



Durability and CMAS Resistance of Advanced Environmental Barrier Coatings Systems for SiC/SiC Ceramic Matrix Composites

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Environmental Barrier Coating - CMAS Interaction Research Efforts

- Advanced EBC development – composition design and developments for improved CMAS resistance; thermomechanical-CMAS Interactions and durability – Zhu et al
- NASA-Air Force Venture and Viper Turbine Coating-CMAS Collaborative programs - Zhu, James Smialek, Robert A. Miller, Bryan Harder
- Formal NASA Intern Undergraduate Students – Nadia Ahlborg and Dan Miladinovich
- Fundamental NASA in-house CMAS properties - Narottam Bansal and Valerie Weiner



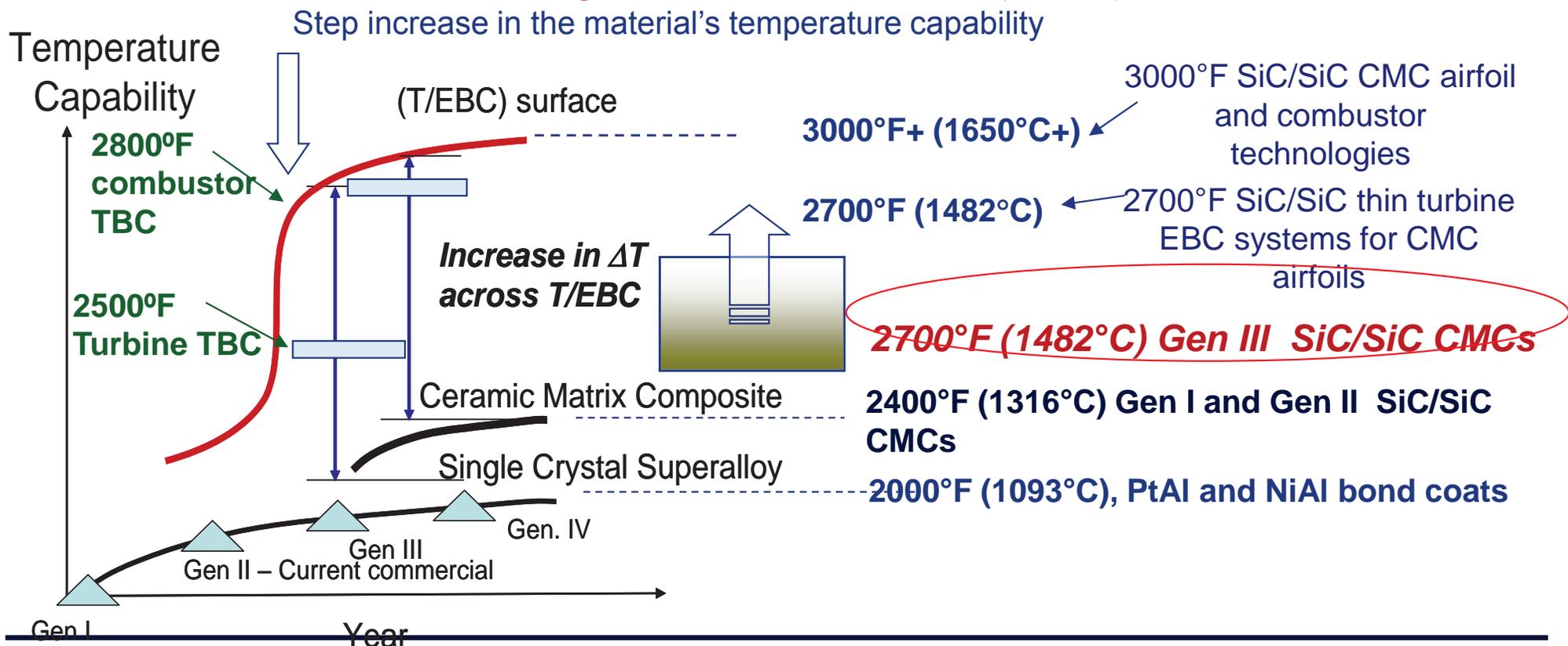
Outline

- **Environmental barrier coating (EBC) development: the CMAS relevance**
- **Some generalized CMAS related failures**
- **CMAS degradation of environmental barrier coating (EBC) systems: rare earth silicates**
 - Ytterbium silicate and yttrium silicate EBCs
 - Some reactions, kinetics and mechanisms
- **Advanced EBCs, HfO₂- and Rare Earth - Silicon based 2700°F+ capable bond coats**
- **Summary**



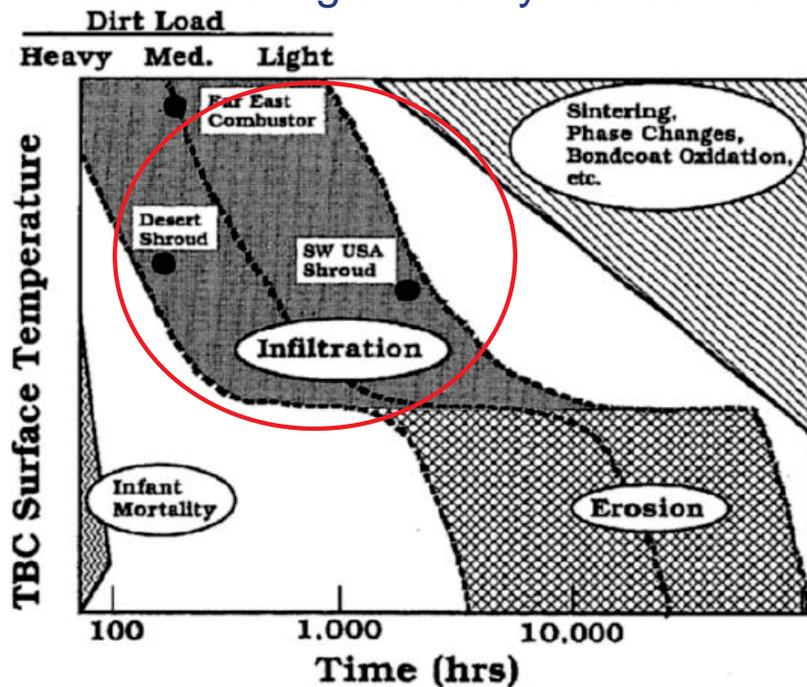
NASA Environmental Barrier Coatings (EBCs) and Ceramic Matrix Composite (CMC) System Development

- **Emphasize material temperature capability, performance and long-term durability-** Highly loaded EBC-CMCs with temperature capability of 2700°F (1482°C)
 - 2700-3000°F (1482-1650°C) turbine and CMC combustor coatings
 - 2700°F (1482°C) EBC bond coat technology for supporting next generation
 - Recession: <5 mg/cm² per 1000 h
 - Coating and component strength requirements: 15-30 ksi, or 100- 207 Mpa
 - **Resistance to Calcium Magnesium Alumino-Silicate (CMAS)**

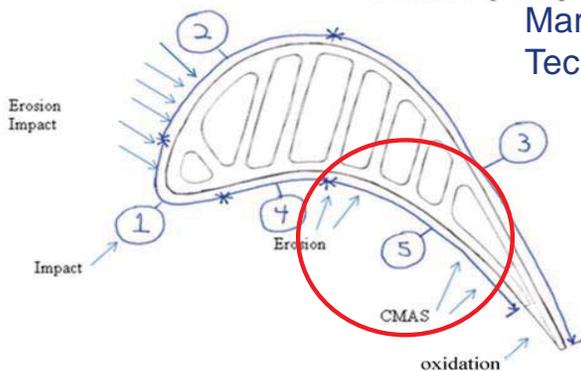


EBC-CMAS Degradation is of Concern with Increasing Operating Temperatures

- Emphasize improving temperature capability, performance and *long-term* durability of ceramic turbine airfoils
- Increased gas inlet temperatures for net generation engines lead to significant CMAS - related coating durability issues – CMAS infiltration and reactions



Marcus P. Borom et al, Surf. Coat. Technol. 86-87, 1996



Current airfoil CMAS attack region - R. Darolia, International Materials Reviews, 2013

Calcium Magnesium Alumino-Silicate (CMAS) Systems Used in Laboratory Tests

- Synthetic CMAS compositions, in particular, NASA modified version (NASA CMAS), and the Air Force Powder Technology Incorporated PTI 02 CMAS currently being used
- Saudi Sands used for past turbine coating studies
- CMAS SiO₂ content typically ranging from 43-49 mole%; such as NASA's CMAS (with NiO and FeO)
- Collaborations on-going with the Air Force; also planned DLR, ONERA etc on Volcanic Ash

Composition selections
ARFL PTI 11717A 02 used at NASA for CMAS studies

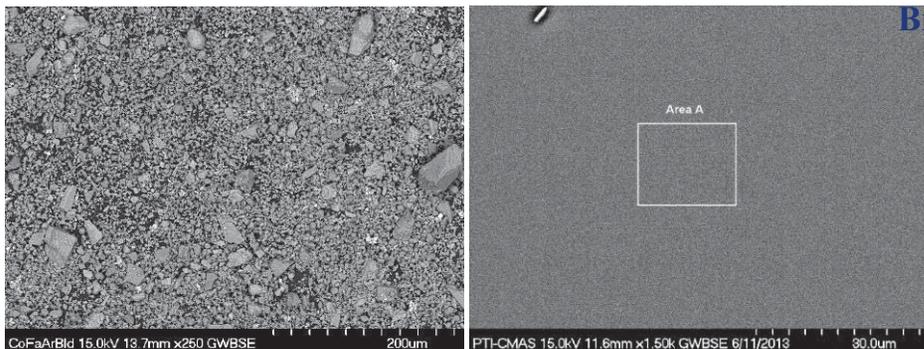
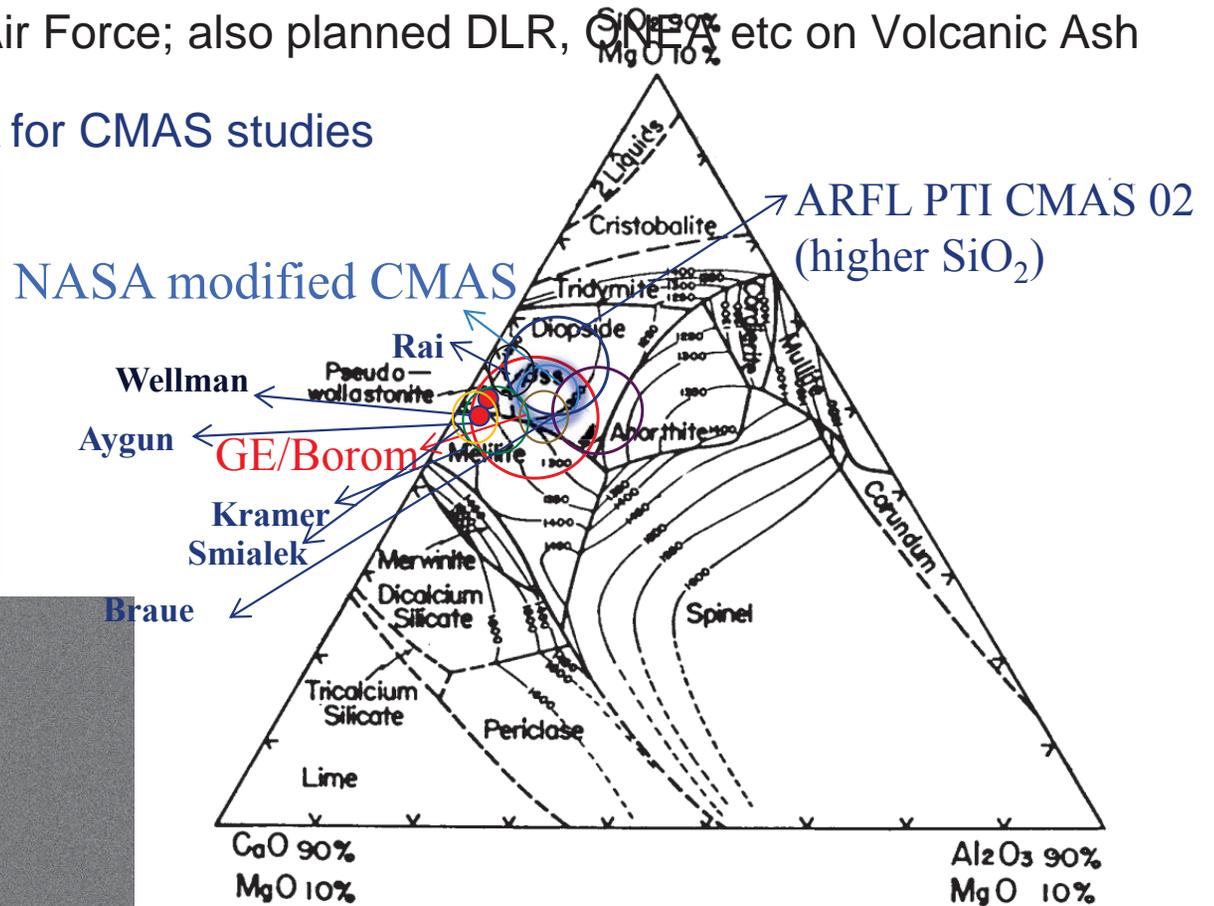
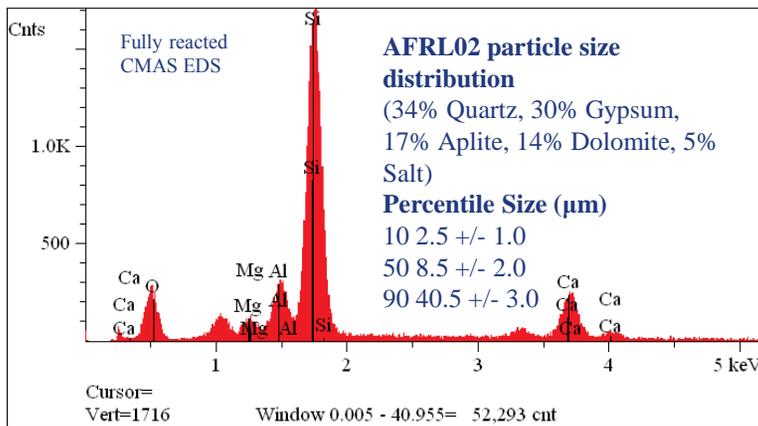
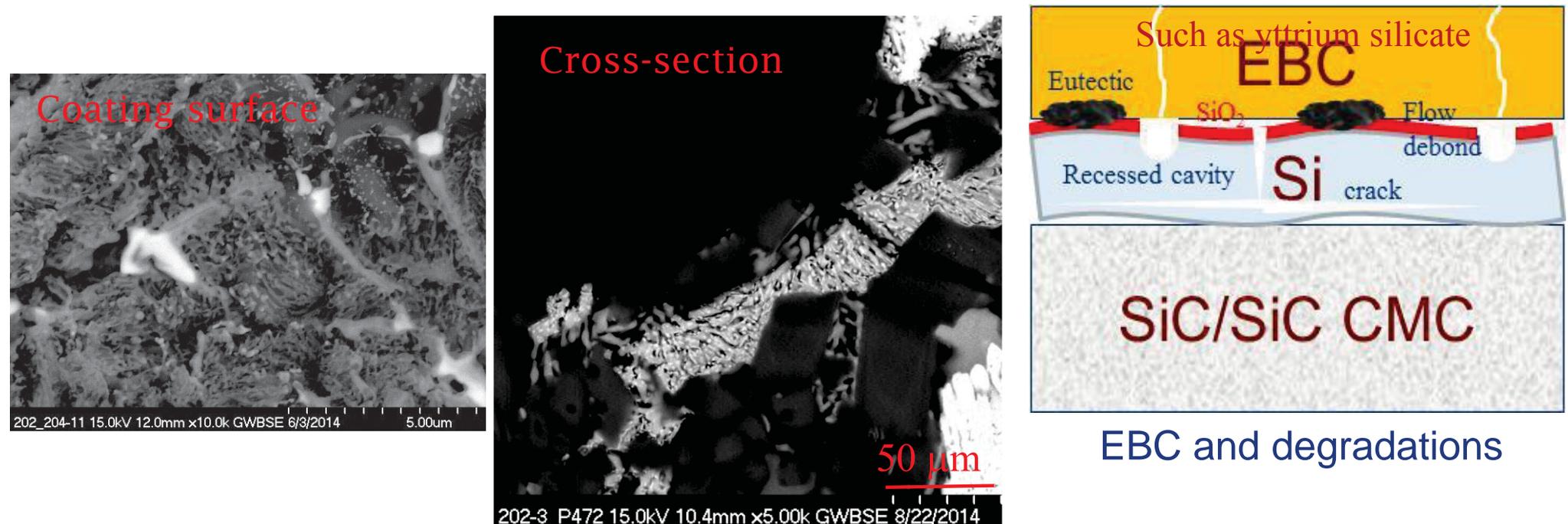


Fig. 4. The 10% MgO plane of the system CaO-MgO-Al₂O₃-SiO₂ showing the isotherms and fields of primary crystallization. A.T.Prince, J.Amer.Ceram.Soc., 37(9)1954 p402-408

CMAS Related Degradations in EBCs

– CMAS effects

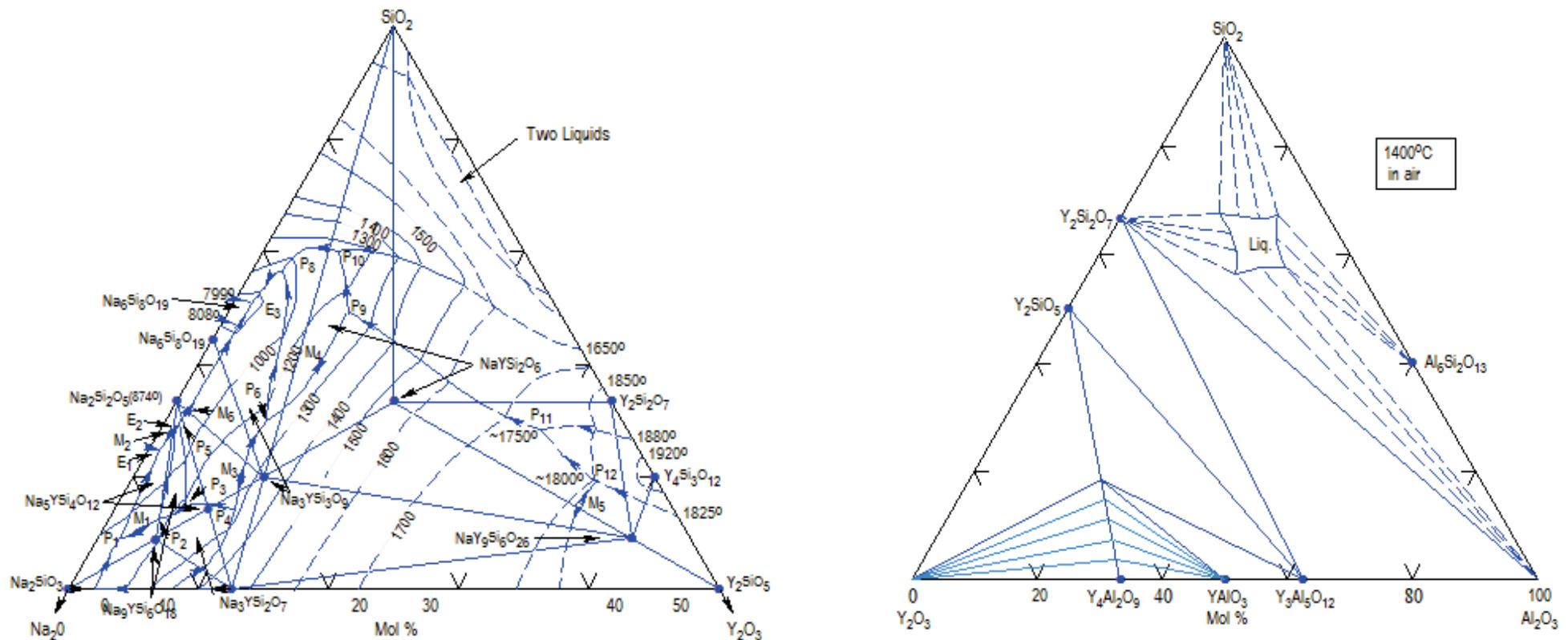
- Significantly reduce melting points of the EBCs and bond coats
- Cause more severe degradations with thin airfoil EBCs
- CMAS increase EBC diffusivities and permeability, thus less protective as an environmental barrier
- Reduced mechanical properties: such as strength and toughness reductions
- Leads to grain boundary attack thus disintegrate EBCs
- CMAS interactions with heat flux, thermal cycling, erosion and thermomechanical fatigue



CMAS induced melting and failure

CMAS Related Degradations in EBCs - Continued

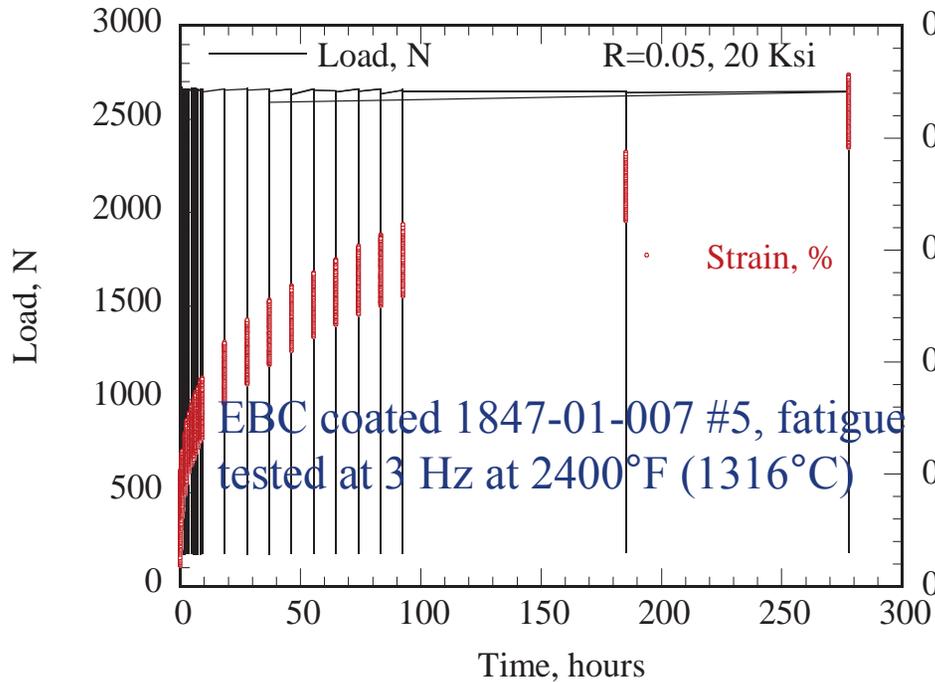
- **CMAS effects on EBC temperature capability**
 - Silicate reactions with NaO_2 and Al_2O_3 silicate



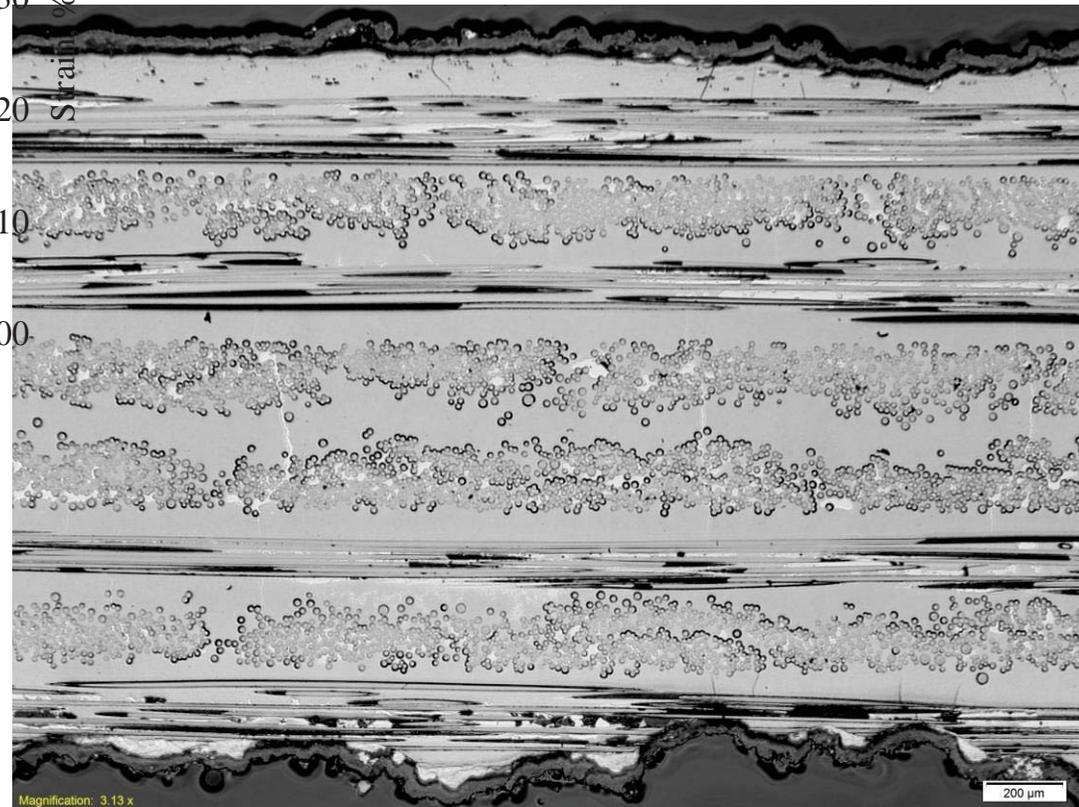
Phase diagrams showing yttrium di-silicate reactions with SiO_2 , NaO and Al_2O_3

CMAS Related Degradations in EBCs

- **Fatigue – environmental interaction is of great concern**



A 20 micrometer thick EBC bond coated Prepreg SiC/SiC CMC after 40 hr, 20 Ksi, stress ratio R=0.05 fatigue testing in air



Environmental Barrier Coating Development Limitations and Requirements

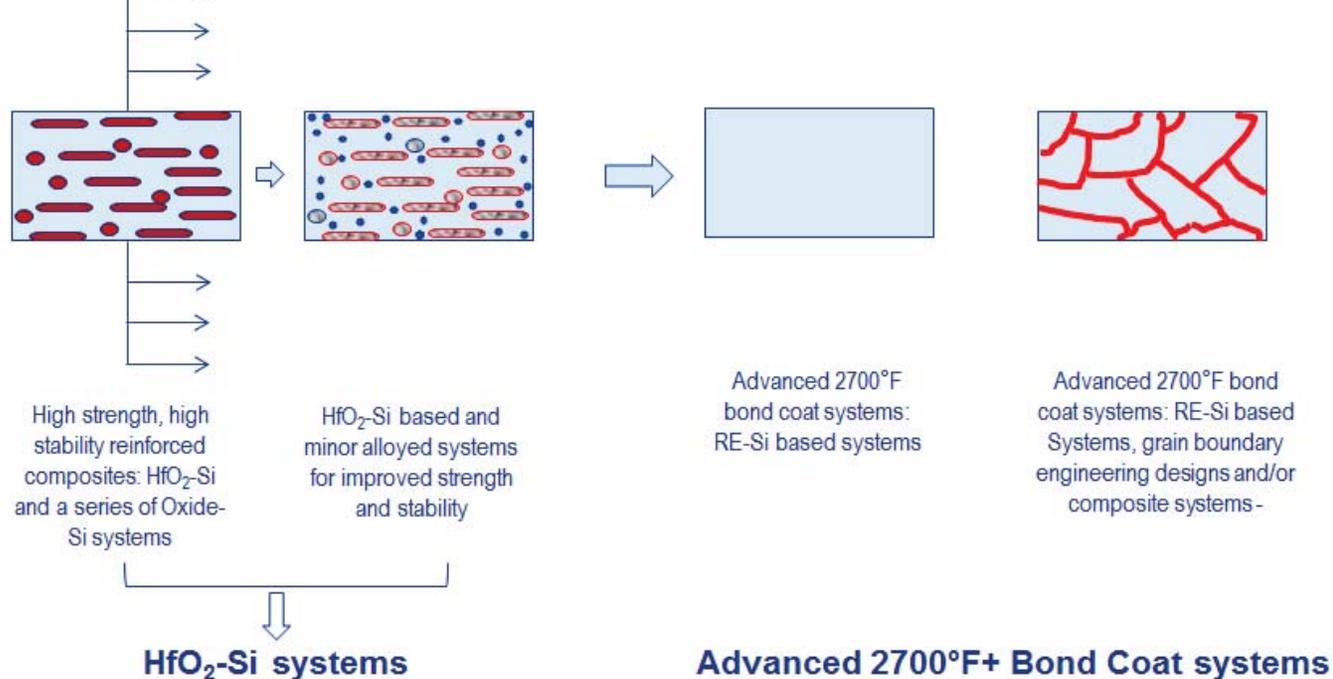


- Current EBCs limited in their temperature capability, water vapor stability and long-term durability, especially for advanced high pressure, high bypass turbine engines
- Advanced EBCs also require higher strength and toughness
 - In particular, resistance to combined high-heat-flux, engine high pressure, combustion environment, creep-fatigue, loading interactions
- EBCs need improved erosion, impact and calcium-magnesium-alumino-silicate (CMAS) resistance and interface stability
 - Critical to reduce the EBC Si/SiO₂ reactivity and their concentration tolerance
- EBC-CMC systems need advanced processing for realizing complex coating compositions, architectures and thin turbine configurations for next generation high performance engines
 - Advanced high temperature processing of high stability cluster and nano-composites

NASA EBC Systems

NASA EBC Systems

- HfO_2 - RE_2O_3 - SiO_2 / $\text{RE}_2\text{Si}_{2-x}\text{O}_{7-2x}$ environmental barrier systems
 - Controlled silica content and transition element and rare earth dopants to improve EBC stability and toughness
 - Develop HfO_2 -Si based + X (dopants) and more advanced rare earth composite compound composition systems for 2700°F+ long-term applications
 - Develop prime-reliant composite EBC-CMC interfaces for fully integrated EBC-bond coat systems
- RE_2O_3 - SiO_2 - Al_2O_3 Systems
- Develop advanced NASA high toughness alternating layered systems
- Advanced 1500°C bond coats

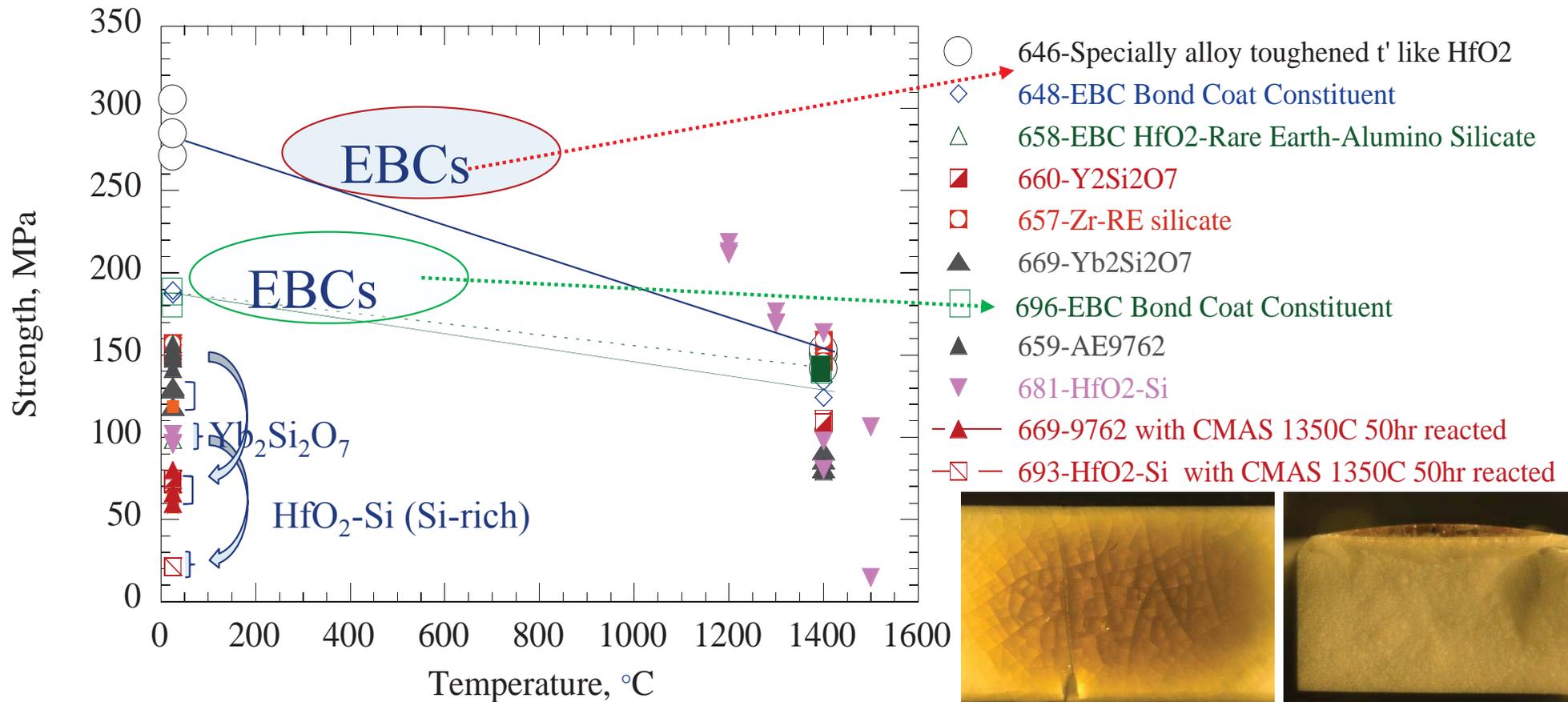


Strength Results of Selected EBC and EBC Bond Coats

- CMAS Reaction resulted in Strength Reduction in Silicates

Selected EBC systems

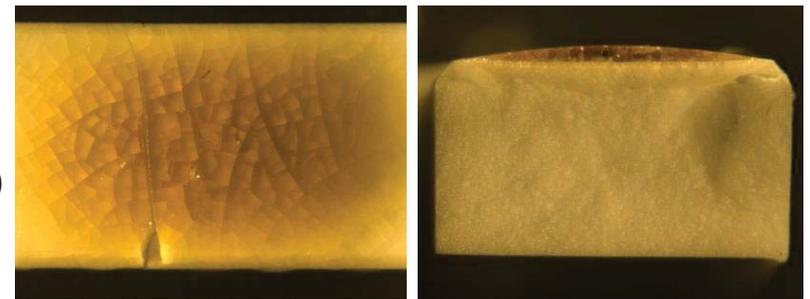
- HfO_2 -RE-Si, along with co-doped rare earth silicates and rare earth aluminosilicates, for optimized strength, stability and temperature capability
- CMAS infiltrations can reduce the strength



Strength test data compared

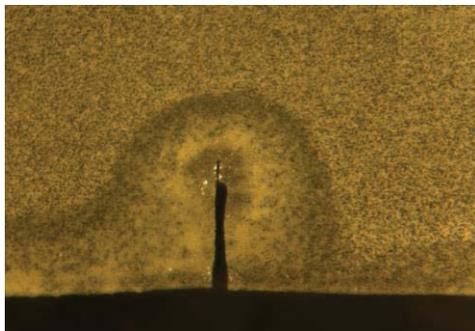
$\text{Yb}_2\text{Si}_2\text{O}_7$ CMAS reacted tensile surface

$\text{Yb}_2\text{Si}_2\text{O}_7$ CMAS reacted specimen fracture surface

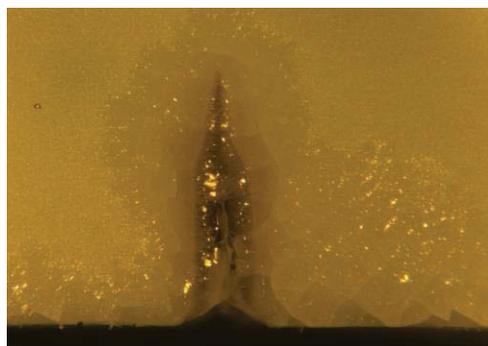


Effect of CMAS Reaction on Toughness of HfO_2 -Si Bond Coat and $\text{Yb}_2\text{Si}_2\text{O}_7$ EBC

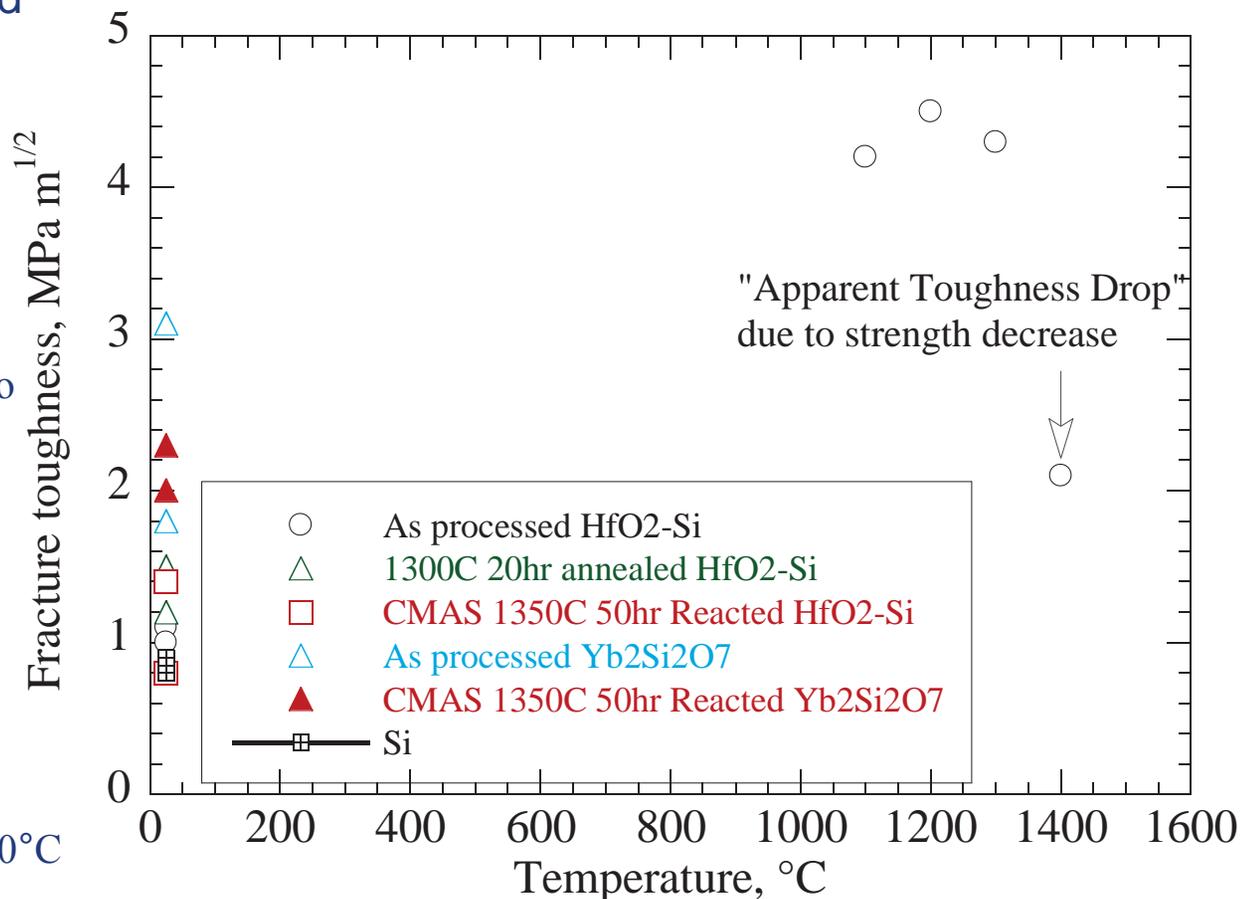
- HfO_2 -Si bond coat and ytterbium di-silicate fracture toughness studied
 - HfO_2 -Si toughness $>4\text{-}5 \text{ MPa m}^{1/2}$ achieved at higher temperature
 - Annealing heat treatments at 1300°C improved lower temperature toughness
 - CMAS effect unclear due to the compounded effects of possible 1350°C CMAS reaction degradation and annealing
- Ytterbium silicate EBC toughness may also be reduced due to CMAS reactions
 - More measurements are needed



HfO_2 -Si illustrating notch distortion due to CMAS exposure at 1350°C for 50 hrs

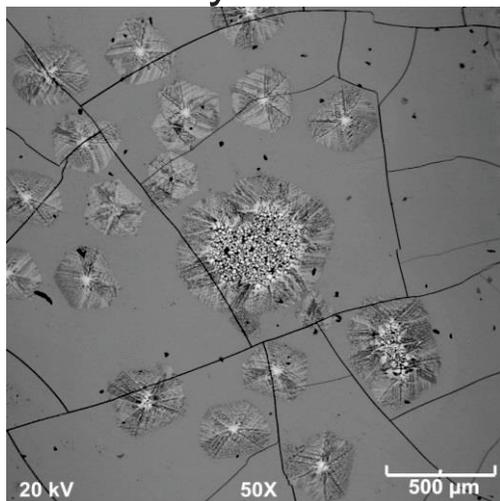


$\text{Yb}_2\text{Si}_2\text{O}_7$ notch after CMAS exposure at 1350°C for 50 hrs

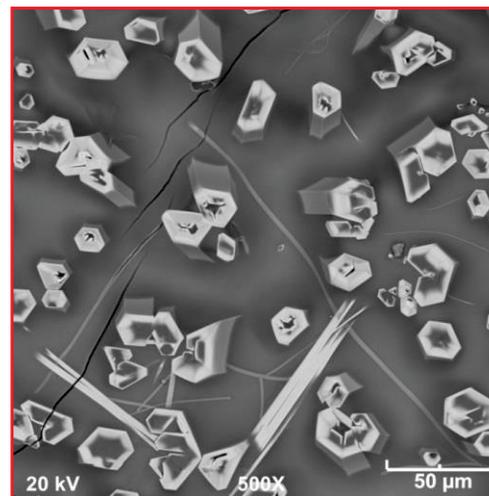


EBC CMAS Surface Reactions

- Ytterbium- and yttrium-disilicate silicates reactions and dissolutions in CAMS



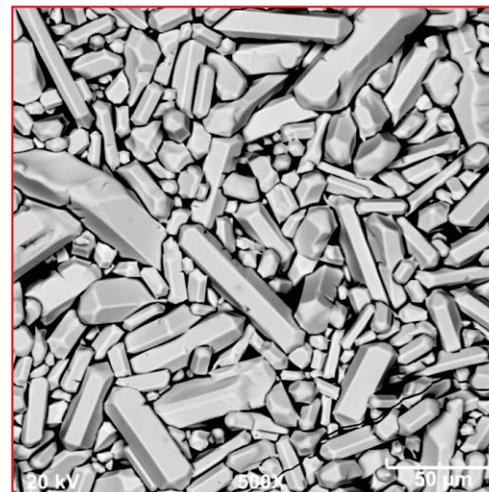
Ytterbium silicate surface CMAS melts: 50 hr
1300°C



Ytterbium silicate surface CMAS melts: 5 hr
1500°C



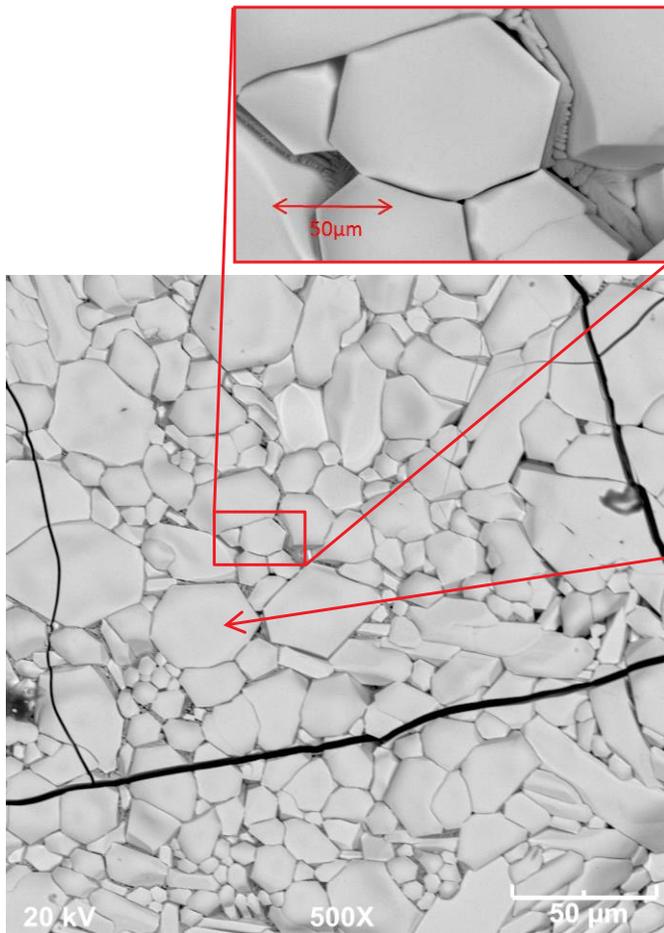
Yttrium silicate surface CMAS melts: 50
hr 1300°C



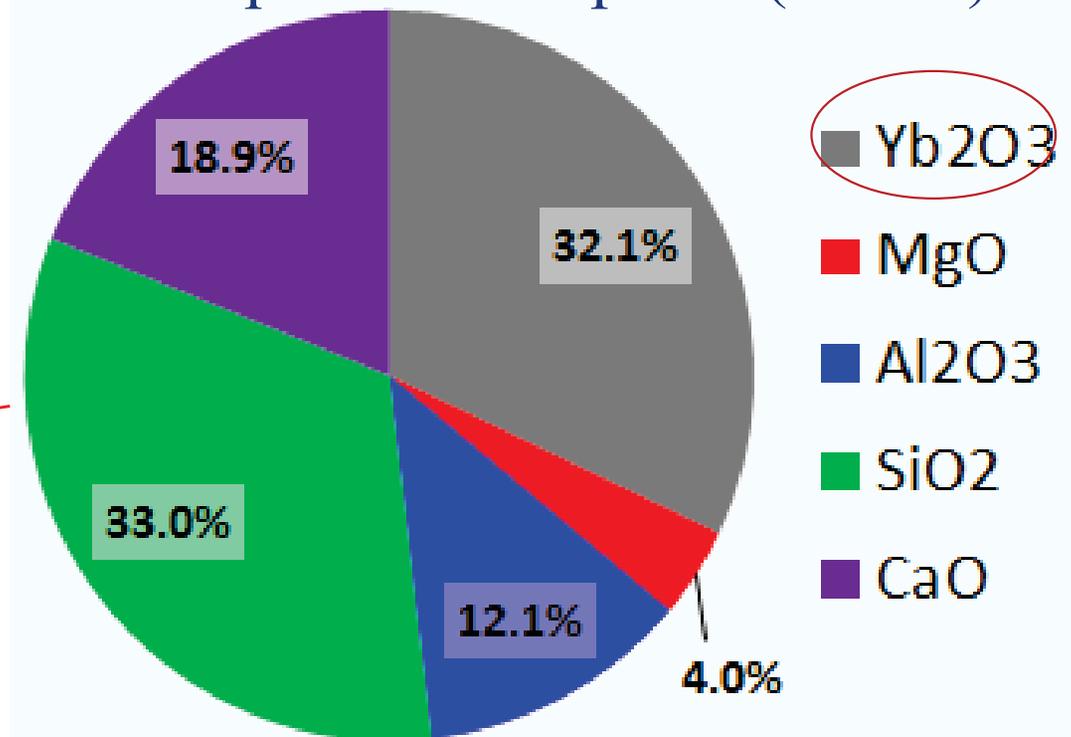
Yttrium silicate surface CMAS melts: 5 hr
1500°C

EBC Reacted Apatite Phases under Long-Term Testing at 1500°C – Ytterbium silicate EBC

- Non stoichiometric characteristics of the CMAS – rare earth silicate reacted apatite phases
- Difference in partitioning of ytterbium vs. yttrium in apatite



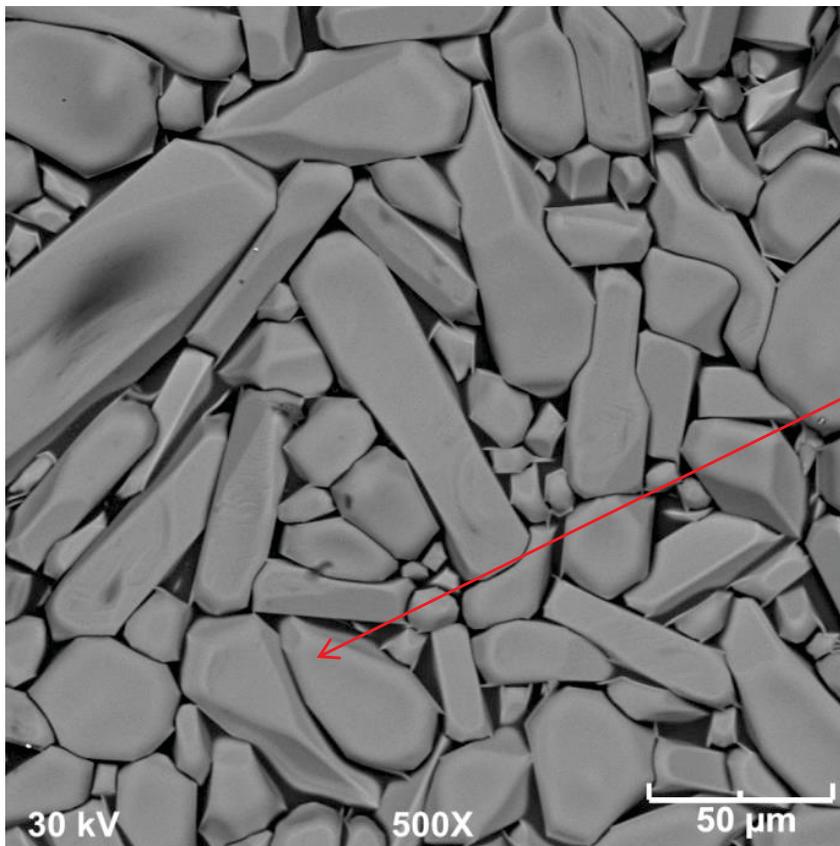
Composition in apatite (100 hr):



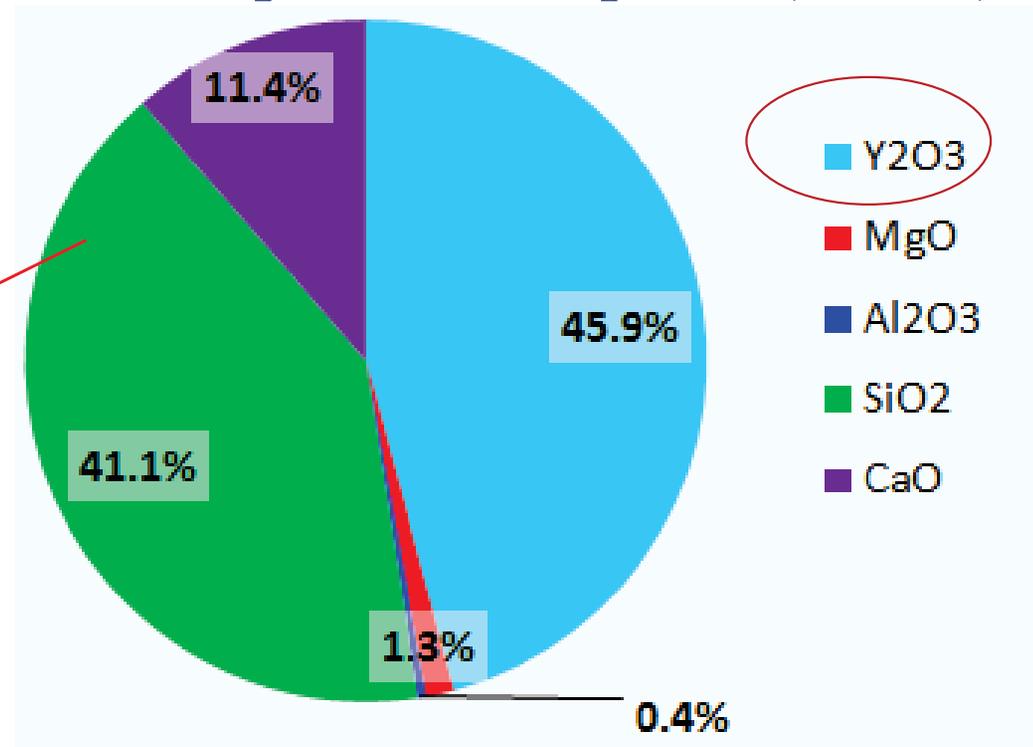
EBC Reacted Apatite Phases under Long-Term Testing at 1500°C: Yttrium Silicate EBC



- Non stoichiometric characteristics of the CMAS – rare earth silicate reacted apatite phases
- Difference in partition of ytterbium vs. yttrium
 - Average AEO/RE₂O₃ ratio ~ 0.68 for ytterbium silicate – CMAS system
 - Average AEO/RE₂O₃ ratio ~ 0.22 for yttrium silicate – CMAS system

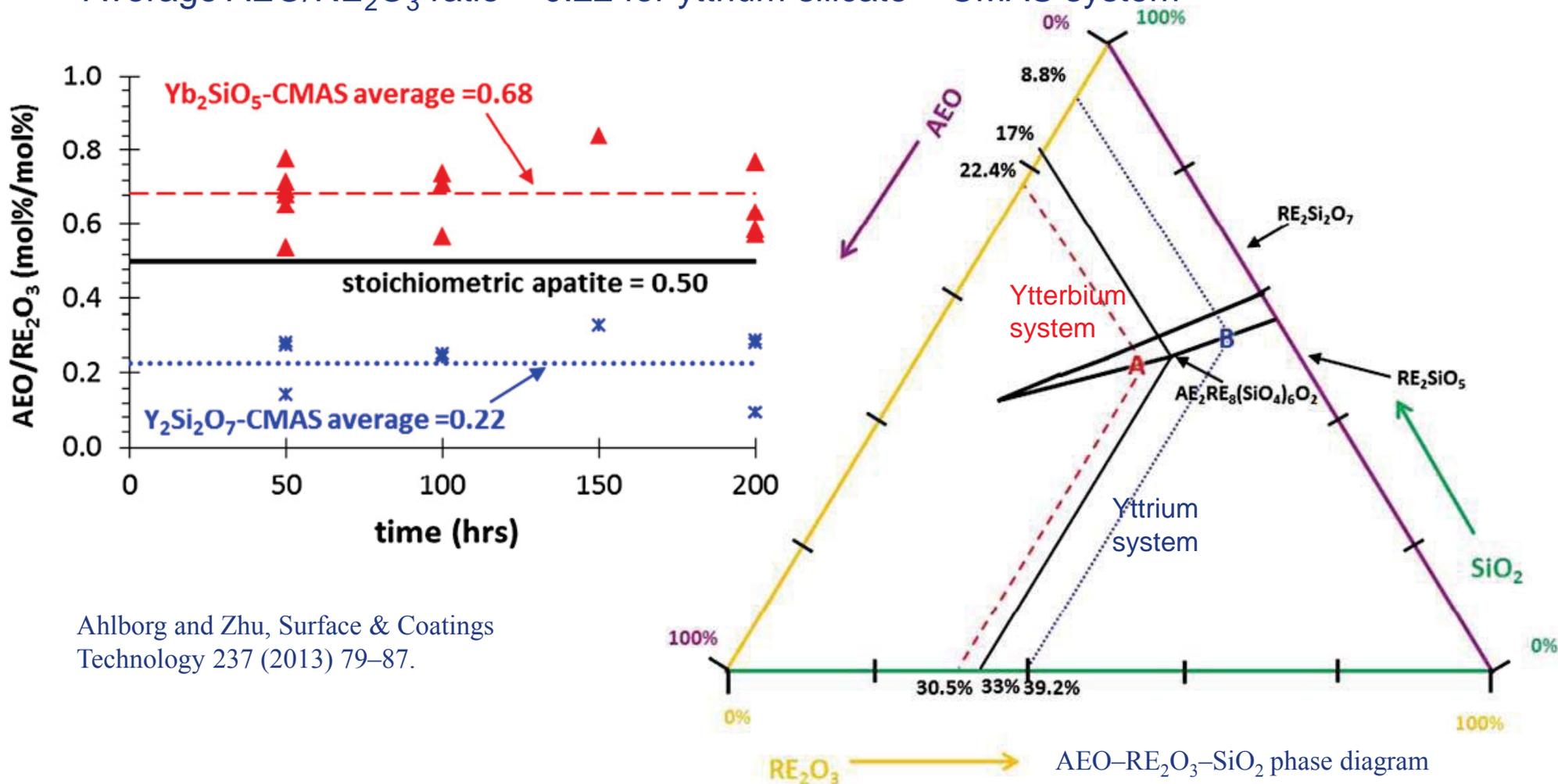


Composition in apatite (100 hr):



Stoichiometry of the Reacted Apatite Phases under Long-Term Testing at 1500°C

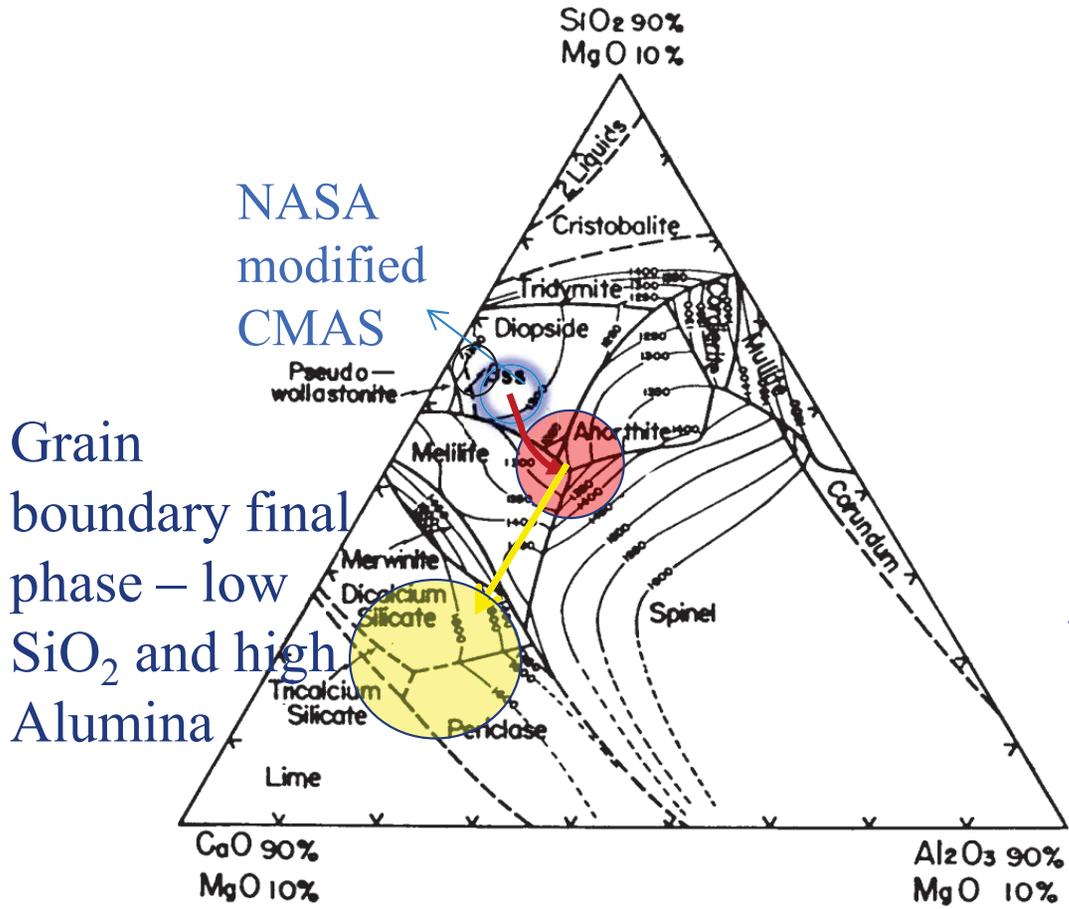
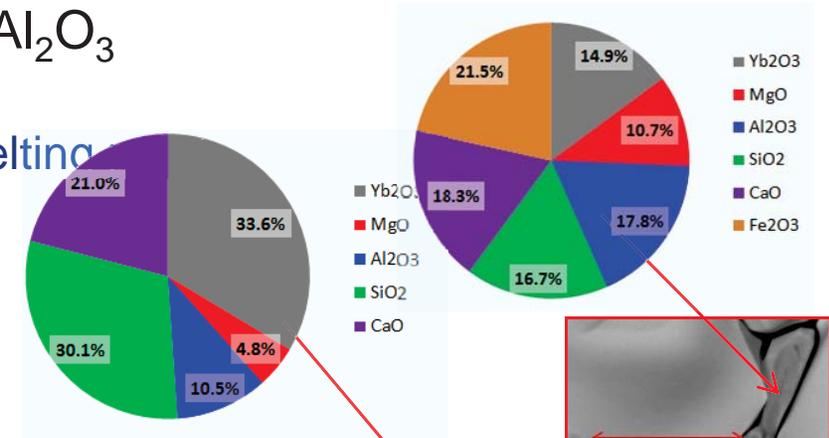
- Non stoichiometric characteristics of the CMAS – rare earth silicate reacted apatite phases – up to 200 hr testing
- Difference in partitioning of ytterbium vs. yttrium in apatite
 - Average AEO/RE₂O₃ ratio ~ 0.68 for ytterbium silicate – CMAS system
 - Average AEO/RE₂O₃ ratio ~ 0.22 for yttrium silicate – CMAS system



Ahlborg and Zhu, Surface & Coatings Technology 237 (2013) 79–87.

Effect of CMAS Reactions on Grain Boundary Phases

- CMAS and grain boundary phase has higher Al_2O_3 content (17-22 mole%)
 - Eutectic region with high Al_2O_3 content $\sim 1200^\circ C$ melting
 - Loss of SiO_2 due to volatility



200 hr, 1500°C

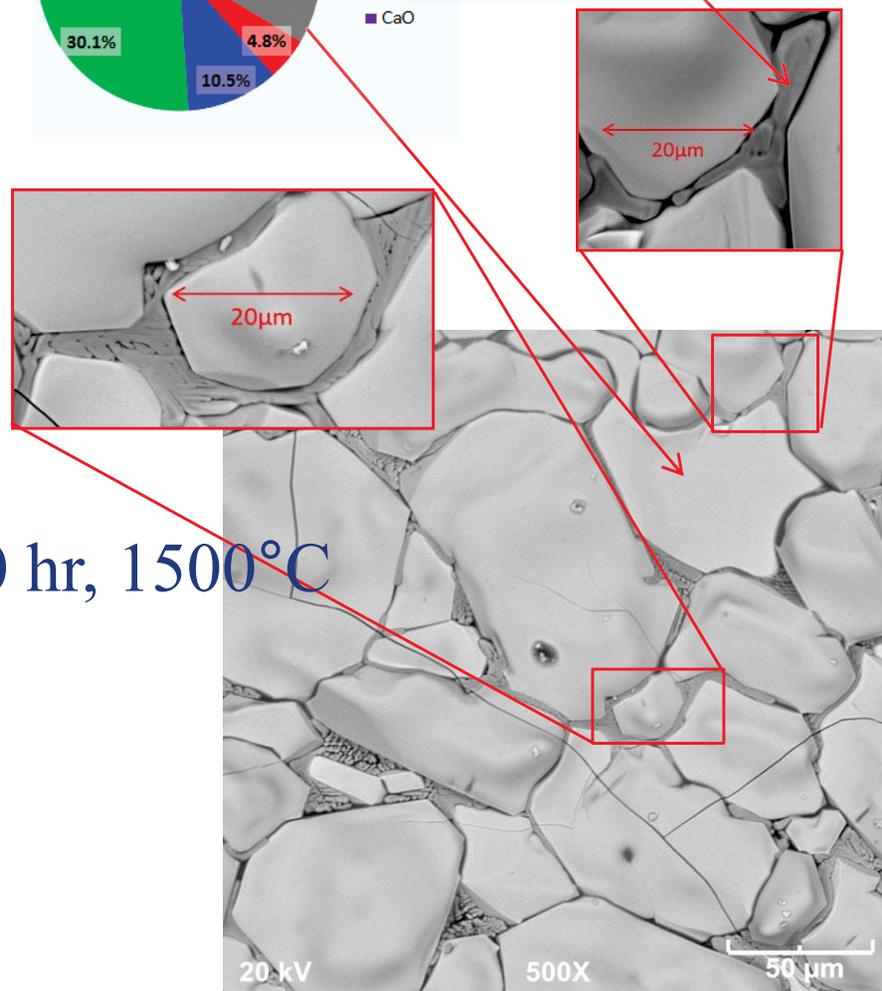
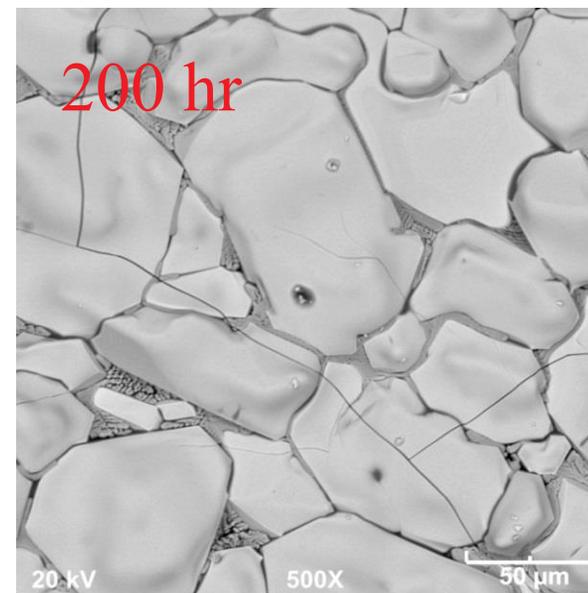
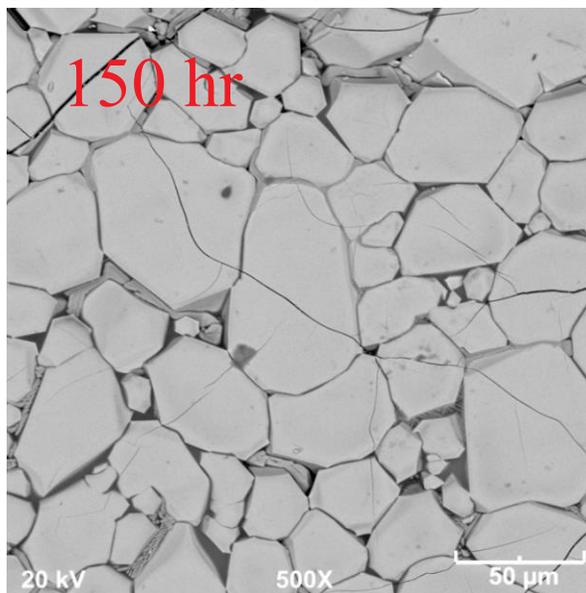
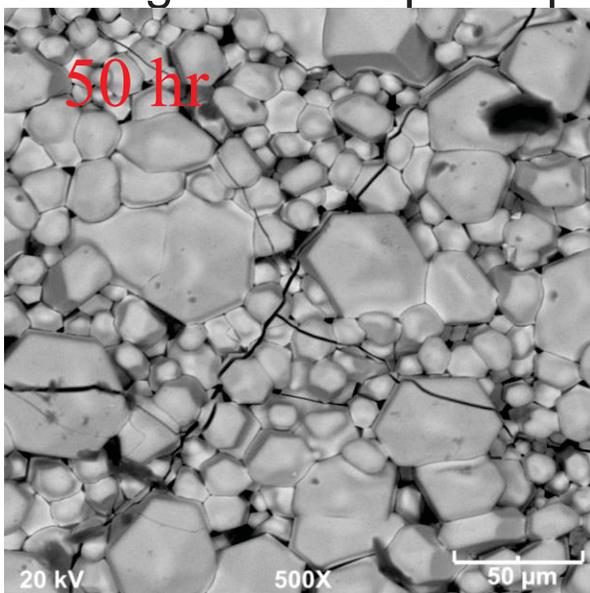


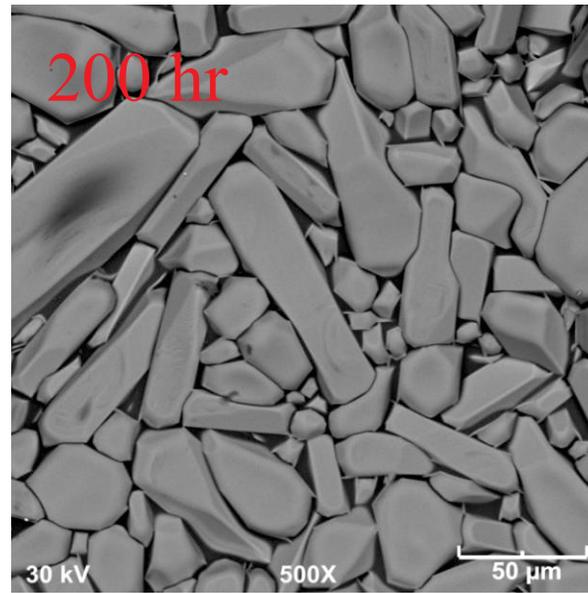
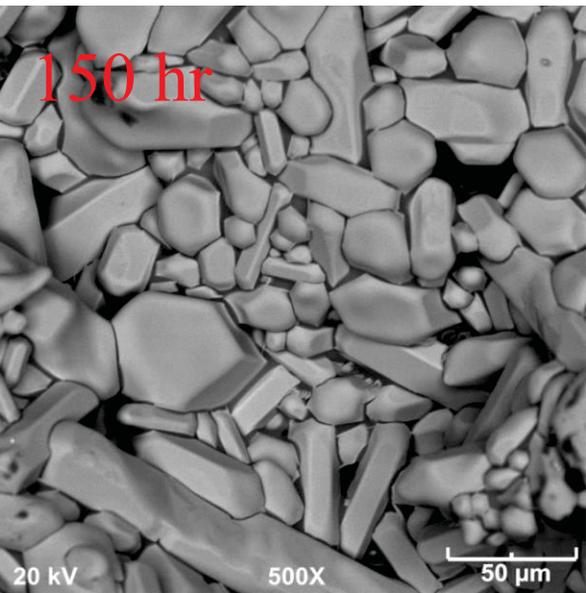
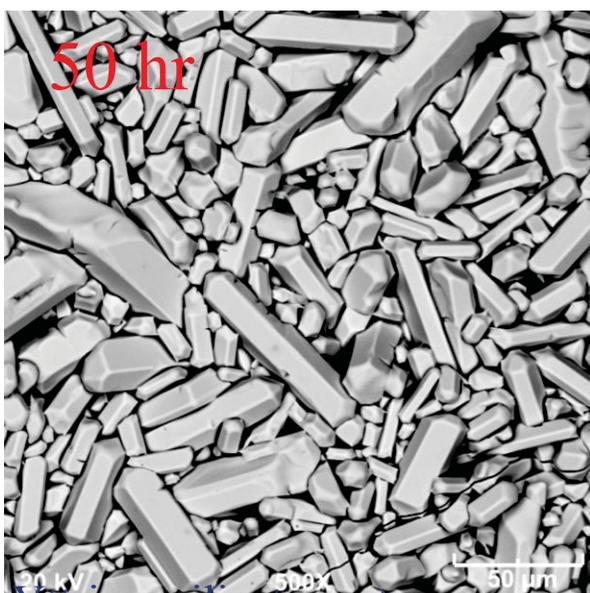
Fig. 4. The 10% MgO plane of the system $CaO-MgO-Al_2O_3-SiO_2$ showing the isotherms and fields of primary crystallization. A.T.Prince, J.Amer.Ceram.Soc., 37(9)1954 p402-408

Rare Earth Apatite Grain Growth

– Grain growth of apatite phase at 1500°C at various times



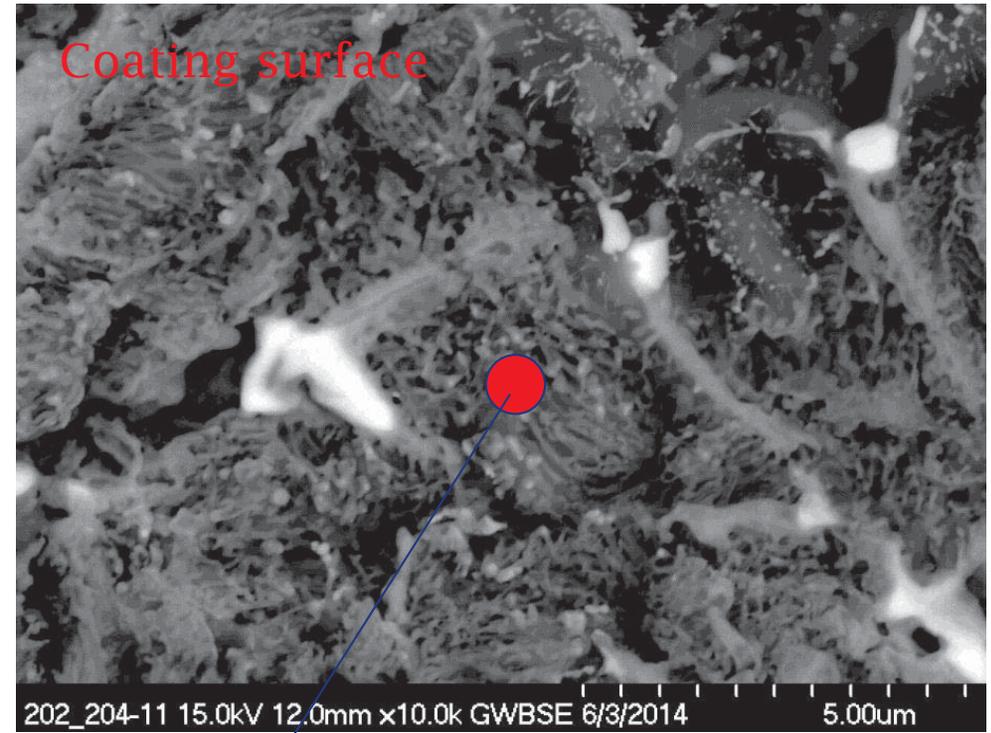
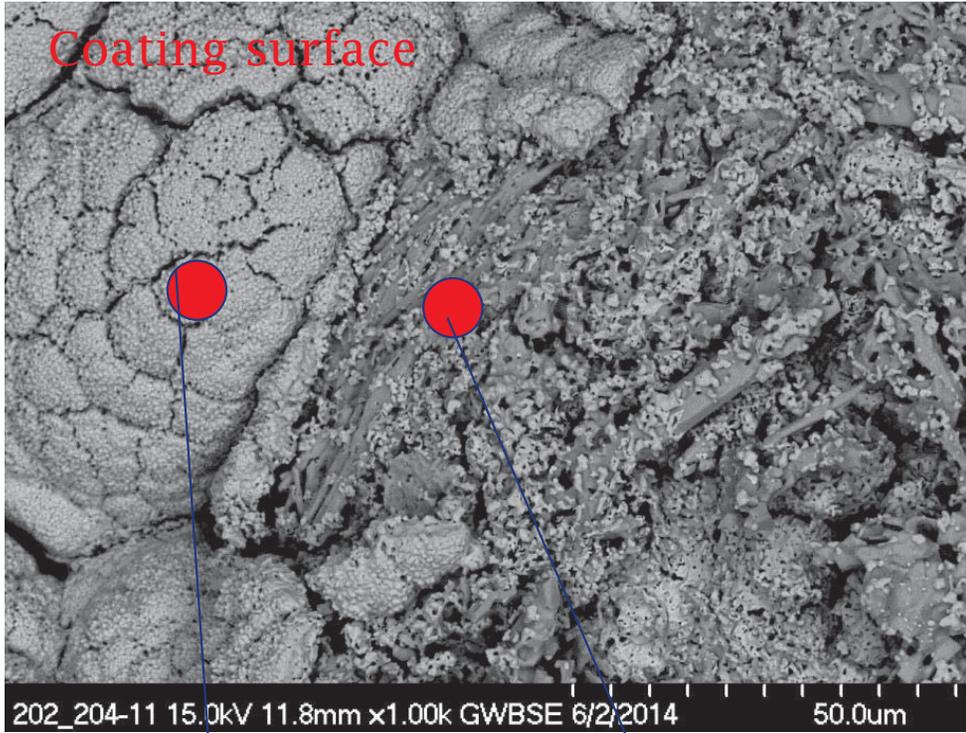
Ytterbium silicate system



Yttrium silicate system

HfO₂-Rare Earth Silicate Composite EBC Systems - Continued

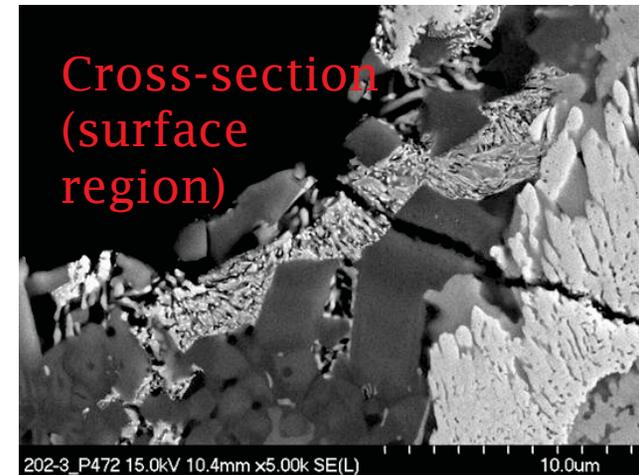
– Silica loss observed in the concentrated CMAS reacted regions



HfO₂ rich phase region

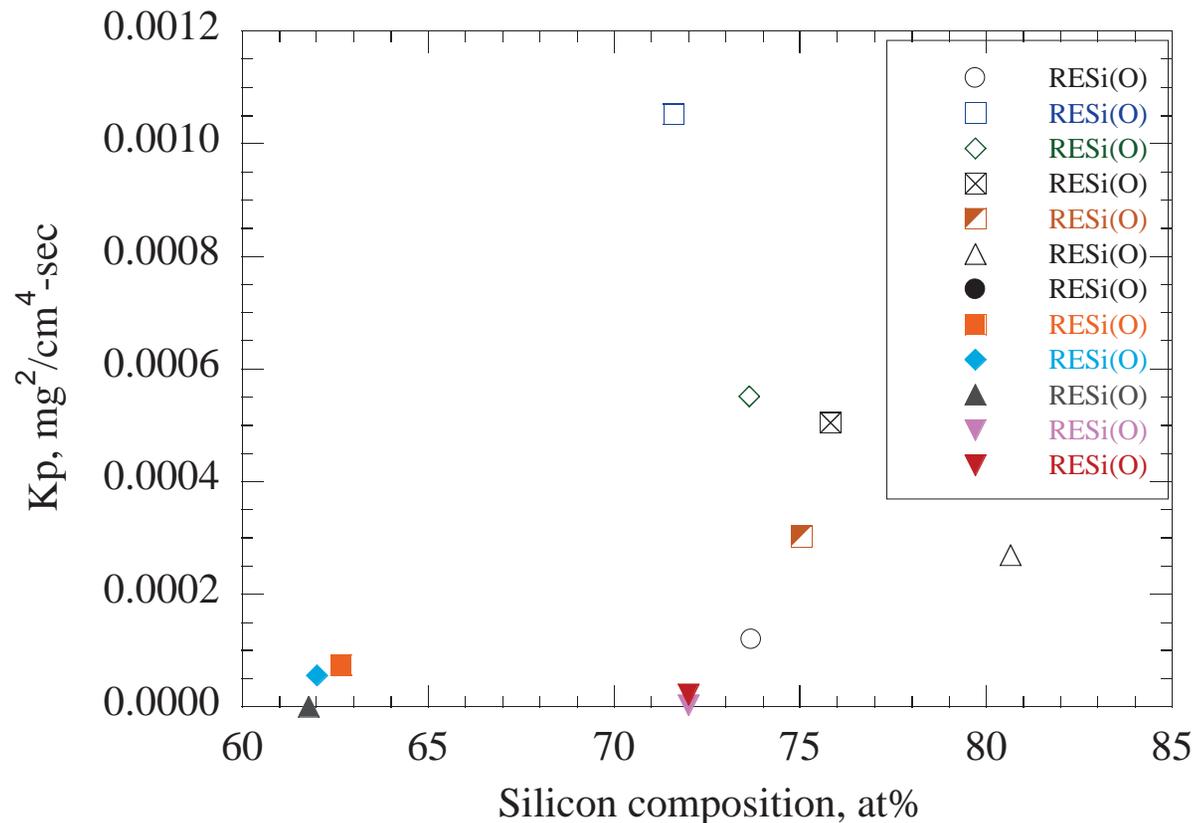
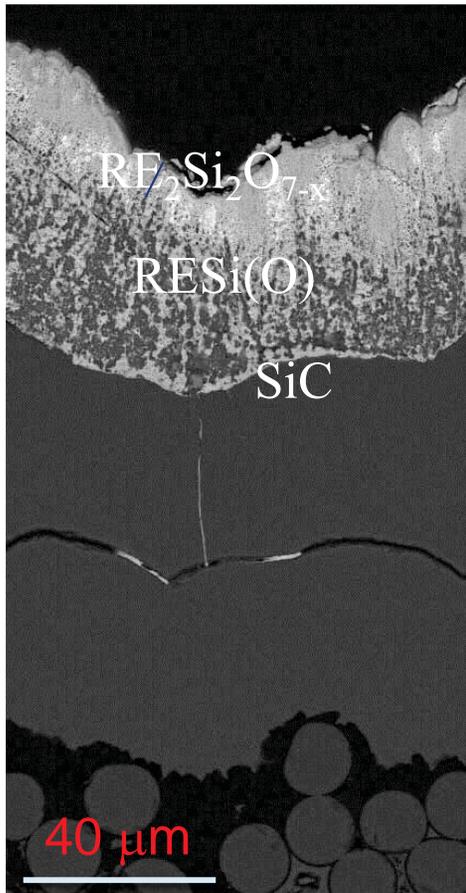
Rare earth silicate - apatite phase rich phase region

CMAS concentrated region, SiO₂ content 20-30 mol% (SiO₂ loss in the steam water vapor tests)



High Stability Rare Earth Silicon Bond Coat with High Melting Point Coating Compositions: Designed with Improved Temperature capability and CMAS Resistance

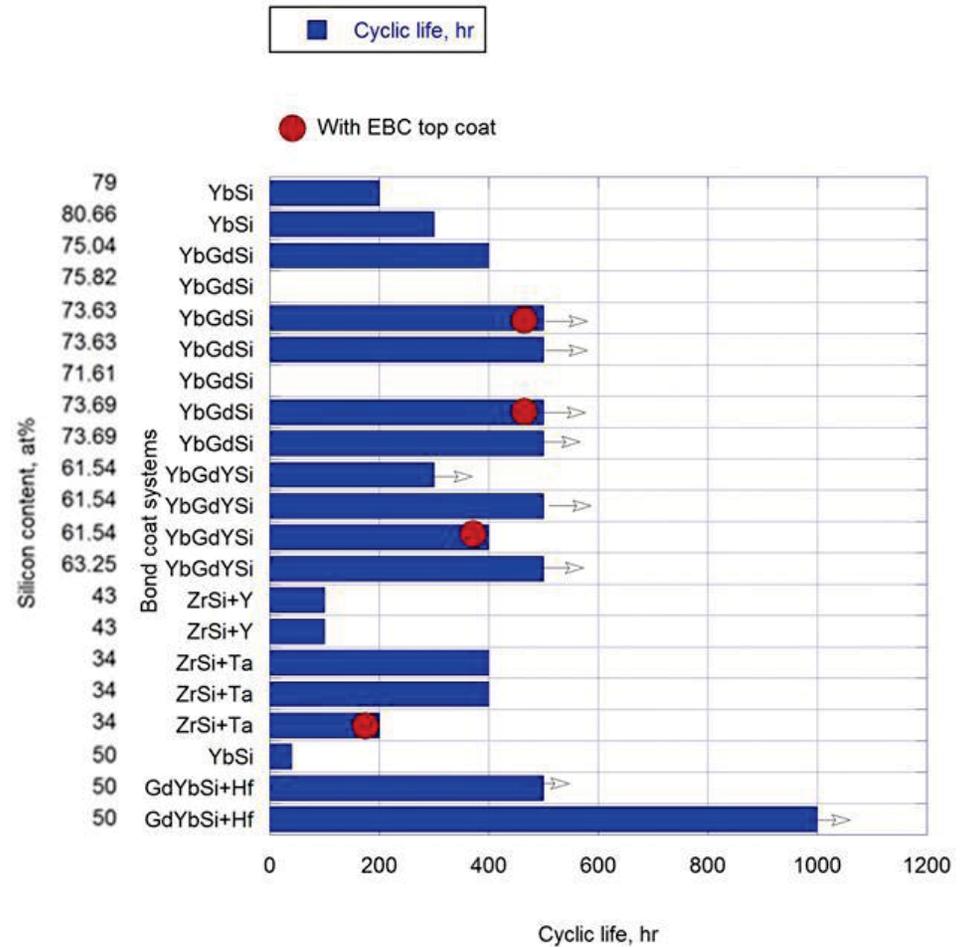
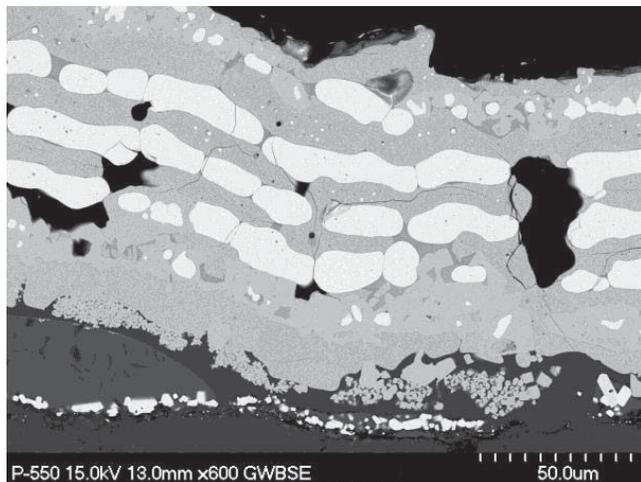
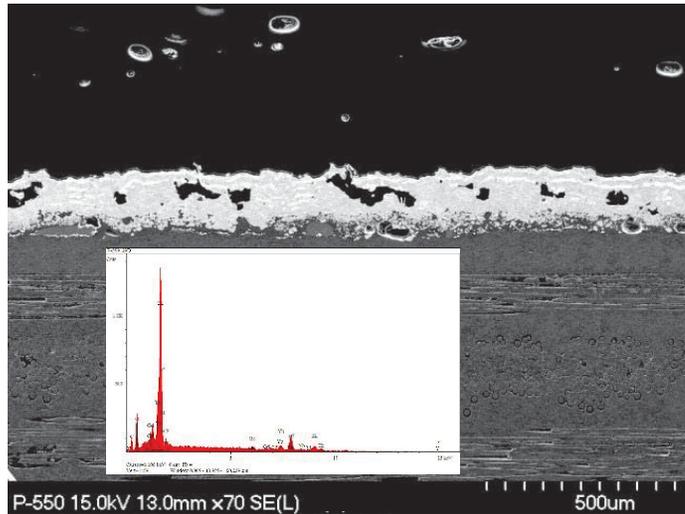
- Thermogravimetric analysis (TGA) in dry O₂ at 1500°C, tested up to 500 hr
- “Protective” scale of rare earth di-silicate formed in oxidizing environments
- Furnace cyclic test life also evaluated at 1500°C



Oxidation kinetics vs Si content

High Stability Rare Earth Silicon Bond Coat with High Melting Point Coating Compositions: Designed with Improved Temperature capability and CMAS Resistance - Continued

- Thermogravimetric analysis (TGA) in dry O₂ at 1500°C, tested up to 500 hr
- “Protective” scale of rare earth di-silicate formed in oxidizing environments
- Furnace cyclic or high heat flux test life evaluated at 1500°C up to 1000 hours with or without CMAS

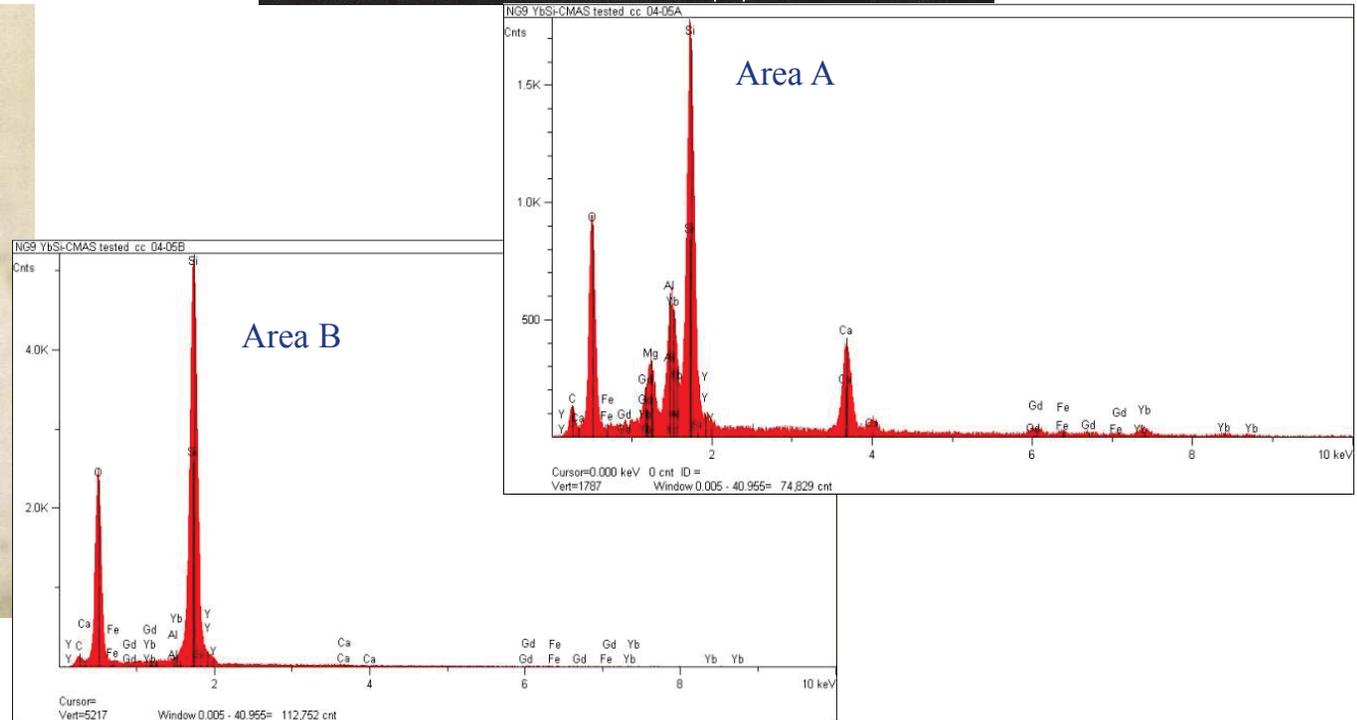
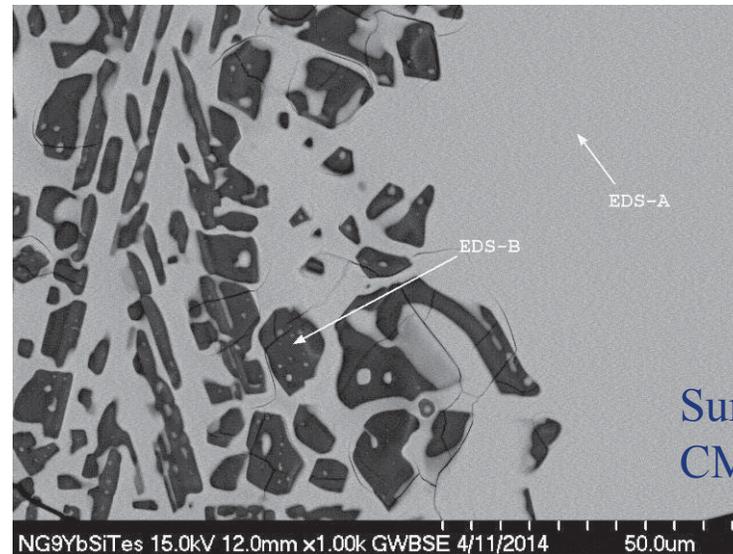


An Yb-Gd2700°F EBC bond coat showed 500hr cyclic durability

FCT life of RE-Si coatings

High Stability and CMAS Resistance Observed from the Rare Earth Silicon High Melting Point Coating Compositions

- Demonstrated CMAS resistance of NASA RE-Si System at 1500°C, 100 hr
- Silica-rich phase precipitation
- Rare earth element leaching into the melts (low concentration ~9 mol%)



CMAS Reaction Kinetics in Bond Coats

- SiO_2 rich phase partitioning in the CMAS melts
- Rare earth content leaching low even at 1500°C
- More advanced compositions are being implemented for improved thermomechanical – CMAS resistance

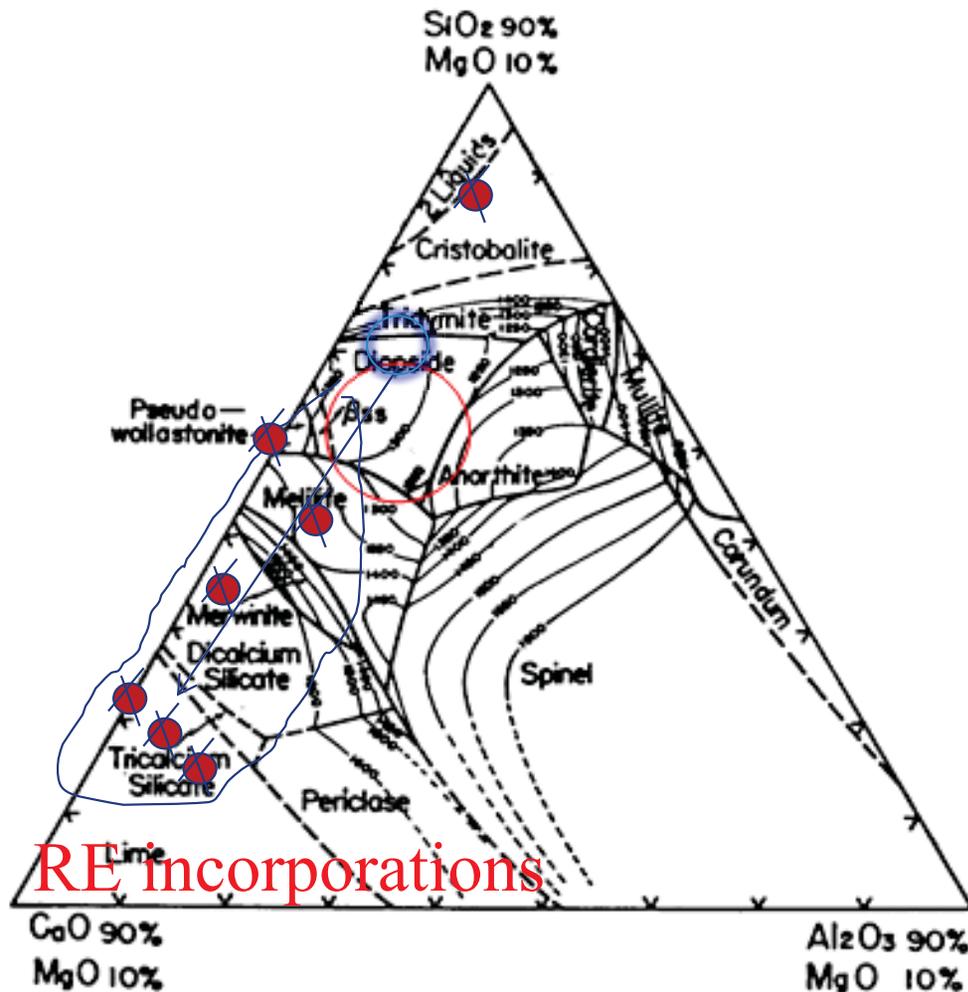
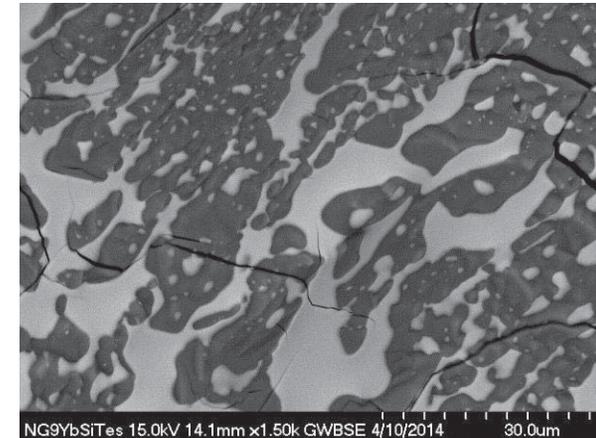


Fig. 4. The 10% MgO plane of the system $\text{CaO-MgO-Al}_2\text{O}_3\text{-SiO}_2$ showing the isotherms and fields of primary crystallization. A.T.Prince, J.Amer.Ceram.Soc., 37(9)1954 p402-408

CMAS Partitioning on RE-Si bond coat, 1500°C , 100hr



Advanced EBC Compositions Improve the Resistance to CMAS



- Controlling CMAS wetting, viscosity, stability and melting points
- Providing better EBC protections for CMCs in CMAS environments
- EBC durability being validated under CMAS-mechanical loading



400 hr, 69 Mpa creep rupture at EBC surface temperature 1400°C



202 hr, 69 MPa creep rupture at EBC surface temperature 1540°C; CMC failure

Advanced EBC Compositions Improve the Resistance to CMAS - Continued

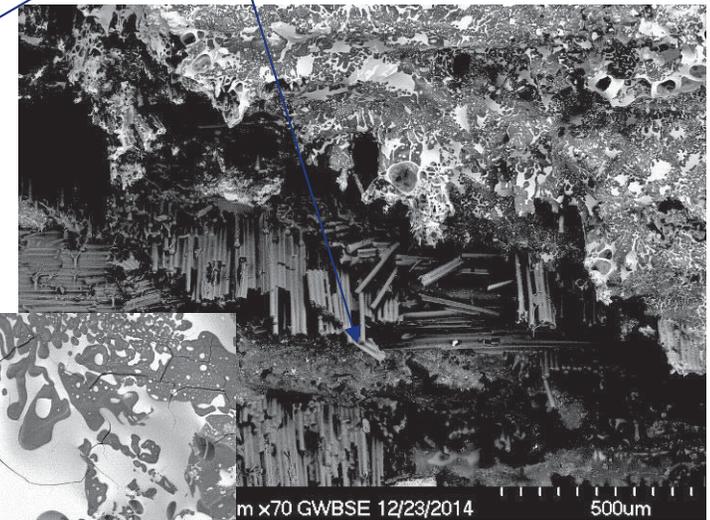
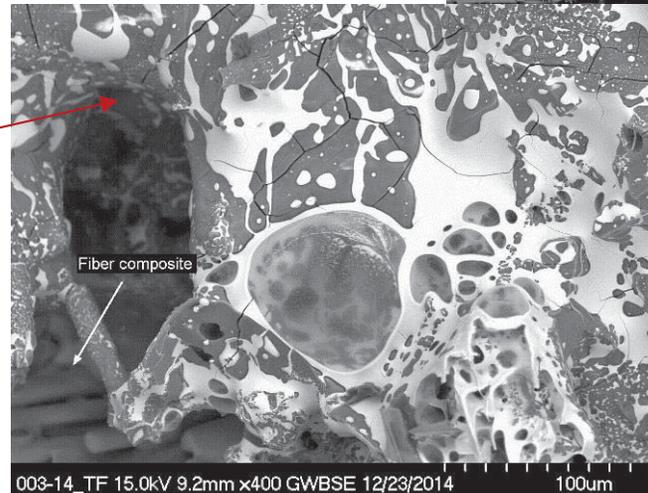
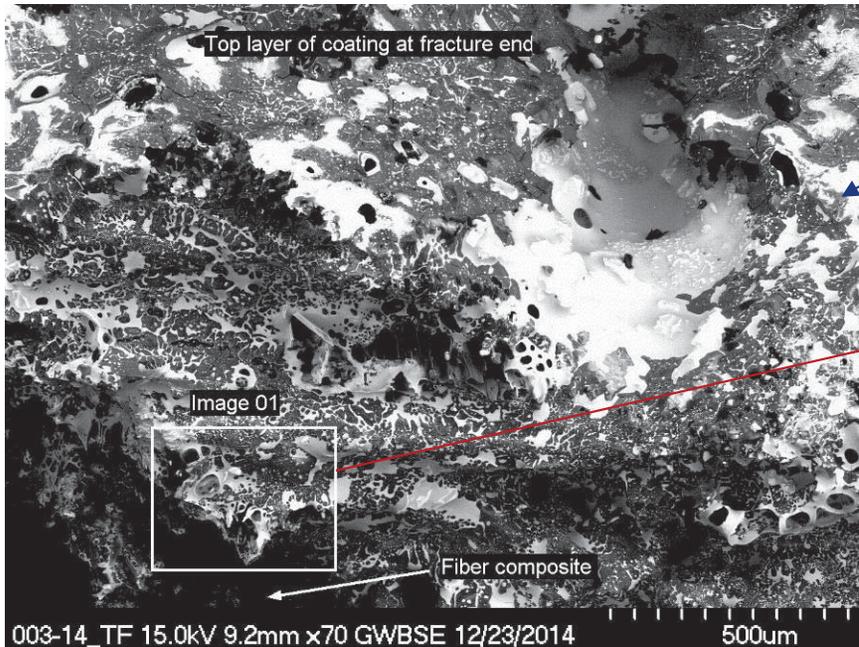
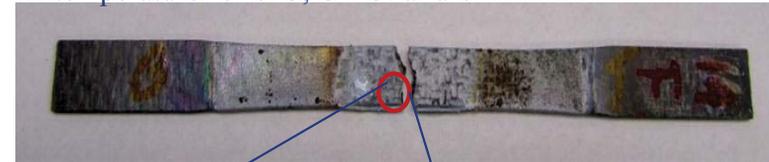


- Controlling CMAS wetting, viscosity, stability and melting points
- Providing better EBC protections for CMCs in CMAS environments
- EBC durability initially validated under long-term CMAS-mechanical loading

400 hr, 69 Mpa creep rupture at EBC surface temperature 1400°C

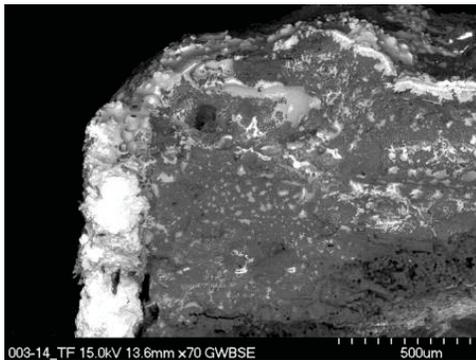


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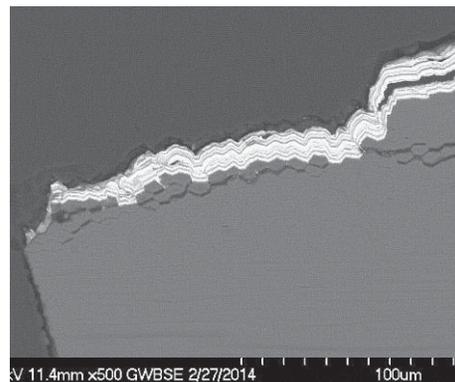


Creep-Fatigue of EBCs-CMCs in Complex Heat Flux and Simulated Engine Environments

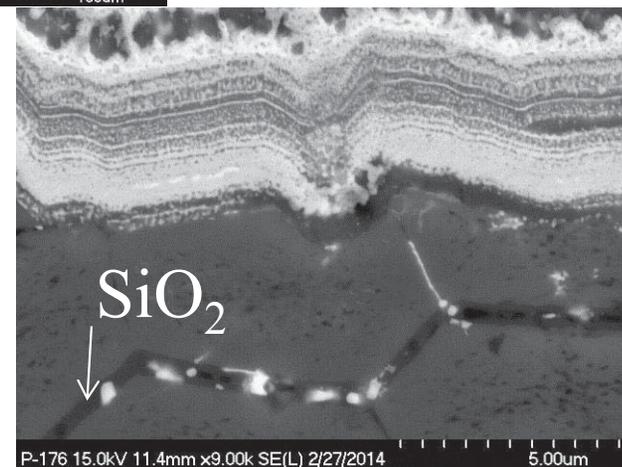
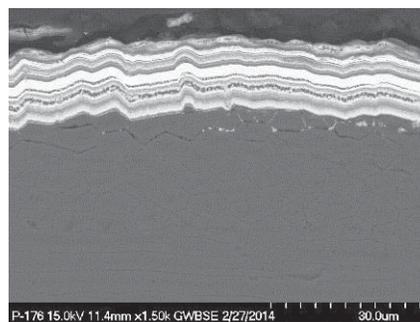
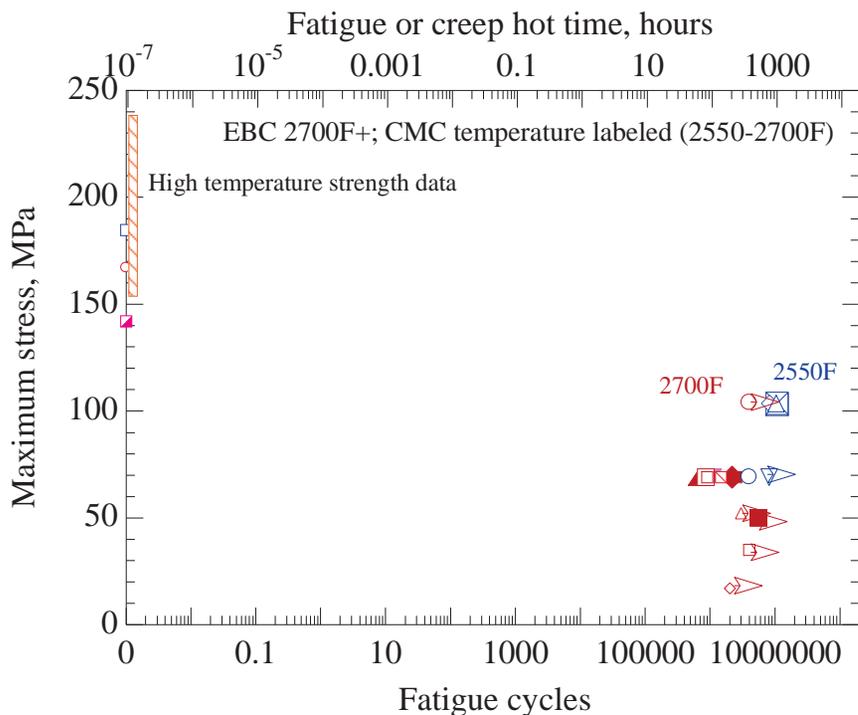
- Long-term creep and fatigue used to validate EBCs at various loading levels
- Demonstrated 2700°F EBC and bond coat capability in complex environments



Fracture surface; 200+ hr at 2700°F+ creep rupture testing with CMAS; Advanced EBC protected CMCs



Advanced Bond Coat on CMC – intact after fatigue test with 15 ksi load and 2600-2700°F surface temperature for 460 hot hours



Stress-oxidation and stress-CMAS environmental testing

Advanced Bond Coat on CMC – intact after fatigue test with 15 ksi load and 2600-2700°F surface temp for 460 hot hours



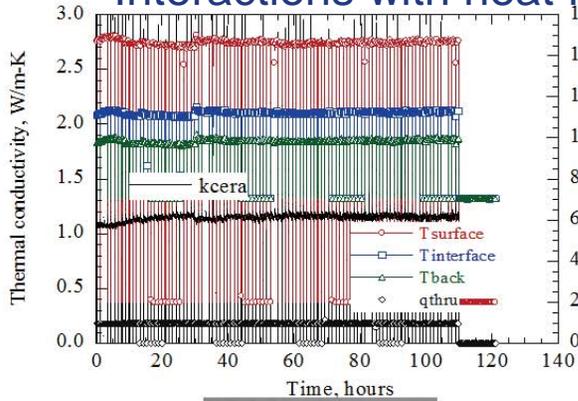
Summary

- CMAS degradation remains a challenge for emerging turbine engine environmental barrier coating – SiC/SiC CMC component systems
- CMAS leads to lower melting point of EBC and EBC bond coat systems, and accelerated degradations
- NASA advanced EBC compositions showed promise for CMAS resistance at temperatures up to 1500°C+, and in combined with mechanical loading
- We have better understanding of CMAS interaction with rare earth silicates, and in controlling the compositions for CMAS resistance while maintaining high toughness
- We are developing better standardized CMAS testing, and working on CMAS induced life reductions, helping validate life modeling

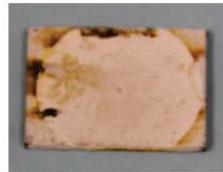
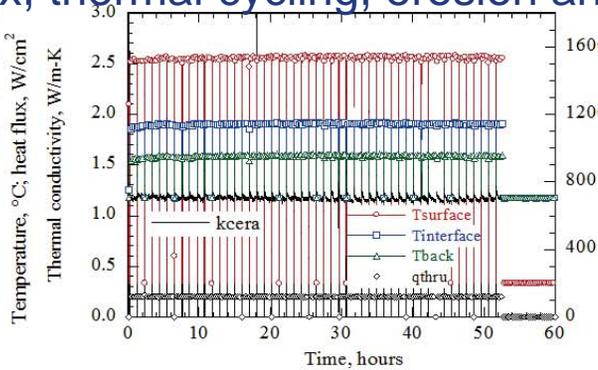
EBC-CMAS Degradation under Thermal Gradients

– Effect of CMAS concentration on EBC-CMC system cyclic durability

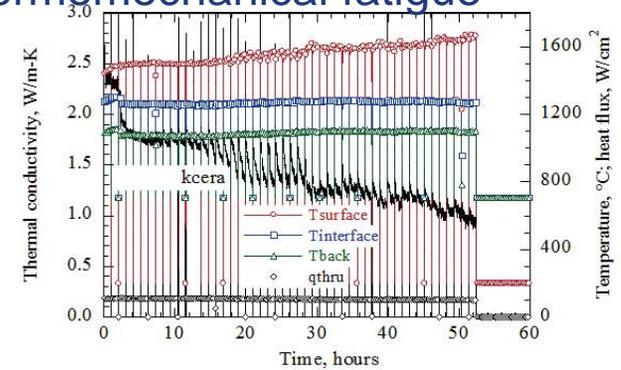
- CMAS reacts with high SiO₂ activity layer and reducing melting point
- Low tough reaction layers such as apatite phases
- Interactions with heat flux, thermal cycling, erosion and thermomechanical fatigue



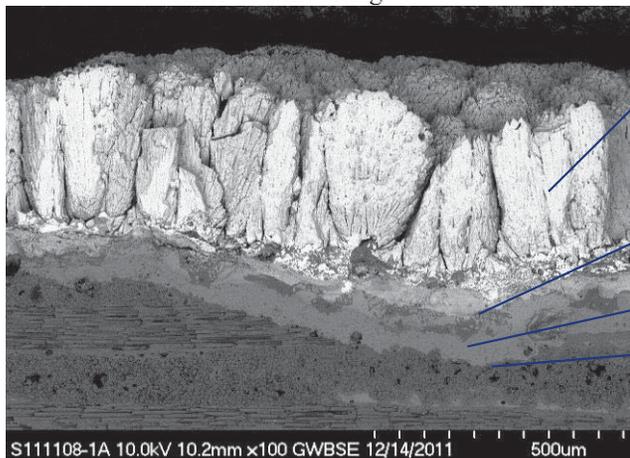
After 100 hr testing



After 50 hr testing



After 50 hr testing



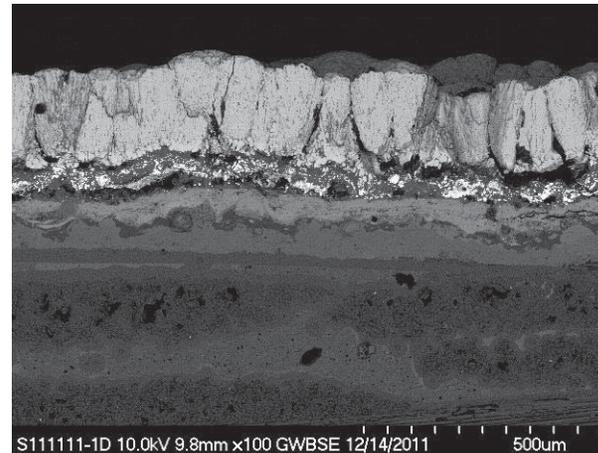
EB-PVD ZrO₂

HfO₂-Yb₂O₃-
Aluminosilicate

Yb₂Si₂O₇

Si

S111108-1A 10.0kV 10.2mm x100 GWBSE 12/14/2011 500um



More severe degradation and delamination:

T_{surface}

1500°C

T_{interface}

1316°C

S111111-1D 10.0kV 9.8mm x100 GWBSE 12/14/2011 500um