Abstract

Submitted to CompTest 2008
4th International Conference on Composites Testing and Model Identification
20 - 22 October, 2008
Dayton, OH

Effect of microscopic damage events on static and ballistic impact strength of triaxial braid composites

Justin D. Littell, Wieslaw K. Binienda, and William A. Arnold
The University of Akron, Akron, OH

Gary D. Roberts and Robert K. Goldberg
NASA Glenn Research Center, Cleveland, OH

In previous work, the ballistic impact resistance of triaxial braided carbon/epoxy composites made with large flat tows (12k and 24k) was examined by impacting 2'X2'X0.125" composite panels with gelatin projectiles. Several high strength, intermediate modulus carbon fibers were used in combination with both untoughened and toughened matrix materials. A wide range of penetration thresholds were measured for the various fiber/matrix combinations. However, there was no clear relationship between the penetration threshold and the properties of the constituents. During some of these experiments high speed cameras were used to view the failure process, and full-field strain measurements were made to determine the strain at the onset of failure. However, these experiments provided only limited insight into the microscopic failure processes responsible for the wide range of impact resistance observed.

In order to investigate potential microscopic failure processes in more detail, quasi-static tests were performed in tension, compression, and shear. Full-field strain measurement techniques were used to identify local regions of high strain resulting from microscopic failures. Microscopic failure events near the specimen surface, such as splitting of fiber bundles in surface plies, were easily identified. Subsurface damage, such as fiber fracture or fiber bundle splitting, could be identified by its effect on in-plane surface strains. Subsurface delamination could be detected as an out-of-plane deflection at the surface. Using this data, failure criteria could be established at the fiber tow level for use in analysis. An analytical formulation was developed to allow the microscopic failure criteria to be used in place of macroscopic properties as input to simulations performed using the commercial explicit finite element code, LS-DYNA. The test methods developed to investigate microscopic failure will be presented along with methods for determining local failure criteria that can be used in analysis. Results of simulations performed using LS-DYNA will be presented to illustrate the capabilities and limitations for simulating failure during quasi-static deformation and during ballistic impact of large unit cell size triaxial braid composites.

Effect of Microscopic Damage Events on Static and Ballistic Impact Strength of Triaxial Braid Composites

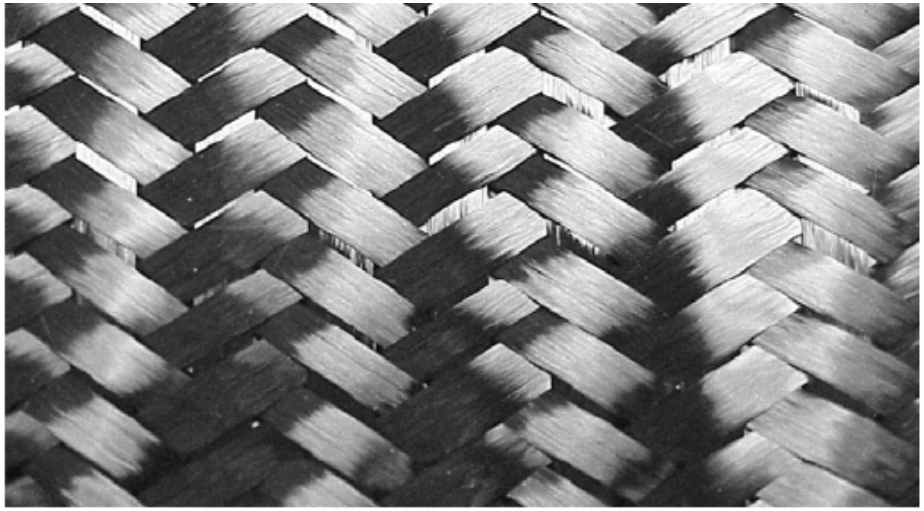
CompTest Conference - Oct. 21, 2008
Dayton, Ohio

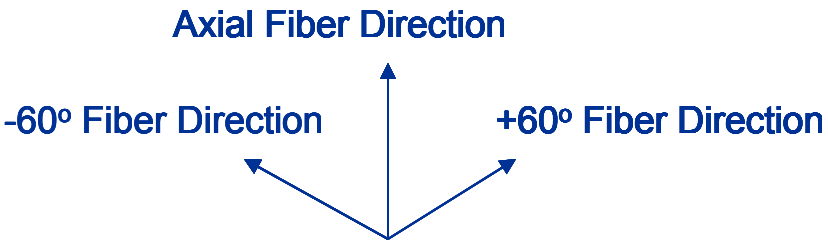
Justin Littell

Ph.D. Student
University of Akron
Akron, Ohio

Wieslaw K. Binienda, William A. Arnold – University of Akron, Akron Ohio
Gary D. Roberts, Robert K. Goldberg – NASA Glenn Research Center, Cleveland Ohio

Triaxial Braided Composite Materials

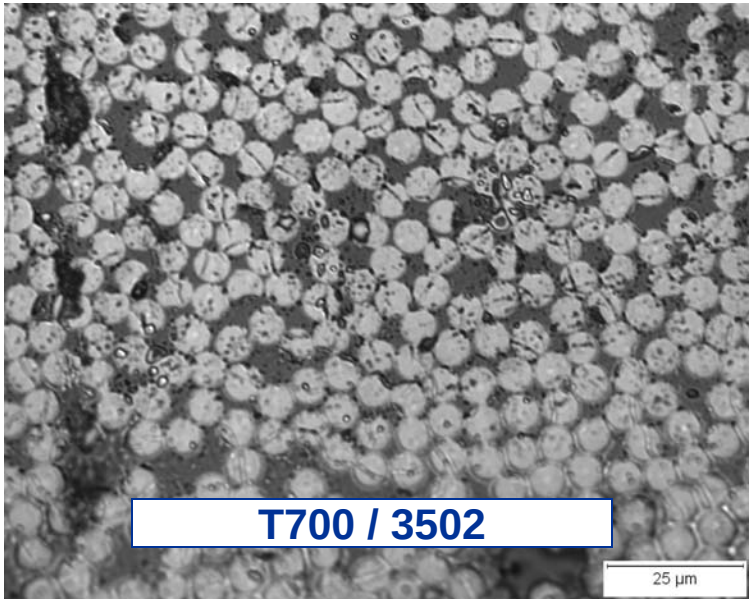


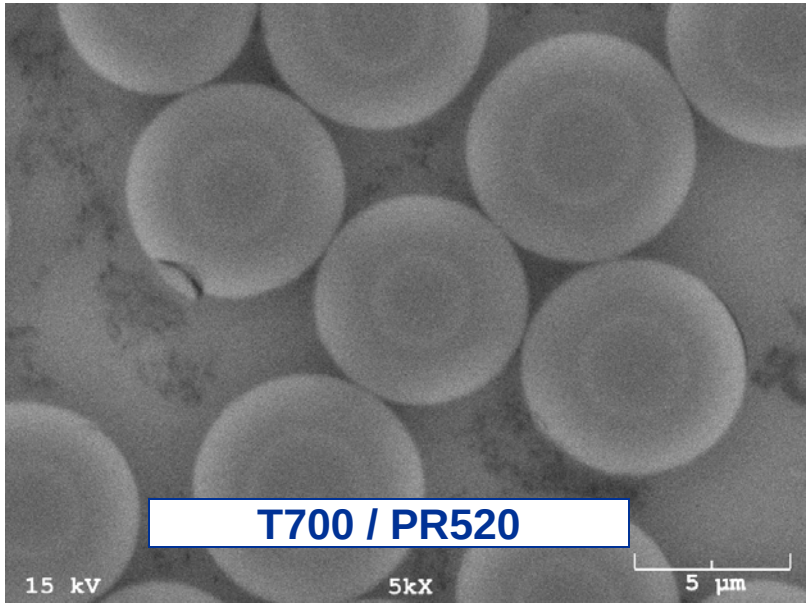


- Two dimensional triaxial braid
- 24k wide axial and 12k wide bias fiber tows
- Layers of +60° and -60° bias fibers braided over a 0° axial fiber
- Quasi-isotropic architecture
- Layup of 6 Layers of braid, total composite thickness 0.125”
- Resin Transfer Molding process (RTM)
- Volume fraction of 56% nominal

Materials

- High Strength, Standard Modulus Fiber – Toray T700
- Two resins
  - Toughened Cytec Cycom ® PR520
  - Untoughened Hexcel 3502



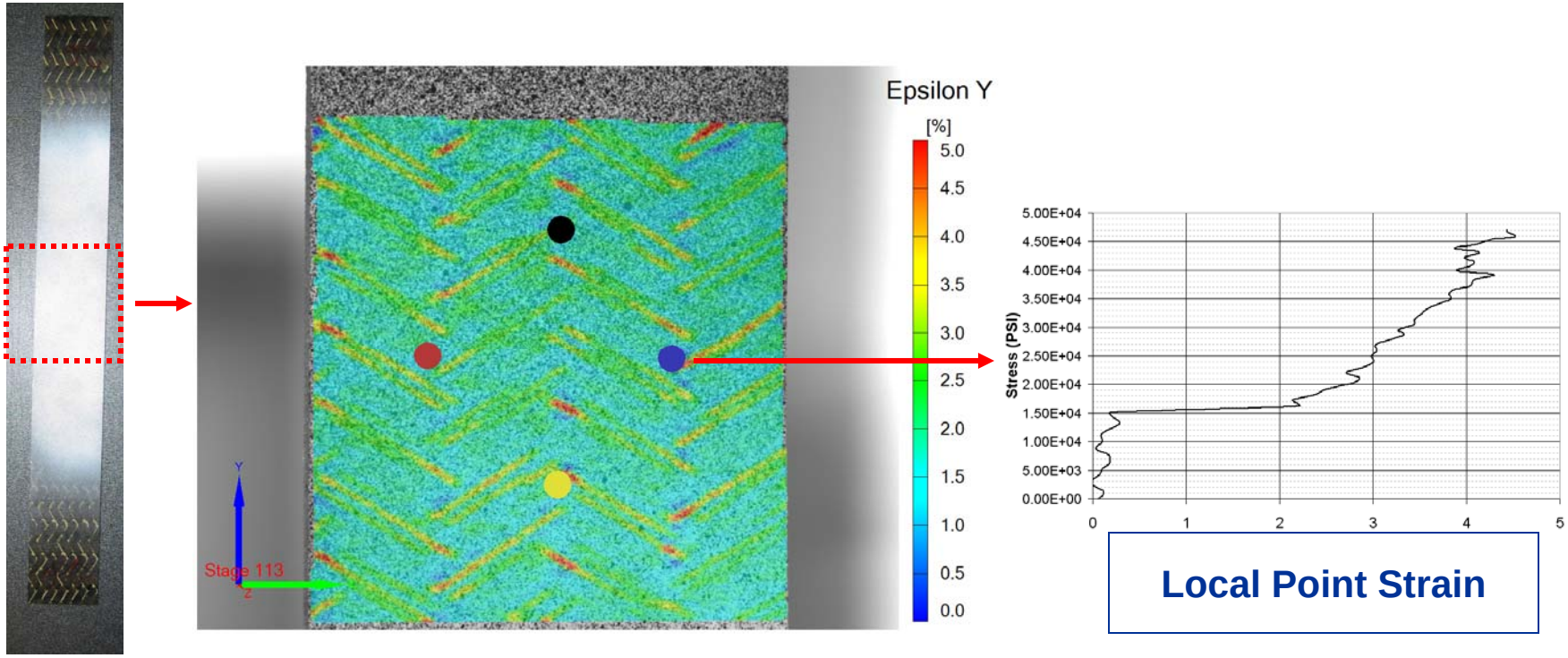


- Presented as examples to cover range of material response

Model Development

- Testing and modeling were done in parallel
  - Test data used optical measurement techniques to obtain full field strain data
    - Quasi-static Testing
  - By examining the test data and using classical composite theory, a new approach was developed to model composites using a novel “Subcell” approach
- Modeling data needs
  - Composite section properties – braid geometry
  - Composite material properties – test data
- Models developed in LS-DYNA
  - Transient, nonlinear, explicit finite element code
  - Primarily impact loading that composites will be subjected

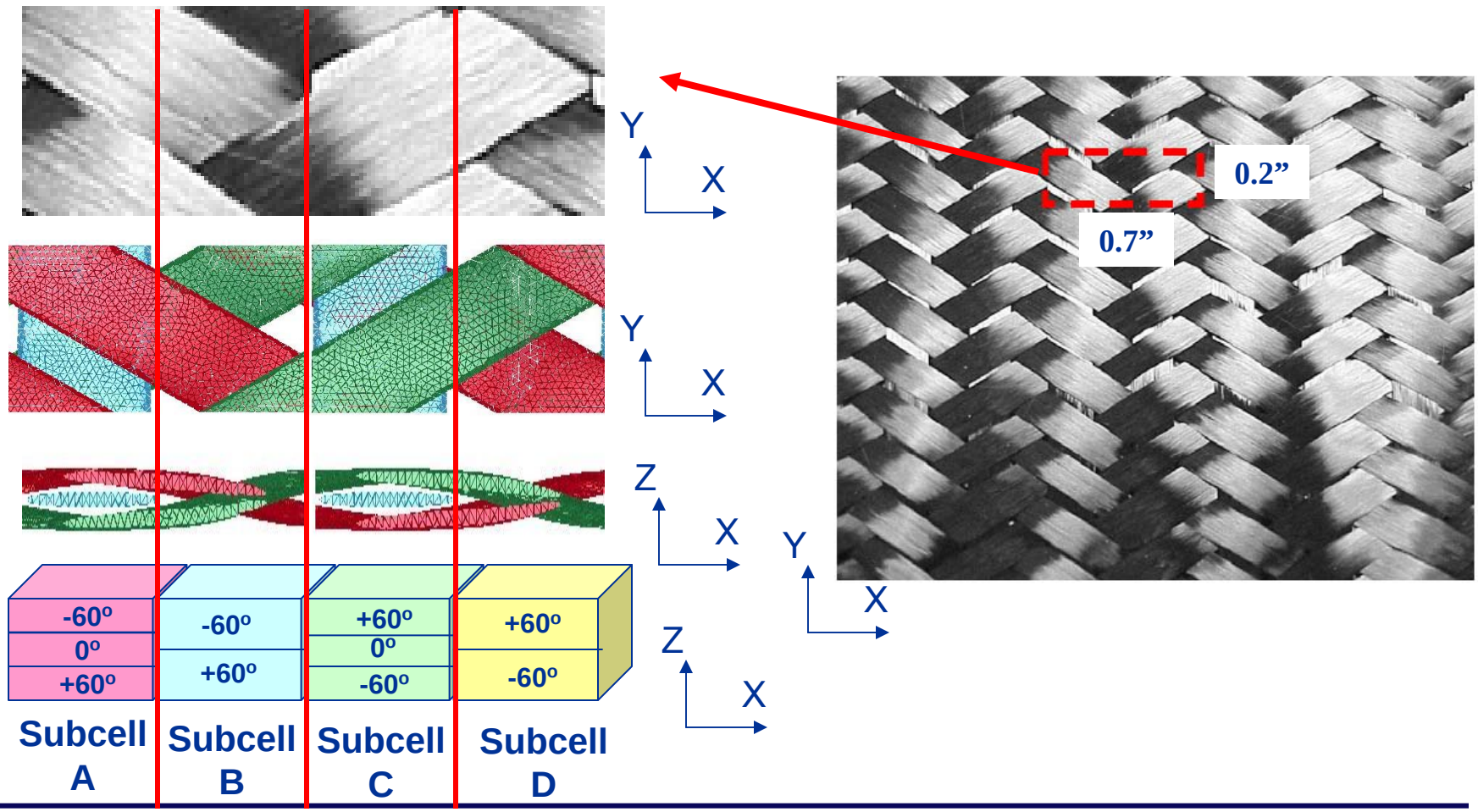
Photogrammetry used for Data Collection



- Global stress vs. strain curves found by creating a “digital strain gage”
- Measures material response in specific areas on specimen
  - Seen by noting lines of high localized strain
- Local failure mechanisms and deformations must be accounted for when developing an analytical method

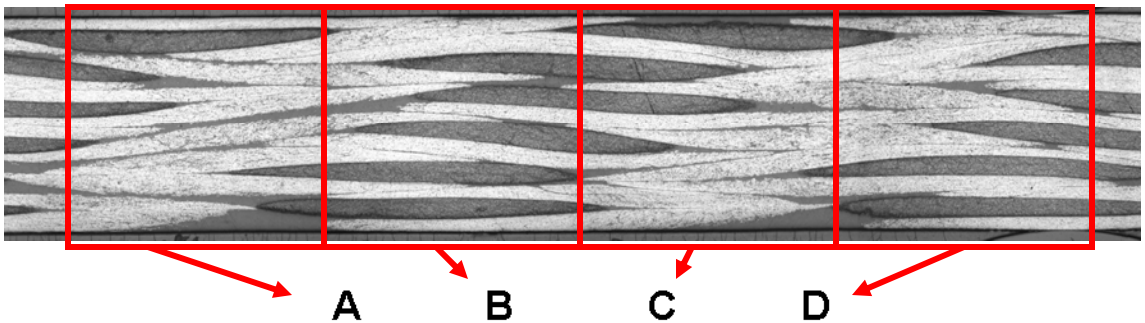
Triaxial Braided Model Methodology

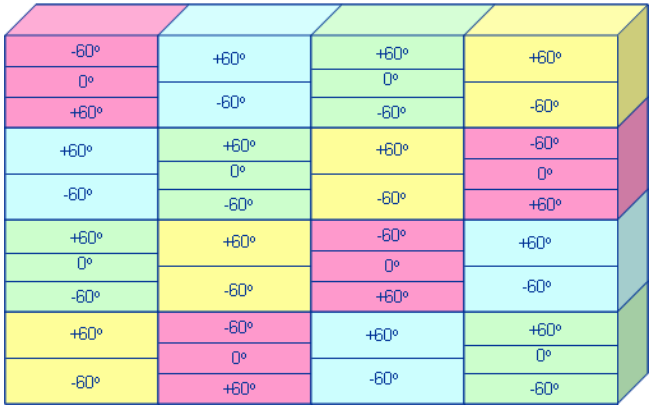
- Develop a macromechanical finite element model capable of capturing the braid architecture and material properties of triaxially braided composites
  - Layers of unidirectional lamina stacked in a “Subcell” configuration
  - Needs local lamina level modulus and failure properties



Defining the Braid Geometry (Section)

- Each part has unique section properties
  - Each section contains information about number of layers (15) and braid angle (Θ) at each layer
- Braid was modeled as layers of unidirectional lamina
  - Shifted (Idealized)





A New Methodology Developed for Implementation of Material Properties

Obtain Fiber, Matrix Properties

↓

Micromechanics
- Develop effective lamina properties

↓

CLPT
- Develop effective laminate properties

↓

Develop a FEM
(One layer, homogenized)

Traditional Approach
(Bottom - Up)

- Items needed
  - Fiber properties - Assumed
  - Matrix properties
- Stackup sequence

Obtain Composite stresses, strains
From optical measurements

↓

CLPT
- Back out effective laminate properties

↓

Micromechanics
- Back out effective lamina properties

↓

Develop a FEM
(includes layers and subcell props)

New Approach
(Top – Down)

- Items needed
  - **COMPOSITE TEST DATA**
- Stackup sequence

Material Card
MAT_RATE_SENSITIVE_COMPOSITE_FABRIC

```
$$$$ USING AS TESTED T700s/PR520 Material Properties and mat 158
*MAT_RATE_SENSITIVE_COMPOSITE_FABRIC
$      MID        RO        EA        EB      (EC)      PRBA      TAU1    GAMMA1
         11.6680E-04   7.45E+6   3.63E+6                .071
$      GAB       GAC       GCA    SLIMT1    SLIMC1    SLIMT2    SLIMC2     SLIMS
  2.75E6                                                  1
$     AOPT     TSIZE     ERODS      SOFT        FS
      2.0                  1                   -1
$                                     A1        A2        A3
                                     0.0       1.0       0.0
$                                     D1        D2        D3
                                     1.0       0.0       0.0
$      E11C      E11T      E22C      E22T       GMS
     0.018    0.0216     0.011    0.0168     0.012
$        XC        XT        YC        YT        SC
  5.469E+4  1.515E+5   5.00E+4   5.25E+4  4.457E+4
$         K
  6.762E+6
```

Controls elastic/plastic behavior of failure

Material Coordinate Definition

Material Response Properties	
Property	Value
EA	Axial Modulus
EB	Trans. Modulus
PRBA	In Plane Poisson
GAB	In Plane Shear Mod.

Failure Properties			
Property	Name	Property	Value
E11C	Comp. strain (Ax.)	XC	Comp. strength (Ax.)
E11T	Tens. Strain (Ax.)	XT	Tens. Strength (Ax.)
E22C	Comp. strain (Trans.)	YC	Comp. Strength (Trans.)
E22T	Tens. strain (Trans.)	YT	Tens. Strength (Trans.)
GMS	Shear strain	SC	Shear Strength

Classical Laminated Plate Theory (CLPT) Use in Reverse for Top-Down Approach

- Stresses/Surface Tractions are related to strains by the following

$$\begin{bmatrix} [N] \\ [M] \end{bmatrix} = \begin{bmatrix} [A] & [B] \\ [B] & [D] \end{bmatrix} \begin{bmatrix} [\varepsilon] \\ [\gamma] \end{bmatrix}$$

- Balanced and Symmetric (B and D matrices, also A16, A26 = 0)

$$\begin{bmatrix} Nx \\ Ny \\ 0 \end{bmatrix} = \begin{bmatrix} A11 & A12 & 0 \\ A12 & A22 & 0 \\ 0 & 0 & A66 \end{bmatrix} \begin{bmatrix} \varepsilon_x \\ \varepsilon_y \\ 0 \end{bmatrix}$$

- Where

$$A11 = \sum \overline{Q}11 = \overline{Q}11^{0\deg} * t^{0\deg} + \overline{Q}11^{60\deg} * t^{60\deg} + \overline{Q}11^{-60\deg} * t^{-60\deg}$$

$$\overline{Q}11 = m^4 Q11 + n^4 Q22 + 2m^2 n^2 Q12 + 4m^2 n^2 Q66$$

$$Q11 = \frac{E11}{1 - v12 * v21}, Q22 = \frac{E22}{1 - v12 * v21}, Q12 = \frac{v21 * E11}{1 - v12 * v21}, Q66 = G12$$

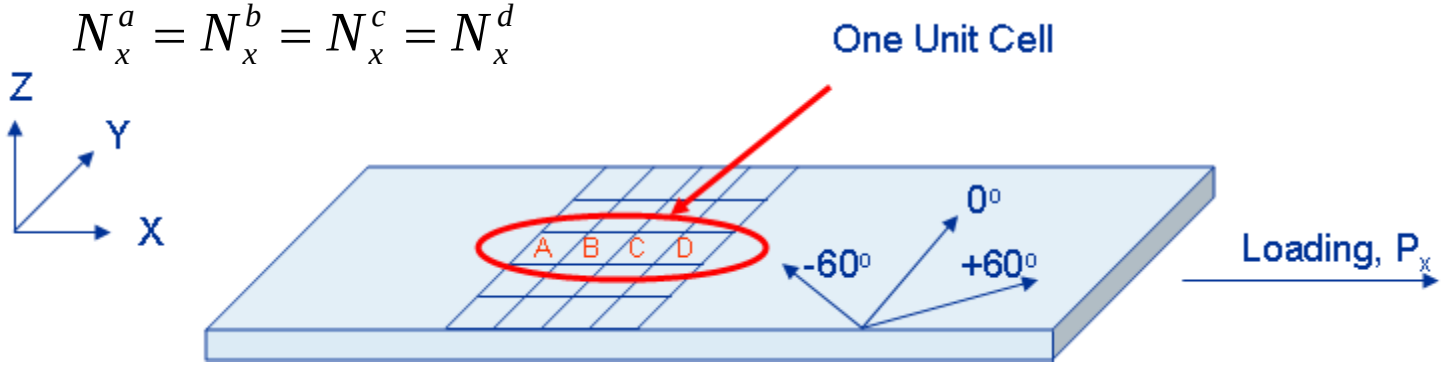
- E11, E22, v12, v21, G12 are parameters needed in LS-DYNA

Equation Development

Micromechanics of Composite Materials assumptions

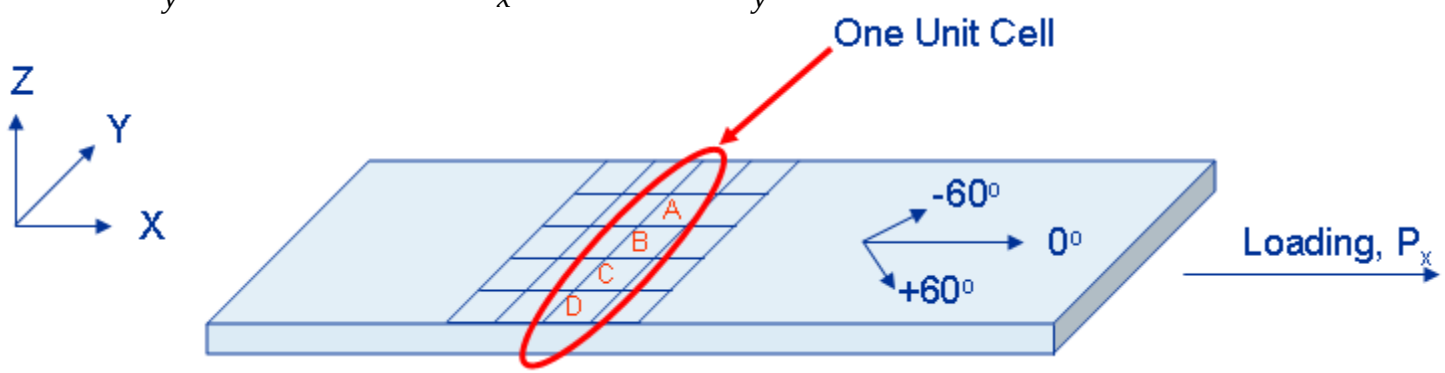
- Transverse ASTM 3039 specimen
  - $V_f^a * N_y^a = V_f^b * N_y^b$

    $N_x^a = N_x^b = N_x^c = N_x^d$

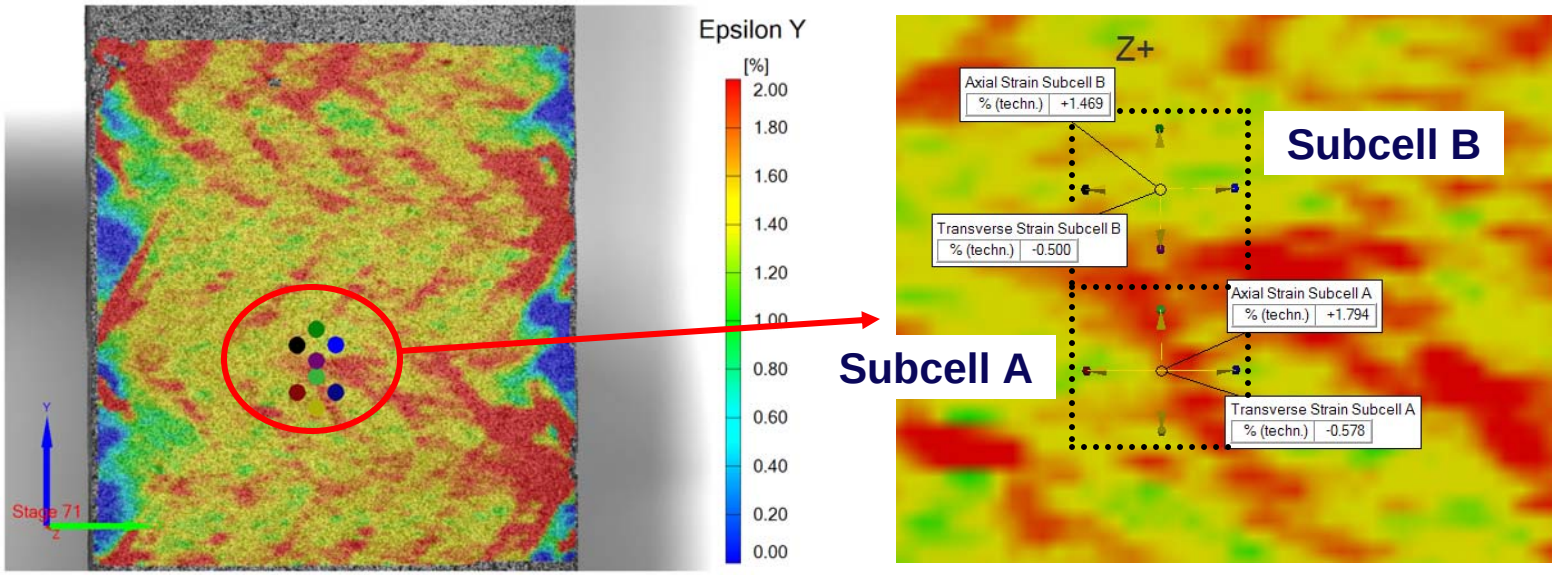


- Axial ASTM 3039 specimen
  - $N_y^a = N_y^b = N_y^c = N_y^d = 0$

    $N_y^a = 0 = A12^a * \varepsilon_x^a + A22^a * \varepsilon_y^a$



Developed Equations (using Subcell Strains)



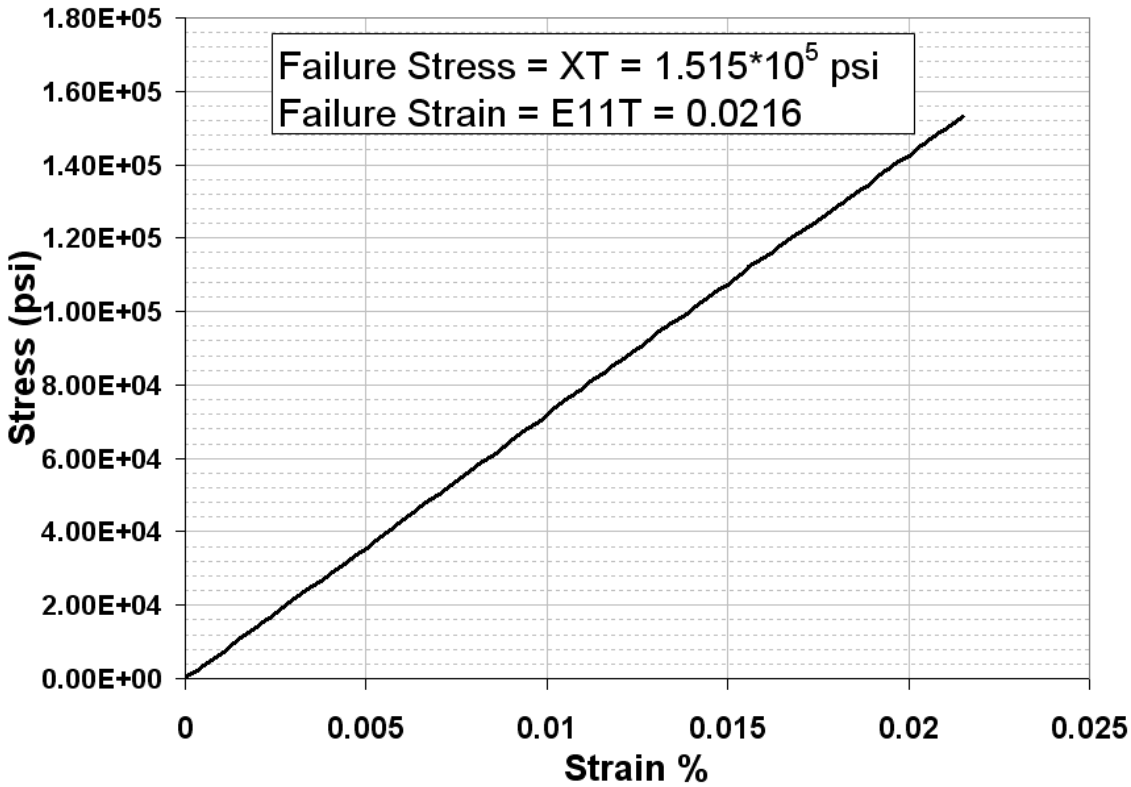
Transverse Tensile Test Global Axial Strain

Local Subcell Strains

- Strains are found using the optical measurement system
- In the end, there are 6 variables (Q11,Q12,Q22,Q66,Ny^a,Ny^b) and 6 equations
  - 2 from TT Subcells A and C (CLPT)
  - 2 from TT Subcells B and D (CLPT)
  - 1 from volume fraction averages (Micromechanics)
  - 1 from AT Subcells A and C (CLPT)
- Solve simultaneously

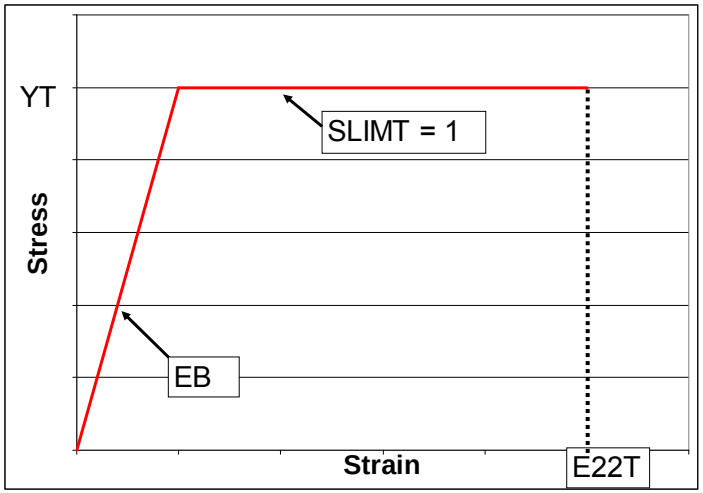
Axial Tensile (AT) Strength

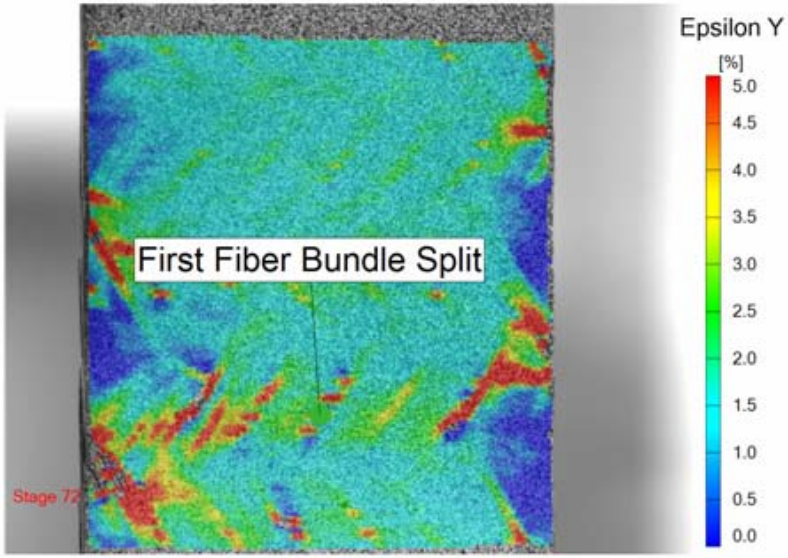
- Assume that in AT tests, the AT fiber carries most of the load
- E11T comes from ultimate strain at AT failure

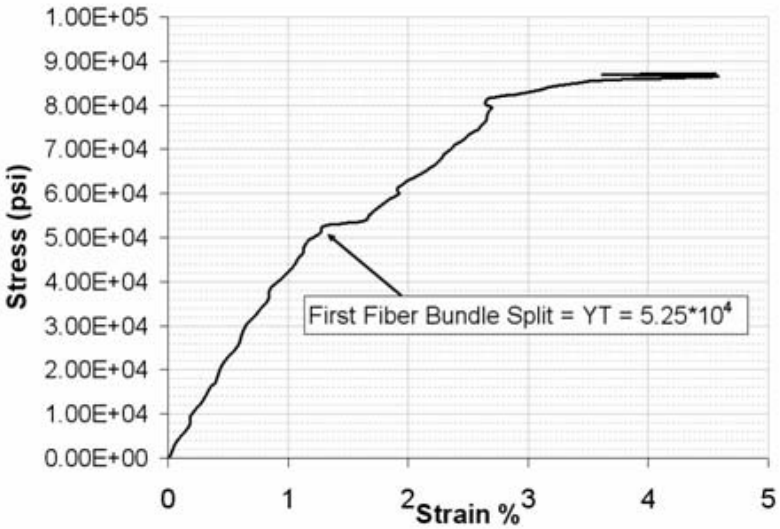


Transverse Tensile (TT) Strength

- Look at fiber splitting on TT specimen
- Load at first split will be YT
- SLIMT will be set to 1
- E22T will be set to failure strain of test

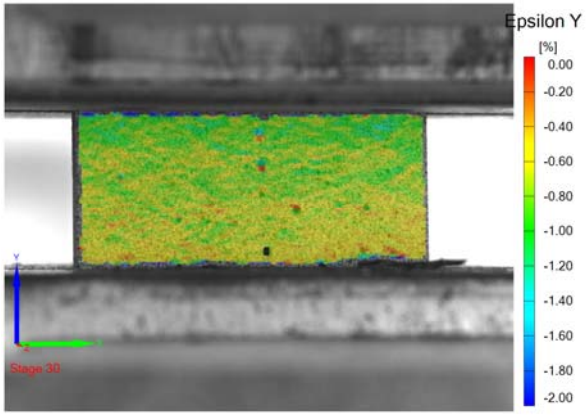


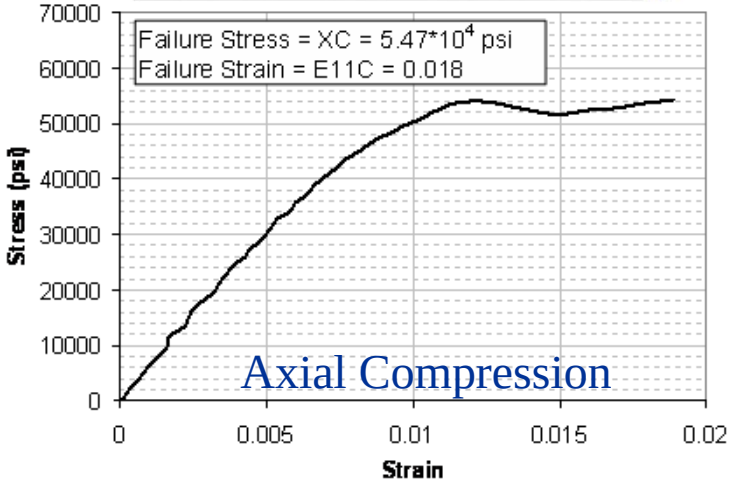


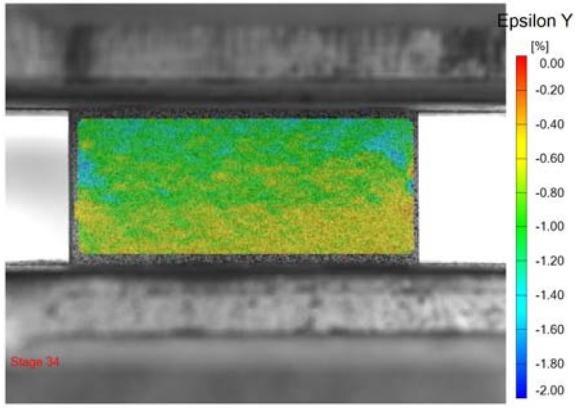


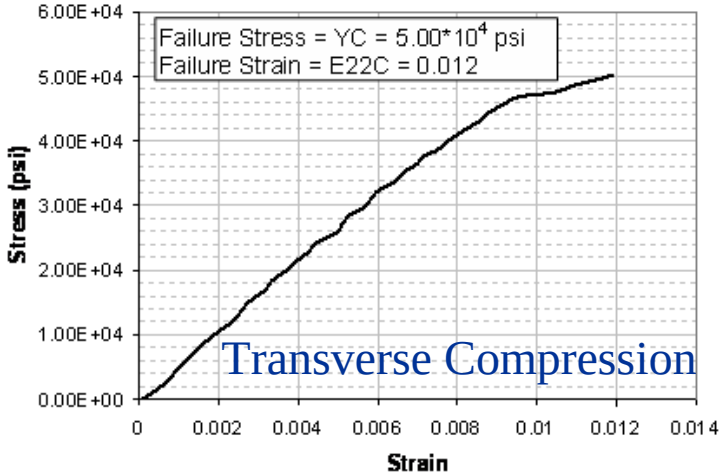
Compressive Strain/Strength

- Material behaves as a homogenous
  - Use strength at failure for both Axial and Transverse tests
  - Use strain at strength for both Axial and Transverse tests



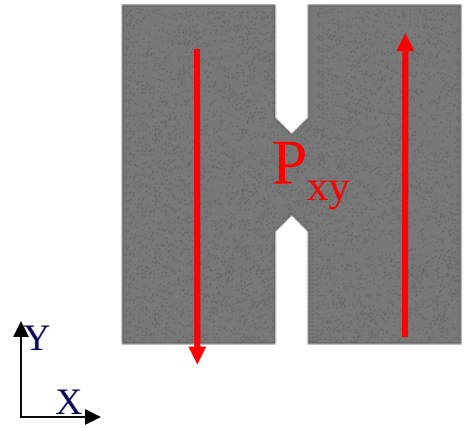


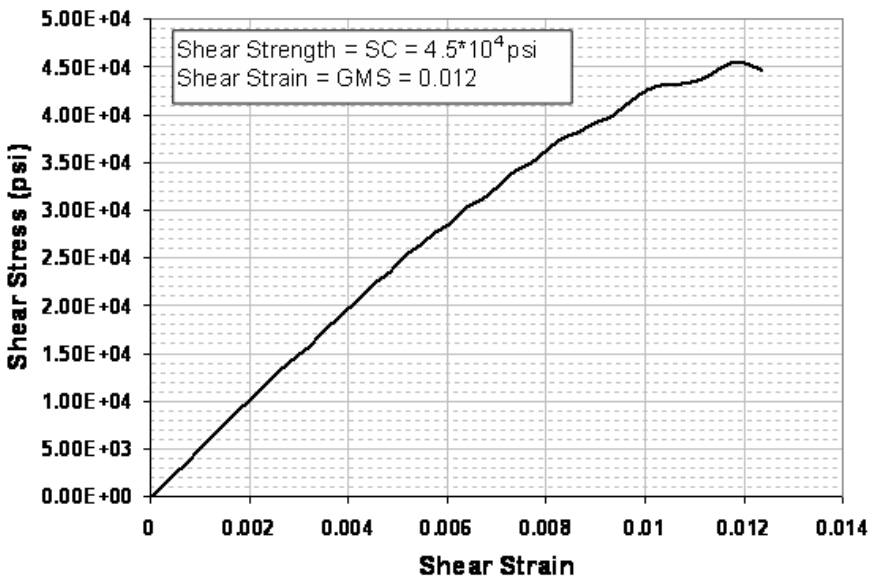


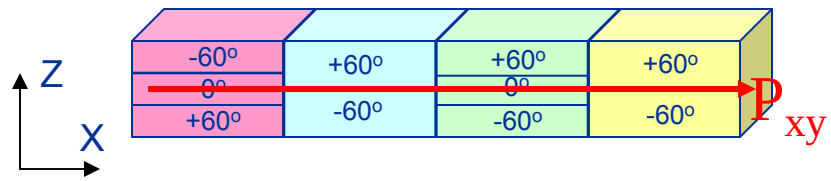


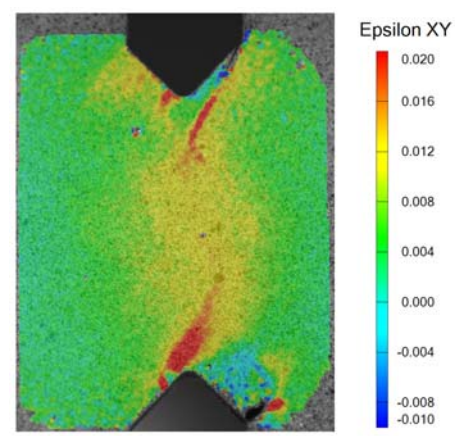
Shear Strength

- Using Modified Shear Specimen Design based on Kohlman
  - **ASTM 5379**
- P_{xy}, N_{xy} is for each of the integration layers
  - Can be directly implemented for shear strength
- LS-DYNA needs ε_{xy} for GMS

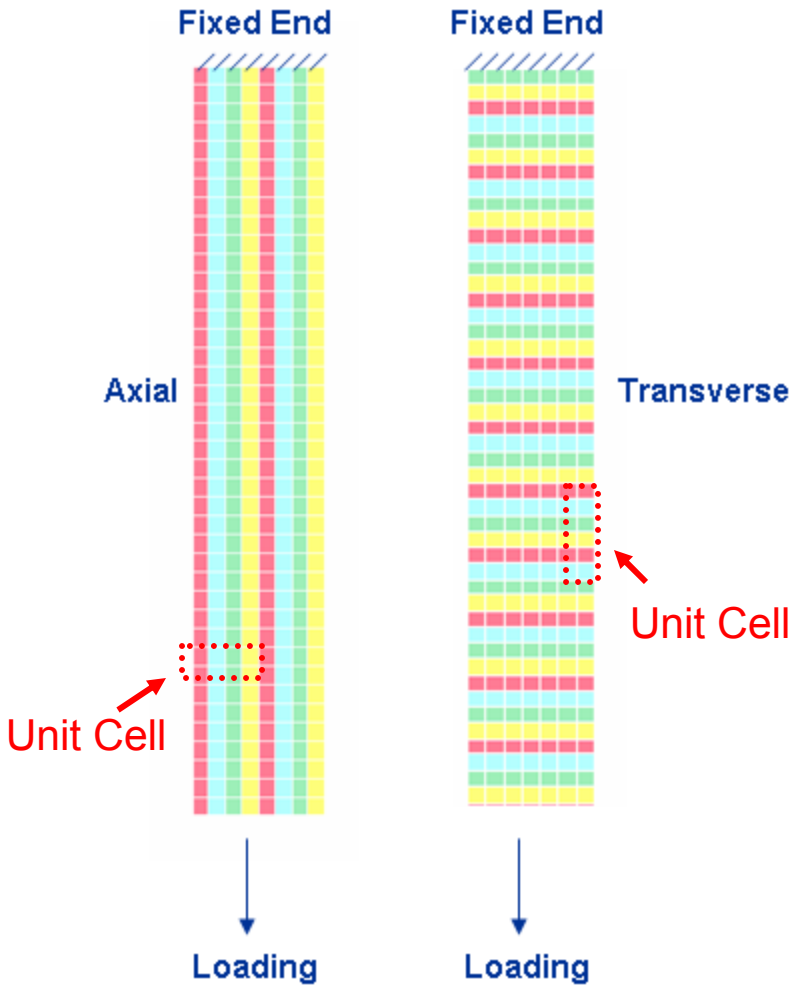








Finite Element Models



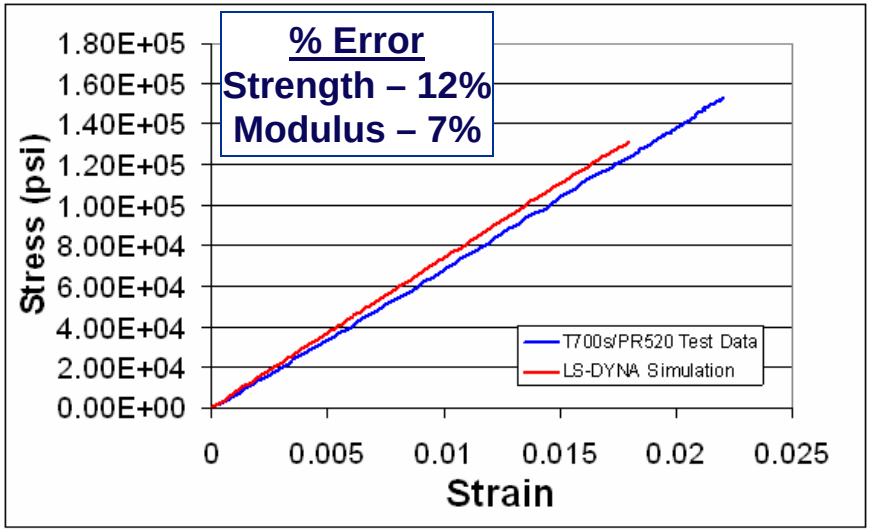
- Both Axial and Transverse Specimens were developed using ASTM D3039 specimen geometries
- Fixed end boundary conditions were used to simulate the fixed grip
- Loading was applied at the opposite end using enforced displacement

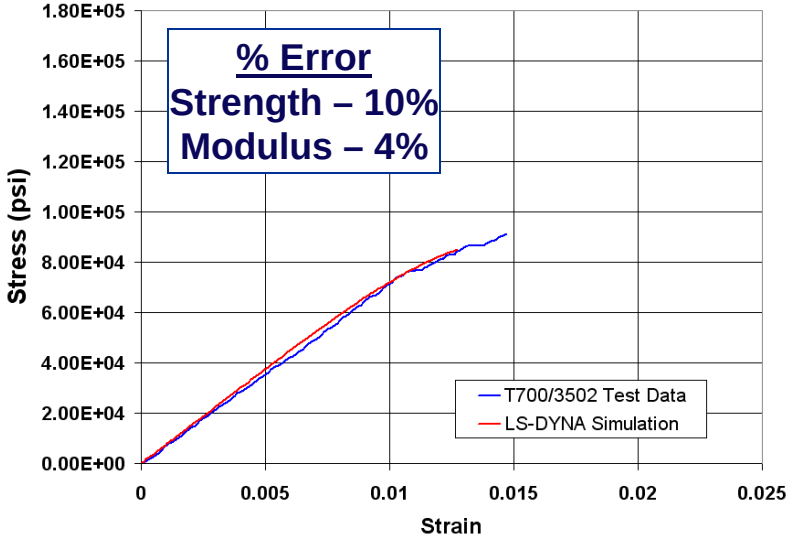
Static Results

T700 / PR520

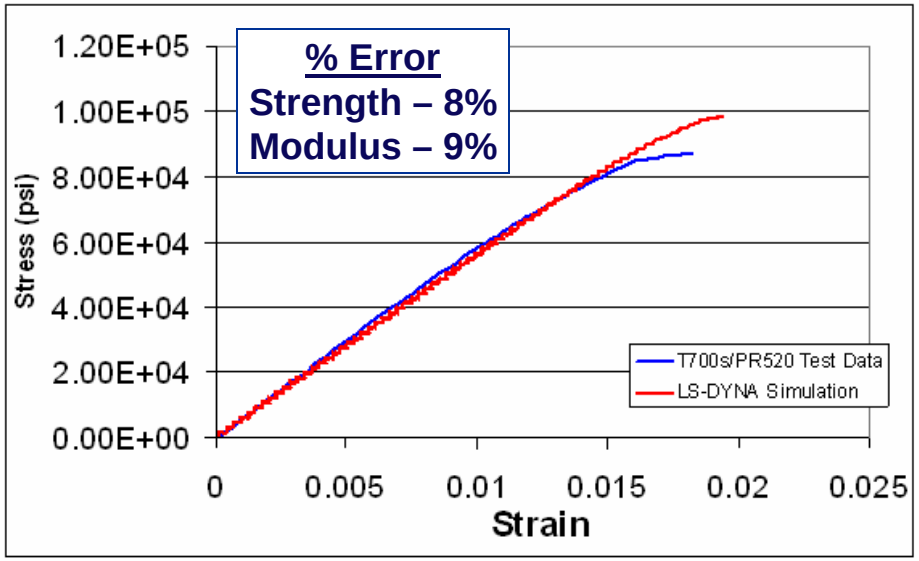
T700 / 3502

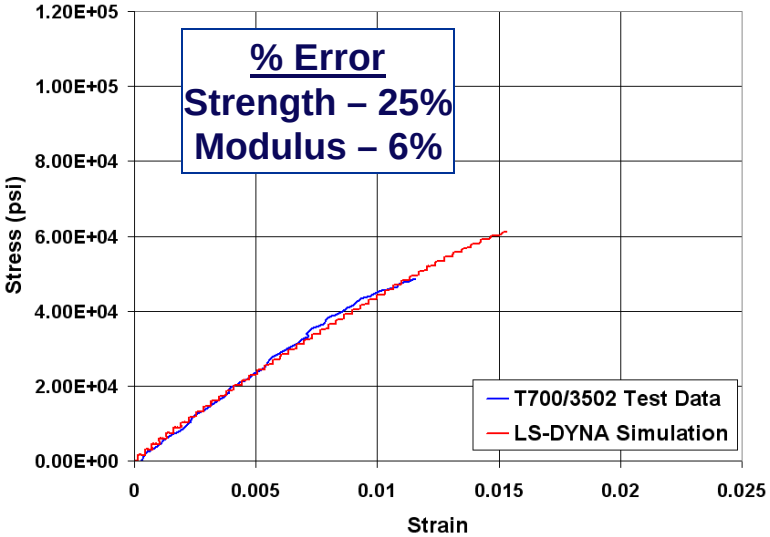
Axial Tension



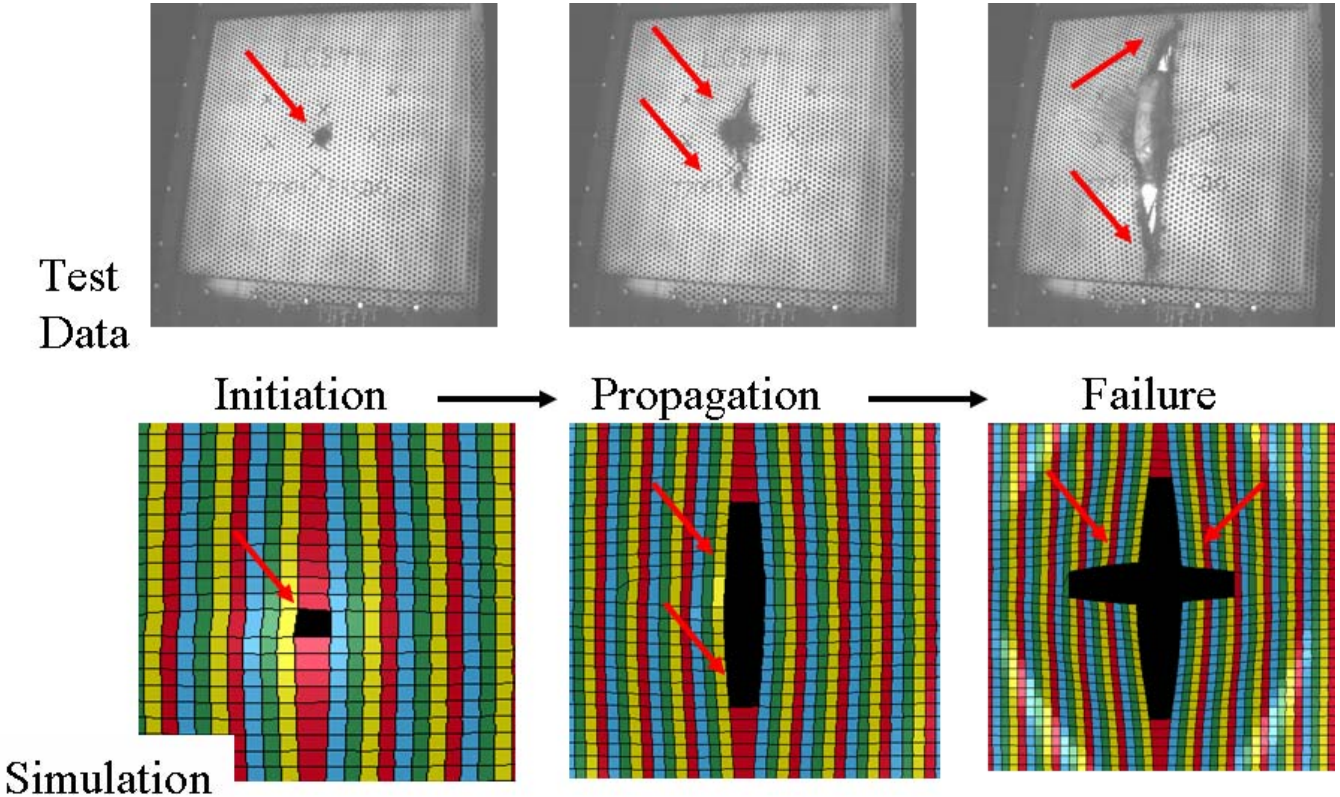


Transverse Tension



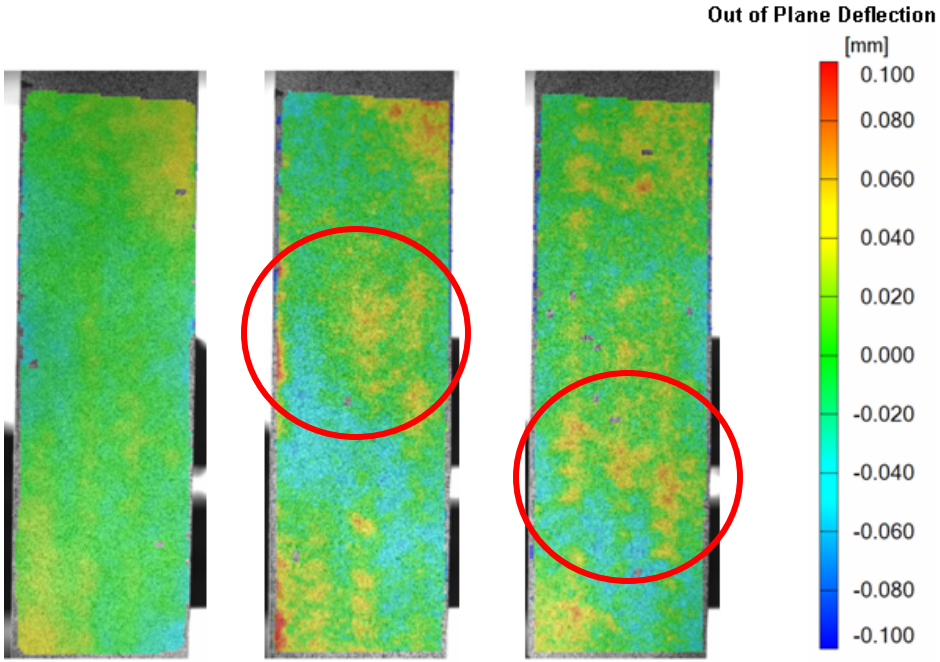


Determination of T700 fiber / PR520 Resin Impact Characteristics



- Simulations were completed Spring 2008
  - Showed penetration threshold at 630 ft / sec
- Used as a starting point for impact tests
- Impact tests conducted Summer 2008
  - Penetration threshold was between 609 and 637 ft / sec

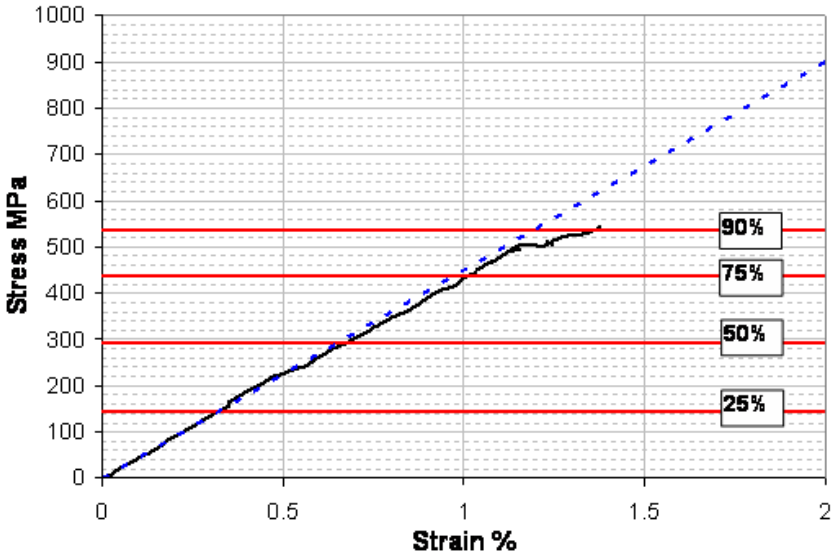
Limitations: Delaminations for T700 fiber / 3502 resin



50%
292 MPa

75%
439 MPa

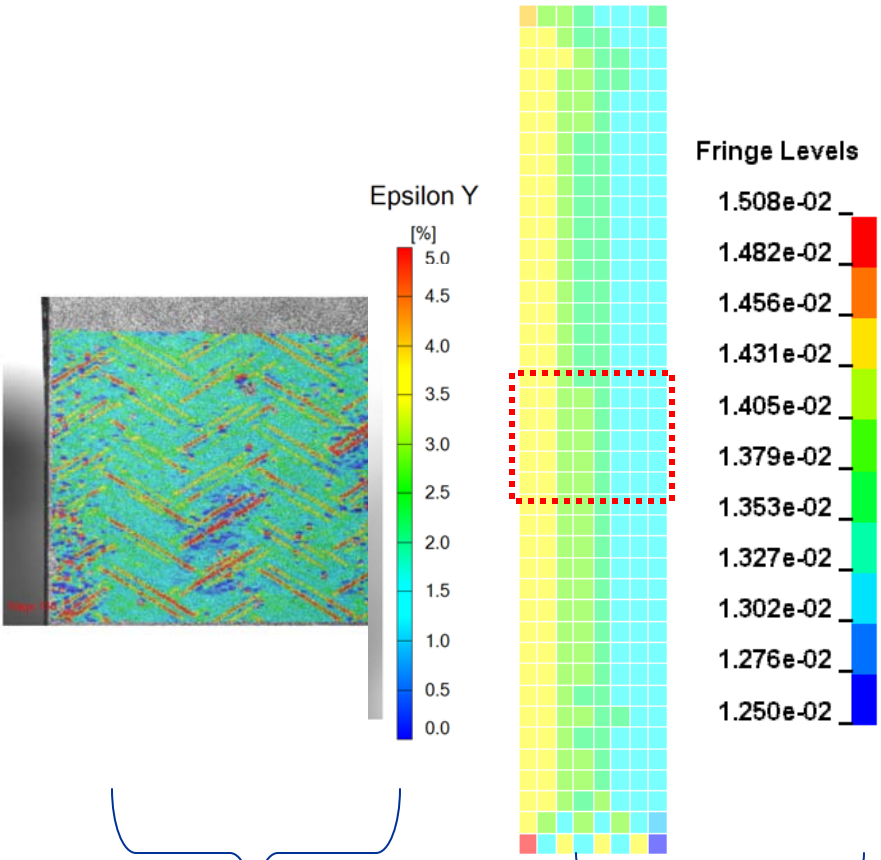
90%
535 MPa

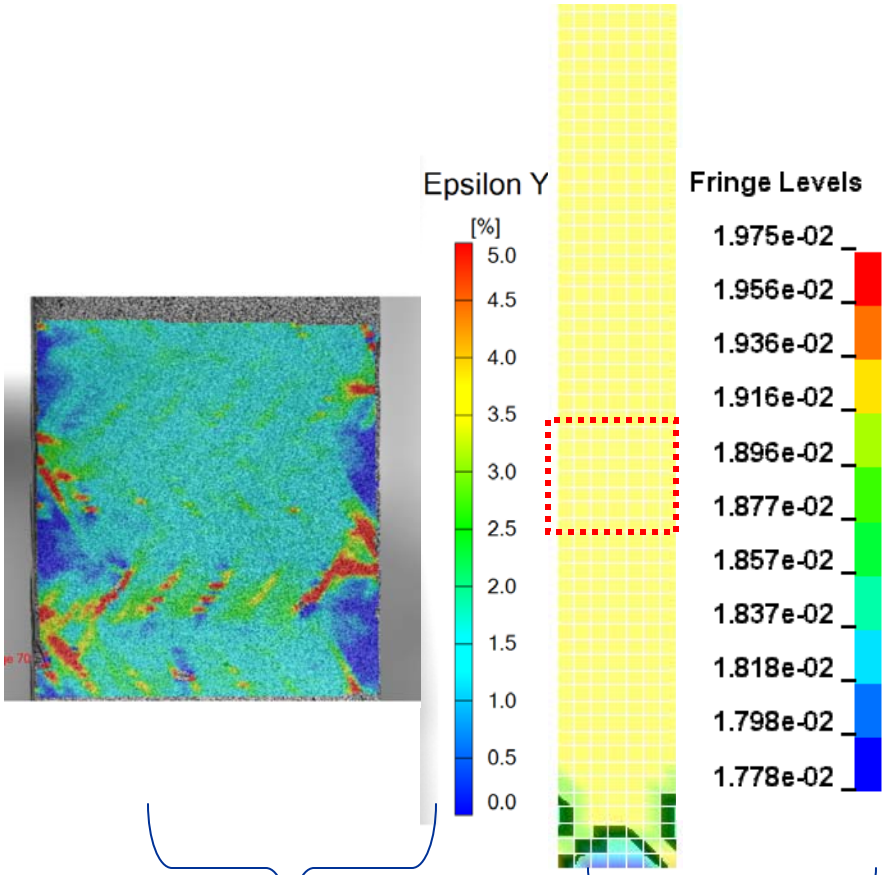


- OOP displacements verified by NDE
- Global Material response curves become non-linear after delaminations occurred
- Due to the nature of integration point formulation, cannot simulate ultimate failure values between layers

Limitations: FEM cannot simulate fiber bundle splitting

- Axial Tension (axial strain)
- Transverse Tension (axial strain)





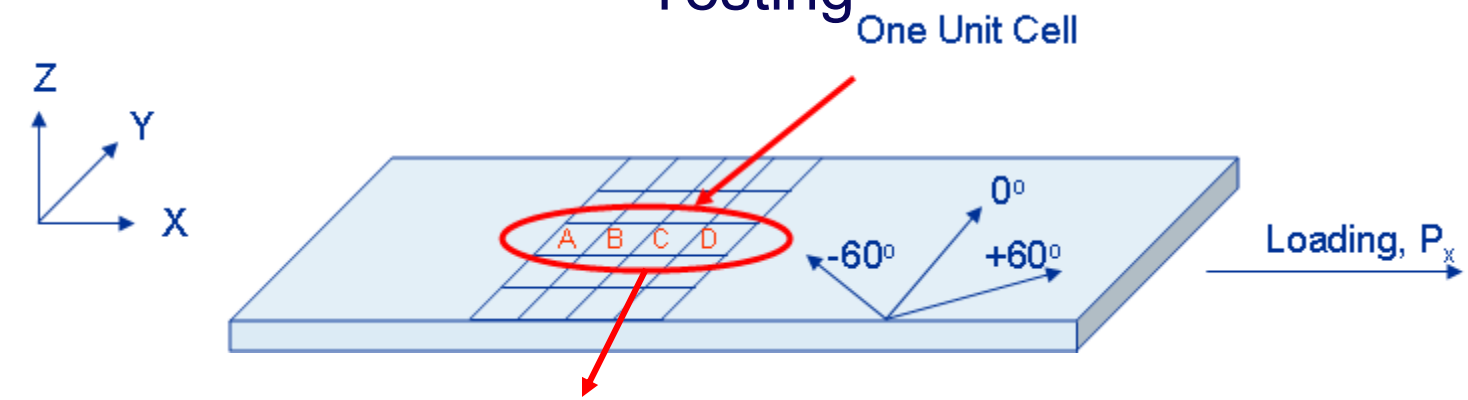
Optical Measurements LS-DYNA Optical Measurements LS-DYNA

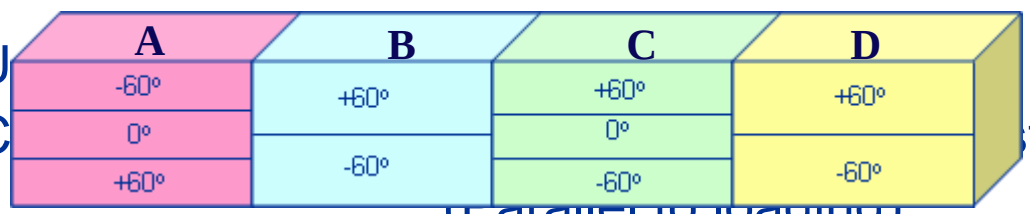
Conclusion

- Standardized test methods in conjunction with an optical measurement system have been used to collect material property data for triaxial braided composite materials
  - Global material response curves
  - Local transverse fiber bundle splitting
  - Local subsurface delaminations
- A hybrid micro-macromechanical computer model has been developed
  - Incorporates braid architecture
  - Incorporates tested material property data
- Comparisons between test and simulation show good agreement
  - 10% in static simulations
  - Penetration threshold in impact simulations
- Factors seen in the test data cannot be simulated as of now
  - Ongoing work

Backup

Equation Development – Transverse Tensile Testing





- U geometry $\rightarrow$ P_x, N_x
- C strain assumption yields
- (Parallel to loading)
- (Perpendicular to loading)

$$N_x^a = N_x^b = N_x^c = N_x^d$$

$$V_f^a * N_y^a = V_f^b * N_y^b$$

Rewriting the equations for each Subcell (Transverse Tensile Testing)

- Subcell A
  - N_x is applied load and all strains are found from optical measurement system

$$N_x^a = A11^a * \varepsilon_x^a + A12^a * \varepsilon_y^a$$

$$N_y^a = A12^a * \varepsilon_x^a + A22^a * \varepsilon_y^a$$

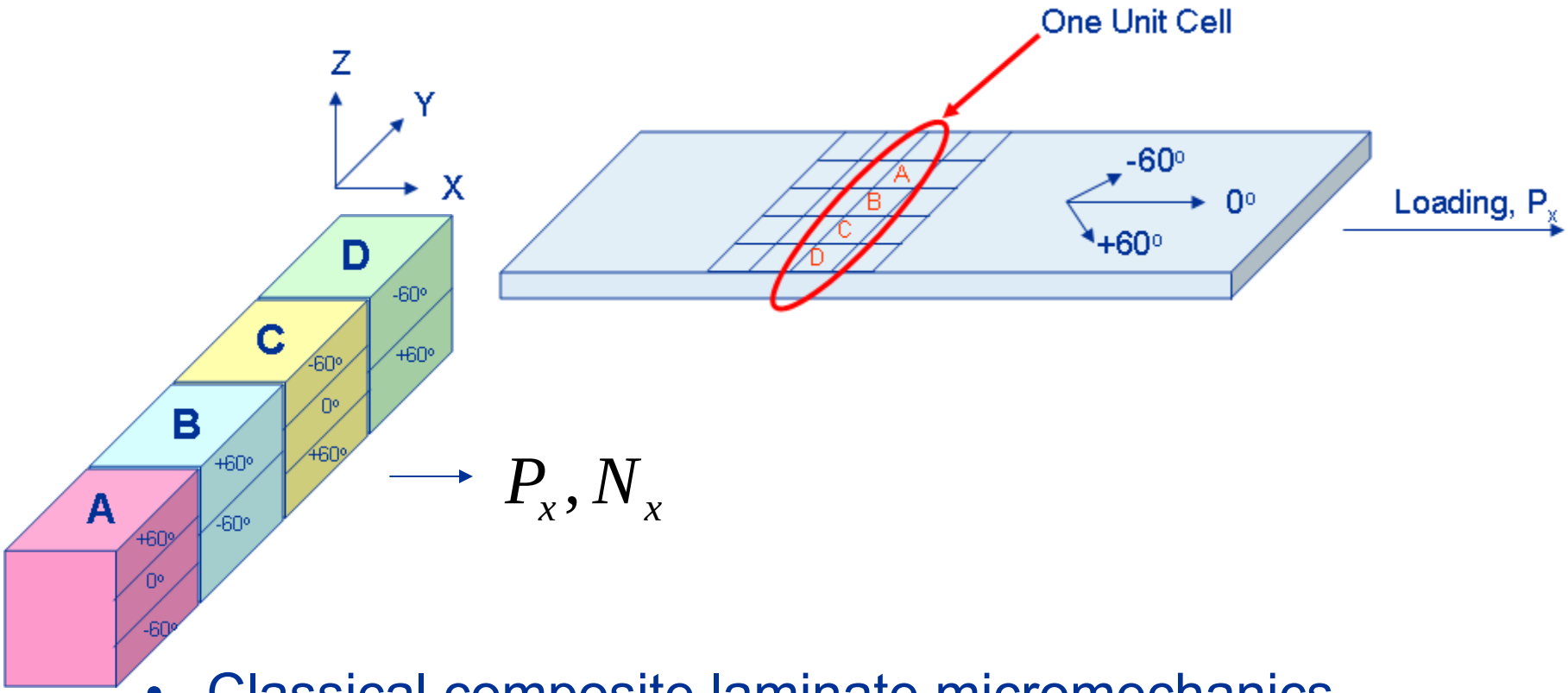
- Subcell B
  - N_x is applied load and all strains are found from optical measurement system

$$N_x^b = A11^b * \varepsilon_x^b + A12^b * \varepsilon_y^b$$

$$N_y^b = A12^b * \varepsilon_x^b + A22^b * \varepsilon_y^b$$

- Four Equations

Equation Development – Axial Tensile Testing



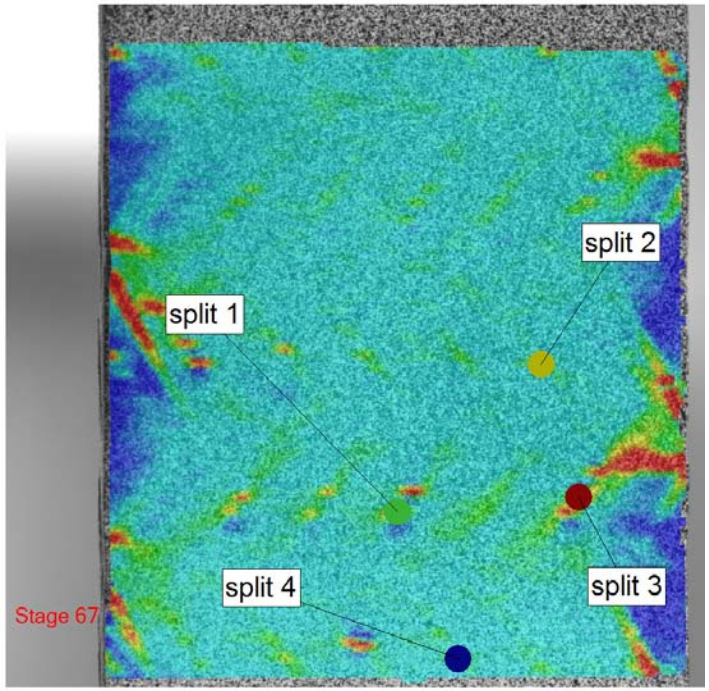
- Classical composite laminate micromechanics assumptions yield

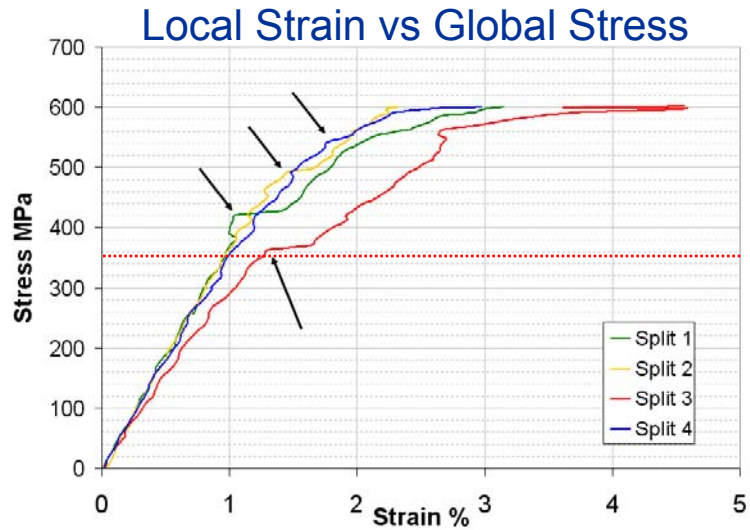
$$N_y^a = N_y^b = N_y^c = N_y^d = 0$$

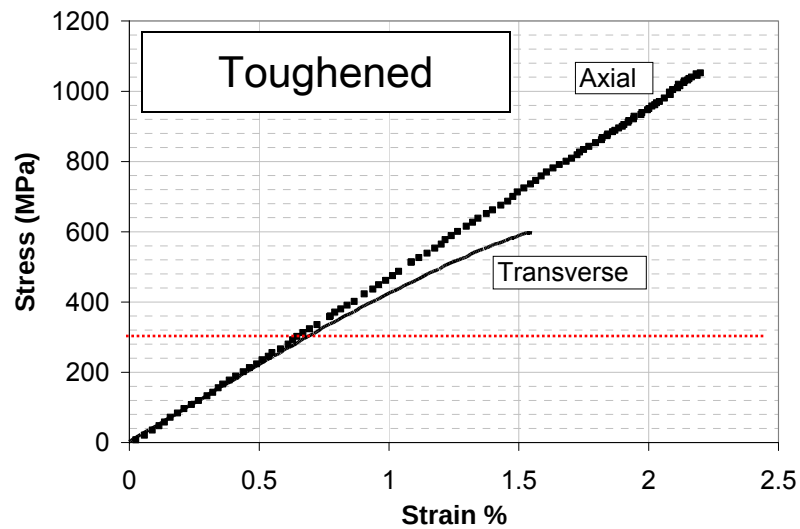
$$N_y^a = 0 = A12^a * \varepsilon_x^a + A22^a * \varepsilon_y^a$$

Advanced Data Analysis

Toughened Fiber Bundle Splitting – Transverse Testing

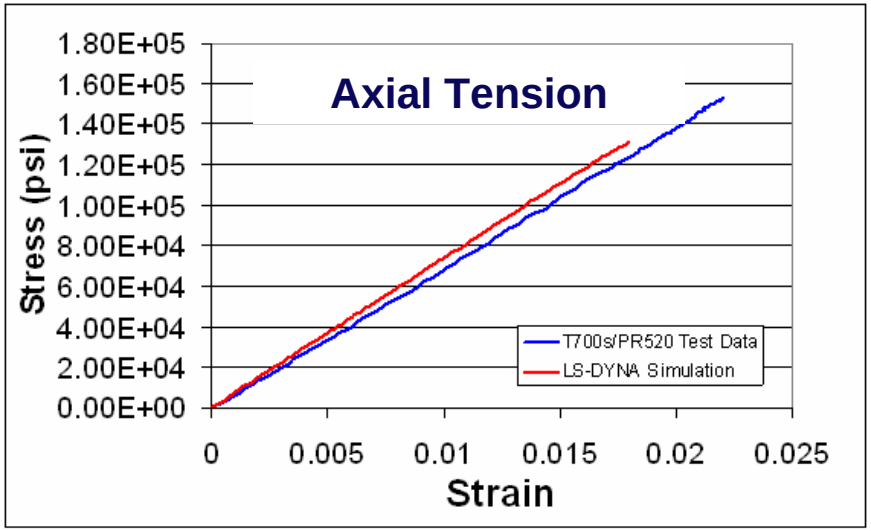


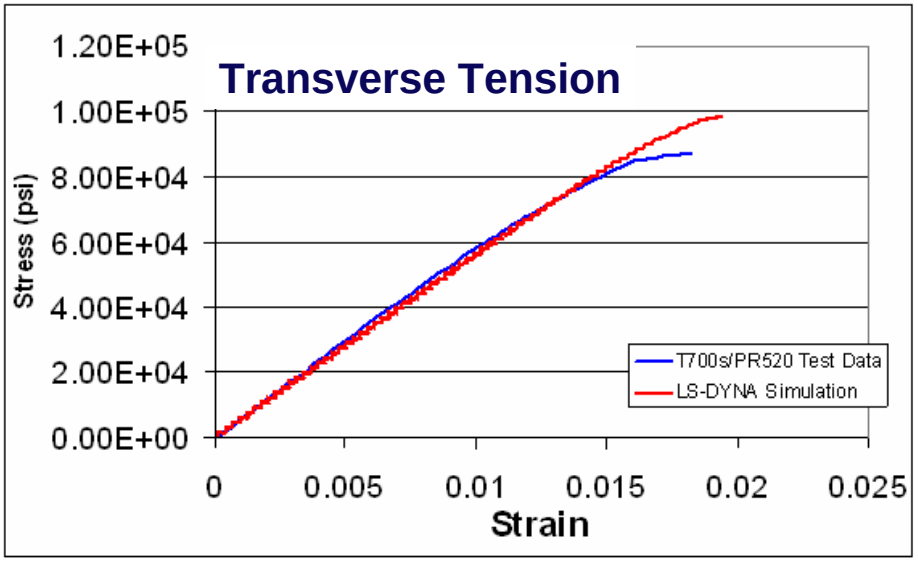




- Identify local failure strain
- Local failure initiation correlations to global non-linearities

T700 Fiber / PR520 Resin Static Results



	Axial Tension Modulus (psi)	*Axial Tension Strength (psi)*
Test	6.8E6±1.6E5	1.52E5±4.9E3
LS-DYNA	7.4E6	1.31E5
% Error	7%	12%

	Transverse Tension Modulus (psi)	*Transverse Tension Strength (psi)*
Test	6.2E6±2.3E5	8.69E4±4.3E2
LS-DYNA	5.6E6	9.38E4
% Error	9%	8%