Failure Analysis of a Helicopter External Fuel-Tank Pylon

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Abstract

An eight-inch-long (0.2 m) crack was found in an external helicopter fuel-tank pylon. The damaged pylon was removed from service and analyzed at NASA Langley Research Center to determine the cause of the crack. Results of the analysis revealed that crack initiation occurred at corrosion pits in a fastener hole and crack propagation was a result of fatigue loading.

Introduction

At the U.S. Coast Guard air station in Elizabeth City, North Carolina, an eight-inch-long (0.2 m) crack was found in an external fuel-tank pylon of a HH-60 “Jayhawk” helicopter. External fuel tanks are used to extend the range and operation times of U.S. Coast Guard (USCG) helicopters and are fastened to pylons attached to the helicopter fuselage. Fuel-tank pylons are fabricated from 7075 aluminum alloy forgings that are machined into the final configuration. The HH-60 is capable of carrying three external pylon-drop fuel tanks: one 80-gallon (300 L) tank and two 120-gallon (450 L) tanks. The crack was found in a 120-gallon tank pylon. Photographs of a Coast Guard HH-60 helicopter are shown in Figure 1. The photograph of the entire helicopter fuselage in part (a) shows the location of a 120-gallon fuel tank. As seen in part (b), the pylon that connects the fuel-tank assembly to the fuselage is normally covered and hidden from view. A photograph of a fuel-tank pylon is shown mounted to the fuselage in part (c); the pylon is visible because the covering has been removed for inspection.

Figure 1. Photographs are shown of the external pylon-mounted fuel-tank assembly on the HH-60 helicopter. (a) A 120-gallon fuel tank is shown on the fuselage. (b) A more detailed photograph of the fuel-tank pylon attachment is shown. (c) A pylon is seen after the fuel-tank assembly was removed for inspection.
The failure analysis reported herein concentrated on finding evidence of fatigue loading and/or corrosion that could lead to cracking of the fuel-tank pylon. Helicopter components are known to experience fatigue (cyclic) loading resulting in possible fatigue cracking (refs. 1, 2). Additionally, USCG helicopters are routinely exposed to corrosive marine environment and cracking may be exacerbated by environmental-load interactions that produce brittle crack growth resulting from stress corrosion cracking and/or corrosion fatigue (ref. 3). After the crack was detected, the damaged pylon was removed from service and shipped to NASA Langley Research Center in Hampton, Virginia for detailed destructive examinations to determine the cause of cracking.

Initial Visual Inspection

A photograph of the cracked pylon is shown as Figure 2a. The left curved-side of the pylon was mounted vertically to the helicopter fuselage; the arrow in the upper-right corner of Figure 2a indicates the vertical-up direction. The fuel-tank assembly was attached to the pylon through the two holes on the far right side of Figure 2a. Loads applied to the pylon are nominally expected in the vertically downward direction, applied at the end of the horizontal (cantilever) portion of the pylon, creating high bending stresses in the corner region of the pylon as indicated in Figure 2a. The crack appears to have initiated at fastener holes in this high stress region and propagated in a vertically downward direction. Eventually the crack propagated into a vertical stiffener and appeared to have arrested. Figure 2b is a more detailed view of the crack region outlined in Figure 2a. The dashed line in Figure 2b parallels the crack from the apparent region of crack initiation (fastener holes in the upper flange) to crack arrest at a vertical stiffener.

![Figure 2. Photographs of the cracked pylon are shown. (a) A photograph of most of the pylon is shown. (b) A more detailed image is given of the cracked region.](image-url)

The top of the pylon upper flange, in the crack initiation region, is shown in Figure 3a. Here, the crack is seen bisecting two fastener holes, each $\frac{3}{16}$-inch (4.8 mm) in diameter; presumably, the crack initiated at one of these holes. The bottom surfaces of the upper flange near these fastener holes are shown in Figures 3b and 3c. The two smaller holes adjacent to the large hole located on the left of Figure 3a were used to attach the bracket shown in Figure 3b. Results presented later in this paper indicate that crack initiation occurred at the fastener hole shown in Figure 3b.
Destructive Examination

To perform detailed examination of the crack surface, the pylon was broken by carefully applying a load at the end of the horizontal cantilever while the curved vertical portion was locked into position. This loading allowed the horizontal cantilevered portion of the pylon to rotate and fracture the remaining ligament without disturbing the crack surfaces. A photograph of the broken pylon is shown in Figure 4a. Only the crack surface associated with the smaller piece of the pylon was examined; the larger piece was preserved for future examination. A photograph of the crack surface in the upper flange (where crack
initiation appeared to occur), and a portion of the web, is shown in Figure 4b. Visual inspection of the crack surfaces revealed dark-colored regions on the crack surfaces, primarily near the fastener holes where the crack appeared to initiate. These dark spots, indicated in Figure 4b, are corrosion products (likely oxide debris) that are typically produced during fatigue crack growth of aluminum alloys in an aggressive environment (refs. 4, 5). A photograph of the crack surface as it propagated from right to left into the vertical stiffener is shown in Figure 5. To the right of the curved final crack front, the fatigue crack surface is covered with a uniform dark colored corrosion product layer. To the left of the crack front, the fracture surface (produced as the cracked pylon was broken) is clean and free of corrosion products.

![Image of crack surface](image)

**Figure 5.** A photograph of the crack surface as the crack propagated into the vertical stiffener is shown.

**Fractography**

A more detailed examination of the crack surface was performed with a scanning electron microscope (SEM). The smaller piece of the broken pylon, shown in Figure 4a, was too large for SEM examination and was cut into smaller pieces. First, a single cut was made (indicated in Figure 6a by the dashed line) parallel to the crack so that the entire crack surface was contained in a strip of material approximately 12-inches (0.3 m) long. Five additional cuts were made normal to the crack surface (indicated in Figure 6b with dashed lines) dividing the crack surface into six pieces, each approximately 2-inches (50 mm) long. The six pieces of the crack surface were small enough to be accommodated in the SEM. Care was taken not to cut through regions containing important crack-surface features.

Crack-surface corrosion products made SEM analyses difficult. Therefore, the crack surface was ultrasonically cleaned in a 50% nitric acid (HNO₃) solution for 60-second intervals. After the crack surface was removed from the acid solution, it was thoroughly rinsed with water, and the cleaning procedure was repeated before the crack surface was thoroughly cleaned. (A third 60-second exposure was not needed.) This technique has been used at LaRC to remove oxides from crack surfaces without altering crack-surface features of interest.

Features and characteristics of the crack surface are used to provide information about the history and cause of the crack. The crack-surface features of primary interest to this study are markings that result from crack-tip deformation; herein, called deformation markings. Deformation markings are useful
Figure 6. Two photographs are shown to illustrate how the broken pylon was cut for SEM analysis. (a) The first saw cut removed the crack surface from the pylon. (b) The crack surface was then cut into six smaller pieces.

because they indicate the location and shape of the crack front at an instant in time. Crack surfaces produced during constant-amplitude fatigue loading typically have periodic deformation markings that are nearly equally spaced (ref. 6). Where cracks are known to advance during each load cycle, the spacing between these markings (commonly called striations) is the increment of crack advance per load cycle, termed as the fatigue crack growth rate. This is useful because the fatigue crack growth rate can be related to the crack-tip load parameter, $\Delta K$. Most structures (including helicopter pylons) experience irregular variable-amplitude loading. Complex load interactions occur during variable-amplitude loading, possibly resulting in irregular fatigue crack growth rates or crack arrest (i.e., cracks become non-propagating). In general, it is difficult to distinguish between striation markings (produced as a crack propagates) and crack arrest marks (produced as a crack arrests). Therefore, it is difficult to determine fatigue crack growth rates with deformation markings produced during variable-amplitude loading (ref. 9). Even in cases of complex variable-amplitude loading, a great deal of information about the crack history can be obtained from crack-surface markings. For example, both striations and crack arrest marks indicate the progression of the crack front, so no distinction between these markings is needed to determine the crack initiation site. During the early stages of crack growth, the crack propagates in radial directions from the crack initiation site producing curved crack-surface markings (concave toward the crack initiation site) that can be used to determine where crack initiation occurred.

SEM examination of the clean crack surface revealed crack-surface deformation markings, evidence that crack propagation was a result of fatigue loading. The majority of the crack surface did not exhibit deformation markings, but those that were found revealed that crack initiation occurred at corrosion pits in the fastener hole of Figure 3b. The photograph of the crack surface in Figure 7a shows the region of crack initiation; crack initiation occurred at the fastener hole in the upper right corner of the figure, highlighted with a box. The crack surface in Figure 7b is a higher-magnification image of the crack initiation region highlighted (box) in Figure 7a. A region of corrosion pitting is noted in Figure 7b along the inside edge of the fastener hole (right side of the photograph). The region of crack initiation noted in Figure 7b is shown at higher magnification in Figure 7c. Corrosion pits are seen as a rough-textured region on the right side of Figure 7c. Deformation markings on the crack surface are barely visible in Figure 7c. The curved dotted line in Figure 7c shows the curvature of the deformation marks (concave to

* For aluminum alloys, striated fatigue crack growth (one striation per load cycle) is known to occur for $\Delta K > 7$ MPa$\sqrt{m}$. Multiple load cycles may occur between crack-surface marks at lower $\Delta K$ (refs. 7, 8).
the right) indicating that crack initiation occurred at the corrosion pits seen on the right side of the figure. These marks are better seen at higher magnification in Figure 7d.

Figure 7. The crack initiation region is shown in a series of images, at increasing magnification. (a) Photograph of the crack surface in the pylon upper flange region is shown. Also shown are micrographs of (b) crack surface at edge of fastener hole surface, (c) crack initiation at corrosion pits, and (d) crack surface markings at high-magnification in crack initiation region.

SEM images of the crack initiation region of Figure 7 are shown at a different angle in Figure 8 to illustrate the complex nature of the crack surface and to highlight regions of pitting that did not result in crack initiation. Here, the crack surface was viewed at an angle 45° from the direction normal to the crack surface, and the specimen is rotated 90° (with respect to the orientation of Figure 7) so the fastener hole is oriented horizontally. In Figure 8a, the intersection of the crack surface with the edge of the fastener hole is indicated with a dashed horizontal line; the surface of the hole and the crack surface are below and above the dashed line, respectively. The upper and lower surfaces of the upper flange (shown in Figure 3) are the edges seen on the extreme right and left sides of Figure 8a, respectively. The boxes in Figure 8a identify two regions of corrosion pitting in the fastener hole: one where cracks did not initiate (left box) and one where cracks did initiate (right box). These regions are shown at higher magnification in Figures 8b and 8c, respectively.
Figure 8. Micrographs of the crack initiation region, viewed at a 45° angle to the crack surface, are shown. (a) The crack surface at the edge of the fastener hole surface is shown. High-magnification micrographs are shown of corrosion pits (b) where cracks did not initiate and (c) where cracks initiation occurred.

Evidence of crack-surface deformation marks at three locations along the crack surface is shown in Figure 9. These three locations, identified in Figure 9a as b, c, and d, are shown at high-magnification in Figures 9b, 9c, and 9d, respectively. The markings observed in Figures 9b and 9c are not equally spaced, possibly suggesting that complex variable-amplitude loading occurred. The deformation markings shown in Figure 9d are equally spaced and exhibit the morphology typical of fatigue striations. The average spacing between deformation markings is determined by counting the number of marks intersecting lines along the crack growth direction (i.e. normal to the marks). In Figure 9b, 8 marks are counted along a line 100 μm long. For Figures 9c and 9d, 10 marks and 20 marks are counted along lines 15 μm and 30 μm long, respectively. If each deformation mark was produced by a single load cycle, the average fatigue crack growth rates corresponding to Figures 9b, 9c, and 9d would be $1.2 \times 10^{-5}$ m/cycle, $1.5 \times 10^{-6}$ m/cycle, and $1.5 \times 10^{-6}$ m/cycle, respectively. The markings in Figures 9b and 9c are not regularly spaced and may not have been produced by single load cycles. Therefore, these markings do not permit a good estimation of fatigue crack growth rates. The crack-surface markings seen in Figure 9d are likely striations, each produced by a single load cycle. Therefore, the calculated fatigue crack growth rate of $1.5 \times 10^{-6}$ m/cycle is likely a good estimate for the crack surface of Figure 9d. Based on laboratory 7075 aluminum data, the fatigue crack growth rate for Figure 9d corresponds to values of $\Delta K$ between 15 MPa√m and 20 MPa√m (refs. 10, 11).

Discussion

The results of detailed destructive examinations show that crack initiation occurred at corrosion pits that formed in a fastener hole. Corrosion problems are greater for USCG helicopters because they operate in a marine environment where exposure to salt-water spray – known to corrode aluminum alloys – is
likely. Fastener holes create a crevice-like environment that entraps water resulting in a corrosive environment where pitting is likely (ref. 12). Pitting damage is difficult to detect because the damage is highly localized (e.g. may be hidden in a fastener hole) and only affects a small amount of material.

The crack found in the fuel-tank pylon initiated at corrosion pits in the region where the highest stresses were expected. As shown in Figure 2a, vertical-downward loading on the pylon creates a high-stress region in the corner where the cantilever meets the vertical portion of the pylon. The stress concentration effect of fastener holes in this region further increases the local stress state near these discontinuities (ref. 13). Finally, corrosion pitting in these fastener holes creates rough surfaces that further increase the severity of the local stress state (refs. 14-16). Due to these three factors (pylon loading, fastener hole geometry, and corrosion pits) the highest local stress occurred at the fastener hole of Figure 3b, the crack initiation site.

Deformation markings on the crack surface were used to identify the crack initiation site. However, care must be taken when using fatigue striations to quantify the fatigue crack growth rate. Researchers have suggested that a single striation may not be the result of a single load cycle (refs. 7, 8). Unpublished research for aluminum alloys by Piascik has revealed similar findings; Paris regime fatigue crack growth ($\Delta K > 7 \text{ MPa}\sqrt{\text{m}}$) exhibits single load cycle per striation behavior, but near the fatigue crack growth threshold ($\Delta K < 5 \text{ MPa}\sqrt{\text{m}}$) multiple load cycles are required per striation because more complex crack-tip damage processes are operative. As the crack propagated through the webbing, clear evidence of striation markings was found on the crack surface indicating a fatigue crack growth rate of approximately $10^{-6} \text{ m/cycle}$ (recall Figure 9d). Test data for similar materials indicates this occurs for values of $\Delta K$ between 15 MPa\sqrt{m} and 20 MPa\sqrt{m}; in the range of $\Delta K$ where each load cycle is expected to produce one striation. Striations in Figures 9b and 9c were not well defined possibly because the values of $\Delta K$ were low, i.e. near the fatigue crack growth threshold.
The striation markings seen in Figure 9d indicate that rapid crack propagation occurred in the pylon webbing (approximately 1.5 mm of crack growth per 1000 load cycles). As the crack grew, increases in $\Delta K$ and fatigue crack growth rates likely occurred. Pylon failure was prevented because a vertical stiffener deflected the crack causing crack arrest. Cracks tend to propagate in the direction perpendicular to the principal (largest) normal stress, the orientation where the mode I stress intensity factor ($K_I$) is at its maximum value and the mode II stress intensity factor ($K_{II}$) is zero (ref. 17). This stiffener likely diverted the crack from principal stress directions, resulting in mixed-mode crack-tip loading (i.e. both $K_I$ and $K_{II}$ are non-zero) and a reduction in $K_I$, the crack driving force. Crack arrest was likely a result of a reduction in crack driving force and crack-face contact due to mode II (sliding-mode) displacements further reducing the fatigue crack driving force (refs. 18, 19). The crack surface corrosion product in Figure 5 was likely produced by sliding-mode contact of rough crack surfaces.

Failure of this pylon was prevented when the vertical stiffener diverted and arrested the propagating fatigue crack. However, prevention of similar cracking problems in the future should focus on preventing the corrosion pitting that lead to crack initiation. In other words, preventing crack initiation (here caused by corrosion pitting) is preferable to, and likely more successful than, relying on techniques to arrest cracks after they develop.

Conclusions

A failure analysis was performed to determine the cause of an eight-inch-long crack found in a U.S. Coast Guard HH-60 helicopter external fuel-tank pylon. Based on the failure analysis results presented in this paper, it is believed the following sequence of events occurred and resulted in the crack.

1. **Corrosion pitting** – The initial damage occurred when corrosion pits formed in a fastener hole. Fasteners are known to provide a crevice-like environment where entrapped water can form a highly corrosive environment where pitting is likely to occur.

2. **Crack initiation** – Crack initiation occurred in a region of high local stress; at corrosion pits in a fastener hole, near the corner with the highest bending stress; all three factors greatly increases local state of stress.

3. **Fatigue crack propagation** – Crack-surface markings indicate that crack propagation was a result of fatigue, or cyclic, loading. Corrosion products on the crack surfaces suggest that fatigue crack growth occurred in a corrosive environment. The presence of fatigue loading and corrosive environment exacerbates cracking; fatigue cracks contained in 7000 series alloys exposed to air with 90% humidity will continue to grow at extremely low loads.

4. **Crack arrest** – After propagating through approximately 80% of the pylon web, the crack intersected a vertical stiffener and was diverted. Diverting the crack likely lowered the crack-tip driving force resulting in crack arrest.

References


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