GERMANIUM BLOCKED
IMPURITY BAND (BIB) DETECTORS

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1. **INTRODUCTION**

- Extrinsic, photoconductive semiconductor detectors cover the infrared spectrum from a few μm up to 250 μm.

- Photoconductors exhibit high responsivity and low noise equivalent power.

- The Si blocked impurity band (BIB) detector invented by M. D. Petroff and M. G. Stapelbroek has a number of advantages over standard bulk photoconductors. These include:
  - smaller detection volume leading to a reduction of cosmic ray interference
  - extended wavelength response because of dopant wavefunction overlap
  - photoconductive gain of unity
2. **Ge BIB**

- The success of Si BIB detectors has been a strong incentive for the development of Ge BIB detectors.

- The advantages of Si BIB detectors stated above should, in principle, be realizable for Ge BIB detectors.

- If Ge BIB detectors can be made to work out to 250 μm with high responsivity and sufficiently low dark current, they could replace stressed Ge:Ga photoconductors.

- Can the dark current be reduced to acceptable levels?
Figure 1(a). Schematic of conventional detector.

Figure 1(b). Schematic of BIB detector.
Figure 2(a). Doping levels in a conventional Ge detector.

Figure 2(b). Doping levels in a Ge BIB detector.
Fig. 3. Schematics of space charge, electric field and potential energy for a reverse biased p-type BIB detector.
3. Ge BIB DETECTOR DEVELOPMENT

3.1. Epitaxial Blocking Layer Devices

3.1.1. Ge epitaxy

- Whereas Si epitaxy techniques have been developed to a very high degree of perfection, Ge epitaxy has been attempted only on a few occasions.

- Ge chemistry is very different from Si chemistry.

- Ultra-pure Ge compounds [Ge(CH₃)₄, Ge(C₂H₅)₄] are being developed for III-V semiconductor technology. They may be useful to Ge epitaxy.
Substrate choice and preparation

- We have used a number of different crystals with various crystallographic orientations in the development of Ge epitaxy:

  - n-type wafers (~$10^{11}$ cm$^{-3}$) are used for the electrical characterization of the epitaxial layers which are typically p-type because of residual copper contamination (junction isolation).

  - p-type wafers (~$10^{15}$ cm$^{-3}$) are used for I-V comparison tests with conventional photoconductors.

  - p-type wafers (~$2 \times 10^{16}$ cm$^{-3}$, low compensation) are used for Ge BIB detectors.
• Wafer polishing process:

  - mechanical planar lapping with alumina slurry.
  
  - mechano-chemical polishing with syton containing $\text{H}_2\text{O}_2$.
  
  - brief etch in $\text{HNO}_3$:HF (3:1) followed by soak in HF (1% in H$_2$O) to remove oxides.

• Epitaxy:

  - first experiments with atmospheric pressure vapor phase epitaxy (VPE). Disadvantage: high substrate temperature, H$_2$ diluted feed gas (contamination, diffusion of dopants into the blocking layer).
  
  - current experiments are performed with low pressure VPE. Advantage: low substrate temperature.
Fig. 4. Schematic of horizontal VPE apparatus. Quartz tube is 5.7 cm O.D. x 75 cm long.
3.1.2. Characterization of epi layers

- Optical micrographs
- Variable temperature Hall effect and resistivity
- Rutherford backscattering (channeling) spectrometry (RBS)
- Secondary ion spectrometry (SIMS)
- Spreading resistance measurements
(100) \( N_D - N_A = 1 \times 10^{14} / \text{cm}^3 \)
580\( ^\circ \)C; 5 sccm GeH\(_4\),
with H\(_2\) reduction step,
polycrystalline deposition

(113) \( N_D - N_A = 2 \times 10^{14} / \text{cm}^3 \)
580\( ^\circ \)C; 5sccm GeH\(_4\),
with H\(_2\) reduction step,
no growth (etching)

(113) \( N_D - N_A = 5 \times 10^{11} / \text{cm}^3 \)
550\( ^\circ \)C; 10sccm GeH\(_4\),
no H\(_2\) reduction step,
single crystal deposition

Fig. 6. Optical micrographs of Ge epi layers.
Fig. 7. Variable temperature Hall effect measurements of a Ge epilayer on an n-type [113] substrate. The hole freeze-out curves indicate a light copper contamination. The two curves (+, *) are measurements of the same sample and demonstrate reproducibility.
Fig. 8. RBS channeling spectra of a Ge epi film (#636). The "cloudy" region shows significant dechanneling indicating a high defect concentration.
Fig. 9. SIMS of LPVPE Epi Films: Oxygen Concentration
Fig. 10. SIMS of LPVPE Epi Films: Carbon Concentration
3.1.3. Preliminary detector test results

- Responsivity
- Dark current
Fig. 11. Spreading resistance as a function of depth from the epilayer surface for: (a) an area of epilayer close to the leading edge of the wafer in II-16 where the growth rate was \( \sim 0.06 \mu \text{m/min} \), and (b) an area of epilayer farthest from the leading edge of the same wafer where the growth rate was \( \sim 0.02 \mu \text{m/min} \). The slight rise in resistivity at the very surface is due to the native oxide.
Fig. 12. Responsivity as a function of bias for detectors 13-2 and 13-3 at 2.3 K under reverse bias. The substrate material is moderately doped ($5 \times 10^{15} \text{ cm}^{-3}$). Such material exhibits hopping conduction but does not have extended wavelength spectral response. Tests were performed with a narrow band filter at $\lambda = 98.9 \mu \text{m}$. 
Fig. 13. Dark current as a function of detector bias for detector 13-2 with an epilayer and for the same "detector" without an epilayer at 2.3 K under reverse bias. Below a bias of ~100 mV, the blocking layer effectively reduces hopping conduction in this moderately doped material ($N_A - N_D = 5 \times 10^{15} \text{ cm}^{-3}$).
Fig. 14. Dark current as a function of bias for detectors 13-2 and 13-3 at 2.3 K under reverse bias.
3.2. Ion Implanted BIB Detectors

• Concept:
  - In case pure and structurally perfect epitaxial layers are hard to produce, we can resort to implantation of dopants into an ultra-pure crystal.

• Low energy B⁺-implantation tests:
  - three B⁺ energies: 150 keV, 95 keV, 50 keV form a 0.4 μm thick layer with \( N_A = 3.5 \times 10^{16} \text{ cm}^{-3} \).
  - annealing at 400°C for one hour in argon.
  - extended wavelength response.
  - responsivity = 0.5 A/W, dark current < 10⁻¹⁴ A, at bias = 100 mV and \( T = 2.0 \text{ K} \). \( \text{NEP} \approx 4 \times 10^{-16} \text{ W/ \sqrt{Hz}} \).
Fig. 15. Responsivity of a Ge BIB detector, low energy B⁺-implant type. Active layer depth = 0.6 μm, [B] = 1 × 10¹⁶ cm⁻³, λ_filter = 98.9 μm, f_chopper = 23 Hz
Fig. 16. Spectral response of Ge BIB detector, low energy B⁺-implantation type.
• High energy B⁻-implantation tests:

- 14 implant energies up to 4 MeV doubly charged boron ions lead to a 5 µm thick layer with \( N_A = 1 \times 10^{16} \text{ cm}^{-3} \).

- Variable temperature Hall effect and resistivity measurements indicate full activation of shallow acceptor dopant B. No deep levels are detectable after annealing. Below 15 K, hopping conduction becomes dominant.

- Infrared transmission measurements and device tests are in progress.
Fig. 17. Ge BIB, high energy ion implant profile
Fig. 18. Free carrier freeze-out of high energy B⁺-implanted layer. Before annealing (+), the slope of the freeze-out curve is steeper than after annealing (*). The latter slope corresponds to ~10.4 meV, the binding energy of shallow boron acceptors. Below 15 K, hopping conduction becomes dominant.
Fig. 19. Resistivity as a function of inverse temperature of the high energy B⁺-implant layer before (+) and after (*) annealing. Hopping conduction becomes dominant below 15 K.
4. **CONCLUSIONS**

- A LPVPE technique for the low temperature growth of epitaxial Ge layers has been developed.

- Hall effect and resistivity measurements indicate that the epi layers are lightly p-type due to residual copper contamination.

- First generation Ge BIB detectors made with moderately doped substrates \( (5 \times 10^{15} \text{ cm}^{-3}) \) exhibit effective blocking of the hopping current.

- First generation Ge BIB detectors exhibit responsivities around 1 A/W.

- Second generation devices using low pressure VPE are being processed.

- Ion implanted active layers are tested.

- It is currently not known what temperatures will be required to reduce the dark current down to levels which are acceptable for SIRTF applications.