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STUDIES OF SILICON PN JUNCTION SOLAR CELLS

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FOREWORD

This report summarizes the major technical findings made in the research program at the University of Florida sponsored by NASA Lewis Research Center under Grant No. NSG-3018. This grant is a continuation of the support began under the same grant number on June 24, 1974. We report here our findings during the year ending September 1977.

The work to be reported came from the efforts of several key people. C.T. Sah suggested to NASA the possibility of placing this grant research at the University of Florida, and afterwards, together with the author, helped make the arrangements and formulate the technical plan of the research. Dr. Sah has contributed significantly to nearly every aspect of the research described in this report. After joining the faculty of the University of Florida in September, 1976, Arnost Neu-groschel became a key contributor to the research program. The technical collaboration of M.P. Godlewski and H.W. Brandhorst, Jr., of NASA Lewis Research Center greatly aided the completion of several phases of the research. During the tenure of the NASA grant, the Energy Research and Development Administration began supporting a separate but related research program at the University of Florida under the author's direction. This support aided the research reported here, and made possible a valuable and continuing technical interaction with J.G. Fossum of Sandia Laboratories (ERDA), Albuquerque, New Mexico. In addition to these senior investigators, two graduate students at the University of Florida participated in the research: P.J. Chen and S.C. Pao.
CHAPTER 1 INTRODUCTION

As Brandhorst first noted, the power conversion efficiency of p-substrate, p-n junction silicon solar cells is limited because $V_{OC}$ falls short of theoretical expectations by about 15 percent [1]. Thus, increasing $V_{OC}$ is important in raising efficiency, doubly so because the fill factor $FF$ increases with increasing $V_{OC}$ [2]. Indeed, as we have recently pointed out [3], essentially in agreement with the earlier projection of Brandhorst [1], if $V_{OC}$ were increased to 700 mV, AMO efficiencies in the range of 18 to 20 percent become possible.

1.1 Studies of Basic Mechanisms

Our previous work at Florida under NASA LeRC grant support has proposed methods, based on experiment [4], for determining which of the many fundamental mechanisms [5,6] that can occur in silicon devices are responsible for this discrepancy in $V_{OC}$ in any given solar cell. This work has demonstrated experimentally the important conclusion that the discrepancy originates in the excessive recombination current occurring in the degenerately doped emitter of the cell. In particular, for cells having a base resistivity of about 0.1 Ohm-cm, we demonstrated that the dark recombination current from the quasineutral emitter region exceeded that from the quasineutral base region by more than an order of magnitude [4]. We demonstrated further that both of these recombination currents exceeded the dark recombination current coming from the junction space-charge region. In these demonstrations, the dark cell was biased in the forward direction with a voltage $V$ near 600 mV, which is approximately the value of $V_{OC}$ produced in these cells. The conclusion implied by these experiments is that the large recombination current occurring
either in the bulk of the degenerately doped emitter or at its surface limits $V_{OC}$ and efficiency well below the projections of classical p-n junction theory [1]. This conclusion, based on experiment, contrasts with the recent theoretical suggestions of other authors that field-induced bandgap narrowing in the junction space-charge region is the dominant contributor to the observed performance, including $V_{OC}$ and efficiency, of p-substrate, diffused p-n junction silicon solar cells [7,8].

As yet, the physical mechanisms causing the anomalously large emitter recombination current have not been identified. However, considerable progress has recently been made. Our work at Florida has shown experimentally [9] the decided superiority of forward-biased admittance measurements over open-circuit-voltage-decay (OCVD) measurements in determining the dominant relaxation time [4], originally called the dominant natural mode [10], of the cell. The accurate determination of the dominant relaxation time is crucial to the determination of the minority-carrier charge stored in the quasi-neutral emitter and of the phenomenological carrier lifetime of the emitter, and thus to the determination of the physical mechanisms responsible for the excess emitter recombination current [4]. Furthermore, we have just reported a new method, based on the temperature dependence of the emitter recombination current [11], for determining the bandgap narrowing in the quasi-neutral emitter region as a function of the emitter doping concentration. Bandgap narrowing is believed to be one of the main physical mechanisms that contribute to the discrepancy in $V_{OC}$ [4,5,6]. This new experimental method developed at Florida, together with work just described by Lanyon and co-workers [12] and by Mertens and co-workers [13], may yield insight needed
to better define the fundamental physical mechanisms underlying the $V_{OC}$ discrepancy.

1.2 Studies of HLE Structures for Increasing $V_{OC}$

In the past year, although our work on fundamental physical mechanisms has continued to some degree, the main attention of the NASA LeRC sponsored research at Florida has shifted toward the demonstration that new device structures can substantially diminish the effects of whatever mechanisms in the emitter may be degrading $V_{OC}$, without requiring a detailed identification of these mechanisms. This new emphasis led to our proposal, from a theoretical standpoint [14,15], of the high-low-junction emitter (HLE) solar cell, in which the emitter region contains a high-low (H-L) junction. We have demonstrated experimentally [3,16] that this H-L junction can suppress the dark emitter recombination current to levels so low that the emitter current no longer presents an obstacle to the achievement of an open-circuit voltage of 700 mV. For test cells we have fabricated at Florida, the open-circuit voltage observed has been mainly determined by the dark base recombination current. Our experiments have shown that large values of $V_{OC}$ result in these HLE test cells, but that values approaching the theoretical limit of 700 mV will require improving the diffusion length of electrons in the quasineutral base region.

The maximum $V_{OC}$ seen to the present has been 640 mV for AMO illumination. This voltage, which compares with the 600 mV maximum normally seen in p-n junction silicon solar cells, has been measured on an oxide-charge-induced (OCI) HLE structure. In this structure, the H-L junction is formed in emitter material of non-degenerate doping concentration by an oxide-charge-induced electron accumulation layer [3,16]. Open-circuit voltages about 10 mV less have been observed in another type of HLE solar
cell, the diffused-HLE structure, in which impurity diffusion forms the 
H-L junction in the emitter region.

1.3 Structure of This Report

This report consists of three parts. In the first part, which is 
Chapter 2, we develop, from a theoretical standpoint, the modifications 
of the basic Shockley equations that result from the random and non-
random spatial variation of the chemical composition of a semiconductor. 
These modifications underlie the existence of the excessive emitter 
recombination current that limits the $V_{OC}$ of solar cells. The discussion 
includes a more rigorous treatment of quasi-fields than has appeared 
before. In the second part, work related to the measurement of parameters 
is presented. Chapter 3 treats the measurement of series resistance. 
Chapter 4 deals with the measurement of the base diffusion length and 
presents two methods for establishing the energy bandgap narrowing in 
the heavily-doped emitter region. The discussion in Chapter 5 relates to 
corrections that can be important in the application of one of these 
methods to small test cells. In the third part, we briefly describe 
OCI-HLE test cells which exhibit a considerably higher $V_{OC}$ than was 
previously seen in n-on-p solar cells.

Insofar as is possible, each of the Chapters are written as in-
dependent units to enable the reader with a main interest in a later chap-
ter to be able to read that one directly without first having to become 
familiar with the earlier chapters.
REFERENCES FOR CHAPTER I


PART I  BASIC MECHANISMS DEGRADING $V_{OC}$: THEORY
CHAP. 2 CARRIER GENERATION, RECOMBINATION, TRAPPING AND TRANSPORT IN SEMICONDUCTORS WITH POSITION DEPENDENT COMPOSITION

2.1 Introduction

Semiconductor structures and materials with position dependent chemical composition have been of central interest in device applications. For structures with abrupt changes in but otherwise uniform chemical compositions, such as the heterojunction interfaces of a Schottky barrier or a MOS capacitance, the mathematical analyses of carrier generation, recombination, trapping and transport (GRIT) are based on ideal and position independent energy band models for each material region [1,2]. The solutions of the device characteristics are then obtained by matching the solutions of each of these uniform regions at their interfaces. These solutions are obtained from the following Shockley equations which were first used to get the characteristics of p-n junction diodes and transistors [3,4]. These were generalized to include the most important generation, recombination, trapping and tunneling mechanisms [5] and high-frequency effect [6] near thermal equilibrium as well as hot carrier effects on transport [7].

\[
\begin{align*}
\dot{J}_N &= q \phi_N V_N + q \mu_N N E \\
&= - q \mu_N N V_N \\
\dot{J}_P &= - q \phi_P V_P + q \mu_P P E \\
&= - q \mu_P P V_P \\
0 &= V \cdot \dot{J}_N + I_{SS}
\end{align*}
\]
\[ 0 = -\nabla \cdot \mathbf{j}_P + I_{SS} \]  
\[ \nabla \cdot \mathbf{E} = -V^2\psi_I = \rho/\varepsilon = (q/\varepsilon)(P - N + N_{DD} - N_{AA} - N_T) \]  

Here, the coefficients which characterize the material properties and the variables have the usual meaning [5].

When the chemical composition of the material or device structure varies continuously and strongly in a regular or random fashion, the Shockley equations need to be modified. A number of fairly distinct situations can be treated and discussed below.

(1) If the composition varies on an atomic scale but the atomic perturbations are widely separated (concentration less than about $10^{17}$ perturbations/cm$^3$), we have the classical case which is contained in the Shockley equations: the mobilities, $\mu_n$ and $\mu_p$, the diffusivities, $D_n$ and $D_p$, and the generation-recombination-trapping rates all vary with position. These material parameters are macroscopic averages taken in regions of sufficiently large volumes to contain many ($\sim 10^6$) particles (electrons, holes and atoms) to make the statistical fluctuation small.

(2) If the composition variations are on a macroscopic scale (concentrations of the foreign atomic species or physical defects are greater than about $10^{17}$ cm$^{-3}$ but smaller than about 1% of the host atomic density), the Shockley equations are still applicable. However, the ideal energy band model of allowed band of nonlocalized energy (Bloch) states or levels with sharp band edges separated by forbidden energy gaps must be modified. The modification comes about from the high concentrations of the spatially random atomic perturbations which shift
the band states of the ideal model into the energy gaps, thereby producing a
distribution of semi-localized energy gap states both in energy and in position.
The deep gap states, which are located far away (many $kT/q$ in energy) from
the ideal band edge positions, are nearly true bound states with negligible
carrier mobility. They can be accurately included in the generation–recombination–
trapping terms of the Shockley equations. However, the shallow gap states,
which are located in energy near the ideal band edges, may have appreciable
mobility due to the large spatial overlaps of the very extended wave functions.
These edge states will modify the macroscopic and thermal averages of the
mobilities and diffusivities in the Shockley equations, giving them position,
temperature and electric field dependences significantly different from those
of the ideal energy band model discussed in case (1). Here, the ideal or
sharp band edges lose their significance. To put the physics of this situation
into a simple perspective, Cohen [8] introduced the concept of mobility edge
to separate the semi-delocalized edge states from the nearly localized tail
states. The modifications of the macroscopic and thermal averages of the
mobility and diffusivity from the ideal case, then, come in two factors: the
integration limit is now the position of the mobility edge instead of the ideal
band edge and the density of state in the integrand is decreased due to the
splitting of the band states into the gap. Van Overstraeten [9] has provided
an analysis of the mobility in this situation. Although he included the tail
states in the mobility and diffusivity averages by assuming that they have the
same carrier scattering power or relaxation time as the band states, the errors
in computing the diffusion and drift current are probably small since the densities
of the electrons or holes trapped at these localized gap states are much smaller
than those in the edge and band states.
The effects of the distributed gap states on the carrier generation-recombination rates are well known. It was demonstrated as the principal source of excess current in tunnel diodes [10]. However, a demonstration of the application of this model for calculating the recombination rates is not available. An example will be given in the next section.

(3) If the composition variations are on a chemical scale and continuous over an extended region, basic modifications of the Shockley equations must be made. These modifications can be analyzed by separating out the random and the nonrandom components of the atomic potentials. The random component will produce the modifications to the Shockley equations similar to those discussed in cases (1) and (2). The nonrandom and continuously varying component will produce a corresponding spatial variation of energy levels of the band or Bloch states as well as the semi-localized gap (edge and tail) states. This produces major modifications of the Shockley equation, giving rise to new terms. The position dependent energy level was first analyzed by Kroemer [11] for device applications in graded or position-dependent band gap materials. He began with the effective mass equation and introduced the concept of quasi-electrical fields for electrons and holes.

Although the graded band gap materials have seen little application in conventional transistor devices in the past, it shows promise for improved efficiency in solar cells. Thus, an extension of the Shockley equation for this case will also be presented.

In the next two sections, we shall obtain quantitative results for the generation-recombination rates in the distributed gap states of case (2) and modifications of the Shockley equations in the graded band gap materials of case (3).
2.2 Generation-Recombination-Trapping in Distributed Gap States

For this case, the basic property of the gap states is that each imperfection center, whether having a point, pair or more complex geometry, is separated from its neighboring centers by a distance large compared with extent of the bound state wave function localized at the center. Thus, the transition rate of an electron (or hole) bound to one center to the adjacent centers is much smaller than the rate to the conduction or valence band and edge states. Then, the capture (or emission) rates of electrons (or holes) at each gap-state center can be added to give the total rate.

The above property of independent or noninteracting gap-state centers makes the arithmetics of the original Shockley-Read derivation [12] of the recombination kinetics directly applicable. If the centers were interacting, then additional rate equations for the bound-to-bound transitions between the centers or between states on one center must be included. This was done previously [5].

For the independent centers, the four electron and hole capture and emission rates between a band state and trap state located at $E_T$ are shown in Figure 1. The transition rates, averaged over the distribution in the conduction or valence band states are just those given by Shockley and Read:

\[
\begin{align*}
  a_T &= c_n(E_T)n[1-f_T(E_T)]\rho_T(E_T)dE_T \\
  b_T &= e_n(E_T)f_T(E_T)\rho_T(E_T)dE_T \\
  c_T &= c_p(E_T)\rho_T(E_T)dE_T \\
  d_T &= e_p(E_T)[1-f_p(E_T)]\rho_T(E_T)dE_T
\end{align*}
\]

In Equation (6), $c_n$ is the macroscopic averaged capture probability rate ($\#$/cm$^3$-sec) of electrons in the conduction band by the gap state located at
an energy of $E_T$. In Equation (7), $e_n(E_T)$ is the corresponding emission rate of electrons. $n$ is the concentration of electron in the conduction band.

$f_T(E_T)$ is the fraction of the gap states occupied by electrons at the energy $E_T$ and $\rho(E_T)dE_T$ is the concentration of the gap states located in the energy range $E_T$ and $E_T+dE_T$. The coefficients in Equations (8) and (9) has the corresponding meanings for holes.

Although the rate equations (6) to (9) are used by Shockley and Read specifically for the thermal transition process, they are also applicable to optical and Auger-impact transition processes using extended definitions for the capture and emission rate coefficients, $c_n$, $c_p$, $e_n$, and $e_p$.[5]

The total capture and emission rates through all the gap-state centers can be obtained by summing or integrating Equations (7) to (9) over the distribution of the gap states in the band gap. However, since the gap states are independent, we may apply the condition of steady state to Equations (6) to (9) before making the summation over the gap states. This will be used as an illustration of the modifications of the recombination rates by the presence of a distribution of gap states. Thus, under the d.c. steady-state condition, we have

$$a_T - b_T - c_T + d_T = (\partial f_T)/\partial t = 0 \quad (10)$$

which gives the well known Shockley-Read steady-state results

$$f_T(E_T) = \frac{c_n(E_T)N + e_p(E_T)}{c_n(E_T) + c_p(E_T) + e_n(E_T) + e_p(E_T)} \quad (11)$$

$$R_T(E_T) = \frac{c_n(E_T)e_p(E_T)Np - c_n(E_T)e_p(E_T)}{c_n(E_T)N + c_p(E_T) + e_n(E_T) + e_p(E_T)} \quad (12)$$
The total net steady-state recombination rate is then the sum of $R_T(E_T)$ over all the gap states:

$$R_{SS} = \int R_T(E_T) = \int_{E_V}^{E_C} \sum_{E} c_n(E) p_p(E) N - e_n(E) e_p(E) \rho_T(E_T) dE_T$$

(13)

Here, the integration is from the mobility edges, $E_V$ to $E_C$, over the density distribution of the gap states, $\rho_T(E_T)$.

Let us now work out a simple example for solar cell junctions which has an explicit result. Consider the forward bias condition in a p-n junction. The constant quasi-Fermi level approximation may be used in the junction space charge layer so that we have

$$P = N = n_i \exp(qV/2kT)$$

(15)

The steady-state recombination rate given by Equation (13) then simplifies to

$$R_{SS} = \frac{n_i(\exp(qV/kT) - 1)}{C \exp(qV/2kT) + 1} \int_{E_V}^{E_C} \frac{e_n(E) e_p(E)}{e_n(E) + e_p(E)} \rho_T(E_T) dE_T$$

(16)

where the ratio $C$ is somewhat dependent on the properties of the gap states as a function of energy but is assumed to be roughly unity in this example. It is unity if $(c_n + c_p)n_i = e_n + e_p$ at all gap-state energy levels. This is an interesting simple result which shows that the junction current due to recombination in the space charge layer for a distribution of gap states is proportional to the geometrical average of the emission rates over the distribution of the gap states.
2.3 Position Dependent Chemical Composition

For this case, we will consider only the effects from the nonrandom component of the atomic potentials since the gap states produced from the random part can be taken into account by the way described in section II. The fundamental starting point of this problem is the many-body Schroedinger equation for the entire crystal which is assumed to be large so that surface effects are unimportant. Due to position dependence of the chemical composition, there is a slowly varying aperiodic crystal potential which is the sum of the potentials of all the atoms in the crystal. Then, in the usual manner, the core electrons attached to each atomic nucleus can be taken into account by a transformation [13] to give an effective aperiodic pseudopotential. The aperiodic part can then be isolated out from the periodic part and treated as an extended perturbation to the periodic potential problem. The pseudo-effective mass equation [14] can then be derived for the perturbed problem. Although a detailed mathematical analysis following the above procedure has not been worked out for the specific problem on hand, a simple physical picture emerges which is certainly valid for slowly varying chemical composition. A quantitative condition of validity is that the perturbation potential in an extended region under consideration should be small compared with the periodic electric field due to atomic ions in that region. An equivalent condition is that the perturbation potential referenced to an average potential in the extended region should be small compared with the energy gap of interest. For such a slowly varying perturbation, the spatial variation can be represented by the position dependences of the electron affinity, $\chi(\mathbf{r})$, and energy gap, $E_{g}(\mathbf{r})$, or alternatively by the position dependences of the conduction and valence band edges, $E_{C}(\mathbf{r})$ and $E_{V}(\mathbf{r})$, which serves as the effective potential energies seen by the electrons and holes in the effective mass equations. In
addition, there are also spatial dependences of the density of state or effective mass which must be taken into account. This band model is illustrated in Figure 2.

The equilibrium one-electron distribution function is then given by

$$f(E) = \frac{1}{1 + \exp(E-E_F)/k_BT}$$

where the total energy is the sum of the potential energy and the kinetic energy and $E_F$ is the Fermi energy. In the conduction band, $E > E_C$ and

$$E = E_C + (\hbar^2k^2/2m_C)$$

and in the valence band, $E < E_V$ and

$$E = E_V - (\hbar^2k^2/2m_V)$$

For nonequilibrium, we can define a quasi-Fermi level for electrons, $F_N$, and one for holes, $F_P$, so that the distribution functions have the form of

$$f(E) = \frac{1}{1 + \exp(E-E)/k_BT}$$

where $F=F_N$ for electrons and $F=F_P$ for holes. Using the standard perturbation procedure to solve the Boltzmann equation [6], the modified current equations corresponding to the ideal case given by Equations (1) and (2) are

$$\dot{J}_N = \frac{qD_N}{n} \nabla N - \frac{q\mu_n N \nabla V}{n}$$

$$= - \frac{q\mu_n N \nabla V}{n}$$

and

$$\dot{J}_P = - \frac{qD_P}{p} \nabla P - \frac{q\mu_P P \nabla V}{p}$$

$$= - \frac{q\mu_P P \nabla V}{p}$$

where $V_C=-E_C/q$, $V_N=-F_N/q$, $V_V=-E_V/q$ and $V_P=-F_P/q$. For the nondegenerate case, the electron and hole concentrations are given by

$$N = N_C\exp[(E_C-F_N)/k_BT]$$

$$P = N_V\exp[(F_P-E_V)/k_BT]$$

The mobilities and diffusivities appearing in Equations (21) to (22a) are now the local values which are averaged over the carrier concentrations $N$ and $P$.
In addition, the effective density of states, $N_C$ and $N_V$, are also local values which are spatially varying due to the position dependences of the density of energy states.

The conventional concepts of an intrinsic sample as well as the intrinsic carrier density and intrinsic Fermi level at thermal equilibrium also lose the familiar meaning since they are functions of the spatial dependences of $E_C(r)$, $E_V(r)$, $E_G(r)$ and $\chi(r)$. For example, the position dependences of the electrostatic potential for a pure semiconductor, i.e. a graded band gap semiconductor without any ionized chemical impurities and defects, can be obtained only by solving the Poisson equation in a given material with known spatial dependences of $E_C$, $E_V$, $E_G$ and $\chi$ and for specified boundary conditions. This electrostatic potential variation with position, however, will not be the same as the spatial dependences of the intrinsic Fermi potential, $E_I(r)$, which can still be calculated in the conventional way using Equation (23): $N=P=n_0=\sqrt{N_C N_V} \exp\left[(E_C-E_V)/k_B T\right]$ and $F_N=F_P=E_I(r)$.

The example just given seems unfamiliar at first glance since it defies our firmly-rooted concepts used in semiconductor device analysis. However, further thought on a familiar situation immediately removes this conceptual bottleneck. For example, the different spatial variations between the intrinsic Fermi potential and the electrostatic potential in a graded band gap material has an analogous counterpart in the contact region between two dissimilar but compositional uniform materials such as the oxide-silicon interface in a MOS capacitor. In this case, if we let the electrostatic potential in the oxide coincide with the intrinsic Fermi potential of the oxide and disallow a discontinuity in the electrostatic potential at the interface plane between the oxide and the semiconductor, then the electrostatic potential in the semiconductor will not coincide with the semiconductor intrinsic Fermi potential but will be displaced from it. From this interface picture, one can then readily
build up the spatially varying band picture shown in Figure 2 for a graded gap semiconductor by placing many macroscopically thin semiconductor slices with different chemical compositions in contact. The conclusion on the difference between the electrostatic and intrinsic Fermi potentials is then immediately obvious. Such a multi-layer structure is in fact the model one implicitly uses to solve the one-electron Schroedinger equation as well as the transport equation in a one-dimensional graded gap semiconductor.

A numerical demonstration of the differences between electrostatic and intrinsic Fermi potentials can also be inferred from the work of Emtage which was undertaken many years ago [15].

An important point to be noted is that the electrostatic field, given by the gradient of the electrostatic potential, loses its fundamental microscopic significance in a graded band gap semiconductor. It is the gradient of the conduction or valence band edges, \( \nabla E_c(\hat{r}) \) and \( \nabla E_v(\hat{r}) \), which appears in the Schroedinger equation and which give the forces experienced by the electron and hole quasi-particles and not the gradient of the electrostatic potential. This point was made clear by Kroemer [11] who coined these effective fields the quasi-electric fields [16]. Thus, the hot electron properties in a high electric field is dependent on the quasi-electric field for electrons, \( \nabla E_c(\hat{r}) \), and not the ordinary electrostatic field, \( \nabla \phi \), obtained from the Poisson equation. On the other hand, the boundary conditions at an interface between two dissimilar materials are to be imposed on the electrostatic field, electrostatic displacement and electrostatic potential or the true macroscopic charge densities, \( N, P, N_{DD} \) and \( N_{AA} \), but not on the quasi-electric fields.
2.4 Summary

The modifications of the Shockley equations by the position dependences of the atomic composition of semiconductors are described. Two modifications are analyzed. The first concerns the random spatial variations which give rise to distributions of highly localized states in the band gap. These band gap states enhance the carrier recombination rates. The total steady-state recombination rate at these gap states is worked out as an example and applied to the space charge region of a pn junction.

The second modification discussed in this paper is a major one. It is shown that change in the form appears in the two current equations of the five Shockley equations for a two-band semiconductor. Instead of a proportionality to the conventional electric field, the electron and hole drift currents are now proportional to the gradient of the conduction and valence band edges respectively. These gradients of band edges are also known as the quasi-electric fields and they are not equal in the graded band gap materials while they are equal in an uniform material. The quantum mechanical basis for the presence of the quasi-electric fields are discussed and shown to come from the aperiodic part of the potential energy of the crystal in the many-body problem and its one-electron representation. The use of the quasi-electric fields for hot carrier effects and the electrostatic field for boundary condition matching are also pointed out.
References for Chapter 2


[16] There are also several more recent papers using Kroemer's quasi electric field concept, such as W. G. Oldham and A. G. Milnes, "n-n semiconductor heterojunctions," Solid-State Electronics, 6, 121-132, (1963) and L. J. Van Ruyven and F. E. Williams, "Electronic transport in graded-band-gap semiconductors," American J. Physics, 35, 705-709 (1967)
Figure Captions

Figure 1  The four electron and hole capture and emission transition processes between a conduction or valence band state and a bound state located at an energy $E_T$.

Figure 2
(a) The one-electron energy band model of a solid plate with position dependent composition giving rise to spatial variations of the energy gap, $E_G$, electron affinity, $\chi$, the band edges of the conduction and valence bands, $E_C$ and $E_V$. The physical crystal potential composed of the sum of the atomic potentials is not shown in the figure. The shaded regions are the conduction and valence bands. Gap states such as those indicated in Figure 1 are not shown here for simplicity. The rise of the potential well (heavy curves) near the crystal surface is highly expanded in relation to the width of the crystal labeled.

(b) The one-electron energy band picture of two semiconductor crystals in contact giving rise to spatial variations of the energy gap, $E_C(y)$, electron affinity, $\chi(y)$, and the conduction and valence band edges, $E_C(y)$ and $E_V(y)$. The two vacuum levels, on the left and right, are shown to illustrate a large area semiconductor plate situation. The dashed $E_C(y)$ and $E_V(y)$ curves with discontinuities are those of the idealized and physically unrealistic situation assumed in the past which overlooked the distortion of the atomic structure and electron distribution in the interface region which give the gradual transition of $\chi$, $E_G$, $E_C$ and $E_V$ shown in solid curves.
Figure 2 (a)
\[ X_2 \cdot FT \cdot Y \cdot ^2 \cdot X(Y) \cdot n \cdot E \cdot C \cdot (Y) \]

**Fig. 2(b)**
PART II: NEW EXPERIMENTAL METHODS
CHAPTER 3 EXPERIMENTAL DETERMINATION OF SERIES RESISTANCE
OF P-N JUNCTION DIODES AND SOLAR CELLS

3.1 Introduction

The series resistance $R_s$ is a parasitic element in diodes and solar cells that can degrade the device performance. In general, the presence of $R_s$ will bend the current-voltage characteristic away from the idealized characteristic [1]. In a solar cell, this bending decreases the power conversion efficiency [2].

The series resistance of a diode or a solar cell consists of the contact resistances of the metal contacts at both faces of the device and of the resistance of the bulk semiconductor. The resistance of the bulk substrate can be calculated taking into account a spreading resistance correction [3]. The contact resistance is difficult to calculate accurately. In solar cells with a very thin diffused layer and contact grid
on the top, the resistance of the diffused layer might dominate. Calculation of this resistance is difficult because of the nonuniform current flow occurring in the diffused layer [4].

Thus, because of the uncertainties and difficulties involved in calculating $R_s$, methods for direct measurement of $R_s$ are desirable. For a solar cell, a method has been described [5] that involves exposing the solar cell to different illumination levels. Here we describe three other methods for measurement of $R_s$ which can be applied to any diode or solar cell. These methods involve:

a) small signal admittance measurement at moderately high frequency;

b) combined measurement of dc and ac conductance;

c) open-circuit-voltage-decay measurement (OCVD).

3.2 Theory

The small signal equivalent model of the ideal junction diode, derived by Shockley [1], consists of a diode conductance $G_{QN}$ and a diode capacitance $C_{QN}$ in parallel. The circuit elements, $G_{QN}$ and $C_{QN}$, result from solving the continuity equation for holes and electrons under small signal excitations including both the quasineutral base and emitter regions. In parallel with these two elements, the circuit representation of a real diode includes two capacitances originating in the junction space-charge region: $C_T$, associated with the ionized impurities, and $C_{SCR}$, associated
with the mobile charges [6]. The total capacitance is 
\[ C_D = C_{QN} + C_T + C_{SCR} \] 
To complete the circuit, \( R_s \) is added, as shown in Figure 1(a).

**Method (a):** As shown by Sah [6], \( C_D \) increases exponentially with increasing bias voltage \( V_D \) whereas, for small forward bias, \( C_D = C_T \) shows a relatively weak dependence on \( V_D \). If the small-signal admittance of the diode is measured at a forward bias small enough and at a frequency \( \omega \) high enough that \( \omega C_D \gg G_D \), then the diode equivalent circuit reduces to \( R_s \) and \( C_D \) in series, as shown in Fig. 1(b) \( R_s \) can then be easily measured using a suitable high-frequency (1 to 10 MHz) bridge.

**Method (b):** In contrast to method (a), we now set the forward bias voltage \( V \) and the frequency \( \omega \) to such values that \( \omega C_D \sim G_D \). Noting that an admittance bridge measures \( G_p \) and \( C_p \) of the parallel combination shown in Fig. 1(c), we see that the dependence of \( G_p \) plotted against \( V \) will differ from the dependence of \( G_D \) against \( V \) because of the presence of \( R_s \). For dc conditions, as illustrated in Fig. 1(d), the measured conductance \( G_{DC} = \left[ R_s + 1/G_D \right]^{-1} \) will show yet another dependence on \( V \). Comparison of \( G_p(V, R_s) \) versus \( G_{DC}(V, R_s) \) then yields \( R_s \). A simple computer program can be written to enable rapid determination of \( R_s \).

**Method (c):** The open-circuit-voltage-decay (OCVD) measurement [7] has been often used to determine minority carrier lifetime of a diode. In this measurement, a voltage step applied in the forward direction causes a flow of forward current I.
The terminals are then open-circuited. The resulting cessation of \( I \) causes a sharp drop of the voltage \( \Delta V \) to occur at the diode terminals. This voltage drop \( \Delta V \) divided by \( I \) gives series resistance \( R_s \).

3.3 Results

Table I shows results from methods (a) and (c) together with data obtained by the illumination method of Wolf and Raushenbach [5]. The devices used in the measurements were \( p^+\text{-}n \) diodes, 30 mil. in diameter and \( 2\times2 \text{ cm}^2 \) \( n^+\text{-}p \) solar cell. The OCVD method was inaccurate for the devices measured since the voltage drop \( \Delta V \) was difficult to determine accurately; this is especially difficult in devices with short lifetimes. Method (b) could not be used for our devices, since the ac measurements have to be taken at a frequency \( \omega \) such that \( \omega \ll 2/\tau \) [1], where \( \tau \) is the minority carrier lifetime in the base. In the devices we studied, \( \tau \) was a few microseconds which requires measurement at frequencies not exceeding about 1 kHz. At such low frequencies, \( G_P \) will differ from \( G_{DC} \) by only about 0.005%, which is much less than the estimated experimental accuracy of 1%. This method can be used, however, for short-lifetime devices. For example, if \( \tau = 0.1\mu\text{sec} \), the ac measurements could be done at a frequency of a few hundred kHz, at which \( G_P - G_{DC} \) would be large enough to enable a determination of \( R_s \).

Method (a) is the simplest approach and gives accurate results that agree reasonably well with those produced by
the illumination method. The accuracy of method (a) is limited by the accuracy of the bridges used, which is better than five percent for the measurements reported here. Measurements were taken at frequencies between 1 and 5 MHz. The results obtained from the illumination method depend on the light intensity [5]; the values of $R_s$ determined by this method fall in the range indicated in Table I. In contrast to the illumination method, method (a) involves only electrical excitation and applies to ordinary diodes as well as to solar cells.

The high-frequency measurements were done on Wayne-Kerr B601 and B801 bridges; low-frequency measurements were done on a Wayne-Kerr B224 bridge.

3.4 Summary

This work demonstrates a simple and rapid method (method (a)) for determining the series resistance of diodes and solar cells based on the measurement of small-signal admittance. The OCVD method (method (c)) provides a rapid but coarse estimate of $R_s$. Method (b) is useful for devices having a short substrate lifetime. Method (b) enables also measurement of $R_s$ at any bias voltage $V$, which allows the determination of any dependence that $R_s$ might have on $V$. 
<table>
<thead>
<tr>
<th>Device</th>
<th>Method (a)</th>
<th>Method (c)</th>
<th>Photovoltaic output [5]</th>
</tr>
</thead>
<tbody>
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<td><strong>p⁺-n diode</strong></td>
<td></td>
<td></td>
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<tr>
<td>10Ωcm</td>
<td>20</td>
<td>19-28</td>
<td>15-23</td>
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<tr>
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<td>4.35</td>
<td>3-10</td>
<td>3.5-4.5</td>
</tr>
<tr>
<td><strong>2x2 cm² n⁺-p solar cell</strong></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>2Ωcm</td>
<td>0.33</td>
<td>~0.6</td>
<td>0.25-0.34</td>
</tr>
</tbody>
</table>

Table I  Series resistance from methods (a) and (c) together with results using light excitation for comparison.
REFERENCES FOR CHAPTER 3


Figure 1  (a) ac equivalent circuit; (b) high-frequency equivalent circuit for $\omega C_D \gg C_D$; (c) parallel equivalent circuit as measured by bridge, where $G_p$, $C_p$ are functions of $G_D$, $C_D$, $R_S$, $\omega$; (d) dc equivalent circuit.
4.1 Introduction

One of the material parameters that most influences the performance of a p-n junction solar cell or diode is the minority-carrier diffusion length \( L \) of the base (or substrate) region. The diffusion length limits the collection efficiency and power conversion efficiency achievable in a solar cell, and it controls the current-voltage characteristic in a long-base (nonilluminated) diode.

Many methods exist for the measurement of \( L \). Of these, the ones of most relevance to p-n junction device studies yield the value of \( L \) in a solar cell or diode after the junction is formed so that the influence on \( L \) of the process used to fabricate the junction is included. Existing measurement methods applicable to this problem [1]-[5] involve various tradeoffs between accuracy and convenience.

The purpose here is to describe a new method [6] applicable to solar cells for measuring \( L \) and thus the minority-carrier lifetime \( \tau \), which is accurate and requires only straightforward capacitance measurements made at the device terminals. The accuracy results because the measurements required employ very accurate capacitance bridges. The accuracy is demonstrated by showing that the values of \( L \) yielded by this capacitance method agree well with those obtained by use of the X-ray method described by Rosenzweig [1]. In contrast, the results determined by the capacitance method disagree sharply with those determined from the open-circuit-voltage-decay method [4], for reasons that are discussed.
In its present form, the method measures \( L \) for a non-illuminated cell. We discuss the conditions under which this value of \( L \) is appropriate to use in describing an illuminated solar cell, and indicate experiments that confirm these conditions.

The method can also determine material properties of the quasi-neutral emitter region. We discuss the determination of the energy-band narrowing and the phenomenological lifetime in the emitter both by this method and by another method involving only dc measurements of emitter current.

4.2 Theoretical Grounds for the Method

Consider a \( p^+-n \) junction diode or solar cell with no illumination applied; analogous results hold for \( n^+-p \) devices. The minority-carrier diffusion length, \( L_p = \sqrt{D_p \tau_p} \), in the n-type base influences the small-signal capacitance \( C_{QNB} \) and conductance \( G_{QNB} \) of the quasi-neutral base region according to [7]

\[
C_{QNB} = \frac{q}{kT} \frac{A q n \frac{L_p^2}{p}}{2N_{DD}} [\exp(qV/kT) - 1] \quad (1)
\]

\[
G_{QNB} = \frac{q}{kT} \frac{A q n \frac{D_p^2}{p}}{N_{DD} (L_p^2)} [\exp(qV/kT) - 1] \quad (2)
\]

for an applied sinusoidal signal whose period greatly exceeds the base lifetime \( \tau_p \). The above results hold for a one-dimensional model of a device having a uniformly doped base and operating under low-injection conditions. The undefined symbols appearing in (1) and (2) have their usual meanings.

If either \( C_{QNB} \) or \( G_{QNB} \) could be determined from measurements, (1) and (2) would then yield \( L_p \), which is the desired quantity. But the
capacitance \( C \) and conductance \( G \) measured at the terminals contain components in addition to \( C_{QNB} \) and \( G_{QNB} \) \[8\]-\[11\]

\[ C = C_I + C_S + C_{SC} + C_{QNE} + C_{QNB} \]  

\[ G = G_S + G_{SC} + G_{QNE} + G_{QNB} \]  

under low-injection conditions. In (3) and (4) the subscript \( S \) refers to the surface region, \( SC \) to the junction space-charge region, and \( QNE \) and \( QNB \) to the quasi-neutral emitter and base regions, respectively. The capacitance \( C_I \) is the capacitance due to the ionized impurities in the junction space-charge region and \( C_{SC} \) is that due to the mobile holes and electrons in that region \[8\]. From the viewpoint expressed in (1)-(4), the problem of determining \( L_p \) becomes the problem of extracting \( C_{QNB} \) from the measured \( C \) versus \( V \) characteristic or of extracting \( G_{QNB} \) from the measured \( G \) versus \( V \) characteristic.

As experiments \[12\], \[13\] including those described in Section III show, \( G_{QNE} \) can be comparable to \( G_{QNB} \) or larger than \( G_{QNB} \) in many p-n junction devices of interest. Thus, except in special cases, one cannot determine \( G_{QNB}^' \) and hence, \( L_p^' \), from the conductance data.

The key to the extraction of \( C_{QNB}^' \) and hence to the determination of \( L_p^' \), from the C-V data lies in making capacitance measurements at relatively high as well as at low frequencies. At low frequencies, the mobile carriers in all regions of the device respond to the signal, and (3) describes the measured low-frequency capacitance. But, at a higher frequency only mobile carriers associated with \( C_I \) and \( C_S \) will follow the signal \[7\] and will contribute to the measured capacitance if the period of the signal is long compared with both the dielectric relaxation
time of the semiconductor and the transit time across the junction space-charge region. Thus by subtracting this higher-frequency capacitance from the low-frequency capacitance, one obtains a capacitance associated with the quasi-neutral regions. This capacitance is \( C_{QNB} \) if the phenomenological lifetime \( \tau_E \) of the emitter [12]-[14] is much shorter than \( \tau_p \), and if the signal frequency \( f \) is chosen so that \( \tau_E \ll 1/f \ll \tau_p \).

If, on the other hand, \( \tau_E = \tau_p \), or if \( 1/f \ll \tau_E, \tau_p \), this subtraction yields the capacitance, \( C_{QNE} + C_{QNB} \), which reduces to \( C_{QNB} \) for the commonly occurring case that \( C_{QNE} \ll C_{QNB} \). Thus as will be discussed and illustrated in the following sections, the method yields \( C_{QNB} \) and, hence, from (1), the value of \( L_p \).

The method applies directly to large-area solar cells. In some cases, however, advantages may result from making the measurements on small-area test devices. For example, this may be desirable if high-frequency admittance bridges having a suitable capacitance range are unavailable in a particular laboratory. For small-area devices, with radius \( a \sim L_p \), the two-dimensional fringing effects illustrated in Fig. 1 invalidate the one-dimensional theory underlying (1) and (2) and corrections must be made. Correction curves were calculated [15] by solving the continuity equation for holes in two space dimensions. One set of correction curves is shown in Fig. 2.

The theory underlying the method, contained in (1) and (2), applies to non-illuminated solar cells. Solar illumination can change the carrier lifetime because the solar spectrum can change the transitional probability rates of the intermediate levels in the forbidden band [16]. In general, the method yields values of lifetime appropriate to an illuminated cell if the shifting approximation holds, that is, if the current under illumina-
tion is the current in the dark shifted by the short-circuit current [17]. Very careful measurements made on various single-crystal silicon p–n junction solar cells demonstrated the applicability of the shifting approximation for AM0 and lower levels of illumination, thus indicating the validity of the method for these cells at these illumination levels. In assessing the applicability of the shifting approximation, corrections may be needed to account for the effects of series resistance.

A quick check for the light-dependence of the lifetime can be made by measuring the forward-biased capacitance in the dark and again with the light on.

4.3 Illustrative Example

To explore the utility of the method, we fabricated a set of small circular p–n junction solar cells having various base doping concentrations between $1.4 \times 10^{14}$ cm$^{-3}$ and $6.5 \times 10^{17}$ cm$^{-3}$. Each diode was circular with an area of $4.5 \times 10^{-3}$ cm$^2$. Three separate runs were made: runs 1 and 3 at 950°C and run 2 at 1000°C. Runs 1 and 2 employed phosphorus gettering from the back face to getter metallic impurities from the substrate. Boron surface concentration ranged from $1.5 \times 10^{19}$ cm$^{-3}$ to $4 \times 10^{19}$ cm$^{-3}$ and the junction depth ranged from 0.3µm to 0.9µm.

Figure 3 shows representative data: the measured $\ln(G)$ and $\ln(C)$ dependencies on $qV/kT$ for device from run 1 having a base doping concentration $N_{DD} = 1.25 \times 10^{15}$ cm$^{-3}$. The data were taken at 299 K and 376 K. The low-frequency capacitance and conductance, $C_{LF}$ and $G_{LF}$, were measured at 3 kHz using a Wayne-Kerr B224 bridge. High-frequency capacitance $C_{HF}$ was measured at 10 MHz using a Wayne-Kerr B801B bridge. The level of the ac signal was kept below 5 mV.
To calculate the diffusion length $L_p$ of the base region, we assume that the minority carrier charge storage in the base greatly exceeds that in the emitter, and, hence, that $C_{QN} \ll C_{QNB}$. For the particular diode under study, these assumptions are likely to hold for three reasons: a) the low surface concentration $N_S$ tends to limit the size of the energy-gap shrinkage effects [12], [18], [19] in the emitter; b) the low doping concentration $N_{DD}$ of the base implies a large minority carrier charge storage in the base; c) the thinness of the emitter region tends to limit the amount of minority-carrier charge it can store.

$C_{QN} = C_{QNB}$ is obtained by subtracting $C_{HF}$ from $C_{LF}$ which yields for $\ln C_{QN}$ versus $qV/kT$ a straight line having unity reciprocal slope. Then, from (1), the minority carrier diffusion length of the base obtained from the capacitance method is $L_{pC} = 80\mu$m. To assess the accuracy of this result, we also determined the diffusion length by the X-ray method described by Rosenzweig [1], which is accurate within about ±5 percent [1], and found $L_{pX} = 84\mu$m. The good agreement between $L_{pC}$ and $L_{pX}$ tends to confirm the utility and accuracy of the capacitance method.

For some devices, the emitter recombination current will be much less than the base recombination current, so that $G_{QNE} \ll G_{QNB}$ and $L_p$ can be deduced from (2) using only the low-frequency conductance data. This is true for the device under consideration in Fig. 3. If one assumes $C_{QNE} \ll C_{QNB}$, one obtains $L_{pG} = 80\mu$m, which agrees well with the values obtained by the capacitance and X-ray methods. $G_{QNB}$ was obtained using a construction technique described elsewhere [13].

But the conductance method fails whenever the emitter recombination current contributes significantly to the current-voltage characteristic, as can be expected to occur in diodes having higher base doping con-
centrations and consequently lower minority-carrier charge storage in the base. This point is illustrated in Fig. 4. For diodes having base doping concentrations above $10^{16} \text{ cm}^{-3}$, $L_{pC}$ differs from $L_{pC}$ and $L_{pX}$ by a factor of two or more. The capacitance and the X-ray methods give results in excellent agreement for all base doping concentrations, however, which illustrates the accuracy and applicability of the capacitance method. Excellent agreement between $L_{pC}$ and $L_{pX}$ also justifies the use of dark $C_{QNB}$ for determination of $L_p$ in solar cells, since X-ray radiation can excite traps in the bandgap in a similar way as light excitation.

For our devices the ratio $a/L_p$ is larger than about 4 and the two-dimensional correction factor is small, as can be seen from Fig. 2. Appropriate corrections were made.

A. Comparison with the Open-Circuit-Voltage-Decay Method

Determination of the base lifetime $\tau_p$ by the open-circuit-voltage-decay (OCVD) method [4] is a commonly used approach. The OCVD method is attractive because it involves the measurement of only a single transient, which can be rapidly done.

To examine the applicability of OCVD to silicon devices, we consider first its theoretical basis. In the theoretical development of the OCVD response, one finds the open-circuit response of a diode following the sudden removal of forward bias by solving the continuity equation for the minority carriers in the base. Under low-injection conditions, the open-circuit voltage decays linearly with time, to a close approximation, and measurement of the slope of this straight-line decay suffices to determine $\tau_p$. This result applies if the minority carriers in the base dominate in determining the response, as was the case for the germanium diodes considered in the original OCVD study [4]. But if the carriers associated with the junction
space-charge region or with the quasi-neutral emitter contribute appreciably to the response, then the OCVD method is not applicable. The junction space-charge region contributes to the measured capacitance and conductance characteristics of Fig. 3, as evidenced by the nonunity slopes of these characteristics, particularly of the capacitance characteristic. At 299 K, the slopes come nearest to unity in the voltage range, 0.45 V ≤ V ≤ 0.52 V. For this range, the minority hole concentration in the base is in low injection. Figure 5 displays the corresponding OCVD response, taken at 299 K and beginning at V = 0.5 V. Note that this response fails to exhibit a simple linear decay. Thus as one could anticipate from the nonideality of conductance and capacitance characteristics, the OCVD method for determining τ_p is inapplicable for this diode at 299 K. A course estimate of τ_p can be inferred from the OCVD response, however, by fitting a straight line to the response and measuring its slope. This yields τ_p (OCVD) ≈ 8μs, which differs by nearly a factor of two from the value, τ_p ≈ 4.4μs, corresponding to L_{PC} = 80μm determined by the capacitance method and corroborated by the X-ray method and by the calculation [7], τ_p = 2C_{QN}/C_{QN}. For comparison, the straight-line decay corresponding to τ_p = 4.4μs is plotted on Fig. 5.

According to theory, the quasi-neutral capacitance and conductance increase more rapidly with temperature than do the components from the junction space-charge region. Thus one might expect the capacitance and conductance characteristics to have slopes that more closely approach unity as the temperature is raised. This is confirmed by the data of Fig. 3 measured at 376K, where the closest approach to unity slope occurs in the voltage range, 0.25V < V < 0.32V. In this range, which corresponds
to low-injection levels of the minority holes in the base, the conductance almost shows an ideal slope of unity, and the capacitance shows a reciprocal slope of 1.07. As Fig. 5 indicates, however, even these slight departures from the ideal characteristics are enough to produce an OCVD response that again fails to exhibit a simple linear decay. The fitting of a straight line to the OCVD response at 376 K results in the estimate $\tau_p (\text{OCVD}) = 9\mu s$, which differs considerably from the value $\tau_p = 7\mu s$, corresponding to $L_{pc} = 80\mu m$ at 376K determined by the capacitance method and corroborated by calculation [7], $\tau_p = 2C_{QN}/G_{QN}$.

Similar deviations from the ideal capacitance and conductance characteristics and the ideal (linear) OCVD response were seen for all of the silicon diodes of Fig. 4. This suggests that the OCVD method for determining $\tau_p$ is questionable for silicon devices, although one might expect its accuracy to increase as temperature is raised. The capacitance method avoids the inaccuracies resulting from the nonideality of the characteristics by combining low-frequency and high-frequency data.

As was discussed previously [13], the junction current recovery method [20] is generally even a poorer method than OCVD for determining carrier lifetime and diffusion length of the base region.

4.4 Discussion of the Capacitance Method

The capacitance method for determining $L_p$ and $\tau_p$ involves measuring the dependence of the capacitance on forward bias at two frequencies: a frequency $f_L$, low enough that $1/f_L > > \tau_p$, and a frequency $f_H$ high enough that $1/f_H < < \tau_p$. Once the low-frequency and higher-frequency capacitance are measured, they are subtracted to give the quasi-neutral capacitance $C_{QN}$. The resulting dependence of $\ln(C_{QN})$ on $qV/kT$ must be a straight line of unity slope, which provides a convenient self-consistency check
on the validity of the results. For diodes for which $C_{\text{QNE}} \ll C_{\text{QNB}}$, which often occurs, the quasi-neutral base capacitance $C_{\text{QNB}}$ is thus determined. Combining $C_{\text{QNB}}$ with (1) then determines $L_p$ and $\tau_p$.

The inequality, $C_{\text{QNE}} \ll C_{\text{QNB}}$, holds for the cells of Fig. 4 because the surface doping concentration is low ($N_S \sim 10^{19}$ cm$^{-3}$) and the emitter junction is thin ($x_j \sim 1 \mu$m). These conditions make the total number of dopant atoms in the emitter small. That $C_{\text{QNE}} \ll C_{\text{QNB}}$ for these devices is substantiated by the demonstration that $L_{\text{PC}} = L_{\text{PX}}$. But if a given device has a large number of dopant atoms in the emitter, then $C_{\text{QNE}}$ can be large because of band-narrowing effects [12], [18], [19]. If, in addition, the device has a large doping concentration in the base, which from (1) implies a small value of $C_{\text{QNB}}$, then $C_{\text{QNE}} = C_{\text{QNB}}$. For such a device, the diffusion length of the base can be determined by the X-ray method, and the method described in this paper can be adapted to enable the determination of the minority-carrier charge $Q_E$ stored in the emitter and the phenomenological lifetime $\tau_E$ of the emitter. From the values of $Q_E$ and $\tau_E$ thus determined one can deduce information about the band-narrowing and recombination mechanisms in the highly doped emitter [12].

The details of such studies for bipolar transistors were published elsewhere [21], [22]. Similar work on solar cells is in progress.

There are several practical limitations that define the scope of applicability of the capacitance method and they are discussed in detail in Ref. [6]. These limitations are not severe and the capacitance method applies to a wide range of silicon devices, as the results given in Section III demonstrate. The capacitance method has the advantage of being accurate and of requiring only straightforward electrical measurements at the terminal using inexpensive equipment. It also enables the
easy determination of the dependence of diffusion length and lifetime on temperature.

The accuracy of low-frequency capacitance bridges, such as Wayne-Kerr B224, is about ±0.1 percent and the accuracy of available high-frequency bridges is about ±2 percent. If the temperature during measurement is controlled within ±0.1 K, which will introduce an error in $n_1^2$ less than 2 percent, the overall accuracy of the method is estimated to be about ±5 percent.

4.5 Bandgap Narrowing in Emitter from Temperature Dependence of Emitter Current

Slotboom and deGraaff [23] and Martinelli [24] recently proposed methods to determine bandgap narrowing $\Delta E_G$ in the base region of bipolar transistors. Because the base doping of transistors is limited to about $10^{19}$ cm$^{-3}$, these methods do not allow investigation of an entire range of high doping densities up to about $10^{21}$ cm$^{-3}$. Although these and similar methods provide useful information about bulk material remote from a surface, they cannot directly yield $\Delta E_G$ of the thin emitter region of any particular device under study because $\Delta E_G$ may depend in part on the proximity of the semiconductor surface near which lattice strain and impurity and defect clustering can occur. In addition, $\Delta E_G$ may also depend on the particular diffusion or ion-implantation processing used to form the emitter region.

To determine $\Delta E_G$ of the emitter, we propose an alternate method which employs the temperature dependence of the emitter current. This method applies over the entire range of emitter doping concentrations present in p-n junction devices. If the emitter doping profile is flat, the interpretation of the results of the method is particularly straightforward.
The simplest version of the method results if the emitter is transparent, i.e., if injected holes (for an n⁺-p device) recombine mainly at the surface rather than in the bulk of the emitter. If the emitter region is thin, the transparency of the emitter can be demonstrated experimentally even for very high emitter doping concentrations. A theoretical and experimental discussion of this issue will appear elsewhere.

If the emitter is transparent and if the surface recombination velocity $S_p$ is large, then [25]

$$J_{E0} = q \left[ \int_0^{W_E} \frac{N(x)}{D_p(x)n_{ie}^2(N)} \, dx \right]^{-1}$$  \hspace{1cm} (5)

where $n_{ie}^2 = C T^3 \exp(-E_{GO}(N)/kT)$ in which $C$ is a constant nearly independent of the doping concentration [23]. $E_{GO}(N)$ is the bandgap extrapolated to 0 K, which equals 1.206 eV for low doping concentrations [23,26]. For a degenerately-doped emitter region, in which the electron concentration $N$ is large, the mobility is assumed to be independent of temperature. The quasineutral emitter thickness is $W_E$. In (5), $n_{ie}^2$ is the effective intrinsic density.

For a flat doping profile, (5) becomes

$$J_{E0} = C_1 T^4 \exp(-E_{GO}(N)/kT)$$ \hspace{1cm} (6)

where $C_1$ is independent of temperature.

Thus a plot of $\ln(J_{E0}/T^4)$ versus $1/T$ yields a straight line from which $E_{GO}(N)$ can be calculated. The emitter saturation current $J_{E0}$ is determined from the measured total saturation current by techniques described elsewhere [12,13].
The method was used to measure $E_G$ on a device with almost flat implanted As profile with $N = 1.5 \times 10^{20} \text{ cm}^{-3}$ and $W_E \approx 0.37 \mu\text{m}$. The plot of $\ln(J_{E0}/T^4)$ versus $1/T$ for this device appears in Fig. 6. From the slope $E_{G0} = 1.046 \text{ eV}$ which gives $\Delta E_{G0} = 1.206 \text{ eV} - 1.046 \text{ eV} = 0.160 \text{ eV}$.

Additional points for the entire range of emitter dopings from $10^{17} \text{ cm}^{-3}$ to $10^{21} \text{ cm}^{-3}$ are being measured and adjusted to include the effects of Fermi-Dirac statistics. The results will be published elsewhere.
REFERENCES FOR CHAPTER IV


Fig. 1 Cross-section of a diffused $p^+-p$ junction diode showing a two-dimensional spreading effect. Boundary conditions for hole concentrations at the junction edge $P(x_j)$ and at the back contact $P(W)$ are indicated. $W$ is the base width and $S_p$ is a surface recombination velocity at the back contact.
Fig. 2 Ratio of two-dimensional current density $J_{2D}$ to one-dimensional current density $J_{1D}$ as a function of the ratio of device radius $a$ and diffusion length $L_p$. 
Fig. 3 Measured conductance and capacitance versus forward bias $V$ for diode with $N_{DD} = 1.25 \times 10^{15}$ cm$^{-3}$, a $T=376$ K; b $T=299$ K. The insert shows the capacitance characteristic for diode with $N_{DD}=8 \times 10^{16}$ cm$^{-3}$, demonstrating the increased importance of $C_{SC}$ and emphasizing the value of the subtraction for determining $C_{QN}$. In the figure, --- stands for the components from the quasi-neutral regions ($m=1$) and ---.--. stands for $G_{SC} + G_S$. 
Fig. 4 Diffusion length $L_p$ determined by capacitance, conductance, and X-ray methods versus base doping concentration $N_{DD}$. 
\[ \tau_{\text{OCVD}} = -\frac{kT}{q} \frac{1}{dV/dt} \]

Fig. 5 OCVD response for device of Fig. 3 at 299 K and 376 K. Dashed lines correspond to the lifetime \( \tau_p \) determined by the capacitance method.
Fig. 6 \( \ln (I_{EO}/T^4) \) versus 1000/T with a slope giving \( E_G = 1.046 \) eV for \( N_D = 1.5 \times 10^{20} \) cm\(^{-3}\).
CHAPTER 5
AN ANALYSIS OF THE DIFFUSION CURRENT AND CAPACITANCE IN A TWO
DIMENSIONAL p-n JUNCTION DIODE FOR DETERMINING
THE DIFFUSION LENGTH

5.1 Introduction

In a previous related publication [1] by the authors a method was proposed for determining the diffusion length in the base of a p-n junction diode from measurements of the diffusion current-voltage and diffusion capacitance-voltage characteristics. As a central approximation in the development of this method, the current and capacitance characteristics were assumed to be those resulting from the conventional one-dimensional model of the diode, in which excess hole and electron distributions are adequately described by a single space variable x, the distance from the junction surface. This approximation describes the characteristics with sufficient accuracy if the area is large, or more precisely, for a circular device, if the radius a of the emitter region greatly exceeds the minority-carrier diffusion length \( L_p \) of the base region. But the bridge used for the high-frequency capacitance measurements required in this method imposes an upper limit on the capacitance that can be measured. Thus diode structures must sometimes be used for which the approximation is invalid and for which the one-dimensional model of the diode introduces intolerable error in the determination of \( L_p \). Figure 5.1
illuminates a p-n junction diode that, if \( L_p \) were large enough, would be poorly described by a one-dimensional model.

The purpose of this chapter is to provide an analysis of the diffusion current and capacitance characteristics for a two-dimensional model of the p-n junction diode that allows the determination of \( L_p \) for any value of \( a/L_p \).

Grimbergen [2] has treated a problem related to the one of interest. Grimbergen studied a p-n-n+ structure. He assumed the surface recombination velocity \( S_p \) at the high-low n-n+ junction to be zero and the base width \( W \) to be much smaller than \( L_p \). Grimbergen reduced the two-dimensional problem to two coupled one-dimensional problems. As will be shown, the result he obtained is close to one of the cases of our two-dimensional calculation, that corresponding to \( S_p = 0 \), and \( W/L_p = 0.1 \) in Fig. 5.2. But, for many diodes, the assumption \( S_p = 0 \) at the back face is inappropriate and the ratio \( W/L_p \) is not necessarily small. Thus, a general solution of two-dimensional problems will be useful.

The chapter begins by defining the boundary-value problem. It resembles boundary-value problems encountered in the conduction of heat but differs in that a volume recombination term appears in the differential equation. The solution of this boundary-value problem is then presented, which yields the current \( I(\alpha) \) and the capacitance \( C(\alpha) \) normalized by their one-dimensional counterparts plotted against the ratio, \( \alpha = a/L_p \). Used in conjunction with current and capacitance measurements, the plot enables the determination of \( L_p \). This is illustrated for two devices with different areas. The diffusion length \( L_p \) determined by this method is shown to be consistent with that determined
by the X-ray method [3] which illustrates the good accuracy to be expected from use of the two-dimensional analysis.

5.2 Theoretical Analysis

The hole diffusion problem in the n-type substrate of a shallow diffused p-n junction (see Fig. 5.1) is similar to the heat conducting problem of a metal disk [4,5], except that a volume recombination term appears in the differential equation, describing the kinetics of the holes. The differential equation originates from the continuity equation, which for the steady-state condition is

\[ \nabla \cdot \mathbf{J_p} = -qP'/\tau_p \quad (5.1) \]

with

\[ J_p = -qD_p \nabla P \quad (5.2) \]

Combining (5.1) and (5.2) yields

\[ \nabla^2 P = \frac{P'}{D_p \tau_p} = \frac{P'}{L_p^2} \quad (5.3) \]

where \( P' = P - P_{no} \) is the excess hole concentration and \( L_p = \sqrt{D_p \tau_p} \) is the diffusion length of holes in the base region. In cylindrical coordinate \((\rho, \phi, z)\), equation (5.3) becomes

\[ \frac{\partial^2 P'}{\partial \rho^2} + \frac{1}{\rho} \frac{\partial P'}{\partial \rho} + \frac{1}{\rho^2} \frac{\partial^2 P'}{\partial \phi^2} + \frac{3}{2} \frac{\partial^2 P'}{\partial z^2} = \frac{P'}{L_p^2} \quad (5.4) \]

Separation of variables is accomplished by substituting \( P'(\rho, \phi, z) = R(\rho)\Phi(\phi)Z(z) \), which leads to
Figure 5.1 Cross-section of a shallow diffused $p^+$-n junction diode.
Normalization with respect to the junction radius \( a \) can be done by introducing \( \rho = ra \). Thus, equation (5.5) becomes

\[
\frac{1}{R} \frac{\partial^2 R}{\partial \rho^2} + \frac{1}{R \rho} \frac{\partial R}{\partial \rho} + \frac{1}{\Phi} \frac{1}{\rho^2} \frac{\partial^2 \Phi}{\partial \rho^2} + \frac{1}{Z} \frac{\partial^2 Z}{\partial \rho^2} - \frac{1}{L_p} \frac{a^2}{Z} = 0
\]  

(5.6)

Now, let

\[
\frac{a^2}{Z} \frac{\partial^2 Z}{\partial \rho^2} - \frac{a^2}{L_p} = k^2
\]

(5.7)

and

\[
\frac{1}{\Phi} \frac{\partial^2 \Phi}{\partial \rho^2} = -\nu^2
\]

(5.8)

Substituting (5.7) and (5.8) into (5.6), we obtain

\[
\frac{d^2 R}{dr^2} + \frac{1}{r} \frac{dR}{dr} + (k^2 - \frac{\nu^2}{r^2}) R = 0
\]

(5.10)

For a problem possessing azimuthal symmetry, \( \nu = 0 \), i.e., \( \Phi = \text{constant} \).

Thus, (5.6) reduces to the two ordinary differential equations (5.7) and (5.10). Solutions of (5.7) and (5.10) are

\[
R = C J_0(kr)
\]

(5.11)

\[
Z = A e^{-\sqrt{k^2 + \alpha^2} \frac{a}{z}} + B e^{\sqrt{k^2 + \alpha^2} \frac{a}{z}}
\]

(5.12)
Therefore, we have

\[ P'(r,z) = \sum_{k=0}^{\infty} R_{k} z = \sum_{k=0}^{\infty} (A e^{k^2} + B e^{-k^2}) J_{0}(kr) \]  \hspace{1cm} (5.13)
For a shallow diffused junction diode, with zero surface recombination velocity at the front side of Si-SiO\(_2\) interface, the boundary conditions require that

\[
P'(r, 0) = P'(x, 0) = \int A(k) \left[ 1 - \frac{D_p \sqrt{k^2 + a^2}}{a} e^{-\frac{k^2 + a^2}{2a^2}} J_0(kr) \right] dk \quad (5.18)
\]

for \(0 < r < 1\)

and

\[
\frac{\partial P(r, 0)}{\partial z} = 0 = \int (-A(k)) \frac{\sqrt{k^2 + a^2}}{a} \left[ 1 + \frac{D_p \sqrt{k^2 + a^2}}{a} e^{-\frac{k^2 + a^2}{2a^2}} J_0(kr) \right] dk \quad (5.19)
\]

for \(1 < r < \infty\).

Such pairs of integral equations are known as "Dual Integral Equations" [6]. We now compare (5.18) and (5.19) with the following integral pairs:

\[
\int G(k)f(k)J_0(kr) = Ar^0 = A \quad 0 < r < 1 \quad (5.20)
\]
and

\[ \int_0^\infty f(k)J_0(kr) = 0 \quad 1 < r < \infty \quad (5.21) \]

Thus

\[ f(k) = A(k) \sqrt{\frac{k^2 + \alpha^2}{a^2}} \left[ \frac{1 - \frac{D}{S} \sqrt{\frac{k^2 + \alpha^2}{a}}}{1 + \frac{D}{S} \sqrt{\frac{k^2 + \alpha^2}{a}}} e^{-\frac{\sqrt{k^2 + \alpha^2}}{2W}} \right] \quad (5.22) \]

and

\[ G(k) = \frac{1}{\sqrt{\frac{k^2 + \alpha^2}{a^2}}} \left[ \frac{1 - \frac{D}{S} \sqrt{\frac{k^2 + \alpha^2}{a}}}{1 + \frac{D}{S} \sqrt{\frac{k^2 + \alpha^2}{a}}} e^{-\frac{\sqrt{k^2 + \alpha^2}}{2W}} \right] \quad (5.23) \]

If we choose

\[ f(k) = k^{1/2} \sum_{m=0}^\infty a_m J_{2m+1}(k) \quad (5.24) \]

\( a_m \) can be determined from the following simultaneous equations:

\[ a_0 + \sum_{m=0}^\infty L_m a_m = \sqrt{\frac{2}{\pi}} P(x_j) \quad \text{for } n = 0 \quad (5.25) \]

\[ a_n + \sum_{n=0}^\infty L_m a_n = 0 \quad \text{for } n > 0 \]

where

\[ (4n + 1)^{-1} L_{m,n} = \int_0^\infty [kG(k) - 1] k^{1/2} J_{2m+1}(k) J_{2n+1}(k) dk \quad (5.26) \]

Numerical methods are used in computing \( L_{m,n} \).
Once $a_m$ is determined, the total hole diffusion current across the junction is

$$I = qD_p \int_0^a \frac{\partial P'(r, 0)}{\partial z} 2\pi \rho d\rho = qD_p a^2 \int_0^1 \frac{\partial P'(r, 0)}{\partial z} 2\pi rdr$$

$$= qD_p a^2 \int_0^1 k^{1/2} \int_0^1 a (\sum_{m=0}^{\infty} a_j \sum_{m=0}^{\infty} J_{2m+2}^m(k) J_0(kn) ) 2\pi rdr \quad (5.27)$$

After integrating (5.27) and dividing it by the one-dimensional current $I_0$, which can be expressed as

$$I_0 = \frac{qD_p \cdot \pi a^2 P(x_j)}{\left( \frac{S}{2} \right) \left( \frac{D}{2} \right) \cosh \left( \frac{W}{L} \right) + \sinh \left( \frac{W}{L} \right) \right) \left( \frac{S}{2} \right) \left( \frac{D}{2} \right) \cosh \left( \frac{W}{L} \right) + \sinh \left( \frac{W}{L} \right) \right) \quad (5.28)$$

the ratio of $I_\alpha/I_0$ appears as plotted as a function of $\alpha$ in Fig. 5.2. In this figure, $W/L_p$ and $S_{L_p}/D_p$ are parameters.

For a long-base diode, i.e., one for which $W >> L_p$, the constant $B$ in (5.16) approaches zero. For this case, the detailed solution of the two-dimensional boundary value problem is given in Appendix II of Ref. [7]. The result is [8,9]

$$f(\alpha) = \frac{I_\alpha}{I_0} = \frac{J_\alpha}{J_0} = \frac{4}{\pi \alpha} (A_0 + 0.A_1 + 0.A_2 + \ldots) = \frac{4}{\pi \alpha} A_0 \quad (5.29)$$

Because the series $\sum_{m=0}^{\infty} A_m$ converges fast, neglecting terms higher than $A_2$ yields $I_\alpha/I_0$ to an accuracy of ±3%. The same procedures can be applied to calculate $f(\alpha)$ for different values of $W/L_p$ and $S_{L_p}/D_p$. 


Experimental Results

- Devices of run No. 15
- Devices of run No. 5

Using Grimbergen's assumption

Figure 5.2.1 Normalized current density factor $f(a) = J_a/J_0$, as a function of the ratio of junction-area radius and diffusion length for $S_p = \infty$ at back contact.
Figure 5.2.2 Illustrating that the detailed solution presented here (curve A) agrees well with Grimbergen's approximation for diodes for which $W/L_p \ll 1, S_p = 0$, and $(W-x_j)/W \approx 1$. 

$\alpha = \frac{a}{L_p}$

- $f(a)$ calculated for $\frac{W}{L_p} = 0.1$
  - A: $\frac{D_p}{S_{L_p}} = \infty$ (i.e. $S_p = 0$)
  - B: $\frac{D_{L_p}}{S_{L_p}} = 1$
  - C: $\frac{D_{L_p}}{S_{L_p}} = 0$ (i.e. $S_p = \infty$)
  
  Grimbergen's approximation (with $S_p = 0$, $W-x_j/W = 1$, $W < L_p$)
Figure 5.2.3 Illustrating that Grimbergen's approximation deviates from the detailed solution (curve A) for $\frac{W}{L_p} = 0.5$. 

\[ f(a) \text{ calculated for } \frac{W}{L_p} = 0.5 \]

- **A:**
  \[ \frac{D_p}{S_p L_p} = (\text{i.e. } S_p = 0) \]

- **B:**
  \[ \frac{D_p}{S_p L_p} = 1 \]

- **C:**
  \[ \frac{D_p}{S_p L_p} = 0 \text{ (i.e. } S_p = \infty) \]

Grimbergen's approximation (with $S_p = 0$, $W - x_f = 1$, $W < L_p$)
Table 5.1 Calculated $f(\alpha)$, ($f(\alpha) = J_\alpha / J_0$), as a function of $\alpha$, $\alpha = a / L_p$; $\beta$, $\beta = W / L_p$; and $\gamma$, $\gamma = D_p / S_p L_p$.

<table>
<thead>
<tr>
<th>$\frac{W}{L_p}$</th>
<th>$\gamma = \frac{D_p}{S_p L_p}$</th>
<th>$0.1$</th>
<th>$0.3$</th>
<th>$1$</th>
<th>$3$</th>
<th>$10$</th>
<th>$30$</th>
<th>$100$</th>
</tr>
</thead>
<tbody>
<tr>
<td>$\approx$</td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>$5$</td>
<td></td>
<td>14.4</td>
<td>3.66</td>
<td>2.07</td>
<td>1.29</td>
<td>1.05</td>
<td>0.07</td>
<td></td>
</tr>
<tr>
<td>$1$</td>
<td></td>
<td>27.04</td>
<td>6.92</td>
<td>2.31</td>
<td>1.37</td>
<td>1.11</td>
<td>1.03</td>
<td></td>
</tr>
<tr>
<td>$0$</td>
<td></td>
<td>10.24</td>
<td>3.37</td>
<td>1.59</td>
<td>1.18</td>
<td>1.06</td>
<td></td>
<td></td>
</tr>
<tr>
<td>$0.5$</td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>$1$</td>
<td></td>
<td>13.48</td>
<td>4.89</td>
<td>2.02</td>
<td>1.31</td>
<td>1.08</td>
<td>1.01</td>
<td></td>
</tr>
<tr>
<td>$0$</td>
<td></td>
<td>7.03</td>
<td>2.9</td>
<td>1.58</td>
<td>1.23</td>
<td>1.06</td>
<td>1.0</td>
<td></td>
</tr>
<tr>
<td>$\approx$</td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>$5$</td>
<td></td>
<td>9.19</td>
<td>3.05</td>
<td>1.59</td>
<td>1.23</td>
<td>1.08</td>
<td>1.01</td>
<td></td>
</tr>
<tr>
<td>$1$</td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
<td></td>
</tr>
<tr>
<td>$0$</td>
<td></td>
<td>5.8</td>
<td>2.2</td>
<td>1.36</td>
<td>1.08</td>
<td>1.03</td>
<td>0.98</td>
<td></td>
</tr>
</tbody>
</table>

-67-
5.3 Experimental Results

Using the experimental set-up of Fig. 5.3, we measured the current-voltage characteristic of p⁺-n junction diodes with different areas, described in Table 5.2. For the measurement, the devices were heated inside a Statham furnace to a high enough temperature that the base recombination current dominates that from the junction space charge region, as evidenced by a unity slope on the (ln I) vs. (qV/kT) characteristics. The existence of a unity slope indicates that the current I is the sum of the recombination current from the quasi-neutral base and emitter regions. From the experiments [10] and reference [1], we know that the base recombination current dominates for our devices having a base doping concentration less than 10¹⁶ cm⁻³.

Table 5.1 shows the current measurement on devices with seven different areas. Measurements are taken at a temperature of 375.5 ± 0.1°K and at a fixed forward bias of 0.3 volts. The base diffusion length of the devices obtained from X-ray measurement is Lᵖₓ = 80 μm, and the substrate doping concentration is 1.25 x 10¹⁵ cm⁻³.

Experimental results are compared with the theoretical predictions in Fig. 5.2. Good agreement is obtained.

5.4 Determination of Diffusion Length

As Chapter 3 has discussed, the diffusion length can be determined by (a) current measurement or (b) capacitance measurement. The methods of Chapter 3 hold directly provided \( \frac{a}{L_p} \gg 1 \), i.e., they hold for the one-dimensional case only. If the device is small enough that \( \frac{a}{L_p} \gg 1 \) is not true, then methods (a) and (b) must be altered as follows to include the dependence of \( P_n \) and \( J_p \) on two space dimensions.
Figure 5.3 Block diagram for current measurement.
The measured correction factor, $J_a/J_0$, shows excellent agreement with the correction factor, $f(a)$, calculated from the solution to the two-dimensional boundary-value problem.

<table>
<thead>
<tr>
<th>Diameters in Mills</th>
<th>100</th>
<th>70</th>
<th>50</th>
<th>30</th>
<th>20</th>
<th>10</th>
<th>7</th>
</tr>
</thead>
<tbody>
<tr>
<td>$I_a$ (amp)</td>
<td>0.30 x 10^{-3}</td>
<td>0.16 x 10^{-3}</td>
<td>0.85 x 10^{-4}</td>
<td>0.32 x 10^{-4}</td>
<td>0.16 x 10^{-4}</td>
<td>0.47 x 10^{-4}</td>
<td>0.30 x 10^{-5}</td>
</tr>
<tr>
<td>$\alpha = \frac{a}{L_p}$</td>
<td>15.50</td>
<td>11.11</td>
<td>7.94</td>
<td>4.76</td>
<td>3.17</td>
<td>1.58</td>
<td>1.11</td>
</tr>
<tr>
<td>$J_a = \frac{J \alpha}{n a}$ (amp/cm$^2$)</td>
<td>5.92 x 10^{-3}</td>
<td>6.44 x 10^{-3}</td>
<td>6.7 x 10^{-3}</td>
<td>7.02 x 10^{-3}</td>
<td>7.89 x 10^{-3}</td>
<td>9.27 x 10^{-3}</td>
<td>1.21 x 10^{-3}</td>
</tr>
<tr>
<td>$f(a)$</td>
<td>1.03</td>
<td>1.06</td>
<td>1.11</td>
<td>1.22</td>
<td>1.36</td>
<td>1.75</td>
<td>2.15</td>
</tr>
</tbody>
</table>
Method (a): For devices for which the base recombination current greatly exceeds the emitter recombination current (for our devices if $N_B \leq 10^{16}$ cm$^{-3}$), the diffusion length can be determined by the following scheme. Suppose currents are measured (see Section 3) on devices 1 and 2 of radius $a_1$ and $a_2$, then

\[
I_j = A_j J_{a_j} = \pi a_j^2 J_{a_j} = A_j J_0 \frac{J_{a_j}}{J_0} = A_j J_0 f(a_j) \quad (j=1 \text{ or } 2)
\]  

(5.30)

where $a_j = \frac{a_j}{L_p}$, and

\[
J_0 = \frac{n_1^2}{N_{DD}} \frac{1}{L_p} \frac{qV}{kT} = \frac{C}{L_p} = g(L_p)
\]  

(5.31)

There are two equations and three unknowns. To find the value of $L_p$, an iterative procedure is used as follows. Try an initial value of $L_p$ and calculate $a_2 = \frac{a_2}{L_p}$, find the $f(a_2)$ from the curve of Fig. 5.2. Plot $f(a_1) = \frac{J_{a_1}}{J_{a_2}}$. Notice whether $f(a_1)$ is on the curve or not. If not, choose another $L_p$, check $f(a_1)$ again. Continue this procedure until $f(a_1)$ and $f(a_2)$ both lie on the curve. The value of $L_p$ producing this condition is the correct value of $L_p$. Note that good accuracy results provided $a_1$ and $a_2$ differ considerably. For example, $a_2/a_1 = 10$ and $a_1 = 1$ yields accurate determination of $L_p$.

Method (b): If the emitter recombination current is comparable to that of the base (for $N_B \geq 10^{16}$ cm$^{-3}$), the above scheme is inapplicable. However, the capacitance method [1] still can be used to determine $L_p$. This method determines the base diffusion capacitance [11] which, for devices 1 and 2, is
\[
C_D = \frac{dQ_D}{dV} = \frac{dQ_D}{dI_{D,j}} \cdot \frac{dI_{D,j}}{dV} = \frac{1}{\tau_j} \frac{dI_{D,j}}{dV} = \frac{1}{\tau_j} \frac{q}{kT} I_{D,j}
\]

\[(5.32)\]

But \(\tau_1 = \tau_2\) for devices fabricated on the same wafer. Thus the determination of \(L_p\) from \(C_{D1}\) and \(C_{D2}\) is essentially the same as the determination from \(I_{D1}\) and \(I_{D2}\) described in (a) above. The theoretical curve of Fig. 5.2 therefore can also be used for the capacitance method of determining \(L_p\).

In contrast to the methods of Section 5.4, which require two or more devices in determining the diffusion length, the diffusion length can be determined from measurements made on a single device by use of the following iterative procedure. First, \(L_p\) is estimated and the corresponding one-dimensional capacitance (or current) is calculated from (3.11) and multiplied by \(f(a)\) from Figures 5.2.1-5.2.3. The result is compared with the measured (two-dimensional) capacitance (or current). The procedure is continued until self consistency results.

5.5 Discussion

The diffusion length \(L_p\) as a function of temperature \(T\) can also be determined by use of Fig. 5.2. In contrast, the X-ray method does not readily yield \(L_p\) as a function of \(T\) because the small short circuit current (due to the small generation rate of X-ray) is obscured by the thermal noise signals existing in the device and contact wires. Thus the X-ray measurement of \(L_p\) is difficult at temperatures higher than room temperature.
Determination of the diffusion length by the current measurement is simple and accurate.

The validity of the theoretical calculation presented here can be checked by letting \( a \to 0 \), and considering the special case in which the volume recombination vanishes. This produces the electric disk problem [5], illustrated in Fig. 5.1. For \( a = 0 \), \( a_0 = 1, a_2 = a_3 = a_4 = \ldots = 0 \).

Now let the earth ground plate be placed near the electric disk. For example, let \( a/h = 20 \), where \( a \) is the radius of the disk and \( h \) is the distance between disk and earth plate. Then, the average electric field calculated using section 5.2 is

\[
E_{\text{ave}} = \frac{V}{h} \times 1.08
\]

where \( V \) is the potential of the disk. But, if \( a/h \gg 1 \), we know from electrostatics that \( E = V/h \). Thus calculations using Section 5.2 are shown to agree with the results of the analogous electric disk problem.

In our theoretical calculation, we have approximated the shallow discussed junction by an infinitesimally thin disk. The current density becomes infinite at the edge of the junction, which is contrary to the condition occurring in actual diodes. The junction depth \( x_j \) shown in Fig. 5.4 is small but not equal to zero. As is shown in Appendix VI of Ref. [7], the current density at the junction edge can be expressed by

\[
J_{\text{EDGE}} = qD_p \frac{P_{n}(x_j)}{L_p} \frac{K_0\left(\frac{x}{L_p}\right)}{K_1\left(\frac{x}{L_p}\right)}
\]

where \( K_0 \) and \( K_1 \) are the modified Bessel functions of the third kind with orders zero and one, respectively.
Figure 5.4 Exploitation of the symmetry in (a) leads to the simple boundary-value problem of (b), which yields an approximate solution for the edge (sidewall) current density.
REFERENCES FOR CHAPTER 5


PART III: INCREASING $V_{OC}$ BY THE HLE STRUCTURE
CHAPTER 6  Emitter Current Suppression in a High-Low-Junction Emitter Solar Cell Using an Oxide-Charge-Induced Electron Accumulation Layer
INTRODUCTION: As Brandhorst first noted [1], the power conversion efficiency $\eta$ seen in silicon p-n junction solar cells is considerably less than the maximum theoretical value of $\eta$ mainly because the open-circuit voltage $V_{OC}$ is smaller than simple p-n junction theory predicts. Experiments [2] on n$^+$-p silicon cells have shown that this discrepancy in $V_{OC}$ results from the dominance, in the non-illuminated (dark) cell, of the emitter recombination current $J_E$ over the base recombination current $J_B$. In cells having base doping concentrations of the order of $10^{17}$ cm$^{-3}$, for which the largest values of $V_{OC}$ are seen, $J_E$ exceeds $J_B$ by about an order of magnitude [2], rather than being several orders of magnitude less than $J_B$ as is predicted by simple p-n junction theory. The excess $J_E$ has been attributed to the mechanisms [3] of energy band-gap narrowing and lifetime degradation that accompany heavy doping concentrations in the n$^+$ emitter.

To suppress $J_E$ and thus raise the achievable value of $V_{OC}$, a new structure, the HLE junction solar cell, containing a high-low (H-L) junction in the emitter, has been proposed and its performance has been calculated on theoretical grounds [4,5]. The purpose of the present paper is to demonstrate, by experiment, that an H-L junction in the emitter created by an oxide-charge-induced electron accumulation layer can completely suppress $J_E$. We call this device the oxide-charge-induced, high-low-junction emitter (OCI-HLE) solar cell. For the particular structure studied, under AM1 or AM0 illumination, this suppression increases $V_{OC}$ considerably over the values seen in a conventional n$^+$-p solar cell of the same base doping concentration. Larger increases in $V_{OC}$ are to be expected. These expectations are discussed together with the advantages inherent in this type of HLE solar cell.
THE STRUCTURE: As is shown in Fig. 1, the H-L junction in the emitter results from an electron accumulation layer induced in the n-type epitaxial emitter region by a positive oxide charge $Q_0$. The charge $Q_0$ is created near the Si-SiO$_2$ interface during a low temperature oxygen heat treatment [6-8]. To assure a good ohmic contact, a shallow n$^+$ diffusion is made under the metallized portion of the top surface area. This n$^+$ diffusion also provides a high-low-barrier and a small effective surface recombination velocity [9] for holes. The p-type base has a doping concentration $N_{AA} = 5 \times 10^{17}$ cm$^{-3}$, which approximately minimizes $J_B$ in a cell without a back-surface field [10]. For good cell performance, the n-type epitaxial region must satisfy four conditions: (a) it must be thick enough and doped heavily enough to provide the sheet conductance needed to make the series resistance negligible; (b) its doping concentration must be low enough (less than about $10^{17}$ cm$^{-3}$) to avoid the effects of energy band gap narrowing and severe hole lifetime degradation; (c) its hole lifetime must be long enough that nearly all the holes injected from the base diffuse through to the surface rather than recombining within the region; and (d) the surface recombination velocity under the oxide should be low.

The mechanisms underlying the suppression of $J_E$ are simple. As the injected holes diffuse into the electron accumulation layer, the partially reflecting boundary condition [9] at the H-L junction depresses the hole density so that the hole recombination current is controlled by the electron concentration $N_S$ near the surface and the hole surface recombination velocity $S_p$: 

$$J_E \approx \left( n_n^2 S_p / N_S \right) \left[ \exp(qV/kT) - 1 \right]$$

(1)
Here $S_p$ is the surface recombination velocity at the Si-$SiO_2$ interface covering the non-metallized portion of the device surface area; $S_p$ can be orders of magnitude less than the surface recombination velocity at an Ohmic contact. As (1) suggests, if the ratio $S_p/N_s$ is small enough, $J_F$ will be suppressed well below $J_B$.

The oxide-charge-induced H-L structure of Fig. 1 is an attractive realization of the HLE solar cell because it avoids high concentrations of donor impurity atoms in the thin H layer near the surface. This eliminates heavy-doping effects [3] such as band-gap narrowing and hole lifetime degradation in the H layer and it may help the achievement of low values of $S_p$. The avoidance of a diffusion step to form a highly-doped H layer eliminates, over the non-metallized portion of the surface area, the diffusion of unwanted impurities and vacancies into the silicon along with the donor atoms. From the viewpoint of cell performance, these properties suggest that the OCI-HLE solar cell can exhibit good blue response in addition to increased $V_{OC}$.

**CELL FABRICATION:** Several runs of test cells were made that demonstrate the desired $J_F$ suppression. In one such run, the starting material consisted of a 10μm thick n-type layer of resistivity $\rho_{epi} = 0.1 \ \Omega \text{cm}$ grown epitaxially on a p-type substrate of 300μm thickness and resistivity $\rho_{base} = 0.1 \ \Omega \text{cm}$.

The wafer was oxidized in dry O$_2$ with 0.3% TCE at 1100°C to grow a 2500Å thick oxide layer, which was later etched to 1100Å to improve the anti-reflection properties. The temperature of 1100°C was chosen for this test cell to assure a good-quality oxide and a low value of $S_p$. After oxidation the wafer was cooled from 1100°C to 1000°C at the rate of 30°C per hour and then cooled to 700°C in three hours. Holes were opened in the oxide for an n+...
contact diffusion done at 900°C for 20 minutes. The device was then heated in dry oxygen for 2 hours at 700°C to increase the oxide charge density $Q_0$ [6-8].

A second run employed similar processing but used starting material with $\rho_{\text{epi}} = 1.4 \ \text{\Omega cm}$ and $\rho_{\text{base}} = 0.14 \ \text{\Omega cm}$.

EXPERIMENTAL DEMONSTRATION OF $J_E$ SUPPRESSION AND LARGE $V_{OC}$: Since the oxide charge $Q_0$ is near the Si-SiO$_2$ interface, the total charge accumulated in the silicon is $Q_s = Q_0$. The electron distribution is described by MOS theory [11]. The electron surface concentration is about $3 \times 10^{18} \ \text{cm}^{-3}$ ($\rho_{\text{epi}} = 0.1 \ \text{cm}$) and about $2 \times 10^{18} \ \text{cm}^{-3}$ ($\rho_{\text{epi}} = 1.4 \ \text{cm}$). Most of the electrons in the accumulation layer reside within about three Debye lengths (about 100A) from the surface.

The base diffusion length $L_n$, measured by X-ray excitation [12], was $L_n = 20 \mu m$ for devices from the first run and $L_n = 30 \mu m$ for devices from the second run. Measurement of the dark I-V characteristics showed for both runs that the base saturation current (determined by subtracting the space-charge-region recombination current [2]) was equal, within measurement error, to the base recombination current $J_B$ calculated from knowledge of $L_n$. This demonstrates that $J_E << J_B$. Thus $V_{OC}$ is determined mainly by $J_B$ in these devices. The results of the measurements show further that the emitter saturation current $J_{EO}$ is suppressed to less than about $5 \times 10^{-14} \ \text{A/cm}^2$, which corresponds to a voltage, $(kT/q) \ln(J_{SC}/J_{EO}) \geq 705 \ \text{mV}$, for $J_{SC} = 35 \ \text{mA/cm}^2$ (measured in these devices). Thus $J_E$ is suppressed to values so low that it presents no barrier to the achievement of $V_{OC} \approx 700 \ \text{mV}$, which is the maximum value of $V_{OC}$ predicted by classical theory for a 0.1 \ \text{\Omega cm base resistivity for AM0 illumination.}$
The performance of the test cells was measured at NASA Lewis Research Center. The cell of the first run showed the largest $V_{OC}$: $V_{OC} = 634$ mV at 25°C. Because this test cell had much larger metal coverage on the illuminated surface (total area $A_T = 1.34 \text{ cm}^2$; non-metallized area $A_N = 0.972 \text{ cm}^2$) than the 5% metallized area of a cell optimized for AM0 illumination, the measured $V_{OC}$ should be corrected to account for the loading effect of the parasitic p-n junction diode under the metallized portion of the surface. The correction, estimated from $\Delta V_{OC} = (kT/q)\ln(A_T/A_N)$, is about 8 mV. Thus this OCI-HLE cell has an open circuit voltage of 642 mV at 25°C for AM0 illumination. This compares with the largest values of $V_{OC} = 610$ mV previously reported for n-on-p silicon p-n junction solar cells [2,13].

PROJECTIONS: As was emphasized, because $J_E$ is completely suppressed, $V_{OC}$ of these OCI-HLE solar cells depends mainly on $J_B$, which is approximately inversely proportional to $L_n$. Cells of about 0.1 $\Omega$cm base resistivity have shown $L_n$ of approximately 100 $\mu$m [2,14] which is much larger than the values of $L_n$ measured for our OCI-HLE test cells. These low values of $L_n$ may result from several origins. First, in the test cells, we grew SiO$_2$ at 1100°C to assure low values of $S_p$ and complete suppression of $J_E$ at the possible expense of $L_n$. Second, the starting epitaxial material had poor values of $L_n$. For example, the value of $L_n$ measured on the epitaxial p-n junction prior to oxidation and other processing steps was only about 30 $\mu$m for the material used to fabricate the test cells of the first run. We are trying to increase $L_n$ and thus $V_{OC}$ by several approaches, including the growth of SiO$_2$ at lower temperatures, the use of better quality starting material, and the use of a back-surface-field [10] in conjunction with
higher values of base resistivity. An additional improvement in \( V_{OC} \) can be expected from improvements in the short-circuit current density \( J_{SC} \), which will be discussed below. When all of the potential improvements are combined, the projected maximum value of \( V_{OC} \) is about 700 mV at about 25°C.

The measured value of the fill factor \( FF \) at 25°C was 0.802. The value of \( FF \) increases as \( V_{OC} \) increases, and the projected maximum \( FF \), corresponding to \( V_{OC} = 700 \text{ mV} \), is about 0.83 \[15\].

The best measured value of \( J_{SC} \) of these test cells was 35 mA/cm\(^2\). This value resulted even though no special attention was given to achieving good anti-reflection (AR) properties and even though the small \( L_n \) implies poor collection from the longer wavelengths of the solar spectrum. Good blue response was measured at a wavelength of 0.5 \( \mu \text{m} \). Improvements in \( J_{SC} \) will result from improvements in \( L_n \) and the AR properties. The maximum \( J_{SC} \) is expected to occur in OCI-HLE cells made with higher base resistivity (about 10\( \Omega \text{cm} \)) and a back-surface field, as in the Helios cell for which \( J_{SC} = 40 \text{ mA/cm}^2 \) \[16\]. The projected maximum \( J_{SC} \) is about 40 mA/cm\(^2\).

These projections correspond to an AMO maximum power conversion efficiency of OCI-HLE solar cells in the 17 to 18 per cent range, which contrasts with the maximum efficiency now seen in the silicon p-n junction solar cells of about 15% \[16\]. Because of spectral differences, the AM1 efficiency is higher than the AMO efficiency. The projected maximum efficiency of the OCI-HLE solar cell for AM1 illumination is about 20%. High efficiencies also can be expected in OCI-HLE solar cells designed for concentrated sunlight illumination in the range of 25 to 100 suns \[17\].
REFERENCES

Fig. 1 Cross section of an oxide-charge-induced high-low emitter (OCI-HLE) solar cell. Electron accumulation layer is induced by a positive oxide charge $Q_0$. 
CHAPTER 7 DISCUSSION

This report contains the results of research in transition. The first two parts of the report deal with theory and experiment aimed toward an investigation of the basic mechanisms responsible for low $V_{OC}$ in n-on-p solar cells. The third part describes a new device structure that suppresses the emitter current and raises $V_{OC}$ well above values seen before. Our NASA LeRC research now is fully focused on the work described in the third part. In subsequent reports, we will describe in more detail the theory underlying the operation of HLE devices, both OCI and diffused, and will give the results supplied by our ongoing experiments in HLE solar cells.