General Disclaimer

One or more of the Following Statements may affect this Document

- This document has been reproduced from the best copy furnished by the organizational source. It is being released in the interest of making available as much information as possible.

- This document may contain data, which exceeds the sheet parameters. It was furnished in this condition by the organizational source and is the best copy available.

- This document may contain tone-on-tone or color graphs, charts and/or pictures, which have been reproduced in black and white.

- This document is paginated as submitted by the original source.

- Portions of this document are not fully legible due to the historical nature of some of the material. However, it is the best reproduction available from the original submission.

Produced by the NASA Center for Aerospace Information (CASI)
DYNAMICAL ELECTROPHOTOCONDUCTIVITY IN POLYCRYSTALLINE THIN FILMS

Final Report

to

NASA Langley Research Center

Under Grant NSG 1602

Stephen T. Kowel
Philipp G. Kornreich

Principal Investigators

Akbar Nouhi

Department of Electrical and Computer Engineering
Syracuse University
Syracuse, NY 13210

December 1982
ABSTRACT

Studies of the modulation of the photoconductivity of polycrystalline films of cadmium sulfide (CdS) by acoustically induced electric fields are presented. The CdS films were deposited on lithium niobate (LiNbO₃) substrates by vacuum evaporation which was followed by annealing to yield films of high photosensitivity. The change in photoconductivity of these films due to the penetration of electric fields associated with elastic waves propagating on their substrates was demonstrated and studied. The origin of this phenomenon was established by investigating the light intensity and temperature dependences of the conductivity change. The elastic waves traveling on the piezoelectric substrate under the film were generated by interdigital transducers.

The relationship between the acoustic electric field and the induced change in film conductivity was found to be a nonlinear one. The fractional change in conductivity is strongly dependent on the light intensity and the film temperature, showing a prominent maximum as a function of these quantities. The largest recorded fractional change in conductivity was about 25% at electric fields of the order of $10^3$ volts per centimeter.

A phenomenological model was developed based on the interaction between the space charge created by the electric field and the electron trapping states in the photoconductor. We have assumed the existence of a quasi-continuous distribution of impurity states in the forbidden gap which act either as traps or as recombination centers depending on the excitation intensity and the film temperature.
1. INTRODUCTION

The electric field-induced effects upon the charge carriers in a semiconductor have long been of considerable theoretical and practical interest to physicists and engineers. For instance, the application of an electric field to a crystal may change the concentration mobility, and/or the energy distribution function of the free carriers in the crystal. Numerous studies have been published dealing with these effects. They reveal much interesting information about the energy band structure in some materials while also being of great help in elucidating the scattering of the carriers by the lattice, impurities and other defects in crystalline solids. A change in characteristics of the free carriers in a crystal usually results in a change in its electrical conductivity. And this has found many applications in solid state science and technology. In fact, many solid state electronic devices are either based on some effect of electric field on the charge carriers or necessarily involve this in their operation.

We here present experimental data and a theoretical model for the electric field induced modulation of the photoconductivity of polycrystalline films of CdS. The CdS films were vacuum deposited on a lithium niobate (LiNbO₃) substrate followed by suitable heat treatment to increase their light sensitivity. In the films used, photoductivity was found to be a superlinear function of light intensity when the film was at room temperature, and becomes a sublinear function of the excitation intensity at relatively low film tempera-
Due to electrical signals representative of the special Fourier components of the light pattern on the film. The electrical signals from the DEFT sensors can be processed for purposes of pattern recognition, motion detection, and electronic focusing in optical systems.

After carefully examining the existing theoretical models concerning the electric field-induced changes in the photoconductivity, we have found them unable to explain the large field modulation of the photoconductivity of our CdS film and its dependence on the various parameters that we have observed. In view of this, we have developed a new model whose predictions are in good agreement with our experimental observations. This model is based on the combined effect of electron trapping states and the space charge, induced by the space dependent acoustic-electric field, on concentration of free carriers in the conduction band.

We assumed a very fast electron exchange between some low lying localized electron trapping states and the conduction band. This means a rapid establishment of a quasi thermal equilibrium between some electron traps and the free electrons.

Franz [3] and Keldysh [4] have shown that the application of a strong electric field can alter the optical absorption coefficient of a crystal. The theory of this phenomenon, known as the Franz-Keldysh effect, has been further developed and studied by Callaway [5] and Dow and Redfield [6]. It has been shown that the optical absorption thresholds of semiconductors and insulators should shift toward the longer wavelengths in the presence of an electric field. Vavilou and Britsyn [7] carried out experiments studying the effect of strong electric fields on the optical absorption of single crystal of silicon.
They observed substantial increase in the optical absorption of these crystals in the wavelength region of 0.8 to 0.9 micrometers for applied electric fields of about $5 \times 10^6$ V/cm. Similar changes have been reported by French(8) for single crystals of GaAs for electric fields of about $10^5$ V/cm. Electric field induced changes in optical absorption and photoconductivity of single crystals of CdS also have been reported by Williams(9,10) for electric fields as big as $10^5$ V/cm.

Unfortunately, we have not been able to reconcile our measurements with the Franz-Keldysh effect. The change in optical absorption of a crystal and the resultant change in its photoconductivity, due to applied electric field, as predicted by this model, is independent of excitation intensity and temperature. However, our experimental observations do not support such predictions. In addition we have also recorded the electric field modulation of the dark conductivity in some of our films which indicates that some other mechanism is at work.

Another mechanism through which the electric field may change the conductivity of a crystal, is the Poole-Frankel(11) effect. This theory states that the application of a strong electric field to a solid increases the thermal excitation probability of the charge carriers from bound localized states by lowering the height of the binding potential of the bound electrons. In fact, the application of the electric field not only can ease the thermal ionization of trapped electrons, but can also decrease the capture cross section of the trapping centers. This, however, results in further increase in the density of free carriers. Indeed, the electric field-assisted changes in thermal emission and capture parameters of trapping centers in single crystals of CdS have been reported by Bube and Dussel(12).
The application of the Poole-Frankel model to our problem predicts an increase in photoconductivity which is an exponentially increasing function of the applied electric field, indicating a nonlinear relationship between the photocurrent and applied biasing drift voltage. However, the current-voltage characteristics of our films are linear over a large range of voltages. Moreover, this model predicts a fractional change in the photoconductivity which is independent of excitation intensity and monotonically increases with decreasing temperature, whereas our observations are quite different.

We also considered the "barrier model" which assumes the existence of potential barriers between the crystallite and intercrystallite regions in a polycrystalline film. This model, first proposed by Petritz(13), and used by other authors to explain field effect conductances in CdS film,(14) indicate that the I-V characteristics of the film, in which the potential barriers exist, ought to be strongly nonlinear. Such nonlinear behavior is absent in our CdS films.

Finally, we will show that the theory of the acousto-electric effect alone is not also capable of fitting our observations. This theory, based on the space charge creation due to the spatial gradient of the electric field, has been successfully used by many authors to explain the harmonic generation and the acoustic amplification in structures such as separated and combined media convolvers.(15) We will demonstrate that the acousto-electric effect can only predict the bias independent signals in our films while the model developed here is compatible with the whole range of our experimental results.
2. CHARGE REDISTRIBUTION AND THE ELECTRON TRAPPING STATE

As in the acoustic-electric effect, the spatial equilibrium distribution of the charge carriers is perturbed when an acoustically induced electric field penetrates into the CdS film. However, we should also take into account the presence of the localized energy states in the forbidden band gap. It is a well known fact that these defect states play a very important role in the mechanism of the photoconductivity. It is known that the existence of impurity states in the band gap is essential to the explanation of many interesting features of photoconductivity. In ordinary semiconductors the conductivity, in the absence of any incident radiation or particle bombardment, is basically due to the thermally freed charge carriers that are in thermal equilibrium with the conduction band, valence band and impurity states in the band gap. But, in the case of photoconductivity the electrons are excited from the valence to the conduction band by means of the photon absorption and not by thermal emission. This means that in a photoconductor the conduction and valence bands are not in thermal equilibrium any more. However, after a steady state is established, the phot-generated electrons will reach a quasi-thermal equilibrium state with some shallow electron trapping states in the forbidden gap. Such an equilibrium, which is provided by the fast scattering processes in a solid, requires a very fast electron exchange between the conduction band and the shallow electron trapping states \(7,16\). Indeed, the time needed for a change in the number of electrons in the conduction band to be seen by the electron trapping states, or vice versa, is
of the order of scattering times in solid, about $\tau_e = 10^{-13}$ seconds. For example, the photoelectrons generated in the initial stages of illumination are captured by the electron trapping centers after staying only about $10^{-13}$ seconds in the conduction band. Conversely, these states also start to supply electrons to the conduction band within about $10^{-13}$ seconds after the illumination is stopped.

On the other hand, the electron lifetime which is a measure of the speed of recombination processes through which the electron is finally transferred from the conduction to the valence band, is about $\tau_L = 10^{-3}$ seconds, much longer than the typical scattering times in solid. Now, consider a disturbance in the steady state density of the conduction electrons, such as an interruption in the illumination. Let us assume that the duration of such a change induced in the steady state density of free electrons be much shorter than $\tau_L$ but significantly longer than $\tau_e$. Now, it is quite reasonable to assume that a change of this nature will not to be seen in the recombination channel but change the occupancy of the electron trapping states.

Of course, what was said about the electrons is also true for holes. Nevertheless, in our n-type CdS films the majority carriers are electrons and the electrical conductivity due to the holes is negligibly small. Therefore, from now on we will consider only the contribution of the electrons to the current.

Let us now consider a thin photoconducting CdS film acted on by a space-varying electric field which also oscillates rapidly in time. Such an electric field will push some of the free electrons out of some regions of the film and accumulate them in some other regions.
This will produce a space-varying charge distribution through the film as described by Coulomb's law

\[ \nabla \cdot \vec{D} = q\delta n \]  

(2-1)

where \( D \) is the displacement vector, and \( \delta n \) is the excess or scarce electrons per unit volume. Therefore, \( \delta n \) can be positive or negative and in general oscillates with the frequency of oscillating acoustic field. In our experiments, we used sinusoidal acoustic-electric fields with the frequency range of several megahertz corresponding to time periods of about \( 10^{-6} \) seconds. This means that when the space varying electric field accumulates \( \delta n \) excess electrons in a region of the film these electrons stay there for about \( 10^{-6} \) seconds which is much longer than \( 10^{-13} \) seconds, the time scale for fast communication between the trapping states and the conduction band electrons, so that some of these electrons will be captured by the trapping centers. This will leave less than \( n_0 + \delta n \) free electrons available for electrical transport, where \( n_0 \) is the concentration of free photogenerated electron in the absence of the perturbation \( \delta n \). On the other hand there will be more than \( n_0 - \delta n \) electrons (per unit volume) available for conduction in the region from which \( \delta n \) free electrons per unit volume have been taken away, because some of the previously trapped electrons will be excited to the conduction band.

Now, one may ask the following question: What is the actual local density of free electrons available to conduction in the presence of such an oscillating electric fields? Therefore, in the following we will attempt to calculate the electric field induced perturbation
in the local density of the free electrons by taking into consideration some of the concepts discussed here along with some of the mechanisms governing the phenomenon of photoconductivity. First, we will study the simple case of a photoconductor in which there is only one kind of electron trapping states in the band gap. Then, the method of calculation developed for this simple case will be applied to the more general and realistic case of a photoconductor with a quasi-continuous distribution of electron trapping state in the forbidden gap.
3. A PHOTOCONDUCTOR WITH ONLY ONE TRAP LEVEL

Let us consider a photoconductor in which the shallow electron trapping centers of density \( N_t \) have only one energy level, \( E_t \), as shown in Fig. 1. In the presence of light when the quasi-Fermi level has an energy \( E_{fn} \), the density, \( n_o \), of photo-generated electrons in the conduction band is given to a good approximation by

\[
n_o = N_c \exp[-(E_c - E_{fn})/kT] \quad (3-1)
\]

where \( N_c \) is the effective density of states at the bottom of the conduction band

\[
N_c = \frac{2 \hbar^2}{m^* kT} \quad (3-2)
\]

Here \( m^* \) is the electrons effective mass. The average number \( n_t \) of electrons in the shallow electron trapping states of energy \( E_t \) are determined by the Fermi probability function and are given by

\[
n_t = N_t f(E_t) = N_t/[1+\exp(E_t - E_{fn})/kT] \quad (3-3)
\]

The last expression for \( n_t \) can be put into a simpler form as follows

\[
n_t = N_t/[1+\exp(E_t - E_c + E_c - E_{fn})/kT] \quad (3-4)
\]

where \( E_c \) is the lowest energy state in the conduction band. Substitution for \( \exp[(E_c - E_{fn})/kT] \) in Eqn. (3-3) from Eqn. (3-1) yields
Figure 1  Schematic energy diagram for a photoconductor whose electron trapping centers of density \( N_e \) have only one energy level \( E_c \): (a) situation in the dark (b) at intermediate light level.
\[ n_t = \frac{N_t}{1 + \left(\frac{N_c}{N_o}\right)\exp\left(E_t - E_c\right)kT} \]  \hspace{1cm} (3-5)

On the other hand, \( \frac{N_c}{N_o}\exp\left(E_t - E_c\right)/kT \) is the density of electrons in the conduction band when the light intensity is such that the electron Fermi level associated with the shallow trap coincides with the trapping energy level \( E_t \). Let us denote this constant quantity by \( n_f \). i.e.

\[ n_f \equiv \frac{N_c}{N_o}\exp\left[-\frac{(E_c - E_t)}{KT}\right] \]  \hspace{1cm} (3-6)

Substitution of Eqn. (3-5) into Eqn. (3-4) results in the following simpler form for \( n_t \):

\[ n_t = \frac{n_o N_t}{(n_o + n_f)} \]  \hspace{1cm} (3-7)

This equation, which relates the density of trapped electrons \( n_t \) to the steady-state concentration of free photogenerated electrons \( n_o \), is a consequence of the assumption that there is always a quasithermal equilibrium between the electrons in the conduction band and the electron shallow trapping states in the band gap.

Now, let us consider the case in which some \( \delta n \) electrons are pushed away from one region and are accumulated in another region. Let these excess electrons be kept there for a very short period of time, much shorter than the time needed for recombination of electrons, before returning to original places. As we discussed earlier, during such a short period of time the probability for these excess electrons to drop to the valence band or to the recombination centers is negli-
gibbly small. However, some of these accumulated electrons can still be captured by the shallow trapping states because of the fast communication between these states and the conduction band. Therefore, these extra electrons will be distributed only over the conduction band and the electron traps. If \( n \) and \( n_t \) are the new local densities of free and trapped electrons respectively, then one can write the conservation of electron number density as

\[
\begin{align*}
    n + n'_t &= n_0 + n_t + \delta n \\
\end{align*}
\]

(3-8)

Let the new electron Fermi level, after the perturbation \( \delta n \) is induced, be \( E_{fn}' = E_{fn} + \Delta E_{fn} \), where \( \Delta E_{fn} \) is a displacement in the position of electron Fermi level caused by the perturbation \( \delta n \). Now, one can repeat the calculations previously presented for \( n_t \) to find a similar relationship between the new densities of trapped and free electrons.

\[
\begin{align*}
    n_t' &= n N_t'/(n + n_f') \\
\end{align*}
\]

(3-9)

Substitution for \( n_t \) and \( n_t' \) from Eqns. (3-6) and (3-9) into Eqn. (3-8) results in

\[
\begin{align*}
    n^2 + n(n_f + N_t - A - \delta n) + n_f \delta n + A n_f \\
\end{align*}
\]

(3-10)

Where

\[
A = n_o + n_o N_t/(n_o + n_f) \\
\]

(3-11)

It is clear from Eqn. (3-10) that \( n \) is a non-linear function of \( \delta n \).

Let us expand \( n \) in powers of \( \delta n \) around \( \delta n = 0 \). Note that \( n = n_o \) when \( \delta n = 0 \). Therefore, one can write
We calculate the first two derivatives of $n$ with respect to $\delta n$ from equations (3-10). We obtain,

$$\frac{dn}{d(\delta n)} \bigg|_{\delta n=0} = \left( n_0 + n_f \right)^2 / \left[ (n_0 + n_f + N_f)(n_0 + n_f) - n_o N_f \right]$$

(3-13)

$$\frac{d^2n}{d(\delta n)^2} \bigg|_{\delta n=0} = \frac{2n_f N_f (n_0 + n_f)/[(n_0 + n_f)^2 + n_f N_f]^3}{(3-14)}$$

Suppose the photoconductor that we have discussed here is a thin film of CdS deposited on a piezoelectric substrate which is acted on by an electric field accompanying a surface acoustic wave traveling on the film's substrate. Then, as discussed earlier, the perturbation $\delta n$ is exactly the space charge created in the film by the acousto-electric field and oscillates with the frequency of the acoustic wave. For a traveling acousto-electric field in the form of

$$E = E_1 \cos(k \cdot \vec{r} - \omega t)$$

(3-15)

the space charge density, $\delta n$, can be approximated by

$$\delta n = \alpha_0 E_1 \sin(k \cdot \vec{r} - \omega t)$$

(3-16)
where $E_1$ is the amplitude of the traveling electric field in the film and $\alpha$ is given to a good approximation by

$$\alpha = \frac{\mu}{v_o} \gamma ([\gamma^2 + (\omega/\omega_D)^2]^3)$$

(3-17)

where $\mu$ is the electron mobility, $v_o$ and $\omega$ are respectively the velocity and frequency of the acoustic wave carrying the electric field and

$$\gamma = 1 + \left(\frac{v_d}{v_o}\right)$$

(3-18)

in which $v_d$ is the electron drift velocity induced by dc biasing voltage applied to the film through the contact lines. The electron drift velocity in our films is much smaller than the acoustic velocity because of the small electron mobility. Thus, $\gamma$ is approximately equal to one. The electron diffusion frequency $\omega_D$ is given by

$$\omega_D = \frac{v_o}{D_n}$$

(3-19)

where $D_n$ is the electron diffusion coefficient. The dependency of $\delta n$ on $E_1$ and $n_o$, predicted by equation (3-16), has been justified by our experimental results. This will be discussed in detail in the next section. Now, let us substitute for $\delta n$ into Eqn.(3-12) from Eqn.(3-16) to obtain

$$n = n_o + \frac{dn}{d(\delta n)} \left|_{\delta n=0}^{\alpha n_o E_1 \sin(k \cdot \hat{r} - \omega t)} + \right.$$

$$\frac{1}{2} \frac{d^2n}{d(\delta n)^2} \left|_{\delta n=0}^{(\alpha n_o E_1)^2 \sin^2(k \cdot \hat{r} - \omega t)} + \right.$$
Let us denote by, \( \Delta n(E^2) \), the second order term in the series of Eqn. (3-20) or Eqn. (3-12) and calculate the ratio \( \Delta n(E^2)/n_0 \). This ratio will be referred to as the fractional change in photoconductivity. Therefore, substitution for \( \frac{d^2n}{d(\delta n)^2} \bigg|_{n=0} \) into Eqn. (3-20) from Eqn. (3-14) yields

\[
\frac{\Delta \sigma}{n_0} = \Delta n(E^2)/n_0 = \alpha^2 E_1^2 < n_e n_e ^* \left( \frac{n_0 + n_f}{(n_0 + n_f)^2 + n_f^* n_e} \right)^3 N > \sin^2 (k \cdot r - \omega t)
\]

(3-21)

It is clear from Eqn. (3-21) that the fractional change in photoconductivity, \( \frac{\Delta n(E^2)}{n_0} \), is a function of the light intensity through \( n_0 \). By setting the derivative with respect to \( n_0 \) of the quantity inside the curly bracket in Eqn. (3-21) to zero, one can show that the fractional change in conductivity has a maximum at some value of \( n_0 \) corresponding to a finite light intensity. In Fig. 2 we have plotted the ratio \( \Delta n(E^2)/n_0 \), given by Eqn. (3-21) versus \( n_0 \) for three different reasonable values of \( n_f \) and \( N_e \), while \( \alpha E_1 = 0.7 \) was assumed for all three curves.

All three curves indeed show a maximum at some finite light intensity as the density and the energy level of the electron trapping states change. The effect of concentration and energy level of the electron trapping centers on the maximum fractional change in photoconductivity will be discussed in detail in the next section.
Figure 2 Fractional change in electron number density as a function of electron density for a simple photoconducting model in which all the electron traps have only one energy level at $E_c$. 

\[ \frac{\Delta n(E^2)}{n_e} = 10^{-3} \]

\[ 10^{15} \quad 10^{16} \quad 10^{17} \quad 10^{18} \quad 10^{19} \quad 10^{20} \]

\[ N_t = 10^{17} \]
\[ N_t = 10^{18} \]
\[ N_t = 10^{19} \]
We previously discussed the mutual interaction between the bunched electrons and electron shallow traps in a photoconductor considering only a single level of electron trapping states. But in reality there is always a variety of impurity and defect states distributed in the band gap. The existence of such a distribution of states in photoconducting CdS films has already been reported by other authors (17, 18). A typical form of such an impurity state distribution is shown in Fig. 3. Let \( g_t(E) \) be the number of impurity states per unit volume per unit energy interval at energy level \( E \) close to the conduction band. Let the density of free photo-generated electrons in the film be \( n_0 \) and the electron quasi-Fermi level be at the energy level \( E_{fn} \) in the absence of any acoustic-electric field. Here \( n_t \) is given by

\[
E_c \n_t = \int_{E_{fn}}^{E_c} g_t(E)f(E)dE
\]

where \( f(E) \) is the quasi-Fermi distribution function associated with the shallow traps. According to the arguments in the previous section, after applying the acousto-electric fields to the film the density of free and trapped electrons will change to \( n \) and \( n_t' \). Consequently, the electron Fermi level will shift to some other energy \( E_{fn}' \), such that

\[
E_{fn}' = E_{fn} + \Delta E_{fn}
\]

where \( \Delta E_{fn} \) is the amount of energy shift in position of the electron quasi-Fermi level caused by the electron bunching. If the change in
Figure 3  Typical density of states distribution of a CdS film showing impurity state distribution and quasi Fermi energies for electrons and holes.
local density of electrons by the acoustic-electric field is $\delta n$. Then, the statement of the conservation of the electron number density is

$$n + n'_C = n_o + n'_t + \delta n$$  \hspace{1cm} (3-24)

We intend to calculate $n$ as a function of $n_o$ and $\delta n$. Combination of the Eqns. (3-1) and (3-23) yields

$$n = N_c \exp \left(-\frac{(E_c - E_{fn}')}{kT} + \frac{\Delta E_{fn}}{kT} \right)$$  \hspace{1cm} (3-25)

By combination Eqn. (3-1) and Eqn. (3-23) and taking the logarithm of the result we obtain

$$\Delta E_{fn} = KT \ln\left(n/n_o\right)$$  \hspace{1cm} (3-26)

The density of trapped electrons in the presence of space charge, induced by acoustic-electric field in the film, is given by an expression similar to Eqn. (3-22)

$$n'_t = \int_{E_{fn}}^{E} g'_C(E)f'(E)dE$$  \hspace{1cm} (3-27)

where $f'(E - E_{fn}')$ is the new Fermi function. Let us expand $f'(E)$ in powers of $\Delta E_{fn} = E_{fn}' - E_{fn}$ at point $E_{fn}' = E_{fn}$. The mth derivative of $f'(E)$ with respect to $E_{fn}'$ is proportional to $(1/KT)^m$. Thus, the mth order term in the series will be proportional to $(\Delta E_{fn}'/KT)^m$. For $\Delta E_{fn}$ much smaller than KT we will consider only the contribution of the first two terms of this series

$$f'(E) = f(E) + \left[\frac{\partial f'(E)}{\partial E_{fn}'}\right] \Delta E_{fn}$$  \hspace{1cm} (3-28)
Substitution for $f'(E)$ into Eqn. (3-27)

$$n'_t = \int_{E_{fn}}^{E_c} g_t(E)(f(E) + R(E)\Delta E_{fn})dE$$

(3-29)

Where

$$R(E) = \left[ \frac{\partial f'(E)}{\partial E_{fn}} \right]_{E_{fn}}$$

One can eliminate $n'_t$ and $n'_c$ from Eqn. (3-21) using Eqns. (3-29) and (3-22) to obtain

$$E^c_{\text{ct}} - n_o + \Delta E_{fn} \int_{E_{fn}}^{E_c} g_t(E)R(E)dE = \delta n$$

(3-30)

We define

$$L \equiv \int_{E_{fn}}^{E_c} g_t(E)R(E)dE$$

(3-31)

and substitute for $\Delta E_{fn}$ from Eqn. (3-26) into Eqn. (3-30) to obtain

$$\delta n = n_n - n_o + KTL \log(n/n_o)$$

(3-32)

We pointed out earlier that $\delta n$ is linearly dependent on the space varying acoustically induced electric field in a CdS film. Thus, it is clear from Eqn. (3-32) that $n$ is a non-linear function of the acoustic field. Now, let us expand $n$ in powers of $\delta n$, around $\delta n=0$, as we did in the previous section,

$$n = n_o + \left[ \frac{dn}{d(\delta n)} \right]_{\delta n=0} \delta n + \frac{1}{2} \left[ \frac{d^2n}{d(\delta n)^2} \right]_{\delta n=0} (\delta n)^2 + ...$$

(3-33)
The first and the second derivatives of \( n \) with respect to \( \delta n \) are found by taking the corresponding derivatives of the both sides of the Eqn. (2-30). The results of such calculations are

\[
n = n_o + n_o \delta n/(n_o + kTL) + \left[ n_o KTL/2(n_o + KTL)^3 \right] (\delta n)^2 + \ldots
\]  

(3-34)

Let us use again the approximate value given for \( \delta n \) by Eqn. (3-16), and denote by \( \Delta n(E^2) \), the term proportional to the second power of the electric field in the expansion for \( n \) in Eqn. (3-34). Then the fractional change in photoconductivity induced by the bunching of electrons with acoustic-electric field, \( \Delta n(E^2)/n_o \), is given by

\[
\Delta n(E^2)/n_o = 1/n_o \left[ n_o KTL/2(n_o + KTL)^3 \right] (\delta n)^2
\]  

(3-35)

Substitution for \( \delta n \) from Eqn. (3-16) yields

\[
\Delta n(E^2)/n_o = (\alpha^2 E^2_1/2)[n_o KTL/2(n_o + KTL)^3 \sin^2(k \cdot \mathbf{r} - \omega t)] (3-36)
\]

Both \( n_o \), the free electron concentration, and \( L \) given by Eqn. (3-31) are functions of the intensity of light incident on the film. \( n_o \) monotonically increases with increasing intensity of illumination. But the value of \( L \) and its light dependency is mainly determined by the shallow electron trapping states concentration through the location of the electron Fermi level, \( E_{fn} \), associated with these traps as can be seen in Eqn. (3-31). Consequently, the fractional change in free electron concentration is a function of light intensity through the quantity in square brackets in Eqn. (3-36). Its variations with light
light intensity is such that it stays close to its maximum value as long as the value of KTL is comparable to the steady state density of free photogenerated electrons \( n_0 \). Fractional change in density of photoelectrons falls much below its maximum value when KTL becomes much smaller (or much larger) than \( n_0 \). On the other hand, the value of KTL can roughly be approximated (See Eqn. (3-31)) by the density of electron trapping states at the location of electron Fermi level associated with the shallow traps, \( E_{fn} \). Thus, one can conclude that the form of the impurity states distribution functions \( g_t(E) \) determine the variation of the fractional change in concentration of the free electrons with light intensity.

The fractional change in density of free photo-generated electrons induced by the acoustic-electric field was calculated from Eqn. (3-34) for various values of \( n_0 \), equivalent to different illumination intensities on the film. In these calculations \( g_t(E) \) given below was employed to represent the impurity state distribution in the film.

\[
g_t(E) = g_0 \exp[-(E_C - E)/\beta] \quad (3-37)
\]

The results of these calculations for several different values of \( \beta \) and \( g_0 \) are shown in Fig. 4 and 5 as plots of fractional change in electron concentration versus steady state electron density \( n_0 \).

We will show in the next section that the characteristics of these curves, representing the behavior of the fractional change in photoconductivity of a film by acoustic electric fields, are in fairly good quantitative agreement with our experimental observations of the electrophotoconductivity in CdS films.
Figure 4  Calculated fractional change in conductivity versus electron concentration.
Curve I for \( g_o = 10^{20} \text{ m}^{-3} \), \( \beta = 0.08 \text{ ev} \) and Curve II for \( g_o = 5 \times 10^{21} \text{ m}^{-3} \).
\( \beta = 0.08 \text{ ev} \).
Figure 5: Calculated fractional change in conductivity versus electron concentration for
$\frac{\delta \sigma}{\sigma_0} = 5 \times 10^{-12}$ m$^3$, $\beta = 0.042$ ev.
4. EXPERIMENTAL OBSERVATIONS

The CdS films used were fabricated by vapor deposition. These films exhibited a poor photosensitivity, slow responses and very small ratio of light conductivity to dark conductivity (hereafter will be referred to as light to dark ratio, L/D). The CdS films were subsequently heat treated. It is only after heat treatment in the presence of O₂, N₂, Cu and Cl that the film acquires high photoconductivity and fast response to the light. It turns out that these changes come about essentially from the introduction of imperfection states in the forbidden gap as a result of doping and also due to the crystallization effect of the high annealing temperature. What follows is a description of our experimental observations.

In all the experimental descriptions that follow the electrical currents were always measured between two interdigital metal electrodes on the film. These electrical contact patterns, were made by first evaporating a thin layer of indium onto the CdS film and then a layer of aluminum on top of the indium. Indium was used since it is believed to make a good Ohmic contact to the n-type CdS films. The separation between individual electrodes were about 60 micrometers, and there were 200 strips of CdS between the metal contact lines.

The spectral response of the photoconductivity of these films was the first to be studied. Monochromatic light was obtained from a white light source using a grating monochromator. Fig. 6 displays a typical spectral distribution of the photocurrent, normalized to equal numbers of incident photons, for a CdS film. It is seen that the photons of
Figure 6 Spectral distribution of photocurrent at 24°C. Normalized to equal numbers of incident photons.
wavelengths of about 500 nm have the maximum efficiency for free carrier generation corresponding to a band gap of \( E_g = 2.47 \text{ ev} \). A small amount of photocurrent (about 12% of the maximum photocurrent at 500 nm) was detected for photons having wavelengths in the range of 510 nm to 610 nm. Clearly the photocurrent in this wavelength region, which is still considerably larger than the dark current, arises from electrons excited to the conduction band from impurity states at 2.03 electron volts (ev) below the conduction band or 0.43 ev above the valence band. These states which are negatively charged in the dark because they are occupied with electrons, may act as hole traps in the presence of light. These states are most probably introduced by Cu impurities.

Next we measured the rise time and decay time of the photocurrent at various light intensities. Here we have defined the rise time as the time it takes for the photocurrent to rise to 90% of its steady state value from the moment the film is exposed to the light. Likewise, the decay time is the time needed for the photocurrent to drop to 10% of its previous steady state value after the light is removed. Fig. 7. shows the rise time for the photocurrent as a function of light intensity (white light) for a CdS film deposited on a LiNbO_3 substrate at room temperature. It is shown that the rise time of the photocurrent decreases from about 0.5 second to 4 milliseconds as the light intensity on the film increases from 2 foot candles to 150 foot candles (the light meter that we used calibrated in terms of foot candles where one foot candle is equivalent to 15.83 mW/m^2).
Figure 7  Rise time of photocurrent as a function of light intensity at 24°C.
Such a light dependency for the rise time of the photoconductivity is an indication of the presence of electron trapping states in the band gap close to the conduction band. The steady state concentration of the photo-generated electrons in the conduction band is reached only after these electron trapping states are filled with electrons. On the other hand, the lower the excitation intensity the longer it takes to fill out the electron traps. Therefore, a longer rise time should be expected at lower excitation intensities if strong electron trapping states are present.

The decay time of the photocurrent in a typical CdS film is plotted in Fig. 8 as a function of light intensity. It is seen that the decay rate of the photocurrent after the illumination is terminated decreases with increasing excitation intensity. For a photoconductor without electron trapping states the decay time of the photocurrent will be equal to the lifetime of the photogenerated electrons. However, the presence of the electron traps elongates the decay time, since the trapped electrons are gradually thermally excited back to the conduction band after the excitation is stopped. These electrons stay in the conduction band and maintain a conductivity higher than dark conductivity until being removed from the conduction band by recombination. This, of course, will continue until all the trapped electrons are recombined through their transfer to the conduction band.

Fig. 9 displays a plot of photocurrent versus applied voltage, for a typical CdS film fabricated on a LiNbO₃ substrate. During the experiment the light intensity on the film was kept constant and the photocurrent was measured for different values of the applied drift voltage.
Figure 8  Decay time of photocurrent as a function of light intensity at 24°C.
Figure 9. Current versus voltage applied to the electrodes on a photoconducting CdS film.
Such a linear I-V characteristic, as we discussed in Section 1, eliminates the possibility of potential barrier formation at crystalline boundaries in our polycrystalline CdS films.

Also studied was the dependency of the photoconductivity of the CdS films on the light intensity. This was done by measuring the photocurrent through the film at various excitation intensities. The result of such a measurement is shown in Fig. 10 as a plot of the photoconductance of the film as a function of light intensity for a typical CdS film at room temperature (about 24°C).

The photoconductivity of the CdS films was always found to be proportional to some power, \( \gamma \), of the light intensity, I, on the film. Where it was found that \( 1 < \gamma < 2 \) for photoconducting CdS films on LiNbO₃ substrate at room temperature. For example, we found \( \gamma = 1.7 \) for the film corresponding to the plot in Fig. 10. It was demonstrated that the photocurrent in these films increases superlinearly (i.e. \( \gamma > 1 \)) with increasing light intensity when the film is at room temperature.

The effect of temperature on the photoconduction properties of the CdS film was also studied. In these measurements the sample was suitably mounted in a vacuum system in order to prevent water vapor from condensing on the sample when cooled and atmospheric oxygen being absorbed when it was heated. We studied the effect of the temperature on the photoconduction of the CdS film by measuring its resistance at various temperatures while the intensity of light on the film was kept constant. In Fig. 11 we have shown plots of the photoconductance against the reciprocal of temperature for the same film that was used in measurements corresponding to Fig. 10. It is seen that, as the temperature of the film decreases,
Figure 10  Photoconductance of a Cds film as a function of white light intensity at 24°C.
Figure 11. Photoconductance of a CdS film as a function of temperature at various light intensities.
its photoconductance increases and finally reaches a region, at relatively low temperature, in which it becomes almost insensitive to a further decrease in temperature. The rate of change of photoconductivity of the film with temperature decreases with increasing light intensity. For low level excitation the photoconductivity of the film increases by a factor of more than $10^2$ as the temperature of the film drops from room temperature to about, $-100^\circ C$. The electron mobility cannot change significantly over this temperature range $^{(31)}$. Therefore, it is concluded that the density of conduction electrons rises as the temperature drops.

A CdS film with superlinear photoconductivity at room temperature (Fig. 15) was cooled down to $-160^\circ C$, and then its photoconductance at various light intensities was measured. The result of these measurements is plotted in Fig. 12. It is clear that the dependency of the photoconductivity, $\sigma$, of the film on the light intensity, $I$, has changed from $\sigma \propto I^{1.7}$ at room temperature to $\sigma \propto I^{0.9}$ at $-160^\circ C$.

The experimental justification and studies of the modulation of photoconductivity by acoustically induced electric field in our CdS films were made possible through the measurement of electrical signal currents indicative of such changes. These are the signal currents produced by part of the photoconductivity modulation which is proportional to the square of the acoustic electric fields. These signals, expected from the anticipated non-linear relationship between the change in conductivity and the acoustic-electric field, can be calculated either by using the Eq. (3-34) or starting from the series expansion of the photoconductivity in terms of the electric fields. A complete discussion of this prescription is
Figure 12 Photoconductance of a CdS film as a function of white light intensity at -160°C.
presented elsewhere\(^1,2\), therefore, only a brief review is given here. To this end, we will start with considering the case of two surface acoustic waves traveling in the same direction, \(x\), on the substrate under the film. We will assume, for simplicity, that the acoustic loss is negligible and that the electric field effects are uniform through the thickness of the film. Consequently, for two piezoelectric surface waves propagating in the \(x\)-direction, the associated electric fields inside the film can be approximated by the following plane waves:

\[
E_x = E_1 \cos(k_1 x - \omega_1 t) + E_2 \cos(k_2 x - \omega_2 t) \tag{4-1}
\]

\[
E_y = 0 \tag{4-2}
\]

\[
E_z = E_1 \cos(k_1 x - \omega_1 t) + E_2 \cos(k_2 x - \omega_2 t) \tag{4-3}
\]

Due to the spatial gradient of these electric fields, a space charge is created in the film. A linearized solution to problem will approximate the density of bunched electrons, \(\delta n\), also by a plane wave such as

\[
\delta n = \alpha_1 E_1 n_0 \sin(k_1 x - \omega_1 t) + \alpha_2 E_2 n_0 (k_2 x - \omega_2 t) \tag{4-4}
\]

\(\alpha_1\) and \(\alpha_2\) are constants that we introduced in previous sections. Let us assume that the electrical contacts to the film are provided by an interdigital finger pattern whose lines are running parallel to \(x\)-direction on the plane of the film. Application of a CD voltage to these contact electrodes will generate a DC biasing field \(E_d\) which will result in an electrical current flowing in \(y\)-direction. The current density \(J\) at any point in the film is given by

\[
J = \sigma E_d = q\mu n E_d \tag{4-5}
\]
where $q$ and $\mu$ are respectively the electronic charge and mobility, and the free electron density $n$ in a film under the influence of acoustic-electric fields is obtained from Eqn. (3-35) and (4-4). Substitution for $n$ in Eqn. (4-5) reveals the fact that the current density contains components oscillating with frequencies of $\omega_1 - \omega_2$, $\omega_1$, $\omega_2$, $\omega_1 + \omega_2$ etc. We are interested only in the component of the current that oscillates with the difference frequency of the two acoustic waves, $\omega_1 - \omega_2$. This current, denoted by $i_{ac}$, will be referred to as the electrophotoconductance (EPC) signal. The EPC signal $i_{ac}$ for the case which is presently discussed can easily be calculated to be

$$i_{ac} = \frac{12J(\omega_1 - \omega_2)dx}{d} C E_0 \frac{n_0^3}{(n_0 + KTL)^3} \cos(kx - \omega t)dx$$

(4-6)

where

$$C \approx \frac{q^2 \alpha_1 E_2 E_1 h}{4}$$

$$k = k_1 - k_2$$

$$\omega = \omega_1 - \omega_2$$

(4-7)

$h$ is the film thickness and integration in Eq. (4-6) is over the whole film area contributing to current. For a uniformly illuminated film the density of photogenerated electrons, $n_0$, is independent of $X$ and $Y$ coordinates and Eq. (4-6) can be simplified to:

$$i_{ac} = C E_0 \frac{n_0^3}{(n_0 + KTL)^3} \int \cos(kx - \omega t)dx$$

(4-8)

From Eqs. (4-6) and (4-8) it is expected that the EPC signals $i_{ac}$ to be proportional to the biasing field $E_d$ as well as being proportional to the Fourier components of the light distribution on the film when
it is illuminated by a uniform light or a periodic light-dark pattern. Both of these expectations were satisfied by our experimental observations. One can also calculate the total DC current through the film in the absence of any acoustic electric field to obtain:

\[ i_{dc} = \int q \mu n_0 E \text{d}x \text{d}y = q \mu E n_0 h \text{d}x \text{d}y \quad (4-9) \]

Dividing the Eq. (4-8) by Eq. (4-9) yields the following

\[ \frac{i_{ac}}{i_{dc}} = \frac{n_0^2 KTL}{(n_0 + KTL)^3} \propto \frac{\Delta n(E^2)}{n_0} \quad (4-10) \]

The constants of proportionality in Eq. (4-10) are independent of \( n_0 \) and \( KTL \). Thus the light intensity and temperature dependency of the measured \( \frac{i_{ac}}{i_{dc}} \) is expected to be the same as that of the calculated ratio \( \frac{\Delta n(E^2)}{n_0} \).

So far we have shown that when two surface acoustic waves are propagating on the piezoelectric substrate of the film, an EPC signal oscillating at the difference frequency of the two waves will be detected at the metal contact lines on the film. These are the signal currents that we have measured in all the experiments to be described in the rest of this section.

The surface acoustic waves were created by interdigital transducers laid down on the LiNbO\(_3\) substrate. Usually six transducers were fabricated, two on each of three sides of the Cds film/contact region. These transducers were made by vacuum evaporation of aluminum on the LiNbO\(_3\) substrate, using the proper masks and photolithographic techniques. In most of our experimental measurements we used the
transducers that had center frequencies of about 30 MHz and a useable
bandwidth of about 4 MHz. The transducer terminals were simply con-
nected to the output of an rf signal generator. The driving voltage
was typically about 1 to 5 volts peak-to-peak. The efficiency of
the transducers could have been improved by compensating for its
capitance using an inductor of approximately 1 micro henry.

A complete schematic diagram of the experimental arrangement used
to measure the EPC signals is shown in Fig. 13. Two surface acoustic
waves (SAW) traveling in the same direction on the LiNbO₃ substrate
were generated by exciting the two adjacent transducers on the same
side of the CdS film/contact region (the central square in Fig. 13).

From all of the modulation product signals generated in the film
we wished to extract only the EPC signal at the difference frequency
of the two waves. Thus, a very selective low-pass filter was inserted
directly after the metal contacts. This filter, with 60 dB gain differ-
ence between 30 MHz and 1 MHz effectively eliminates the feed-through
and all of the high sum frequency signals. Thus, only the signals oscil-
lating at the difference frequency of the two acoustic waves, could pass
through the filter.

The signal was then fed into a current amplifier for detection.
This amplifier was designed to maintain a constant dc biasing photo-
current through the film, i_{dc}. This meant that the dc voltage applied
to the metal contacts would change with light intensity. For ac signals
the input impedance of the amplifier was very low, only about 11 ohms.
The low impedance implies that the amplifier was a "current amplifier".
Figure 13  Schematic diagram of experimental arrangement for measuring electrophotoconductance signals
If the input current into a current amplifier is \( i_{ac} \), and if its output voltage is \( V_{ac} \), then the transresistance is defined as

\[
R_T = \frac{V_{ac}}{i_{ac}} \quad (4-11)
\]

\( R_T \) can be determined by injecting a known ac current into the amplifier and measuring the output voltage. Once \( R_T \) is known, Eqn. (4-11) can be used to calculate \( i_{ac} \) from the observed values of the output voltage, \( V_{ac} \). We used an amplifier with a transresistance of 160 kilo-ohms. The substrate, low pass filter, and the amplifier were all mounted in an aluminum box that was suitably designed to minimize the electromagnetic feed-through from the transducer drivers to the pick-up amplifier.

Usually, one transducer was driven by a signal generator at fixed frequency of about 30 MHz while the other transducer was excited by a generator sweeping from 25 MHz to 35 MHz. This way, all the EPC signals \( i_{ac} \) representative of the spatial Fourier components of the image projected on the film, with frequencies of up to 5 MHz corresponding to spatial frequencies up to 13 cycles per centimeter, could be displayed on an oscilloscope screen for simultaneous comparison of their amplitudes. In all the experiments to be described here, we projected either a uniform light or a simple light-dark bar pattern onto the Cds film. For such simple light patterns, the amplitudes of the EPC signals at the difference frequencies of the two acoustic waves were found to be exactly proportional to the Fourier components of the light intensity distribution on the film.

We first studied the effect of the dc biasing current on an EPC signal. This was done by measuring an EPC signal for various applied biasing currents while the intensity and the distribution of light on the film and the voltages applied to the transducers were all kept constant.
A typical result of such a measurement is displayed in Fig. 14 as the measured peak-to-peak value of a signal at 2 MHz is plotted against the biasing current. It is seen that the size of the signal is linearly proportional to the biasing current. This proportionality relationship between $i_{ac}$ and $i_{dc}$ justifies the existence of the proposed electric field modulation of the photoconductivity in our CdS films.

According to the model that we have developed for electrophotoconductivity, explained in detail in the last two sections, it is the local density of the conduction electrons in the CdS film which is altered by the acoustic electric fields. The modulated density of the free photo-generated electrons, n, is given by Eqn. (3-30).

All the quantities in Eqn. (3-36) have already been introduced previously. Let us define the new parameter, $\eta$, to be

$$\eta = \frac{KTL}{n_0} \quad (4-12)$$

Substitution of Eqn. (4-12) in Eqn. (4-10) yields

$$\frac{i_{ac}}{i_{dc}} \propto \frac{\Delta n(E^2)}{n_0} \propto \left[\frac{\eta}{(1 + \eta)^3}\right] \quad (4-13)$$

In the remaining part of this section we will describe the observed characteristics of the EPC signals, a measure of the electric field modulation of photoconductivity of the CdS films. This will be done in parallel with an attempt to demonstrate the consistancy of these experimentally observed characteristics with predictions of our theoretical models. Of course, we do not intend to investigate the exact quantitative match between the experimentally measured values of the EPC signals, $i_{ac}$, and their calculated values based on our theoretical model. This really re-
Figure 14  EPC signal, $i_{ac}$, as a function of DC biasing current, $i_{dc}$. 
quires a detailed knowledge of the distribution and concentration of the
impurity states just below the conduction band edge, the mobility of
electrons, and other such parameters. In addition, it also requires an
exact and complete solution to the problem of surface acoustic wave pro-
pagation in the presence of the photoconducting CdS film. Instead, we
will demonstrate a good qualitative agreement between the theory and
experiment, using reasonable approximations for the quantities involved.

We have demonstrated in Section 3 that the fractional change in
electron concentration, $\Delta n(E^2)/n_0$, is a function of the light intensity
on the film through the quantities $n_0$ and $L$ inside the square brackets
in Eqn. (3-34). While $n_0$ is proportional to the excitation intensity,
the value of $L$ at any excitation level is mainly determined by the con-
centration of the impurity states close to the quasi-Fermi level, of the
electrons in rapid communication with the conduction electrons. As a
function of $\eta$, given by Eqn. (4-12), the quantity inside the square brackets
in Eqn. (4-13) stays close to its maximum value as long as the ratio $\eta$
remains close to $\eta_{\text{max}}$. Where $\eta_{\text{max}}$ is the value of $\eta$ for which the quan-
tity inside the square brackets in Eqn. (4-13) is at its maximum. It can
easily be shown that $\eta_{\text{max}} = 0.5$. Accordingly, the fractional change in
conductivity, $\Delta n(E^2)/n_0$, decreases from its maximum when $\eta$ deviates appre-
cialby from $\eta_{\text{max}}$. This happens when $n_0$ is much smaller or much bigger
than the value of KTL. Thus, it is the profile and size of the impurity
state distribution function, $g_t(E)$, which determines the light dependency
of $\Delta n(E^2)/n_0$. Impurity distributions having high concentrations at
energies close to the conduction band, and decaying toward the middle
of the band gap, have been reported for photoconducting CdS thin films similar to ours. Therefore, quite reasonably, the existence of such impurity states in our CdS films also was assumed and approximated by the distribution function $g_t(E)$ given in Eqn. (3-37). This distribution function was employed to calculate, $L$, from Eqn. (3-31), and then the dependence of $\Delta n(E^2)/n_o$ on the steady state electron density $n_o$, from Eqn. (3-36).

Some results of such calculations are shown in Fig. 3 and Fig. 4 in which the fractional change in $n_o$ is plotted versus $n_o$. In these plots the parameters $g_o$ and $\beta$, defined in Eqn. (3-33), are kept constant over each individual curve, but at least one of them changes from one curve to the other. It is clear that the electrically induced fractional change in free electron concentration becomes a maximum for some value of $n_o$, and that the position of this maximum is shifted as a result of a change in the profile and the magnitude of the impurity state distribution function, $g_t(E)$.

On the other hand, the dependency of $i_{ac}/i_{dc}$ on the steady state concentration of electrons, $n_o$, should be the same as that of $\Delta n(E^2)/n_o$. Since $n_o$ is proportional to the intensity of light on the film, therefore the light intensity dependency of $i_{ac}/i_{dc}$ is expected to follow a pattern similar to the curves displayed in Fig. 3 and Fig. 4.

The experimental investigation of the effect of excitation intensity on the fractional change in photoconductivity of the CdS films was made possible by measurements of the EPC signals for various intensities of white light on the film. During these measurements the driving voltages
and frequencies of the transducers, the biasing current, the film temperature, and the light pattern projected on the film were all kept unchanged. Thus, the only varying element was the excitation intensity, which generates free carriers in the film. Similar measurements were repeated for several different Cds films. A typical result obtained from these experiments is shown in Fig. 15 in which the measured $i_{ac}/i_{dc}$ is plotted against light intensity for two different Cds films. At each light level the free electron density in the film was calculated from its measured photocconductance. Curves in Fig. 15 are redrawn in Fig. 16 as a plot of $i_{ac}/i_{dc}$ versus free electron density, $n_0$, in the film. Comparison of the curves in Fig. 16 and those in Fig. 3 shows a remarkable consistancy between theoretically predicted and experimentally observed modulation of photocconductivity in a Cds film. Fig. 15 reveals that, starting at very low light intensities, the normalized EPC signals, $i_{ac}/i_{dc}$, increases with increasing light intensity and reaches its maximum value, and then decreases. Such a behavior was found to be universal for all the heat-treated Cds films. However, the position of the maximum was found to differ from one film to another. This, can presumably be attributed to variations of the density of impurity states from film to film. Indeed, we are aware of the fact that our impurity diffusion and film curing technique does not produce identical Cds films. This can be shown by means of very simple film characterization techniques. Of course, a drastic difference between different Cds films is not expected either, as long as the same recipe is employed.
Figure 15  Normalized signal current as a function of light intensity for two different CdS films at room temperature.
Figure 16  Measured fractional change in conductivity for two different CdS films.
In Fig. 17 the measured normalized signal current, $i_{ac}/i_{dc}$, is plotted as a function of excitation intensity for a CdS film that was subjected to a heat-treatment quite different than the usual one. The film was simply heated in the furnace at about 650°C for about an hour in the absence of all the dopants that we use in our usual film annealing process.

This film had a clear yellowish color, unlike the ordinarily heat treated CdS films which have a dark brownish appearance. The density of the impurity states, at least in this film, is expected to be quite different from those described earlier. Therefore, according to the predictions of our previously discussed model, the light intensity dependency of the normalized signal current in this film is expected to be considerably different from those in the other films. This is clearly the case as is seen from the comparison of Fig. 15 with Fig. 17.

We explained earlier that the quantity inside the square brackets in Eqn. (4-13) is really a function of the ratio $\eta = KTL/n_0$, and has a maximum. Its maximum occurs when $\eta$ becomes equal to $\eta_{max} = 0.5$.

On the other hand, it was pointed out that $\eta$ itself is a function of the radiation level on the film through $n_0$ and $L$. Any change in light intensity will also change the parameter $\eta$, but its dependency on light is mostly determined by $g_\epsilon(E)$. Therefore, $i_{ac}/i_{dc}$ approaches its maximum when the light intensity is such that the electron density $n_0$ becomes comparable to $KTL$ so that $\eta$ assumes a value close to $\eta_{max}$.

It might be more practical to identify the position of the maximum in $i_{ac}/i_{dc}$ in terms of the corresponding values of $n_0$ and $L$ at this maximum rather than specifying it with the related light intensity.
Figure 17 Normalized signal current as a function of light intensity for a film which was heat treated in the absence of all the dopants that we use in our conventional heat treating method.
If we call them \((n_0)_{\text{max}}\) and \(L_{\text{max}}\), then the maximum in \(i_{\text{ac}}/i_{\text{dc}}\) is expected to be attained whenever \((n_0)_{\text{max}}\) and \(L_{\text{max}}\) are achieved. Moreover, a set of \(n_0\) and \(L\), corresponding to some light intensity on the film at room temperature, can be obtained at much lower light levels if the film is cooled down to lower temperature. This means that for a film at a temperature below room temperature, \(n_{\text{max}}\) will occur at a light intensity lower than the one for which \(n\) becomes maximum at room temperature. According to this argument the position of the maximum in \(i_{\text{ac}}/i_{\text{dc}}\) should shift toward lower excitation intensities by lowering the film temperature. It was indeed experimentally observed that the maximum in \(i_{\text{ac}}/i_{\text{dc}}\) moves toward lower light intensities as the film temperature decreases. For example, in Fig. 18 curve I is a plot of measured normalized signal current, \(i_{\text{ac}}/i_{\text{dc}}\), as a function of light intensity when the film was kept at room temperature while curve II is the result of similar measurements made when the film was at \(-160^\circ\text{C}\). It is clear that the excitation level corresponding to the maximum has moved from about 100 foot-candles at room temperature to about 5 foot candles at \(-160^\circ\text{C}\), as we expected. The flatness of the curve II around its maximum can partly be due to the change in light dependency of the photoconductivity of the Cds films from superlinear at room temperature to sublinearity at lower temperatures.

We have also studied the effect of the change in film temperature on the normalized EPC signal current, \(i_{\text{ac}}/i_{\text{dc}}\) at constant excitation intensity. This was done by measuring the normalized EPC signal at various temperatures (usually below room temperature) while the intensity of the light pattern projected on the film was kept constant. A typical result
Figure 18 Normalized signal current, $\frac{i_{AC}}{i_{DC}}$, as a function of light intensity. Both curves correspond to the same Cds film, film was at room temperature for Curve I and at $-160^\circ$C, for Curve II.
is given in Fig. 19. It demonstrates the fact that starting at room
temperature, the fractional change in photoconductivity of the film in-
creases to an almost steady-state regime (the flat portion of the curve).
The maximum in $\frac{i_{ac}}{i_{dc}}$ shifts to higher temperatures as the intensity of
light on the film increases. Interestingly enough, this is also consis-
tent with, and indeed justifies, our foregoing discussions, and should
have been expected. For instance, consider the point corresponding to
room temperature on curve I in Fig. 19. This point is actually the
same as the point corresponding to 10 foot-candles on curve I in Fig. 18.
Consequently, starting from this point the decrease in temperature
(curve I in Fig. 18) ought to have the same effect on $n_0$ and $L$ as does
the increase in light intensity (curve I in Fig. 18). This is due to
the fact that the increase in light intensity and decrease in temperature
have similar effects in displacing the quasi-Fermi levels in the band
gap. Increasing the light intensity or decreasing the temperature have
the same effect on $\frac{i_{ac}}{i_{dc}}$.

Though the steady-state concentration, $n_0$, of the photogenerated
electrons is always proportional to some power of the illuminating light
intensity, the value of $L$ is almost entirely decided by the density of
imperfection states at the electron Fermi level. Let us assume that the
impurity state distribution, $g(E)$, is such that the change in $L$, as the
electron Fermi level moves toward or away from the conduction band, is
smaller than the change in $n_0$. This means that $L$ will not be as sensitive
to light intensity changes as is $n_0$. In a situation like this, any change
in the position of the electron Fermi level will alter more through the
changes in $n_0$ rather than $L$. In this case the effects of the variation
Figure 19 Normalized signal current, I/I₀ versus reciprocal temperature when the light intensity was kept unchanged at 10 and 20 foot candles.
intensity and temperature on $i_{ac}/i_{dc}$ are mostly influenced by their effect on $n_0$ rather than on $L$ or the location of the electron Fermi-level. This means that the normalized EPC signal, $i_{ac}/i_{dc}$, becomes maximum whenever $n_0$ reaches a particular value, regardless of the level of excitation and temperature. In other words the fractional change in conductivity remains unchanged as long as there is no change in the density of free electrons in the conduction band. This can clearly be seen from the flat portion of the curves in Fig. 18. In this regard we would like to emphasize the fact that our model for electric field modulation of the photoconductivity is based only on the existence of the fast communicating electron traps in the band gap. It is only the interaction of these fast trapping states with conduction electrons which brings about the kind of modulation that we have studied here. According to this model the electric field induced change in photoconductivity does not depend on the detail of the steady-state photoconductivity characteristics like its superlinear or sublinear light dependency. The modulation is governed only by the density of free electrons in the conduction band and the parameters corresponding to the fast electron traps.

For instance the free electron concentration $n_0$ at which the maximum in curve I (Fig. 18) has occurred is almost equal to the electron concentration corresponding to the maximum in curve II in the same figure. This can easily be seen from Fig. 10 and Fig. 12 by realizing the fact that the maximum in curve I corresponds to 100 ft. cads. at room temperature while the light intensity and temperature for the maximum point on curve II are
respectively about few foot-candles and -160°C. Moreover, the steadiness of $I_{ac}/I_{dc}$ on the flat portions of the curves in Fig. 17 is due to the fact that the density of the photogenerated electrons is almost constant, and does not change due to a further drop in film temperature. This is evident from the temperature dependence of the photoconductance of the CdS films shown in Fig. 11. Thus, we may conclude that $L$ is indeed a slowly varying function of the excitation intensity in our photoconductive CdS films.

Further justification for this may come from the fact that for the theoretically calculated curve I in Fig. 3, when the electron concentration changes from $10^{15}/cm^3$ to $10^{19}/cm^3$, the value of $KTL$ changes only from about $5 \times 10^{18}/cm^3$ to $10^{20}/cm^3$. This means that over a light intensity range for which the density of photogenerated electrons, $n_o$, changes by a factor of $10^6$, the value of $L$ changes only by a factor of 20. One should recognize that the theoretical curve I in Fig. 3 resembles the experimental curve I in Fig. 18.

We explained in Section 1 how the penetration of the acoustically induced electric fields into a photoconductive CdS film produces a space charge in the film due to the redistribution of the free charge carriers. The change in local density of free electrons is proportional to the intensity of the inducing acoustic-electric field $E_a$, and the steady state density of free electrons $n_o$, i.e. $n_oE_a$. This requires the existence of a signal current flowing through the film and oscillating with the different frequency of the two acoustic waves, even in the absence of an applied drift voltage. These signals which we have called "unbiased signals" should obviously be proportional to the steady-state density of photo-
electrons \( n_0 \). This demands an identical light intensity dependency for both the ordinary photoconductance and the unbiased signals in a CdS film. Since these quantities both are proportional to \( n_0 \) which is a function of excitation intensity. We were able to detect and measure some unbiased signals in some of our films at relatively high excitation intensities (usually above 60 foot-candles). These signals were usually large enough to be detected when both surface acoustic waves were traveling in the direction perpendicular to the contact electrode lines. In Fig. 20 and Fig. 21 we have plotted the ordinary photoconductance and an unbiased signal as a function of light intensity for two different CdS films. It is clear from these plots that the photoconductance and the unbiased signals in a CdS film both have exactly the same light intensity dependencies.
Figure 20 The photoconductance and unbiased signal as a function of light intensity in a CdS film.
Figure 21. The photoconductance and the unbiased signal as a function of light intensity in a CdS film.
5. CONCLUSIONS

Our studies demonstrated that the photoconductivity of polycrystalline CdS films deposited on a LiNbO$_3$ substrate is modulated by electric fields associated with surface acoustic waves (SAW) traveling on the piezoelectric substrate of the film. This change in conductivity of the film in the presence of light proved to be a large effect and a non-linear function of the acoustically induced electric field. For instance, conductivity changes up to 25% and proportional to the square of the acoustic-electric field were detected. It is also evident that the fractional change in conductivity of the CdS films is strongly dependent on the excitation intensity and the film temperature.

Our conclusions tend to reject the possibility that the existing models, concerning the electric field induced changes in the photoconductivity of materials, can explain our experimental observations. On the other hand, the theoretical model that we have developed which was presented in Section 3 seems to be successful in explaining and predicting the electrophotoconductance in our CdS films. This model is based on the presence of shallow trapping states in the photoconductor and redistribution of photocarriers by electric fields accompanying the SAW waves propagating on the film's piezoelectric substrate. According to this model the important parameters affecting the size of the change in photoconductivity of the films are the intensity of the acoustic electric field, the mobility of free carriers in the film, and the density and energy distribution of electron traps in the band gap. For instance, at fixed values of the acoustic-electric fields, the fractional change in photoconductivity approaches its maximum value whenever the density of free photogenerated electrons becomes comparable to the density of electron trapping states at the position of the
position of the quasi-Fermi level associated with the electrons in shallow trapping states. However, the density of photogenerated electrons and the position of the electron quasi-Fermi level both are determined by the level of excitation and the film temperature. Therefore, the light intensity and the temperature dependence of the electrophotoconductance which we observed in our experiment are expected.