Fatigue Crack Closure Analysis Using Digital Image Correlation

William P. Leser  
North Carolina State University, Raleigh, North Carolina

John A. Newman  
Langley Research Center, Hampton, Virginia

William M. Johnston  
Lockheed Martin Engineering and Sciences  
Langley Research Center, Hampton, Virginia

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Abstract

Fatigue crack closure during crack growth testing is analyzed in order to evaluate the criteria of ASTM Standard E647 for measurement of fatigue crack growth rates. Of specific concern is remote closure, which occurs away from the crack tip and is a product of the load history during ΔK-reduction fatigue crack growth testing. Crack closure behavior is characterized using relative displacements determined from a series of high-magnification digital images acquired as the crack is loaded. Changes in the relative displacements of features on opposite sides of the crack are used to generate crack closure data as a function of crack wake position. For the results presented in this paper, remote closure did not affect fatigue crack growth rate measurements when ASTM Standard E647 was strictly followed and only became a problem when testing parameters (e.g., load shed rate, initial ΔK, etc.) greatly exceeded the guidelines of the accepted standard.

Introduction

Fatigue crack growth (FCG) test data are most commonly presented in plots of FCG rate (da/dN) as a function of the cyclic crack-tip stress intensity factor, ΔK, herein called the crack driving force. A schematic of a typical FCG rate curve is shown in Figure 1. Using logarithmic axes, typical engineering alloys exhibit a linear FCG-rate-versus-ΔK relation at intermediate ΔK values (Paris regime), with crack growth rates rapidly increasing at higher ΔK values (near the fracture toughness) and with rapidly decreasing rates as a FCG threshold (ΔKth) is approached (ref. 1). The near-threshold region, characterized by slow crack growth rates, is of primary concern to the authors. Accurate determination of ΔKth is of practical importance for life prediction of many components due to the extremely slow FCG rates, and the tendency of the crack driving force, ΔK, to increase with increasing crack length. The majority of fatigue life for many aircraft components is consumed near the threshold and small changes in ΔKth can significantly affect the predicted fatigue life. Therefore, it is important to characterize the crack growth threshold (ΔKth), and understand mechanisms that influence threshold, most notably crack closure, the crack-face contact that occurs during cyclic loading (refs. 2-5).

Naturally-occurring cracks initiate under near-threshold loading conditions and propagate under increasing-ΔK conditions. Development of FCG rate curves is problematic in that test cracks can not be practically started in the threshold region, mimicking service conditions by generating the entire FCG curve under increasing ΔK conditions. For practical reasons, tests are typically started in the Paris regime (intermediate ΔK values) while the applied loads are gradually reduced such that ΔK values decrease as the crack propagates, as indicated by the dashed-red curve in Figure 1. Care must be taken to ensure that this artificial loading does not affect the fatigue crack growth rate data. ASTM standard E647 ("Standard Test Method for Measurement of Fatigue Crack Growth Rates") was developed to ensure satisfactory test results are obtained (ref. 6).

Recent research suggests that performing crack growth tests under ΔK-reduction conditions can adversely affect the FCG data (refs. 7, 8) due to a test-history-induced crack closure phenomenon. During cyclic loading, crack faces can contact before the load reaches the minimum value, a phenomenon called crack closure, which is naturally occurring, and typically occurs near the crack tip (see Figure 2a). However, when crack closure is a product of the load reduction method, such that contact away from the crack tip occurs first (termed "remote closure"; see Figure 2b), the FCG data are affected in an artificial
way, i.e., the results would be a product of the test history and not an indication of the mechanical performance. Here, remote closure is assumed to be a product of high crack-tip plasticity created at high ΔK near the start of the FCG test.

Remote closure will likely result in artificially high values of ΔK_th, as shown schematically in Figure 3, because the reduction in the crack-tip driving force is a result of the crack-wake load history rather than the mechanical performance of the material. In the absence of remote closure, the FCG threshold would be ΔK_th1, but due to remote closure that may develop as a threshold test progresses, a threshold value of ΔK_th2 is obtained. In this example, if the non-conservative threshold value of ΔK_th2 is used in the design of a component, premature failure may occur.

Although ASTM standard E647 appears to produce good data for a wide range of materials, it was developed to be material independent. However, the level of crack-tip plasticity and, therefore, the likelihood of remote closure, is dependent on material behavior (e.g., yield stress). Further, a significant practical concern with the standard is that as a test approaches the threshold region, the FCG growth rates become very slow, which requires extensive amounts of machine time. Accelerating the test, especially near the threshold region, would greatly improve testing efficiency if it could be done without sacrificing the quality of the results (i.e., no remote closure). Therefore the objective of this study is to investigate the possibility of exceeding the ASTM standards, using aluminum alloys, without producing remote closure. Ultimately this approach should be used as an experimental approach to obtain an optimized testing procedure. In this paper, data from FCG tests done, both in accordance with, and in violation of, ASTM Standard E647 will be analyzed to obtain detailed fatigue crack closure profiles to characterize crack closure during the tests.

* Here, optimization means determining the most rapid (in terms of machine time) load history that will achieve FCG threshold conditions without remote closure.
Figure 2. Schematics of fatigue crack closure scenarios: (a) Crack tip closure and (b) Remote closure.

Figure 3. FCG Rate curves illustrating likely changes in $\Delta K_{th}$ when remote closure occurs ($\Delta K_{th2}$) and in the absence of remote closure ($\Delta K_{th1}$) (ref. 5).
Testing

Fatigue crack growth tests were performed using closed-loop servo-hydraulic test machines with constant amplitude sinusoidal loading. With only noted exceptions, testing was conducted in accordance with ASTM standard E647 using eccentrically-loaded single-edge notch tension (ESE(T)) specimens (ref. 9) having width, W, and thickness, B, of 38.1 mm and 2.3 mm, respectively. A computer-controlled system (ref. 10) was used to continuously monitor crack length during testing using the back-face compliance technique (ref. 11). This system automatically adjusts loads as the crack grows to ensure that programmed stress intensity factors are applied throughout the tests.

Crack closure data was obtained by analyzing a series of high-magnification (300-700X) images of the crack obtained during cyclic loading. A photograph of the equipment used (microscope, digital camera, and test machine) is shown in Figure 4. A random pattern of 4 μm speckles was deposited on specimen surfaces in the region of crack growth to provide features whose motion can be tracked as a function of load in the digital images. A photograph of a specimen with a field of speckles applied to the surface is shown in Figure 5. At high-magnification (lower portion of Figure 5), individual speckles can be seen. The relative displacement of features on either side of the crack is used to determine when crack closure occurs. The cyclic loading was slowed to 0.01 Hz (one load cycle per 100 seconds) and images were captured every second for more than two complete load cycles (240 total images).

![Figure 4. Servo-hydraulic test stand and digital camera with microscope.](image)
Digital Image Correlation

The digital image correlation system provides 2D displacement data from the series of pictures taken during the test. The digital images captured during cyclic loading were analyzed using VIC-2D (ref. 12), an image processing program developed by Correlated Solutions, Inc.† Digital images of the speckled regions of the specimen were loaded into the image correlation system. An area of interest was selected on the first photograph, which defined the area where displacement data were taken. Over the entire selected area of interest, subsets of pixels are tracked in each image relative to their position in the reference image. This image correlation process results in a series of displacement fields that can be post-processed in order to observe how the strain field changes throughout the load cycle. An example of an image correlation system generated strain field ($\varepsilon_{yy}$, strain in vertical direction) for the wake of a loaded fatigue crack is shown in Figure 6.

† Herein, VIC-2D will be referred to as “image correlation system.”
Closure Analysis

After the digital images were analyzed using the image correlation system, the displacement data were analyzed using the Elber method to determine the load at which the crack closed (ref. 13). In this method, the crack-opening displacement at a point along the crack wake was plotted against load resulting in a compliance plot for that specific crack-wake location. Deviations in linearity on this plot indicate crack closure. A typical plot of crack opening compliance is shown in Figure 7a. To better determine deviations from linearity, a least squares regression analysis was performed on the closure-free portion of the compliance plot. The error was then calculated between the data points and the regression line. These error values were then plotted against load; this is called a reduced compliance plot. A typical plot of reduced compliance is shown in Figure 7b. Two regression lines were then fit to the reduced compliance data. The first is fit to the data that were assumed to be closure free (in the example of Figure 7b, all points above a load ratio of 0.6). The second is fit to data affected by closure (in this example, all points below a load ratio of approximately 0.3). The intersection of these two fitted lines was used as the load at which crack closure occurs for that specific crack-wake location. This method was repeated every five pixels (approximately 4 µm for the example of Figure 7) along the crack to produce a closure profile along the crack wake as in Figure 7c.
Figure 7. Reduced compliance method as implemented in MATLAB.\(^1\)

\(^1\) MATLAB is a registered trademark of The Math Works, Inc.
Results

To validate the image correlation system method, digital images from a test that had been previously analyzed using an older image correlation system were reanalyzed (ref. 5). Here, constant-$\Delta$K fatigue crack growth testing was conducted on an aluminum alloy 8009 specimen at $R = 0.05$ and $K_{\text{max}} = 5.5$ MPa$\sqrt{\text{m}}$ ($\Delta K = 4.95$ MPa$\sqrt{\text{m}}$) until steady-state crack growth conditions were achieved. Testing was momentarily stopped while alumina slurry ($\text{Al}_2\text{O}_3$ particles mixed with methanol) was introduced into the crack mouth to simulate a thick crack-mouth oxide layer. Fatigue loading resumed and the crack wake was analyzed after 130 $\mu$m of crack growth. The results of this previous study are shown in Figure 8 using circular symbols. Crack closure levels in much of the crack wake occur near $R_{\text{contact}}$ (ratio of contact load and maximum load) of 0.5, as indicated by the horizontal red dashed line. Closure loads decrease between the location of the crack-wake particles (130 $\mu$m from the crack tip) and the crack tip. Closure data obtained from the image correlation system (analyzing the same digital images) are also shown in Figure 8 (solid line curve). These data exhibit nearly the same behavior as the previous results suggesting that the image correlation system is able to reproduce the results from reference 5, but with the advantage of generating data at hundreds of additional crack-wake locations.

Figure 8. Comparison of image correlation system (VIC-2D) results with published results.

Preliminary results of this study have demonstrated that remote closure can occur when the testing parameters recommended by ASTM Standard E647 are greatly exceeded. Figure 9 shows the closure profile of an aluminum alloy 8009 specimen tested at constant-$R = 0.1$ conditions (initial $K_{\text{max}} = 11$ MPa$\sqrt{\text{m}}$ and $C = -393.7$/m) obtained using the image correlation system. The experimental results of

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$\Delta K$ during fatigue crack growth testing is characterized by the K-gradient, $C$. Negative values indicate that $\Delta K$ values decrease as the crack propagates. Standard test methods require $C > -78.7$/m. See reference 6 for details.
Figure 9 were taken after approximately 3mm of crack growth ($\Delta K = 3.3 \text{ MPa$\sqrt{\text{m}}$}$), which is approaching the $R = 0.1$ FCG threshold value for this alloy. High-magnification images were taken at three locations; near the crack tip, the location corresponding to the start of the $\Delta K$-reduction test, and further behind the crack tip (corresponding to steady state pre-cracking at $\Delta K = 9.9 \text{ MPa$\sqrt{\text{m}}$}$, $R = 0.1$). This test exceeds the ASTM standard E647 limits on the K-gradient, C, by a factor of 5. Here, remote closure is shown to occur because crack closure occurs in the crack wake before occurring at the crack tip. The horizontal red dashed line in the figure corresponds to the mean normalized closure load ($P_{\text{closure}}/P_{\text{max}} = 0.21$) of the crack wake. In comparison, the crack tip closes later (at a lower load $P_{\text{closure}}/P_{\text{max}} = 0.12$), which is the definition of remote closure, and is assumed to be an artifact of the test procedure.

![Aluminum alloy 8009](image)

**Aluminum alloy 8009**

Constant-$R = 0.1$, $C = -393.7/\text{m}$

$\Delta K_i = 9.9 \text{ MPa$\sqrt{\text{m}}$}$, $\Delta K = 3.3 \text{ MPa$\sqrt{\text{m}}$}$

Figure 9. Closure profile for a specimen of aluminum alloy 8009 with initial $K_{\text{max}} = 11 \text{ MPa$\sqrt{\text{m}}$}$ ($K_i = 9.9 \text{ MPa$\sqrt{\text{m}}$}$) and $C = -393.7/\text{m}$. Remote closure occurs.

However, experimental results also show that the guidelines of ASTM standard E647 are overly-conservative for some load scenarios. Figure 10 shows the closure profile (crack closure loads as a function of distance behind the crack tip) of a specimen of an aluminum alloy 8009 specimen tested at an initial $K_{\text{max}} = 11 \text{ MPa$\sqrt{\text{m}}$}$ and $C = -393.7/\text{m}$. The data presented in Figure 10 correspond to a $\Delta K = 7.4 \text{ MPa$\sqrt{\text{m}}$}$, after 800 $\mu\text{m}$ of crack growth from the start of the test. Here, crack closure occurs in the crack wake at approximately $P_{\text{closure}}/P_{\text{max}} = 0.25$, as indicated by the horizontal red dashed line. Closure loads ($P_{\text{closure}}/P_{\text{max}}$) increase closer to the crack tip (within 100 $\mu\text{m}$ of the crack tip), increasing to approximately $P_{\text{closure}}/P_{\text{max}} = 0.36$ at the crack tip. In this case, crack closure occurs first at the crack tip with closure occurring in the crack wake at lower loads, in a manner characteristic of steady-state crack closure in the absence of load history effects (ref. 14).
A test with the same conditions as in Figure 10 was conducted on an aluminum alloy 2024 specimen, a material with different crack closure characteristics. These constant-$R = 0.1$ FCG data are presented in Figure 11 (initial $K_{\text{max}} = 11$ MPa$\sqrt{m}$ and $C = -393.7/m$). Crack closure data were acquired after approximately 1 mm of crack growth from the start of the test, corresponding to a $\Delta K = 6.7$ MPa$\sqrt{m}$ (nearly the same $\Delta K$ value as for the data of Figure 10). Crack closure occurs in the crack wake at approximately $P_{\text{closure}}/P_{\text{max}} = 0.60$, as indicated by the horizontal red dashed line. Closure loads ($P_{\text{closure}}/P_{\text{max}}$) decrease closer to the crack tip (within 100 $\mu$m of the crack tip), reaching approximately $P_{\text{closure}}/P_{\text{max}} = 0.3$ near the crack tip. Here, the first crack closure event (highest $P_{\text{closure}}/P_{\text{max}}$) occurs in the crack wake (distance from crack tip > 100 $\mu$m) meaning that remote closure occurs, presumably as a product of the load history.

A final example is presented that conforms with ASTM Standard E647. Here, a constant-$R = 0.1$ FCG test is conducted on an aluminum alloy 7055 specimen (initial $\Delta K = 5.94$ ksi$\sqrt{\text{in}}$, $C = -78.7/m$). Crack closure data for this test are presented in Figure 12 after approximately 2.5 mm of crack growth (from the start of the $\Delta K$-reduction test), corresponding to an applied $\Delta K = 4.95$ MPa$\sqrt{m}$. Crack closure in the crack wake, near the start of the test, was determined to be approximately $P_{\text{closure}}/P_{\text{max}} = 0.27$. Within 400 $\mu$m of the crack tip, closure loads are shown to increase from this value ($P_{\text{closure}}/P_{\text{max}} = 0.27$) to approximately $P_{\text{closure}}/P_{\text{max}} = 0.4$ near the crack tip. This crack closure profile, characterized by the first closure event occurring near the crack tip with crack-wake locations closing at lower loads, is similar to that shown in Figure 10 and typical behavior of steady-state conditions, i.e., not affected by load history.
**Aluminum alloy 2024**

Constant-$R = 0.1$, $C = -393.7/m$

$\Delta K_l = 9.9$ MPa$\sqrt{m}$, $\Delta K = 6.7$ MPa$\sqrt{m}$

![Figure 11](image1.png)

*Figure 11. Closure profile for a specimen of aluminum alloy 2024 with initial $K_{max} = 11$ MPa$\sqrt{m}$ ($\Delta K_l = 9.9$ MPa$\sqrt{m}$) and $C = -393.7/m$. Remote closure occurs.*

**Aluminum alloy 7055**

Constant-$R = 0.1$, $C = -78.7/m$

$\Delta K_l = 5.94$ MPa$\sqrt{m}$, $\Delta K = 4.95$ MPa$\sqrt{m}$

![Figure 12](image2.png)

*Figure 12. Crack closure profile for aluminum alloy 7055 with initial $K_{max} = 6.6$ MPa$\sqrt{m}$ ($\Delta K_l = 5.94$ MPa$\sqrt{m}$) and $C = -78.7/m$. Remote closure does not occur.*
Discussion of Results

Digital image analysis (VIC-2D) has been used to characterize crack closure along the wake of a propagating crack during constant-R ΔK-reduction testing. Although unnatural (meaning that this is not a load history typical of service cracks), generating FCG data using this test method is practical as long as the load history does not affect the resulting FCG test results. Despite some studies that suggest following ASTM Standard E647 can result in remote closure and artificially-high values of ΔKth (refs. 7, 8), the results presented in this paper suggest otherwise. Two examples were shown (recall Figures 10 and 12) where the ΔK-reduction test did not result in remote closure, despite the fact that in one case (Figure 10) the ASTM criterion on the K-gradient \((C \geq -78.7/m)\) was violated by a factor of 5. Therefore, the results presented in this paper suggest that the ΔK-reduction testing prescribed by ASTM standard E647 is not only capable of producing good data (meaning not affected by remote closure), but is likely overly conservative in that tests may be able to operate at faster ΔK-shed rates without producing remote-closure-affected data. Further study is needed to determine an optimal load history.

An optimal load history for ΔK-reduction testing will likely be material dependent. Consider the difference in the observed crack closure characteristics between Figures 10 and 11, despite being subjected to nearly identical load histories. In this case, the most significant difference between these tests is the alloys. The tests corresponding to Figures 10 and 11 were performed on aluminum alloys 8009 and 2024, respectively. Aluminum alloy 8009 is a very fine-grained (typical grain diameter approximately 0.5 μm) powder-metallurgy alloy having an elastic modulus and yield stress of 87.6 GPa and 419.9 MPa, respectively (refs. 15-17). Aluminum alloy 2024 has a relatively coarse microstructure (typical grain size 70-150 μm) and has elastic modulus and yield stress of 71.7 GPa and 350.3 MPa, respectively (refs. 18, 19). Although these alloys have similar tensile properties, the fine-grain alloy (8009) was free of remote closure while the coarse-grain alloy (2024) was affected by remote closure, at least for the test conditions associated with Figures 10 and 11. Differences in crack-path roughness might be a factor, but additional study is needed before conclusions can be made.

Summary

This research has shown that remote closure may occur, especially when the criteria of ASTM standard E647 are violated. However, results presented in this paper suggest that the current standard is overly conservative for some test and material combinations such that the criteria of ASTM E647 can be violated without inducing remote closure. Any procedure for optimized load shedding (allow most rapid load shedding without sacrificing data integrity) should consider specific material properties, but more research is needed to determine which material properties are important.

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\[\text{Here, optimal means the fastest } \Delta K\text{-reduction test that does not result in remote closure.}\]
References


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Leser, William P.; Newman, John A.; Johnston, William M.

NASA Langley Research Center
Hampton, VA 23681-2199

National Aeronautics and Space Administration
Washington, DC 20546-0001

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