 | 0.02 | 5200 | 8300 |
|       | 172.4 | 25.00 | 57.2 | 8.30 | 1.0 | 60 | 0.50 | 1270 | 1810 |
|       | 137.9 | 20.00 | 46.2 | 6.70 | 1.0 | 60 | 0.50 | 3650 | 5650 |
|       | 103.4 | 15.00 | 34.5 | 5.00 | 1.0 | 60 | 0.50 | 6100 | 11000 |
|       | 68.8 | 9.975 | 22.9 | 3.325 | 3.0 | 180 | 0.50 | 52000 | 77000 |

| 0.50  | 12.7 | 105.5 | 15.30 | 101.4 | 14.70 | 1.0 | 60 | 0.02 | 690 | 1140 |
|-------|-----|-------|-------|-------|-------|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|-----|
|       | 70.3 | 10.20 | 67.6 | 9.80 | 1.0 | 60 | 0.02 | 2475 | 4300 |
|       | 52.7 | 7.65 | 50.7 | 7.35 | 1.0 | 60 | 0.02 | 5600 | 8725 |
|       | 35.2 | 5.10 | 33.8 | 4.90 | 1.0 | 60 | 0.02 | 9650 | 15250 |
|       | 172.4 | 25.00 | 57.2 | 8.30 | 1.0 | 60 | 0.50 | 12750 | 25000 |
|       | 137.9 | 20.00 | 46.2 | 6.70 | 1.0 | 60 | 0.50 | 3560 | 6250 |
|       | 103.4 | 15.00 | 34.5 | 5.00 | 1.0 | 60 | 0.50 | 10000 | 18000 |
|       | 68.8 | 9.975 | 22.9 | 3.325 | 3.0 | 180 | 0.50 | 13000 | 22500 |

---

*Except as noted.

*Crack was initiated and propagated to a = 3.81 mm (0.15 in.) at S_max = 96.53 MN/m² (14 ksi) to expedite testing; cycles listed are number required to propagate crack from a = 5.08 mm (0.20 in.).

*Crack was initiated and propagated to a = 3.81 mm (0.15 in.) at S_max = 103.42 MN/m² (15 ksi) to expedite testing; cycles listed are number required to propagate crack from a = 5.08 mm (0.20 in.).
### TABLE III - AVERAGE NUMBER OF CYCLES REQUIRED TO EXTEND CRACKS FROM A HALF-LENGTH OF 2.54 mm (0.10 in.) TO VARIOUS LENGTHS - Concluded

(b) 7178-T6

<table>
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<th>t (hr)</th>
<th>$S_m$ (MN/m²)</th>
<th>$S_a$ (MN/m²)</th>
<th>Loading frequency (Hz)</th>
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<th>12.70 mm (0.50 in.)</th>
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<th>20.32 mm (0.80 in.)</th>
<th>22.86 mm (1.00 in.)</th>
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\*Except as noted.
\*Crack was initiated and propagated to \(a = 3.05\) mm (0.12 in.) at \(S_{\text{max}} = 68.95\) MN/m² (10 ksl) to expedite testing; cycles listed are number required to propagate crack from \(a = 5.08\) mm (0.20 in.).

\*Crack was initiated and propagated to \(a = 3.05\) mm (0.12 in.) at \(S_{\text{max}} = 86.18\) MN/m² (12 ksl) to expedite testing; cycles listed are number required to propagate crack from \(a = 5.08\) mm (0.20 in.).
TABLE IV. VALUES OF $K_{cn}$ FROM FRACTURE-TOUGHNESS TESTS

(a) 7075-T6

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<td>88</td>
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| 5.1    | 0.20  | 297   | 43.1 x $10^3$ | 37.4   | 34.0 x $10^3$ |
| 6.4    | .25   | 306   | 44.4      | 43.0   | 39.1       |
| 7.9    | .31   | 291   | 42.2      | 46.0   | 41.8       |
| 9.1    | .36   | 276   | 40.0      | 46.2   | 42.0       |
| 11.4   | .45   | 248   | 35.9      | 47.0   | 42.8       |
| 15.0   | .59   | 243   | 35.2      | 53.1   | 48.3       |
| 20.3   | .80   | 218   | 31.6      | 55.8   | 50.8       |
| 29.7   | 1.17  | 179   | 26.0      | 56.5   | 51.4       |
| 37.8   | 1.49  | 155   | 22.5      | 55.6   | 50.6       |
| 53.6   | 2.11  | 133   | 19.3      | 59.3   | 53.9       |
| 62.2   | 2.45  | 114   | 16.5      | 56.6   | 51.5       |
| 78.0   | 3.07  | 101   | 14.7      | 61.3   | 55.7       |

| 4.8    | 0.19  | 230   | 33.3 x $10^3$ | 28.2   | 25.6 x $10^3$ |
| 6.9    | .27   | 207   | 30.0      | 30.3   | 27.6       |
| 9.1    | .36   | 184   | 26.7      | 31.1   | 28.3       |
| 13.5   | .53   | 154   | 22.4      | 32.0   | 29.1       |
| 15.0   | .59   | 154   | 22.3      | 33.7   | 30.7       |
| 22.4   | .88   | 123   | 17.8      | 32.8   | 29.9       |
| 32.5   | 1.28  | 111   | 16.1      | 36.7   | 33.4       |
| 48.5   | 1.91  | 90    | 13.0      | 37.6   | 34.2       |

*Guide plates used.*
### TABLE IV.- VALUES OF $K_{cn}$ FROM FRACTURE-TOUGHNESS TESTS — Concluded

(b) 7178-T6

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<th>$K_{cn}$</th>
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*aGuide plates used.*
Figure 1.- Specimen configuration. All dimensions in mm (in.).
(a) $R = 0.02$.

Figure 2. - Fatigue-crack-growth curves for 7075-T6 specimens having different thicknesses.
Figure 2. - Concluded.
(a) $R = 0.02$.

Figure 3.- Fatigue-crack-growth curves for 7178-T6 specimens having different thicknesses.
Figure 4. Fatigue-crack-growth curves for 7075-T6 and 7178-T6 specimens of about the same thickness and tested at the same values of $S_{\text{max}}$ and $R$. 
Typical fatigue-crack-propagation

Typical "pop-in"

(a) 7075-T6.

Figure 5.- Fracture surfaces showing pop-in.
Typical fatigue-crack-propagation

Typical "pop-in"

(b) 7178-T6.

Figure 5.- Concluded.
Figure 6.- Variation of fatigue-crack-growth rate with $\Delta K$ for various thicknesses.
(b) 7178-T6.

Figure 6.- Concluded.
Figure 7.- Variation of $K_{cn}$ with $a_i$ for specimens having different thicknesses.
Figure 8.- Variation of $K_{cn}$ with $a_i$ for 7075-T6 and 7178-T6 specimens of about the same thickness.
Figure 9 - Schematic diagram of loading system for 1334-kN (300 000-lb) testing machine.

Function generator

Mean-load potentiometer

Alternating-load potentiometer

Calibrated load-readout unit

Servo loop

Summing point

Initial amplifiers

Power amplifier

Hydraulic pump

Load cell

Actuator

A. Grips
B. Specimen
C. Servovalve
D. LVDT

NASA-Langley, 1973
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